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HIGHLIGHTS

Selective CO₂ Electroreduction to Multi-Carbon Products on Organic‑Functionalized CuO Nanoparticles by Local Micro‑Environment Modulation

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- Developing surface-functionalized catalysts with exceptional activity and selectivity for the electrocatalytic CO₂RR to C_{2+} products under neutral electrolyte.
- A remarkable C₂₊ FE of nearly 90% at an unprecedented current density of 300 mA cm⁻² and maintain high FE (>80%) at the wide current density performance (100–600 mA cm⁻²) in neutral environments using a flow cell as well as MEA electrolyzer.
- Mechanical study reveals that reducing the coverage of coordinated K·H₂O water increased the probability of intermediate reactants (CO) interacting with the surface, thereby promoting efficient C–C coupling and enhancing the yield of C_{2+} products.

ABSTRACT Surface functionalization of Cu-based catalysts has demonstrated promising potential for enhancing the electrochemical CO₂ reduction reaction (CO_2RR) toward multi-carbon (C_{2+}) products, primarily by suppressing the parasitic hydrogen evolution reaction and facilitating a localized CO₂/CO concentration at the electrode. Building upon this approach, we developed surface-functionalized catalysts with exceptional activity

and selectivity for electrocatalytic CO₂RR to C_{2+} in a neutral electrolyte. Employing CuO nanoparticles coated with hexaethynylbenzene organic molecules (HEB-CuO NPs), a remarkable C₂₊ Faradaic efficiency of nearly 90% was achieved at an unprecedented current density of 300 mA cm⁻², and a high FE (>80%) was maintained at a wide range of current densities (100–600 mA cm⁻²) in neutral environments using a flow cell. Furthermore, in a membrane electrode assembly (MEA) electrolyzer, 86.14% FE_{C2+} was achieved at a partial current density of 387.6 mA cm⁻² while maintaining continuous operation for over 50 h at a current density of 200 mA cm−2. In-situ spectroscopy studies and molecular dynamics simulations reveal that reducing the coverage of coordinated K⋅H2O water increased the probability of intermediate reactants (CO) interacting with the surface, thereby promoting efficient C–C coupling and enhancing the yield of C_{2+} products. This advancement offers significant potential for optimizing local micro-environments for sustainable and highly efficient C_{2+} production.

KEYWORDS CO_2 electroreduction to C_{2+} ; Neutral electrolyte; Organic-functionalized CuO nanoparticles; Local micro-environment modulation

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1 Introduction

Anthropogenic emissions of $CO₂$ resulting from the utilization of fossil fuels have exacerbated the greenhouse efect within the Earth's atmosphere, leading to concentrations exceeding 420 ppm by 2023 and intensifying the pace at which climate change is unfolding [[1](#page-10-0), [2\]](#page-10-1). To address this ongoing warming trend, it is imperative to dedicate extensive efforts to advancing technologies for $CO₂$ sequestration and conversion [\[3](#page-10-2)–[5\]](#page-10-3). Electrochemical CO_2 reduction (CO_2RR) in the presence of water offers a promising avenue for driving the chemical conversion of $CO₂$ by harnessing renewable electricity from sustainable sources near ambient temperature and pressure $[6, 7]$ $[6, 7]$ $[6, 7]$ $[6, 7]$. In comparison with CH₄, CO, and HCOOH, which are the major C_1 products of the CO₂RR $[8-15]$ $[8-15]$, C₂₊ hydrocarbons and oxygenates (consisting of two or more carbon atoms) exhibit higher energy densities (both volumetric and gravimetric) and greater economic value, thereby experiencing higher global demand [[16–](#page-11-1)[19\]](#page-11-2). Compared to other catalysts, Cu-based materials demonstrate remarkable selectivity for the conversion of CO_2 into C_{2+} products [[18,](#page-11-3) [20](#page-11-4)[–23](#page-11-5)]. However, the selectivity and activity of the $CO₂RR$ are severely limited by parasitic proton reduction (HER) occurring at similar overpotentials and high C–C coupling energy barriers [\[24–](#page-11-6)[26\]](#page-11-7). Consequently, there is a pressing demand for the rational design of advanced Cubased electrocatalysts capable of facilitating these processes and enabling efficient synthesis of C_{2+} products.

