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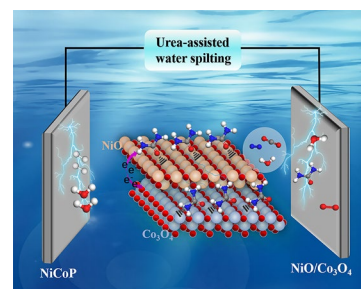
# Scalable Electrocatalytic Urea Wastewater Treatment Coupled with Hydrogen Production by Regulating Adsorption Behavior of Urea Molecule

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## HIGHLIGHTS

- The heterogeneous interface of NiO/Co<sub>3</sub>O<sub>4</sub> was constructed for regulating adsorption behavior of urea functional groups.
- The regulation mechanism of urea molecular adsorption behavior was verified by temperature-programmed desorption experiments and density functional theory calculations.
- Efficient and scalable electrocatalytic urea wastewater treatment coupled with hydrogen production was realized in anion exchange membrane water electrolyzer AEMWE, which can continuously treat urea wastewater at an initial current density of 600 mA cm<sup>-2</sup>, with about 53% in average urea treatment efficiency.

**ABSTRACT** Electrocatalytic urea wastewater treatment technology has emerged as a promising method for environmental remediation. However, the realization of highly efficient and scalable electrocatalytic urea wastewater treatment (SEUWT) is still an enormous challenge. Herein, through regulating the adsorption behavior of urea functional groups, the efficient SEUWT coupled hydrogen production is realized in anion exchange membrane water electrolyzer (AEMWE). Density functional theory calculations indicate that self-driven electron transfer at the heterogeneous interface (NiO/Co<sub>3</sub>O<sub>4</sub>) can induce charge redistribution, resulting in electron-rich NiO and electron-deficient Co<sub>3</sub>O<sub>4</sub>, which are superior to adsorbing C=O (electron-withdrawing group) and -NH<sub>2</sub> (electron-donating group), respectively, regulating the adsorption behavior of urea molecule and accelerating the reaction kinetics of urea oxidation. This viewpoint is further verified by temperature-programmed desorption experiments. The SEUWT coupled hydrogen production in AEMWE assembled with NiO/Co<sub>3</sub>O<sub>4</sub> (anode) and NiCoP (cathode) can continuously treat urea wastewater at an initial current density of 600 mA cm<sup>-2</sup>, with the average urea treatment efficiency about 53%. Compared with overall water splitting, the H<sub>2</sub> production rate (8.33 mmol s<sup>-1</sup>) increases by approximately 3.5 times. This work provides a cost-effective strategy for scalable purifying urea-rich wastewater and energy-saving hydrogen production.



**KEYWORDS** Urea wastewater treatment; Hydrogen production; Adsorption behavior; Heterogeneous interface

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## 1 Introduction

Urea is a vital active nitrogen compound in the nitrogen cycle, playing a crucial role in water, energy and food domains [1, 2]. The large quantities of urea-rich domestic and industrial wastewater without treatment will produce harmful toxins, threatening water ecological balance [3, 4]. Traditional urea decomposition methods, such as enzymatic hydrolysis, biomass degradation and chemical oxidation, often have low economic benefits, complex technology and harsh working environments, making it difficult to meet industrial requirements [5]. It is important to note that electrochemical urea oxidation reaction ( $\text{CO}(\text{NH}_2)_2 + 6\text{OH}^- \rightarrow \text{N}_2 + 5\text{H}_2\text{O} + \text{CO}_2 + 6\text{e}^-$ , UOR) is considered to be a cost-effective method for treating urea-rich wastewater [6]. In particular, UOR has a wide range of applications in energy conversion and storage, including urea-assisted hydrogen production, direct urea fuel cells, photoelectrochemical urea decomposition and wastewater treatment [1, 7]. Ideally, urea-assisted hydrogen production, combined with urea wastewater treatment, can achieve purification of urea-rich wastewater and energy-saving hydrogen production [8, 9]. At present, research on this strategy was limited to laboratory scale, and the realization of highly efficient and scalable electrocatalytic urea wastewater treatment (SEUWT) is still an enormous challenge under industrial current densities in water electrolysis equipment such as anion exchange membrane water electrolyzer (AEMWE) [10, 11].

Urea molecule contains two electron-donating groups ( $-\text{NH}_2$ ) and one electron-withdrawing group ( $\text{C}=\text{O}$ ), which tend to adsorb in the electron-deficient and electron-rich regions of the catalyst, respectively [12, 13]. Due to the influence of functional groups in urea on adsorption behavior, it is of great significance to reveal the regulation mechanism of urea adsorption behavior and search for suitable catalysts for UOR [13]. The investigation of semiconductor physics suggests that by utilizing two semiconductors with different energy structures, it is possible to construct a heterojunction, where an internal electric field and two opposing charge distribution regions can be formed at the heterojunction interface [14–16]. Transition metal-based (such as Ni, Co and Fe metals) sulfides, selenides and nitrides are highly efficient catalysts for UOR [17]. However, the strong polarization and fast reaction rate at high current density can

lead to the dissolution of nonmetallic elements, affecting the stability of the system [18, 19]. Inversely, heterojunctions formed by transition metal oxides (TMOs) are considered potential catalysts for UOR due to their simple synthesis, stable structure and suitability for high currents [20, 21].

