Supporting Information for

NH₃-Induced In-Situ Etching Strategy Derived 3D-

Interconnected Porous MXene/Carbon Dots Films for High

Performance Flexible Supercapacitors

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Supplementary Figures



Fig. S1 Schematic illustration of the fabrication process of CDs and the structure of CDs



Fig. S2 The TEM image of CDs



Fig. S3 Zeta potential of pure MXene and CDs



Fig. S4 Digital images of a pure MXene and b p-MC. c, d Fhe flexibility of p-MC



Fig. S5 Raman spectra of CDs, pure MXene, and p-MC



Fig. S6 XRD patterns of Ti₃AlCN, pure MXene, and p-MC



Fig. S7 HRTEM images of a pure MXene and b p-MC films

Discussion 1. X-ray diffraction (XRD) and HRTEM characterization were carried to investigate the influence of CDs intercalation for the microstructure of MXene. As shown in Fig. S6, the 002 peaks for pure MXene and p-MC films are all located at approximately 6.3°, implying that the interplanar spacing of MXene are very similar in pure MXene and p-MC films. The HRTEM images of the pure MXene and p-MC films also exhibit the similar interplanar spacing of approximately 1.3 nm, which is in accordance with the XRD results. Thus, we can conclude that the intercalation of CDs into the MXene films can enlarge the interlayer spacing between MXene sheets but not changes the interplanar spacing of MXene.



Fig. S8 The N 1s XPS spectra of a Pure MXene and b p-MC films



Fig. S9 The Ti 2p XPS spectra for **a** pure MXene and **b** p-MC film. The O 1s, XPS spectra for **c** pure MXene and **d** p-MC film

Discussion 2. The Ti 2p and O 1s XPS spectra were carried to investigate the changes of functional groups on the surface of p-MC (Fig. S9). As the Ti 2p XPS spectra shown in Fig. S9a and S9b, the p-MC film shows weaker peak intensity of Ti-F bond, indicating the decrease of F-containing functional groups on the surface of MXene [S1-S2]. In the O 1s XPS spectra (Fig. S9c, d), the p-MC film also exhibits a weaker C-Ti- O_x peak and a stronger TiO₂ peak than pure MXene, suggesting the annealing treatment can remove part of the O-containing functional groups and cause more oxidation of surface Ti atoms for p-MC film [S3-S6]. Thus, we conclude that the annealing treatment can remove part of O-containing and F-containing functional groups and lead to partial oxidation for Ti atoms on the surface of MXene.



Fig. S10 Top-view SEM images of **a** pure fresh MXene and **b** annealed pure MXene films



Fig. S11 Top-view SEM images of a fresh MC and b p-MC films



Fig. S12 The 3D infrared spectra for CDs



Fig. S13 SEM image of pure MXene annealed at NH3 atmosphere



Fig. S14 Electrochemical performance for MC-n films and annealed MC-n films. **a** Capacitance of the fresh and annealed film electrodes as a function of CDs content at 2 A g^{-1} . **b** Rate performance of fresh MC-n films from 2 A g^{-1} to 50 A g^{-1} . **c** Rate performance of annealed MC-n films from 2 A g^{-1}



Fig. S15 CV curves of pure MXene film from 5 to 500 mV s^{-1}



Fig. S16 GCD curves of a p-MC and b pure MXene films from 2 to 20 A g^{-1}

Sample	Equivalent circuit model	$R_{s}\left(\Omega ight)$	$R_{ct}\left(\Omega ight)$
Pure MXene		1.202	1.049
p-MC		1.134	0.032

Table S1 Equivalent circuit model and corresponding fitting values of pure MXene film

 and p-MC film



Fig. S17 CV curves of p-MC based flexible solid-state supercapacitor



Fig. S18 GCD curves for p-MC based supercapacitor at various current density





Supplementary References

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