Supporting Information for

# Boosting Zn||I2 Battery's Performance by Coating a Zeolite-Based

## **Cation-Exchange Protecting Layer**

Wenshuo Shang<sup>1, #</sup>, Qiang Li<sup>2, #</sup>, Fuyi Jiang<sup>1, \*</sup>, Bingkun Huang<sup>1</sup>, Jisheng Song<sup>1</sup>, Shan Yun<sup>3</sup>, Xuan Liu<sup>1</sup>, Hideo Kimura<sup>1</sup>, Jianjun Liu<sup>2, \*</sup>, Litao Kang<sup>1, \*</sup>

<sup>1</sup>College of Environment and Materials Engineering, Yantai University, Yantai 264005, P. R. China

<sup>2</sup>State Key Laboratory of High-Performance Ceramics and Superfine Microstructure, Shanghai Institute of Ceramics, Chinese Academy of Sciences, Shanghai 200050, P. R. China

<sup>3</sup>Key Laboratory for Palygorskite Science and Applied Technology of Jiangsu Province, Huaiyin Institute of Technology, Huai'an 223003, P. R. China

<sup>#</sup>W.S. Shang and Q. Li contributed equally to this paper and are co-first authors.

\*Corresponding authors. E-mail: <u>fyjiang@ytu.edu.cn</u> (F.Y. Jiang); <u>jliu@mail.sic.ac.cn</u> (J. Liu); <u>kanglitao@ytu.edu.cn</u> (L.T. Kang)

## **Supplementary Figures and Table**



**Fig. S1 a** XRD patterns, **b** SEM and EDS elemental mapping, **c** thermogravimetric (TG) curve of the pristine or  $Zn^{2+}$ -exchanged zeolite.

As evidenced by XRD analyses (Fig. S1a), the employed zeolite powder consists of FAU framework types (JCPDS No. 38-0241) and ETR framework types (JCPDS No. 71-1557) compounds, according to the classification recommended by International Zeolite Association

(IZA) (C. Baerlocher and L.B. McCusker, Database of Zeolite Structures: <u>http://www.iza-structure.org/databases/</u>). The Si/Al molar ratio of this zeolite powder is determined to be 6.1 by using the inductively coupled plasma optical emission spectroscopy (ICP-OES) measurement. After 6 h soaking in 1M ZnSO<sub>4</sub> solution, a new XRD peak appears at 14.6°, due to the successful exchange of Na<sup>+</sup> with Zn<sup>2+</sup> (Fig. S1b). TG test confirms the high thermal stability of the zeolite up to 500 °C (Fig. S1c). The weight loss from 50 to 200 °C can be attributed to the release of adsorbed and crystal water.



**Fig. S2** SEM and EDS image of **a** activated carbon and **b-d**  $I_2@AC$ . **e** XRD patterns of  $I_2$ , the activated carbon matrix and the  $I_2@AC$  composite. **f**  $N_2$  adsorption and pore size distributions isothermal of the activated carbon matrix and the  $I_2@AC$  composite

Figure S2a-d show that the iodine (I<sub>2</sub>) had been uniformly infused into the activated carbon (AC) matrix in the I<sub>2</sub>@AC composite cathode. After loading, the I<sub>2</sub> transformed from crystalline into amorphous (Fig. S2e), reflecting strong interactions between the matrix and the infused I<sub>2</sub>. The bare AC matrix possesses a high BET quantity adsorbed of 566.2 cm<sup>3</sup> g<sup>-1</sup>, which decrease down to 178.7 cm<sup>3</sup> g<sup>-1</sup> after I<sub>2</sub>-loading (Fig. S2f). These BET results also indicate the successful penetration of I<sub>2</sub> into the pores of the AC matrix.



Fig. S3 TG curves of the dried I<sub>2</sub>@AC cathode slurry in a dynamic nitrogen atmosphere within  $30 \sim 800$  °C



Fig. S4 a, b The cross-sectional and c, d top-view SEM image of Zn-based zeolite coating Zn foil



Fig. S5 SEM images of statically corroded **a-c** bare Zn or **d-f** Zeolite-Zn in 1 M ZnSO<sub>4</sub> electrolyte



**Fig. S6** Structure of **a** FAU and **b** ETR type zeolite constructed and optimized using the first principles<sup>7</sup>. DFT calculation of migration barrier of **c**, **d**  $Zn^{2+}$  and **e**, **f** hydrate  $Zn^{2+}$  within **c**, **e** FAU framework and **d**, **f** ETR framework zeolite lattice



Fig. S7 DFT calculation of migration barrier of  $a I^{3-}$  and b hydrate  $I^{3-}$  within FAU framework



Fig. S8 Fitting results of the a initial and b steady state EIS plots showing in Fig. 2



**Fig. S9** Voltage profiles of the **a** bare Zn//Cu and **b** Zeolite-Zn//Zeolite-Cu asymmetric cells with 1 M ZnSO<sub>4</sub> electrolyte at a constant Zn striping/plating current density of 0.5 mA cm<sup>-2</sup>. The corresponding Columbic efficiencies of these striping/plating processes are shown in Fig. 2d in the main text



**Fig. S10** Enlarged voltage profiles of Zeolite-Zn||Zeolite-Zn symmetric cells under the test condition shown in Fig. 2e



Fig. S11 The different cycle overpotential: a 50 - 51 cycles, b 250-251 cycles

The depth of discharge (DOD) of the symmetric cells was calculated by following equation:

$$DOD = \frac{3.6XM}{\rho N_A nel} \times 100\%$$
(S1)

 $\rho$  represents zinc density  $\rho = 7.14 \text{ g cm}^{-3}$ ;

*n* represents the number of electrons lost for Zn0 converting to  $Zn^{2+}$  n=2;

*M* represents relative molecular mass of Zn  $M = 65.4 \text{ g mol}^{-1}$ ;

*e* represents electric quantity of a charge =  $1.6 \times 10^{-19}$  C;

*l* represents the thickness of Zn foil l = 0.045 mm = 0.0045 cm;

 $N_A$  represents Avogadro constant  $N_A = 6.02 \times 10^{23}$ ;

X represents the capacity in each half cycle during charge/discharge. The deep discharge tasting capacity is 2.5 mAh cm<sup>-2</sup>;

With above parameters, the DOD of this condition is determined to be 9.51%.