Herein, metal organic framework (MOF) derivatives NiO/Co<sub>3</sub>O<sub>4</sub> were constructed using a hydrothermal–calcination method for SEUWT coupled hydrogen production. Density functional theory (DFT) results show that the work functions of NiO and Co<sub>3</sub>O<sub>4</sub> are 8.08 and 6.71 eV, respectively. An internal electric field formed at NiO/Co<sub>3</sub>O<sub>4</sub> interface caused electrons to transfer from Co<sub>3</sub>O<sub>4</sub> to NiO and result in the electron-deficient Co<sub>3</sub>O<sub>4</sub> and electron-rich NiO. The UOR reaction mechanism was revealed through experiment results and temperature-programmed desorption (TPD). The synergistic effect of NiO/Co<sub>3</sub>O<sub>4</sub> heterojunction interface enhances the adsorption capacity of the catalyst for  $-\text{NH}_2$  (electron-donating group) and  $\text{C}=\text{O}$  (electron-withdrawing group), respectively, regulating the overall adsorption behavior of urea and exhibiting excellent reaction kinetics for UOR. In AEMWE, the SEUWT coupled hydrogen production (UOR||HER) was constructed using NiO/Co<sub>3</sub>O<sub>4</sub> and NiCoP as anode and cathode, respectively. Compared to overall water splitting, the UOR||HER system can continuously treat urea wastewater at an initial current density of 600 mA cm<sup>-2</sup>, with about 53% in average urea treatment efficiency and approximately 3.5-fold H<sub>2</sub> yield.

## 2 Experimental Section

### 2.1 Materials

Nickel chloride hexahydrate (NiCl<sub>2</sub>·6H<sub>2</sub>O) and terephthalic acid (C<sub>8</sub>H<sub>6</sub>O<sub>4</sub>, TPA) were purchased from Beijing Innochem Technology Co., Ltd. Cobalt chloride hexahydrate (CoCl<sub>2</sub>·6H<sub>2</sub>O) and N–N dimethylformamide (DMF) were purchased from Shanghai Aladdin Biochemical Technology Co., Ltd. Nickel foam (NF) was purchased from Tianjin Annuo New Energy Technology Co., Ltd. Urea (CH<sub>4</sub>N<sub>2</sub>O) and sodium hypophosphite monohydrate (NaH<sub>2</sub>PO<sub>2</sub>·H<sub>2</sub>O) were bought from Shanghai McLean Biochemical Technology Co., Ltd. Absolute ethyl alcohol (C<sub>2</sub>H<sub>5</sub>OH) was purchased from Tianjing YongSheng Fine Chemical Co., Ltd. Deionized water (resistivity > 18.2 MΩ cm) was used to prepare all solutions, and all the chemicals were used as received without treatment.

## 2.2 Synthesis of NiCo MOF

NiCo MOF precursor was synthesized on NF by a hydrothermal method. Firstly, 0.3 mmol  $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$ , 0.3 mmol  $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$  and 1 mmol TPA were dissolved in 35 mL solution composed of 2.5 mL ethanol, 2.5 mL water and 30 mL DMF. The mixture was stirred in the reactor for 30 min until all the reagents were dissolved. Subsequently, the prepared NF (sonicate in 3 M HCl for 15 min) was transferred into the above solution. After 6 h of hydrothermal treatment at 125 °C and cooling to room temperature, the sample was washed several times with  $\text{C}_2\text{H}_5\text{OH}$  and deionized water, and then dried to obtain the precursor of NiCo MOF. Under the same conditions, Ni MOF and Co MOF precursors were prepared by changing the molar feed ratio of the reagent. Here, Co MOF was prepared as powder.

## 2.3 Synthesis of NiO/Co<sub>3</sub>O<sub>4</sub>

NiCo MOF, Ni MOF and Co MOF precursors were heated at a rate of 5 °C min<sup>-1</sup> in a muffle furnace to 350 °C calcined for 2 h and then dropped to room temperature to obtain NiO/Co<sub>3</sub>O<sub>4</sub>, NiO and Co<sub>3</sub>O<sub>4</sub>.

## 2.4 Synthesis of NiCoP

In a tubular furnace, 0.5 g  $\text{NaH}_2\text{PO}_2 \cdot \text{H}_2\text{O}$  was uniformly dispersed as a P-source at the bottom of the porcelain boat, and the NiCo MOF was placed above another porcelain vessel, locating them upstream and downstream, respectively. In argon atmosphere, heating at a rate of 5 °C min<sup>-1</sup> to 350 °C and maintain at 350 °C for 2 h. After cooling to room temperature, the obtained sample was washed several times with  $\text{C}_2\text{H}_5\text{OH}$  and deionized water, and dried to obtain NiCoP.

## 2.5 Material Characterization

The crystal structure was obtained by X-ray diffraction (XRD, 700SHIMADZU) with Cu K $\alpha$  radiation source. The morphology was characterized by field emission scanning electron microscopy (SEM, JSM-7610F), transmission electron microscopy (TEM, JEOL F200) and energy-dispersive X-ray spectroscopy (EDS). The elemental composition of samples was analyzed using X-ray photoelectron

spectroscopy (XPS, PHI-5000CESCA, Al K $\alpha$  source and  $h\nu = 1253.6$  eV). All the binding energies were calibrated by using C 1s spectrum at 284.8 eV. Normal Raman spectroscopy ( $\lambda = 532$  nm) is obtained on HORIBA (France) without infrared correction. The ultraviolet–visible (UV–Vis) absorbance spectra were measured by spectrophotometer (Beijing Purkinje General T6 new century) for measuring the absorbance of the sample. The UV–Vis diffuse reflectance spectrum was collected on a UV spectrophotometer (Shimadzu, UV-2000) for studying optical properties.