To achieve industrial-scale production of C_{2+} , it is essential to improve $CO₂$ conversion while suppressing the competing HER. Therefore, optimizing the coverage of *CO and *H on the catalyst surface should be prioritized because they are crucial indicators for predicting C–C coupling in C_{2+} production and dihydrogen evolution for $H₂$ generation. However, strong electrostatic interactions resulted in a substantial increase in the adsorbed *H concentration on the catalyst surface as the applied potential increased. Consequently, this leads to a decrease in *CO coverage and impedes C_{2+} production during industrial current density electrolysis.

To address this issue, several studies have proposed diverse strategies, including regulating crystal facets [\[27,](#page-11-8) [28](#page-11-9)], optimizing lattice tension [\[29](#page-11-10)], manipulating oxidation states [[8,](#page-10-6) [30](#page-11-11)], doping with heteroatoms [[31](#page-11-12)], and molecule modifcation [[32\]](#page-11-13) to optimize the *CO and *H coverage balance. The surface functionalization of Cu-based catalysts not only allows for the modulation of the hydrophilic nature of the catalyst surface but also enables the optimization of the local $CO₂/H₂O$ concentration, thereby influencing the surface coverage of $*CO$ and $*H$, which in turn affects the reaction pathways leading to C_{2+} products. This suggests that the manipulation of micro-environmental conditions can precisely control the adsorption behavior of ions and molecules on the catalyst, thereby efectively regulating the $CO₂RR$. For example, thiol-functionalized pyridine fixation on the Cu surface signifcantly impedes the formation of intermediates associated with CO production, thereby constraining the generation of C_{2+} products [[33](#page-11-14)]. After the application of an adequate coating of polyvinylpyrrolidone polymer onto the Cu nanocrystals, complete conversion of C_{2+} to CH₄ was observed in the CO₂RR products, emphasizing the substantial impact of the molecular surface coverage on product selectivity [[8\]](#page-10-6). However, the adopted strategies primarily focus on the electrolyzer design and optimization of the working conditions, with limited emphasis on catalyst engineering to tailor the intermediate binding energy to optimize both *CO and *H coverage. Hence, a comprehensive understanding of the adjustable *CO/*H coverage at the atomic level is imperative to achieve the $CO₂RR$ of $C₂₊$ products with an industrial-level current density.

With these motivations, we report a hexaethynylbenzene organic molecule (HEB)-modifed Cu nanoparticles (HEB-Cu NPs) catalyst synthesized by in-situ electroreduction of HEB-CuO NPs under $CO₂RR$ conditions and further investigate the role of *CO and H^* coverage in C_{2+} production. Consequently, the HEB-Cu NPs catalyst yielded a remarkable C_{2+} Faradaic efficiency (FE) of nearly 90% at an unprecedented current density of 300 mA cm−2 and maintained a high FE $(>80\%)$ over a wide current density range (100–600 mA cm⁻²) in neutral environments using a flow cell. Furthermore, in a membrane electrode assembly (MEA) electrolyzer, 86.14% FE $_{C2+}$ was achieved at a partial current density of 387.6 mA cm⁻², while maintaining continuous operation for over 50 h at a current density of 200 mA cm−2, surpassing the performance of the pristine Cu NPs catalyst without HEB modifcation. In-situ Raman spectroscopy and molecular dynamics (MD) simulations demonstrated the substantial infuence of HEB molecules on the distribution of hydrogen-bonded water coordinated at the surface of Cu NPs, increasing the coverage of *CO intermediates to further promote efficient $C-C$ coupling

and enhance the yield of C_{2+} products. The aforementioned advancement presents remarkable potential for optimizing the local micro-environment of the reaction, enabling the sustainable and highly efficient production of C_{2+} .

2 Experimental Section

2.1 Materials and Chemicals

Hexakis[(trimethylsilyl)ethynyl]benzene (HEB-TMS, purity>98%) was purchased from Zhengzhou Alfachem Co., Ltd. Copper(I) chloride (CuCl, purity>99.99%) was purchased from Sigma-Aldrich. *N*,*N*-Dimethylformamide (DMF, AR) and ethanol absolute (C_2H_5OH, AR) were purchased from Sinopharm Chemical Reagent Co., Ltd. Potassium chloride (KCl, purity > 99.5%) and Nafion perfuorinated resin solution (5 wt%) were purchased from Shanghai Aladdin Biochemical Technology Co., Ltd. Potassium hydroxide (KOH, purity>95%) was purchased from Shanghai Macklin Biochemical Technology Co., Ltd. Deuterium oxide (D₂O, purity > 99.9 atom%D) was provided by Hebei Bailingwei Superfne Materials Co., Ltd. Cation exchange membrane (N117) and carbon paper (29 BC) were provided by Suzhou Sinero Technology Co., Ltd. All chemicals were purchased from commercial sources and used without further purifcation.