Fig. S12 Nyquist plots of the symmetric cells with either Zn or Zeolite-Zn electrodes

Tuble bit i tung fesaris of the regulate pions using the equitation encourt					
Electrode	Rs	CPE <sub>1</sub>	CPE <sub>2</sub>	R <sub>ct</sub>	W
	$(\Omega \text{ cm}^{-2})$	(S s <sup>1/2</sup> cm <sup>-2</sup> )	(S s <sup>1/2</sup> cm <sup>-2</sup> )	$(\Omega \text{ cm}^{-2})$	(S s <sup>1/2</sup> cm <sup>-2</sup> )
Bare-Zn	1.41	3.57×10 <sup>-4</sup>	0.64	157.26	0.52
Zeolite-Zn	2.32	2.45×10 <sup>-3</sup>	0.53	140.50	0.43

Table S1 Fitting results of the Nyquist plots using the equivalent circuit



Fig. S13 GCD cycling tests of symmetric Zn||Zn cells at a large DOD of 40%



**Fig. S14** Surficial SEM images of the **a**, **b** bare-Zn and **c**, **d** Zeolite-Zn electrodes after cycling test. To clearly show the morphology of underneath Zn, the zeolite layer on the Zeolite-Zn had been peeled off before SEM observation



**Fig. S15** Cross sectional SEM images of Zn foils in the bare- and Zeolite-Zn electrodes after cycling test. On the surface of the bare Zn electrolyte, there are obviously detaching debris that indicate the formation of "dead Zn"



**Fig. S16** Typical GCD curves of the Zeolite-Zn $||I_2|$  battery at 0.1 A g<sup>-1</sup>. The green and reds lines with arrows show the charge and discharge medium voltages (1.278 and 1.224 V, respectively)



**Fig. S17** Cycling performance of a pure activated carbon cathode with Zeolite-Zn anode at a current density of **a**  $0.2 \text{ A g}^{-1}$  and **b**  $2 \text{ A g}^{-1}$ 



Fig. S18 The voltage - time curves of Zn-I<sub>2</sub> battery failure with  $\mathbf{a}$ ,  $\mathbf{b}$  0.2A g<sup>-1</sup> and  $\mathbf{c}$ ,  $\mathbf{d}$  2 A g<sup>-1</sup>



Fig. S19 Cycling performance and Coulombic efficiencies (CEs) of the Zeolite-Zn-I<sub>2</sub> battery with 1 M Zn(Ac)<sub>2</sub> at a current density of 2 A  $g^{-1}$ 



**Fig. S20 a** The voltage - time curves (200-207 cycles) and **b** cycling performance of Zeolite-Zn-I<sub>2</sub> batteries with 13.3 mg cm<sup>-2</sup> I<sub>2</sub> mass loading at 0.2 A  $g^{-1}$ 



Fig. S21 a Charge/discharge curves of  $Zn||I_2$  batteries connected in different way. b A LED light board powered by four  $Zn||I_2$  batteries in series connection

The voltage and capacity output can be facilely tailored through battery connections. A single battery delivers a voltage of 1.22 V. Connecting two batteries in series or in parallel can double the voltage or capacity, respectively.

### **Supplementary References**

- [S1] C. Bai, F. Cai, L. Wang, S. Guo, X. Liu et al., A sustainable aqueous Zn-I<sub>2</sub> battery. Nano Res. **11**(7), 3548-3554 (2018). <u>https://doi.org/10.1007/s12274-017-1920-9</u>
- [S2] W. Li, K. Wang, K. Jiang, A high energy efficiency and long life aqueous Zn-I<sub>2</sub> battery. J. Mater. Chem. A 8(7), 3785-3794 (2020). <u>https://doi.org/10.1039/C9TA13081K</u>
- [S3] K. Okhotnikov, T. Charpentier, S. Cadars, Supercell program: a combinatorial structure-generation approach for the local-level modeling of atomic substitutions and partial occupancies in crystals. J. Cheminform. 8(1), 17 (2016). <u>https://doi.org/10.1186/s13321-016-0129-3</u>
- [S4] P.E. Blöchl, Projector augmented-wave method. Phys. Rev. B 50(24), 17953-17979 (1994). <u>https://doi.org/10.1103/PhysRevB.50.17953</u>
- [S5] J.P. Perdew, K. Burke, M. Ernzerhof, Generalized gradient approximation made simple. Phys. Rev. Lett. 77(18), 3865-3868 (1996). <u>https://doi.org/10.1103/PhysRevLett.77.3865</u>
- [S6] G. Kresse, J. Furthmüller, Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set. Phys. Rev. B 54(16), 11169-11186 (1996). <u>https://doi.org/10.1103/physrevb.54.11169</u>
- [S7] H. Cao, X. Cheng, H. Zhang, Beryllium carbide as diffusion barrier against Cu: firstprinciples study. Chin. Phys. B 29(1), 016601 (2020). <u>https://doi.org/10.1088/1674-1056/ab593e</u>