## 2.6 Electrochemical Characterization

All electrochemical tests were performed on Koster workstations (CS301M) using a standard three-electrode system with the prepared catalyst ( $1 \times 1$  cm<sup>2</sup>) as working electrode, graphite rod as counter electrode and Hg/HgO as reference electrode. All high-current data tests are conducted in AEMWE. All experimental data for hydrogen evolution reactions (HER) and oxygen evolution reaction (OER) were tested in 1 M KOH electrolyte solution, while all UOR data were tested in 1 M KOH containing 0.33 M urea. Linear sweep voltammetry (LSV) was tested at a scan rate of 5 mV s<sup>-1</sup>. Electrochemical impedance spectroscopy (EIS) measurements were taken in a frequency range of 100 kHz to 0.1 kHz. The stability of the catalyst was tested by chronopotentiometric ( $i-t$ ), frequency sweep–potentiostat mode, frequency sweep–constant current mode and continuous cyclic voltammetry (CV) method. In the scanning rate range of 10–100 mV s<sup>-1</sup>, electrochemical double-layer capacitance ( $C_{dl}$ ) was evaluated by CV curve, which CV curve was linearly proportional to the electrochemically active surface area (ECSA).

## 2.7 AEMWE Experiment

AEMWE is mainly composed of six parts: anode plate, anode electrode NiO/Co<sub>3</sub>O<sub>4</sub> ( $3 \times 3$  cm<sup>2</sup>), anion exchange membrane, cathode electrode NiCoP ( $3 \times 3$  cm<sup>2</sup>), cathode plate and sealing gasket. NiO/Co<sub>3</sub>O<sub>4</sub> was used for anodic urea oxidation (electrolyte solution with 0.33 M urea of 600 mL), and NiCoP was used for cathodic hydrogen evolution (electrolyte solution with 1 M KOH of 600 mL). The urea oxidation coupled hydrogen evolution properties of NiO/

Co<sub>3</sub>O<sub>4</sub>||NiCoP were tested at room temperature in the voltage range of 1–3 V.

## 2.8 Theoretical Calculations

All calculations were conducted using the Vienna Computational Simulation Package (VASP) based on DFT. The differential charge density calculations are based on the generalized gradient (GGA) of PBE functional and the basis set of flat wave expansion, with a cutoff energy of 450 eV. The Brillouin zone integration was conducted by using a 4 × 4 × 1 k-point mesh for structural relaxation and free energy calculation. The accompanying material includes calculation information.

## 3 Results and Discussion

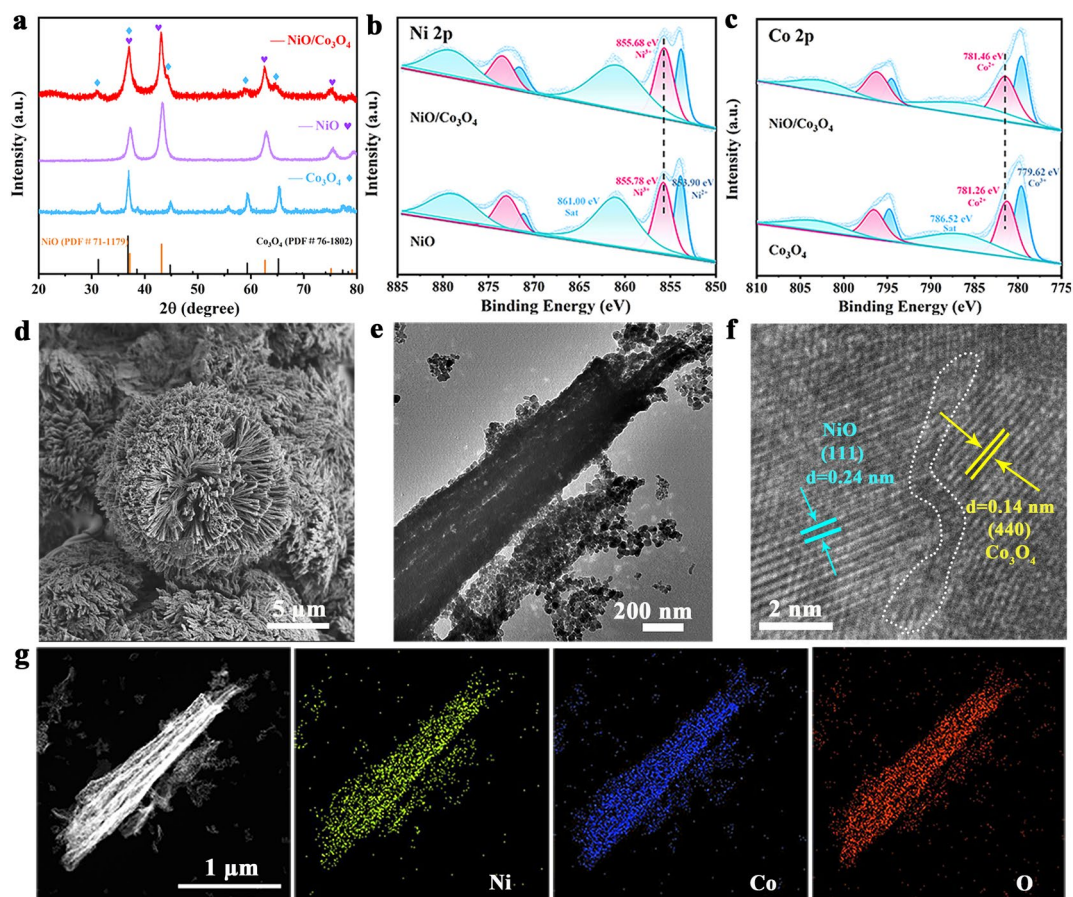
### 3.1 Synthesis and Characterization of NiO/Co<sub>3</sub>O<sub>4</sub>