2.2 Materials Characterizations

Scanning electron microscopy (SEM, HITACHI 8100) was used to examine the surface morphologies of HEB-CuO electrode. The morphology of HEB-CuO and CuO NPs was acquired by transmission electron microscopy (TEM, HITACHI HT7700). Scanning transmission electron microscopy (STEM) measurements and energy-dispersive X-ray spectroscopy (EDS) elemental mapping were taken using Talos F200S-G2. Aberration-corrected high-angle annular dark-feld scanning transmission electron microscopy (AC-HAADF-STEM) was carried out on aberration-corrected JEM-ARM300F. X-ray difraction (XRD, SmartLab 9kw, Cu-K α (λ = 1.5405 Å) radiation) was adopted to characterize the crystalline structures of the synthesized NPs. The scattering range of 2*θ* was from

20° to 80° at a scanning rate of 2° min⁻¹. X-ray photoelectron spectroscopy (XPS, PHI 5000 VersaProbe IV, Al Kα radiation as an exciting source) was used to determine the surface states of the HEB-CuO and CuO NPs. The Raman spectral analysis was conducted using a LabRAM HR Evolution Confocal Laser Micro-Raman spectrometer operating at a wavelength of 785 nm. ATR-SEIRAS experiments equipped with a liquid nitrogen-cooled MCT detector were performed on a Bruker INVENIO-S. X-ray absorption spectra of Cu K edges were acquired at the beamline 14 W1 of Shanghai Synchrotron Radiation Laboratory (China).

2.3 Synthesis of HEB‑CuO Nanoparticles (NPs)

10, 50, 100, and 150 mg HEB-TMS and 12 mL DMF were added in a glass bottle, respectively, to form transparent solution through ultrasonication for 5 min at room temperature, then 44.8 mg CuCl was added to the above solution and ultrasonicated for 2 min. Afterward, the bottle was sealed and heated to 70 °C in an oven for 24 h. After the reaction, the samples were centrifugated and washed with ethanol absolute. Finally, the samples were dried in a vacuum drying oven, and a black-brown powder was obtained [\[34](#page-11-15)].

2.4 Synthesis of CuO NPs

44.8 mg CuCl was dissolved into 12 mL DMF in a glass bottle and ultrasonicated for 4 min (the procedure was the same). Then the above dispersion was put into a 70 °C oven for 24 h. The fnal product was collected by centrifugation, washed with ethanol for several times, and then dried in vacuum overnight.

3 Results and Discussion

3.1 Synthesis and Characterizations of HEB‑Cu NPs Catalyst

Figure [1](#page-4-0)a shows the procedure for the synthesis of the HEB-Cu NPs catalyst (details in the "Supporting Information") via the direct coupling reaction of hexakis[(trimethylsilyl) ethynyl]benzene (HEB-TMS) in *N*,*N*-dimethylformamide (DMF) using CuCl as the catalyst [\[34\]](#page-11-15). HEB-TMS and CuCl were added to DMF and subsequently transferred to a hermetically sealed reaction vessel for ultrasonic dispersion. After undergoing a 24-h reaction at 70 °C in the presence of oxygen, a black precipitate was formed at the bottom of the container (Fig. S1). The generated solid was recovered by centrifugation and washed consecutively with ethanol and H₂O. The transmission electron microscopy (TEM) image in Fig. [1](#page-4-0)b illustrates a well-organized and uniformly spherical structure measuring approximately 5 nm in diameter, which is consistent with previous fndings [[34](#page-11-15)]. The XRD pattern demonstrates that the HEB-CuO exhibits a crystal phase, which can be assigned to the monoclinic CuO (space group = $C2/c$) with $a \times b \times c = 4.688 \times 3.423 \times 5.132 \text{ Å}^3$, in comparison with that of the precursor of CuCl, providing strong evidence for the formation of a new arrangement (Fig. S2). The lattice fringe distance obtained from the highangle annular dark-feld scanning TEM (HAADF-STEM) image was 0.252 nm, corresponding to the (11-1) planes of monoclinic CuO (Fig. [1](#page-4-0)c), in agreement with the XRD results. The EDS results revealed the presence of Cu, O, and C throughout the nanoparticles (Fig. [1](#page-4-0)d). Because EDS mapping lacks sensitivity in detecting light elements, the presence of HEB molecules on the surface of CuO NPs was further confrmed through electron energy loss spectroscopy (EELS) characterization. Compared to the CuO NPs synthesized without the addition of the HEB (Fig. S3), the HEB-CuO catalyst exhibited a distinct carbon peak at 288 eV, which corresponded to the presence of HEB molecule (Fig. S4). Meanwhile, the presence of the HEB molecule on the surface of CuO NPs was further confrmed through Fourier transform infrared (FTIR) spectroscopy, as shown in Fig. S5. Specifcally, the characteristic peaks observed at 1400–1500, 1500–1600, 893, and 3068 cm−1 are attributed to the asymmetric stretching mode of the C–C bonds, skeletal vibrations of the aromatic ring, and bending and stretching vibrations of the aromatic C–H bonds, respectively [[35,](#page-11-16) [36](#page-11-17)]. The XPS is further employed to study the surface elemental states of the Cu species. As shown in Fig. [1e](#page-4-0), the Cu 2*p*3/2 peak observed at a binding energy of approximately 934.6 eV in the HEB-CuO can be attributed to the presence of Cu^{2+} [[8](#page-10-6)], which is analogous to that detected in CuO. Moreover, X-ray absorption near-edge structure (XANES) and extended X-ray absorption fne structure (EXAFS) were utilized to better evaluate the electronic state of Cu in the CuO and HEB-CuO catalysts. The absorption spectrum of the Cu K-edge was observed (Fig. [1](#page-4-0)f), with the Cu electronic state of HEB-CuO falling between that of the Cu foil and

commercial CuO, while the Cu electronic state of HEB-CuO approached that of CuO. The Fourier transform (FT) EXAFS spectrum in Fig. [1g](#page-4-0) also displays similar characteristic peaks of CuO and HEB-CuO in the R-space, mainly in the form of Cu–O bonds. These results indicate that HEB modifcation had no discernible impact on the electronic structure of the CuO surface.

3.2 Catalytic Performance of CO₂RR in Flow Cell

Considering the experimentally and theoretically verifed high CO₂RR activity showcased by Cu-based materials, our study aimed to investigate the activity and selectivity for the neutral condition of HEB-CuO (optimal concentration of HEB-TMS, Fig. S6) and CuO NPs electrocatalysts, which are painted onto the hydrophobic gas difusion layer carbon paper (Sigracet 29BC) with a thickness of approximately 10 μm (Fig. $S7$) in a flow cell configuration (Fig. S8). The cathode was equipped with a peristaltic pump, while the anode was ftted with a gas–liquid mixed flow pump to ensure consistent electrolyte pH and optimal electrode contact. This configuration facilitates the efficient removal of liquid or gas byproducts while promoting $CO₂$ gas reactions at the triple-phase boundaries within the cell device. Hence, after activation of the electrocatalyst, representative polarization curves for HEB-CuO and CuO were obtained in a neutral 1M KCl electrolyte under diferent atmospheres (Figs. [2a](#page-5-0), S9). The HEB-CuO electrocatalyst exhibited a signifcantly enhanced current–voltage response in the presence of $CO₂$ compared to Ar, with a lower onset potential and higher current density. To improve the performance, a collaborative assessment was conducted on the HEB-CuO and CuO NPs. As shown in the linear scanning voltammeter (LSV) curves (Fig. [2a](#page-5-0)), the HEB-CuO electrocatalyst has lower onset potentials at \sim -0.7 V in neutral electrolyte than that of CuO (~−0.9 V), demonstrating that the surface modifcation of CuO catalyst with HEB organic molecule can signifcantly reduce the overpotential required for $CO₂$ activation. To ensure a more accurate analysis of the FE for all gas and liquid $CO₂RR$ products, it is crucial to employ real-time monitoring of the outlet fow rate using a Defender 520 flow meter. Both the catholyte and anolyte were collected for ¹H nuclear magnetic resonance (NMR) analysis. Gas chromatography (Shimadzu 2014C) and liquid ¹H NMR