The catalytic performance of different MOF precursors and their oxide is shown in Fig. S1. Although MOF precursors have a relatively low initial potential, their oxide have better performance at high potentials (1.6–1.8 V<sub>RHE</sub>), so we choose them for the further study. The synthesis schematic diagram of NiO/Co<sub>3</sub>O<sub>4</sub> is shown in Fig. S2. NiO/Co<sub>3</sub>O<sub>4</sub> was prepared by a hydrothermal–calcination method, using NF as a scaffold template. As shown in Fig. 1a, XRD diagram indicates that all diffraction peaks can be clearly indexed to the standard cards of NiO (PDF#71-1179) and Co<sub>3</sub>O<sub>4</sub> (PDF#76-1802). The XPS survey spectra of NiO, Co<sub>3</sub>O<sub>4</sub> and NiO/Co<sub>3</sub>O<sub>4</sub> are shown in Fig. S3. In regard to the high-resolution Ni 2p spectrum of NiO (Fig. 1b), the peaks observed at 853.90 and 871.14 eV are attributed to Ni<sup>2+</sup>. Meanwhile, the peaks observed at 855.78 and 872.99 eV belong to Ni<sup>3+</sup>. In addition, the two accompanying satellite peaks are located at approximately 861.00 and 878.87 eV, respectively [20, 22]. The Co 2p spectrum of Co<sub>3</sub>O<sub>4</sub> shows three double peaks at 779.62/794.78, 781.26/796.51 and 786.52/803.17 eV, belonging to Co<sup>3+</sup>, Co<sup>2+</sup>, and satellite peaks, respectively (Fig. 1c) [20, 22]. The binding energy of Ni 2p in NiO/Co<sub>3</sub>O<sub>4</sub> is negatively shifted by 0.28 eV as opposed to NiO, while it is positively shifted by 0.30 eV for Co 2p in contrast to Co<sub>3</sub>O<sub>4</sub>, which marks a strong electronic coupling with electrons being transferred from Co<sub>3</sub>O<sub>4</sub> to NiO. It can be seen that the NiCo MOF precursor (Fig. S4) and NiO/Co<sub>3</sub>O<sub>4</sub>

(Fig. 1d) display nanowire array structures, while NiO shows to the nanosheet structure (Fig. S5). TEM image further confirms that NiO/Co<sub>3</sub>O<sub>4</sub> has a nanowire structure (Fig. 1e). The HRTEM image of NiO/Co<sub>3</sub>O<sub>4</sub> shows plane spacing of 0.24 and 0.14 nm, belonging to the (111) plane of NiO and the (440) plane of Co<sub>3</sub>O<sub>4</sub>, respectively (Fig. 1f), with a clear interface between them. In addition, EDS mapping images revealed the presence and uniform distribution of Ni, Co and O elements in the entire nanocomposite material (Fig. 1g). The contents of Co and Ni elements in NiO/Co<sub>3</sub>O<sub>4</sub> were determined by ICP-OES (Table S1), which shows the catalyst loading on NF is 1.81 mg cm<sup>-2</sup>.

### 3.2 Electrocatalytic OER and HER Performance

The OER performance of NF, NiO, Co<sub>3</sub>O<sub>4</sub> and NiO/Co<sub>3</sub>O<sub>4</sub> was evaluated in 1 M KOH using a standard three-electrode system. According to LSV curves, NiO/Co<sub>3</sub>O<sub>4</sub> with a Ni and Co content ratio of 1:3 was selected for further investigation owing to its excellent OER properties (Fig. S6). The LSV curve shows that NiO/Co<sub>3</sub>O<sub>4</sub> exhibits the best oxygen evolution performance (Fig. 2a), reaching 500 mA cm<sup>-2</sup> with only a low overpotential of 470 mV, significantly lower than NiO (600 mV), Co<sub>3</sub>O<sub>4</sub> (710 mV) and NF (990 mV) (Fig. 2c). Tafel slope is obtained by fitting the steady-state LSV curves (Fig. S7), and Tafel slope of NiO/Co<sub>3</sub>O<sub>4</sub> (61.80 mA dec<sup>-1</sup>) is much lower than that of NF (134.67 mV dec<sup>-1</sup>), NiO (109.83 mV dec<sup>-1</sup>) and Co<sub>3</sub>O<sub>4</sub> (92.80 mV dec<sup>-1</sup>) (Fig. 2b), implying that NiO/Co<sub>3</sub>O<sub>4</sub> has excellent reaction kinetics [23, 24]. Charge transfer resistance (R<sub>ct</sub>) was fitted through EIS (Fig. 2c and Table S2), and the smallest R<sub>ct</sub> value is NiO/Co<sub>3</sub>O<sub>4</sub> (1.14 Ω), declaring that NiO/Co<sub>3</sub>O<sub>4</sub> has excellent charge transfer ability [25]. The interface behavior of electron distribution during the OER process can be further explored using in situ EIS [4]. Nyquist plots at different potentials were collected in Fig. S8. As the applied potential increases, the arc of NiO/Co<sub>3</sub>O<sub>4</sub> significantly decreases compared with NiO and Co<sub>3</sub>O<sub>4</sub>, meaning that the speed of electron transfer is gradually being accelerated. Figure S9 shows the Bode plot of EIS measurements during the OER process. Compared with NiO and Co<sub>3</sub>O<sub>4</sub>, the phase angle of NiO/Co<sub>3</sub>O<sub>4</sub> in the medium–high-frequency region is less than that in the low-frequency region, which verifies that the electron conduction velocity inside the catalyst is faster