Fig. 1 Characterization of the morphology and structure of catalysts. **a** Schematic illustration of the synthesis of HEB-CuO and CuO NPs catalysts. **b** TEM, **c** HAADF-STEM, and **d** EDS elemental mapping of HEB-CuO NPs catalyst. **e–g** Cu 2*p* XPS, K-edge XANES, and K-edge FT-EXAFS spectra of HEB-CuO and CuO NPs catalysts

spectroscopy (Bruker 400 M) enabled us to obtain representative patterns that exclusively detected H_2 , C_1 (CO, CH_4 , and HCOOH), and C_{2+} (C₂H₄, C₂H₅OH, CH₃COOH, and C_3H_7OH) products in the gas and liquid phases (Fig. S10). The collection time for each product at diferent current densities was no less than 1000 s (Fig. S11), to ensure data accuracy. As shown in Figs. [2](#page-5-0)b, S12, and Table S1, the selectivity of HEB-CuO toward C_{2+} exhibited an increasing trend with progressive augmentation in current density, while the selectivity toward H_2 and C_1 demonstrated a decreasing tendency. A high plateau of C_{2+} selectivity>80% was consistently maintained over a wide current density range of 100–600 mA cm^{-2} with a maximum C_{2+} selectivity of 88.62% at 300 mA cm⁻² in a neutral solution, and the counter HER and C_1 were suppressed below 12%. Additionally, HEB-CuO achieves a maximum $j_{C_{2+}}$ up to 480.5 mA cm⁻² at a FE_{C₂₊} of nearly 80%, which is far surpassing the industrial current density requirements for CO₂RR to C₂₊ (> 200 mA cm⁻²). Figure [2c](#page-5-0) presents the single-pass carbon efficiency (SPCE) of the C_{2+} products at different CO_2 flow rates measured at 300 mA cm⁻². At higher flow rates, the SPCE was lower owing to the excessive $CO₂$ input compared to its consumption. By gradually lowering the $CO₂$ flow rate from 100 to 5 sccm, the SPCE

Fig. 2 Electrocatalytic CO₂RR performance in a flow cell. **a** LSV curves of HEB-CuO and CuO NPs in CO₂ and Ar atmospheres. **b** FE and partial current density of C_1 and C_{2+} products at different current densities. **c** CO₂RR-based SPCE_{C₂₊} versus CO₂ flow rate performance of HEB-CuO catalyst in 1M KCl solution. **d** FE of H₂, C₁, and C₂₊, e C₂₊/C₁ selectivity and **f** the half-cell energy efficiency of C₂₊ products for HEB-CuO and CuO NPs. **g** Stability of HEB-CuO at 300 mA cm−2 in 1M KCl solution

achieved a high value of $\sim 1\%$ to $\sim 30\%$ for C₂₊ production under neutral conditions. Through gas-phase exchange experiments $(CO₂/N₂)$, we further validated that the formation of C_{2+} products resulted from CO_2 molecules rather than the decomposition of the ligand (Fig. S13). The FE of the C_{2+} products for the CuO electrocatalyst was evaluated under the same conditions with that of HEB-CuO (Figs. [2d](#page-5-0), S14). The performance of CO₂RR to C_{2+} exhibited by the CuO electrocatalyst is deemed unsatisfactory, particularly with regard to the significant $H₂$ evolution observed at higher current densities. The electrochemical impedance spectra (EIS) of HEB-CuO (Fig. S15) exhibited the fastest interface charge-transfer rate, which was benefcial for enhancing the reaction rate. Normalization of partial current densities for reduced C_{2+} products is conducted using the electrochemical active surface area (ECSA) (Fig. S16). HEB-CuO displayed a greater C_{2+} current density normalized by the ECSA value than CuO, indicating higher intrinsic activity toward C_{2+} products. In Fig. S17, HEB-CuO showed a consistent OH_{ad} peak with CuO but with a much stronger OH[−] characteristic peak, which indicates that the local alkalinity of HEB-CuO increased and contributed to the improvement of C–C coupling [[37,](#page-11-18) [38](#page-11-19)]. Moreover, the encapsulation of HEB by CuO efectively improved the selectivity for C_{2+} products, with a C_{2+}/C_1 ratio of 12.2, which is 2.5 times of the bare CuO NPs at a current density of 400 mA cm^{-2} (Fig. [2](#page-5-0)e). Furthermore, the energy efficiency of the half-cell $(EE_{half-cell})$ was calculated based on the cathode (CO_2RR) and anode (oxygen evolution reaction) processes. A high plateau of the $EE_{half-cell}$ of HEB-CuO was consistently maintained at > 40% over a wide current density range of 100–500 mA cm−2 with a maximum EE_{half-cell} of 43.3% at 300 mA cm⁻², which was much higher than that of CuO without HEB modifcation (Fig. [2f](#page-5-0)). These fndings unequivocally illustrate the pivotal contribution of the HEB molecules in signifcantly lowering the activation energy barrier for $CO₂$ and efficiently inhibiting the generation of related byproducts, particularly with regard to attenuating the HER at high current densities. Notably, the HEB-CuO electrocatalyst demonstrated exceptional long-term durability in a flow cell, maintaining a high current density (300 mA cm^{-2} in a neutral electrolyte) throughout 10 h of continuous operation with only a marginal decrease in FE_{C2+} , while the FE of H_2 and C_1 remained stable (Fig. [2](#page-5-0)g). This discovery holds signifcant importance in the field of electrocatalytic $CO₂RR$.