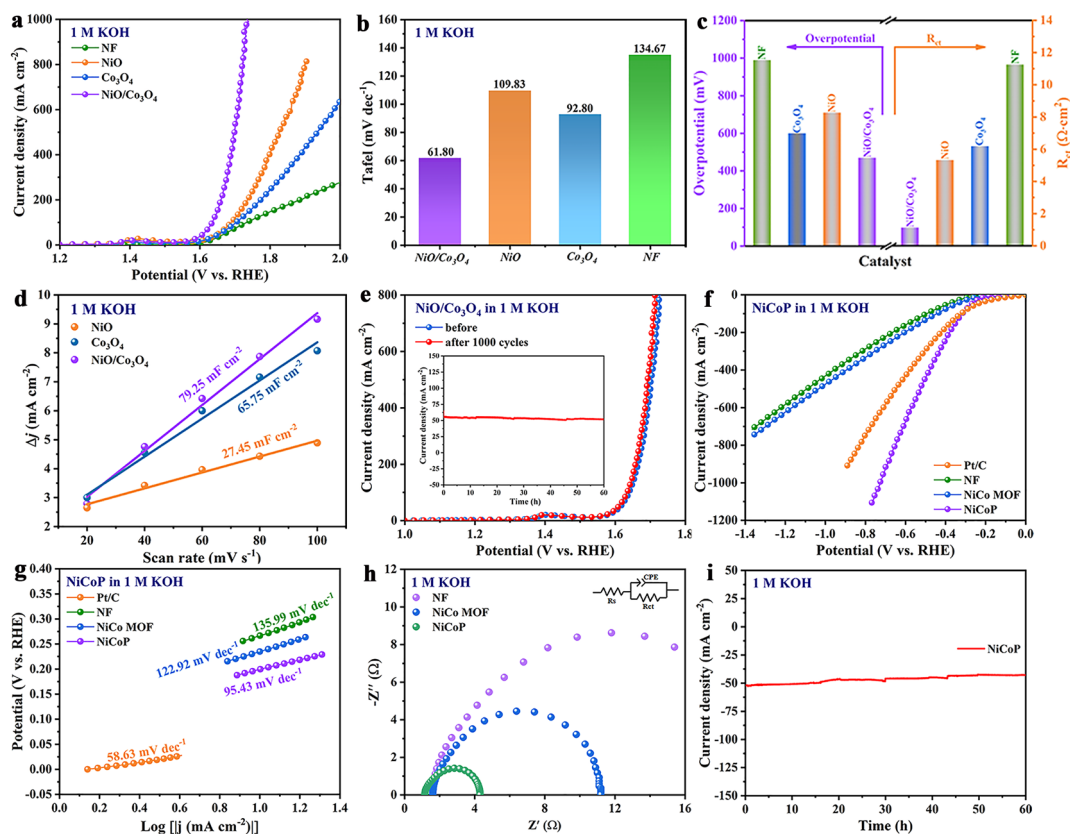
**Fig. 1** **a** XRD patterns of  $\text{Co}_3\text{O}_4$ , NiO and  $\text{NiO}/\text{Co}_3\text{O}_4$ . **b** High-resolution XPS spectrums of Ni 2*p*, **c** Co 2*p*, **d** SEM image, **e** TEM image, **f** HRTEM image and **g** corresponding elemental mapping images of  $\text{NiO}/\text{Co}_3\text{O}_4$

compared with the charge transfer velocity at the catalytic interface [26].

ECSA is estimated based on  $C_{dl}$  obtained from CV curves at different scanning rates [27]. As depicted in Fig. 2d, the  $C_{dl}$  value of  $\text{NiO}/\text{Co}_3\text{O}_4$  ( $79.25 \text{ mF cm}^{-2}$ ) is relatively higher compared with that of NiO ( $27.45 \text{ mF cm}^{-2}$ ) and  $\text{Co}_3\text{O}_4$  ( $65.75 \text{ mF cm}^{-2}$ ), and the results suggest that  $\text{NiO}/\text{Co}_3\text{O}_4$  can expose more catalytic active sites (Fig. S10). The ECSA normalization result shows that  $\text{NiO}/\text{Co}_3\text{O}_4$  maintains the best OER performance (Fig. S11). At different potentials, the TOF value of  $\text{NiO}/\text{Co}_3\text{O}_4$  is higher compared to NiO and  $\text{Co}_3\text{O}_4$  (Fig. S12). Therefore,  $\text{NiO}/\text{Co}_3\text{O}_4$  exhibits excellent intrinsic catalytic activity [28]. In addition to electrochemical activity, the long-term stability of electrocatalysts plays a crucial role in practical applications. Multi-step current and multi-step voltage tests on  $\text{NiO}/\text{Co}_3\text{O}_4$  were performed, with the set bypass voltage and current remaining at 300 s (Fig.