3.3 Catalytic Performance of CO₂RR in MEA

Considering the voltage drop induced by the internal resistance of the solution, particularly at higher kinetic current densities, resulting in a signifcant decrease in energy efficiency and gas diffusion layer flooding in the flow cell configuration, we further employed an anion MEA gas-phase electrochemical electrolyzer with a 1 cm^2 active area to investigate the performance of the $CO₂RR$. The cathode side was supplied with humidified $CO₂$ gas to prevent direct contact between the catalysts and liquid water, thereby limiting reactant difusion. Additionally, a 0.1M KOH solution was circulated on the anode side. The structure and assembly processes are shown in Fig. S18. With a gradual increase in the cell voltage (Fig. [3a](#page-7-0)), the total current density of the HEB-CuO electrocatalyst exhibited a progressive enhancement, ultimately peaking at 400 mA cm^{-2} without iR compensation, significantly surpassing that of CuO. This observation is consistent with the fow cell confguration. More importantly, HEB-CuO exhibits C_{2+} product selectivity with a trend similar to that of the flow cell, effectively suppressing C_1 and H_2 generation. It maintained a low FE for the H_2 and C_1 products while achieving FE_{C2+} above 80% over a wide range of current densities from 200 to 450 mA cm^{-2} (Figs. [3b](#page-7-0), S19, and Table S2). Additionally, HEB-CuO exhibits a peak j_{C2+} of up to 450 mA cm⁻² when FE_{C2+} reaches 86.14%, which far surpasses that of the CuO electrocatalyst for the $CO₂RR$ to C_{2+} (Fig. S20). Figure [3c](#page-7-0) shows the FE and SPCE of C_{2+} with respect to the CO_2 gas flow rate. The C_{2+} FE could be maintained at a high level, whereas the $CO₂$ flow rate persistently declined. The highest SPCE of 13.4% for the C_{2+} products was achieved at 5 sccm, indicating the enormous potential of HEB-CuO for $CO₂RR$ performance. The C_{2+}/C_1 ratio for HEB-CuO was 19.2, which was 2.7 times that of CuO at a current density of 450 mA cm⁻² (Fig. [3](#page-7-0)d). Moreover, the full-cell energy efficiency $EE_{full-cell}$ of the C_{2+} products calculated at different current densities was higher than that of CuO (Fig. [3e](#page-7-0)), further demonstrating that HEB molecule modifcation can signifcantly lower the activation energy barrier for $CO₂$ and efficiently inhibit the generation of related byproducts. Achieving long-term operational stability is a key requirement for practical applications but still presents challenges. Therefore, the stability of $CO₂$ electrolysis over the HEB-CuO catalyst

Fig. 3 Electrocatalytic CO₂RR performance in catholyte-free MEA electrolyzer. **a** Current densities versus applied voltages between 0 and 4 V for HEB-CuO and CuO NPs in CO₂ atmospheres. **b** FE and partial current density of C_1 and C_2 ₊ products at different current densities and **c** SPCE_{C₂₊} versus CO₂ flow rate performance of HEB-CuO electrocatalyst. **d** C₂₊/C₁ selectivity and **e** the full-cell energy efficiency of C₂₊ products for HEB-CuO and CuO NPs electrocatalysts. **f** CO₂RR stability on HEB-CuO at 200 mA cm⁻². **g** Comparison of the HEB-CuO with the reported state-of-the-art Cu-based catalysts. All comparison data points are from the references summarized in Table S3

was further measured at a fxed applied current density of 200 mA cm−2 using a 0.1M KOH anolyte (Fig. [3](#page-7-0)f). The electrolyzer maintained a stable full-cell potential with $\text{FE}_{\text{C}_2\text{H}_4}$ and $\text{FE}_{\text{(CO+CH}_4+\text{H}_2)}$ exhibiting complementary trends. Continuous operation for over 50 h was achieved at 200 mA cm^{-2} with steady $\text{FE}_{\text{C}_2\text{H}_4}$ levels. These results indicate that HEB-CuO demonstrates excellent activity for the electrocatalytic CO_2RR of C_{2+} products in both neutral flow cell and MEA (Fig. $3g$).