S13). In Fig. 2e, the LSV curves of  $\text{NiO}/\text{Co}_3\text{O}_4$  are almost consistent before and after 1000 cycles. Chronopotentiometric result shows that the  $\text{NiO}/\text{Co}_3\text{O}_4$  runs continuously for 60 h without significant attenuation in the current density of  $50 \text{ mA cm}^{-2}$ . After OER stability testing, the nanowire structures of  $\text{NiO}/\text{Co}_3\text{O}_4$  show slight changes, and the high-resolution XPS peaks of Ni 2*p* corresponding to  $\text{Ni}^{3+}$  in  $\text{NiO}/\text{Co}_3\text{O}_4$  are significantly increased, showing the occurrence of reconstruction and generation of high-valence species during the OER process (Fig. S14).

In electrocatalysis, traditional TMOs materials are not suitable for HER, because the TMOs undergo their own reduction reaction under the working potential of HER to become pure metals, affecting the stability of TMOs [29, 30]. Herein, NiCo MOF derivatives NiCoP was synthesized for HER using a hydrothermal–phosphating method. As shown in Fig. S15a, the crystal structure of the sample



**Fig. 2** **a** LSV curves, **b** Tafel slopes and **c** overpotentials at  $500 \text{ mA cm}^{-2}$  and corresponding  $R_{ct}$  of NF, NiO,  $\text{Co}_3\text{O}_4$  and NiO/ $\text{Co}_3\text{O}_4$  toward OER. **d** Calculated electrochemical  $C_{dl}$  for NiO,  $\text{Co}_3\text{O}_4$  and NiO/ $\text{Co}_3\text{O}_4$ . **e** LSV curves of NiO/ $\text{Co}_3\text{O}_4$  before and after 1000 CV cycles, and the stability of NiO/ $\text{Co}_3\text{O}_4$  for OER in illustration without  $iR$  compensation. **f** LSV curves, **g** Tafel slopes of NF, NiCo MOF, NiCoP and Pt/C toward HER. **h** Nyquist plots of different samples for HER. **i** Stability test of NiCoP for HER without  $iR$  compensation. (Note: All LSV curves in this figure were  $iR$  compensation)

was tested by XRD, and the diffraction peak indexes to NiCoP (PDF#71–2336) [31]. The SEM image shows that NiCoP is in a stacked state, which may be caused by the accumulation of nanowires during phosphating process (Fig. S15b). The HER activity of Pt/C, NF, NiCo MOF and NiCoP in 1 M KOH was evaluated using a standard three-electrode system. The LSV curves of each electrode explain the impressive ultra-low onset potential of NiCoP except for Pt/C (Fig. 2f). Remarkably, NiCoP outperforms Pt/C at current densities greater than  $400 \text{ mA cm}^{-2}$  and exhibits astonishing performance at high current densities. Tafel slope of NiCoP ( $95.43 \text{ mV dec}^{-1}$ ) is much smaller than that of NiCo MOF ( $122.92 \text{ mV dec}^{-1}$ ), NF ( $135.99 \text{ mV dec}^{-1}$ ), except for Pt/C ( $58.63 \text{ mV dec}^{-1}$ ), implying that NiCoP has fast HER reaction kinetics (Fig. 2g). The  $R_{ct}$  values of NiCoP, NiCo MOF and NF are  $0.47$ ,  $0.91$  and  $16.46 \Omega$ , respectively, which express NiCoP possesses the

lowest charge transfer resistance (Fig. 2h and Table S3). In Fig. S16, NiCoP provides with the maximum  $C_{dl}$  ( $3.02 \text{ mF cm}^{-2}$ ), exposing still more active sites. According to results of ECSA normalization (Fig. S17) and TOF (Fig. S18), NiCoP exhibits the best intrinsic catalytic activity. Impressively, even when working continuously for 60 h at a potential of  $-1.2 \text{ V}_{\text{RHE}}$ , NiCoP shows no significant change in current density (Fig. 2i). Similarly, the multi-step current and multi-step voltage curves of NiCoP were tested, which maintained at 300 s under different voltages and currents (Fig. S19). The difference in the LSV curve of NiCoP before and after 1000 cycles can be ignored (Fig. S20). After HER testing, there is no significant change in the surface chemical valence state of NiCoP, but the morphological changes are slightly different, possibly due to partial NiCoP dissolution during the HER process (Fig. S21). The P element in the electrolyte of NiCoP after HER electrolysis test was

collected and compared with the initial sample, it was found that about 0.005% of P element was leached (Table S4), and the small leaching amount did not affect the stability of the catalyst [32].

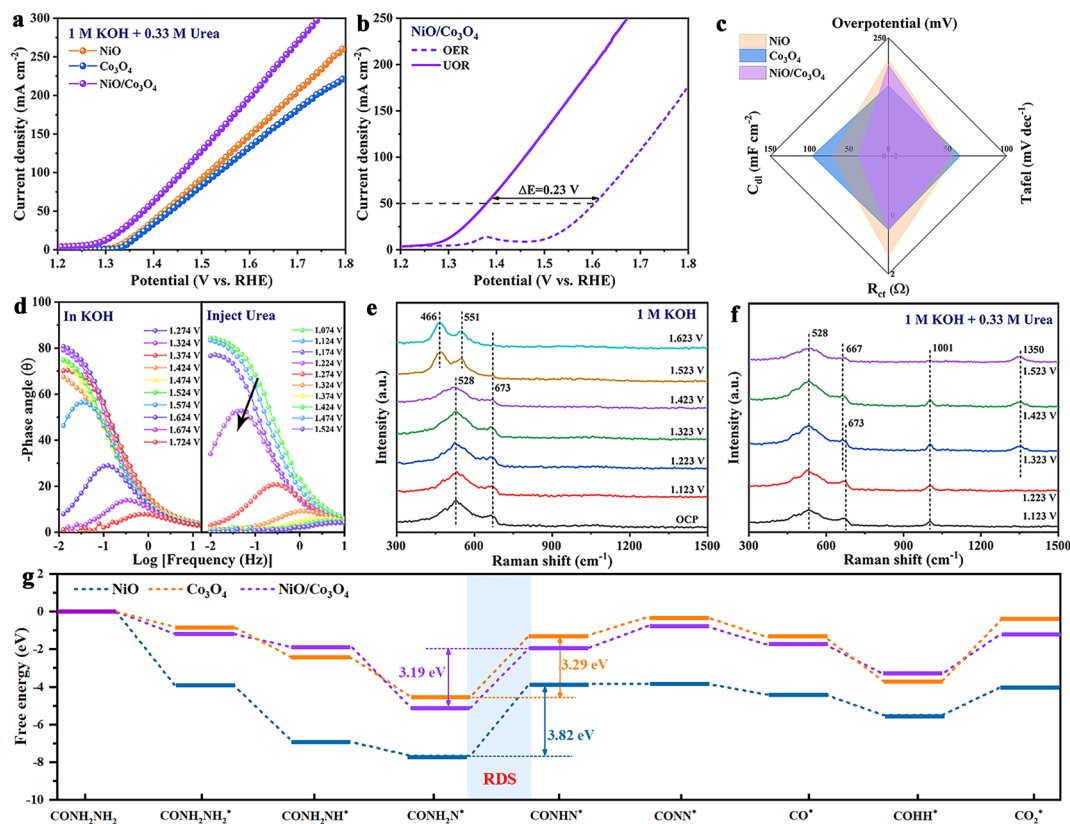
### 3.3 Electrochemical UOR Performance

UOR performances of NiO/Co<sub>3</sub>O<sub>4</sub>, NiO, Co<sub>3</sub>O<sub>4</sub> and metal phosphides (NiP, CoP and NiCoP) were measured in 1 M KOH containing 0.33 M urea using a standard three-electrode system. During the UOR process, metal phosphides have a low initial potential (Fig. S22). Whereafter, their surface may be oxidized to metal oxides/hydroxides, and the unstable nature of the phosphide causes the UOR performance of NiCoP to be lower than NiO/Co<sub>3</sub>O<sub>4</sub> when the potential exceeds about 1.42 V<sub>RHE</sub>. As shown in Fig. 3a, b, NiO/Co<sub>3</sub>O<sub>4</sub> has a lower onset potential in relation to NiO and Co<sub>3</sub>O<sub>4</sub>, superior to most reported oxide catalyst (Table S5), unfolding distinguished UOR performance. Moreover, the corresponding potential of UOR is 1.37 V<sub>RHE</sub> with the current density at 50 mA cm<sup>-2</sup>, which is negatively shifted by 230 mV compared to OER, showing that UOR is thermodynamically superior to OER. The radar plot shows the combined electrochemical properties of NiO/Co<sub>3</sub>O<sub>4</sub>, NiO and Co<sub>3</sub>O<sub>4</sub>, by comparison, NiO/Co<sub>3</sub>O<sub>4</sub> exhibits the best electrocatalytic performance for UOR (Figs. 3c, S23, S24 and Table S6).

Generally, a larger decline in the phase angle indicates a faster charge transfer process. For OER, the phase angle of NiO/Co<sub>3</sub>O<sub>4</sub> in the low-frequency region is stable within the potential range of ~1.474 V<sub>RHE</sub>. When the given potential exceeds 1.524 V<sub>RHE</sub>, the phase angle sharply decreases. In contrast, when the UOR potential of NiO/Co<sub>3</sub>O<sub>4</sub> reached only 1.324 V<sub>RHE</sub>, the phase angle sharply decreased (Fig. 3d). This result demonstrates that NiO/Co<sub>3</sub>O<sub>4</sub> carries a faster electron transfer rate during UOR process. In Fig. S25, NiO/Co<sub>3</sub>O<sub>4</sub> was subjected to 6 long-term experiments at a constant potential. (Each experiment lasted for 6 h, and the electrolyte solution was replaced after testing.) At a constant potential of 1.48 V<sub>RHE</sub>, as urea in the electrolyte is gradually consumed, the current density shows a decreasing trend. After replacing the electrolyte, the current density returned to its initial state. Subsequently, the morphology and structure were well maintained, demonstrating excellent structural stability (Fig. S26). The high-resolution XPS

spectra of the NiO/Co<sub>3</sub>O<sub>4</sub> after UOR are shown in Fig. S27. The Ni 2p and Co 2p deconvoluted into Ni<sup>3+</sup> and Co<sup>3+</sup> showed a significant increase compared to the original sample, and the contents of Ni<sup>3+</sup> and Co<sup>3+</sup> increased from 60.57% and 38.31% to 69.69% and 45.28%, respectively, which indicate that the surface of NiO/Co<sub>3</sub>O<sub>4</sub> was partially oxidized and transformed into high-valence species during the UOR process.