3.4 Discussions

To facilitate a more comprehensive investigation into the in-situ evolution of the derived structures and the underlying mechanism of the electrocatalytic $CO₂RR$, relevant ex-situ and in-situ characterizations were conducted. As shown in Figs. S21 and S22, the HEB-CuO and CuO electrocatalysts after the $CO₂RR$ did not change significantly in morphology, suggesting their structural stability. The EELS mapping and spectra of HEB-CuO after the $CO₂RR$ tests (Fig. S23) also

exhibited a carbon peak at 288 eV with that of the HEB-CuO catalyst, corresponding to the presence of the HEB molecule. Meanwhile, the HRTEM images revealed the presence of the CuO crystal phase, while preliminary color observations following the reaction indicated the formation of metallic Cu, suggesting a reduction in CuO (Fig. S24). The surface of the reacted gas difusion electrode (GDE) was characterized by XPS. The Cu 2*p* spectrum shows that Cu^{0} exists after the reaction (Fig. S25), confirming that $Cu(II)$ was reduced to $Cu⁰$. Further analysis of the Cu LMM Auger spectra (Fig. S26) shows that the Cu^{2+} peak shifted significantly to the Cu^0 peak after the reaction [\[39](#page-12-0)]. Furthermore, the in-situ XANES spectra were obtained in neutral media to validate the phase structure transformation process at various potentials (Figs. [4](#page-9-0)a, S27). Examination of both near-edge and extended-edge structures revealed a progressive decline in the oxidation state of the catalyst from Cu^{2+} to Cu^{0} with increasing potential (Fig. [4](#page-9-0)b). Fourier transform analysis revealed a gradual increase in the strength of Cu–Cu coordination, while the intensity of Cu–O coordination decreased, indicating that Cu^{2+} was completely reduced to metal Cu species, which fully agrees with the color changes. Fourier transform analysis demonstrated a progressive enhancement in the strength of Cu–Cu coordination, accompanied by a decrease in the intensity of Cu–O coordination, suggesting the complete reduction of Cu^{2+} to metallic Cu species (Fig. [4c](#page-9-0)). These fndings are consistent with the observed color alterations. To gain insight into the key intermediates of electrocatalysts for $CO₂RR$ to $C₂₊$ products, we performed in-situ Raman spectroscopy with a flow cell in KCl electrolyte (Fig. S28). Starting at the opencircuit potential (OCP), the Raman spectra of the HEB-CuO and CuO catalysts show several peaks (Figs. [4d](#page-9-0), e, and S29). The peaks at 280, 365, and 2080 cm^{-1} correspond to the stretching mode of Cu–CO and the atop confguration of *CO adsorption, respectively, which are widely employed to assess the *CO coverage on a catalyst [[40,](#page-12-1) [41\]](#page-12-2). As shown in Figs. [4](#page-9-0)d and S29a, the HEB-CuO catalyst appears the obvious Cu–CO restricted rotation and stretching modes at a more positive potential -0.5 V, demonstrating the modifcation of HEB enables the acceleration of reaction kinetics for $CO₂RR$ to $°CO$ intermediate. Moreover, in contrast to the absence of a distinct *CO adsorption peak at atop sites of CuO, a newly emerged atop-adsorbed *CO species is observed on HEB-CuO over a wide potential range from OCP to −0.9 V, providing further evidence for an enhanced

*CO coverage (Figs. [4](#page-9-0)e, S29b). To gain further molecularlevel insight into the *H intermediate of the catalysts, the characteristic peaks of interfacial water at approximately 3400 cm−1 were classifed into three distinct components: 4-HB·H2O, representing water with four coordinated hydrogen bonds at approximately 3240 cm⁻¹; 2-HB·H₂O, representing water with two coordinated hydrogen bonds at approximately 3420 cm⁻¹; and K⋅H₂O, representing water hydrated by K⁺ ions at approximately 3555 cm⁻¹ (Figs. [4](#page-9-0)f, S30) $[32, 42, 43]$ $[32, 42, 43]$ $[32, 42, 43]$ $[32, 42, 43]$ $[32, 42, 43]$. The percentage of the K \cdot H₂O peak generally increased at higher applied negative potentials, demonstrating a decrease in hydrogen bonding and an increase in *H coverage. Upon increasing the applied potential from OCP to -1.1 V, the proportion of K⋅H₂O on the pristine CuO surface signifcantly increased from 24.0% to 31.1%, while only exhibiting a marginal change from 12.8% to 13.6% on HEB-CuO (Fig. [4g](#page-9-0)). This observation implies that pristine CuO exhibits pronounced efficacy for water dissociation and *H adsorption. Considering the divergent trends observed for *CO over HEB-CuO and CuO under increasing cathodic bias, it can be inferred from Fig. [4d](#page-9-0), e that the enhanced CO_2RR to C_{2+} activity exhibited by HEB-CuO compared to that of CuO is likely due to the reduced presence of interfacial water $(K·H₂O)$. Moreover, the adsorption of water molecules on the HEB-CuO and CuO electrocatalysts during the $CO₂RR$ was further investigated via MD simulations conducted in a canonical ensemble at 300 K for a duration of 30 ps (Fig. [4h](#page-9-0)). The Mean Square Displacement (MSD) of water molecules, which serves as an indicator of their mobility, exhibited a declining trend in the presence of the HEB organic molecules coated on the CuO surface (Fig. [4i](#page-9-0)). Furthermore, the number of water molecules for HEB-CuO showed a signifcant decrease over time compared with that of CuO electrocatalyst (Fig. [4j](#page-9-0)). These results provide strong evidence supporting the efficient repulsion of water molecules from the Cu catalyst by the HEB organic molecules, which is consistent with the findings obtained through insitu Raman spectroscopy. Meanwhile, the in-situ attenuated total reflection surface-enhanced infrared spectroscopy (ATR-SEIRAS) (Figs. S31, S32) demonstrates the enhanced clarity of the peaks at 1092 and 1415 cm⁻¹, attributed to *CO*COH and *CO*CO on HEB-CuO compared to pristine CuO, indicating a signifcantly higher *CO coverage on the electrode surface [\[44](#page-12-5)]. This pronounced increase in the C–C coupling probability results in the increased generation of C_{2+} products. Consequently, the HEB-CuO catalyst infers

Fig. 4 Mechanism investigation on the catalytic activity. **a** In-situ XANES spectra at the Cu K-edge. **b** Details of in-situ XANES spectra and **c** in-situ FT-EXAFS spectra in R-space of the HEB-CuO catalyst during CO2RR. **d**, **e** In-situ Raman spectra at diferent ranges of the HEB-CuO catalyst. **f** In-situ Raman spectra of the interfacial water structure on HEB-CuO from OCP to −1.1 V. **g** Quantifcation of the area percentages of K·H2O peaks observed at diferent applied potentials on HEB-CuO and CuO electrocatalysts. **h** Molecular dynamics simulation snapshots depicting the difusion of H2O molecules in the vicinity of a Cu surface, subject to varying levels of stress at 30 ps. **i** Analysis of the mean square displacements of H₂O at the Cu surface of HEB-CuO and CuO. **j** Investigation of the number of surface H₂O observed near the Cu surface at varying strain levels of HEB-CuO and CuO

superior FE_{C2+} values across a wide range of current densities compared to CuO catalysts, which can be ascribed to both limited $*H$ coverage and an elevated CO to H_2O ratio (Figs. [2b](#page-5-0), [3b](#page-7-0)).

4 Conclusions

In conclusion, we developed a surface HEB molecule functionalization method for CuO NPs electrocatalyst, wherein the optimization of local micro-environment modulation, particularly with respect to the surface coverage of *CO and *H, influences the reaction pathways leading to C_{2+} products. This approach enables the simultaneous achievement of a high proportion of *CO and a low concentration of *H, thereby facilitating the efficient conversion of $CO₂₊$ products. The HEB-modifed CuO NPs catalyst exhibits outstanding FE_{C2+} of nearly 90% at an unprecedented current density of 300 mA cm⁻² and maintains high FE ($>80\%$) at the wide current density performance (100 to 600 mA cm^{-2}) in neutral environments using a flow cell and 86.14% FE_{C2+} at a partial current density of 387.6 mA cm⁻², in a MEA electrolyzer. The current study not only provides comprehensive insights into the infuence of *CO and *H coverage on C–C coupling but also lays the foundation for catalyst engineering toward achieving industrial-level $CO₂$ electrolysis in practical MEA devices.

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Declarations

Conflict of interest The authors declare no confict of interest. They have no known competing fnancial interests or personal relationships that could have appeared to infuence the work reported in this paper.

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