Then, in situ Raman spectra are recorded to gain in-depth insight into the electrocatalytic behaviors during UOR catalytic path. When the applied potential is below 1.423 V<sub>RHE</sub> at 1 M KOH, two distinct Raman peaks appear at 528 and 673 cm<sup>-1</sup>, belonging to NiO and Co<sub>3</sub>O<sub>4</sub> (Fig. 3e), respectively [33]. Notably, when the applied potential exceeded 1.423 V<sub>RHE</sub>, the peaks of NiO disappeared significantly, and Raman peaks dominated at 466 and 551 cm<sup>-1</sup> are attributed to the NiOOH species [34]. However, the Raman peak of Co<sub>3</sub>O<sub>4</sub> did not show any new peaks, indicating that Co<sub>3</sub>O<sub>4</sub> may not be reconstructed during OER process [35]. In the 1 M KOH with 0.33 M urea, the Raman peak occurring at 1001 cm<sup>-1</sup> is attributed to urea (Fig. 3f) [36]. When the applied potential exceeded 1.223 V<sub>RHE</sub>, new Raman peaks appeared at 667 and 1350 cm<sup>-1</sup> belonging to CoOOH and Ni(OH)<sub>2</sub> [4, 35]. NiO is rapidly converted to NiOOH during the UOR process, and in electrolytes containing nucleophiles (urea), the resulting NiOOH intermediates will be filled with hydrogen of the nucleophile through spontaneous reduction, resulting in their conversion back to Ni(OH)<sub>2</sub>, which is difficult to be recognized by Raman spectroscopy [4, 37]. These results indicate that CoOOH and NiOOH are considered the true active sites of NiO/Co<sub>3</sub>O<sub>4</sub> in the UOR process. DFT calculations were conducted to further reveal the potential mechanism of the high UOR activity of NiO/Co<sub>3</sub>O<sub>4</sub> (Fig. S28). The Gibbs free energy diagrams show that the dehydrogenation of the intermediate CONNH<sub>2</sub><sup>\*</sup> to CONNH<sup>\*</sup> is the rate-determining step (RDS) in the overall UOR process (Fig. 3g) [38]. Evidently, the Gibbs free energy change (ΔG) value for RDS of NiO/Co<sub>3</sub>O<sub>4</sub> (3.19 eV) is smaller than that of NiO (3.82 eV) and Co<sub>3</sub>O<sub>4</sub> (3.29 eV), which represents that the NiO/Co<sub>3</sub>O<sub>4</sub> heterojunction reduces the energy barrier of RDS, thereby increasing the UOR activity [39, 40]. These results further demonstrate the thermodynamic feasibility of promoting UOR by constructing heterogeneous interfaces.



**Fig. 3** **a** LSV curves of NiO, Co<sub>3</sub>O<sub>4</sub> and NiO/Co<sub>3</sub>O<sub>4</sub> toward UOR. **b** Comparison of OER and UOR LSV curves for NiO/Co<sub>3</sub>O<sub>4</sub>. **c** Integrated electrochemical performance radar chart. **d** In situ impedance diagram of NiO/Co<sub>3</sub>O<sub>4</sub> for OER and UOR. In situ Raman spectra of NiO/Co<sub>3</sub>O<sub>4</sub> for the **e** OER and **f** UOR processes. **g** Free energy profiles of NiO, Co<sub>3</sub>O<sub>4</sub> and NiO/Co<sub>3</sub>O<sub>4</sub> for UOR (Note: All data in this figure were without *iR* compensation)

### 3.4 Electrocatalytic Urea Oxidation Reaction Mechanism

In electrocatalytic reactions, the surface charge state of catalysts plays a crucial role in the adsorption behavior of reactants [41]. In order to further elucidate the interactions at the heterojunction interface of NiO/Co<sub>3</sub>O<sub>4</sub>, corresponding energy levels were established through Mott–Schottky (M–S) plots, VB–XPS spectrum, UV–Vis diffusive reflectance spectroscopy and work functions. The M–S plots of NiO and Co<sub>3</sub>O<sub>4</sub> exhibit negative slopes, illustrating that they are p-type semiconductors with flat band potentials ( $E_{\text{FB}}$ ) of 1.464 V<sub>RHE</sub> and 1.224 V<sub>RHE</sub>, respectively (Fig. S29) [42]. Subsequently, the VB–XPS spectra of NiO and Co<sub>3</sub>O<sub>4</sub> were analyzed, with the valence bands ( $E_{\text{v}}$ ) of 1.636 and 1.431 eV (Fig. S30), respectively, which were consistent with the M–S measurement results. The band gaps ( $E_{\text{g}}$ ) of

NiO and Co<sub>3</sub>O<sub>4</sub> were obtained by UV–Vis with the values of 3.30 and 1.91 eV, respectively (Fig. S31). DFT calculations indicate that the work functions of NiO and Co<sub>3</sub>O<sub>4</sub> are 8.08 and 6.71 eV, respectively, meaning that electrons can transfer from Co<sub>3</sub>O<sub>4</sub> to NiO until the Fermi level reaches equilibrium (Fig. 4a, b). From the analysis of the differential charge density, it can be seen that electrons flow from Co<sub>3</sub>O<sub>4</sub> to NiO, leading to the depletion electrons on the side Co<sub>3</sub>O<sub>4</sub>, electron aggregation on the NiO side (Fig. 4c). Based on the above results, the energy bands of NiO and Co<sub>3</sub>O<sub>4</sub>, as well as the built-in electric field between them, can be constructed (Fig. 4d, e). When a direct contact is established between NiO and Co<sub>3</sub>O<sub>4</sub>, self-driven electron transfer induces charge redistribution at the interface, resulting in Co<sub>3</sub>O<sub>4</sub> and NiO becoming electron-deficient and electron-rich regions, respectively.

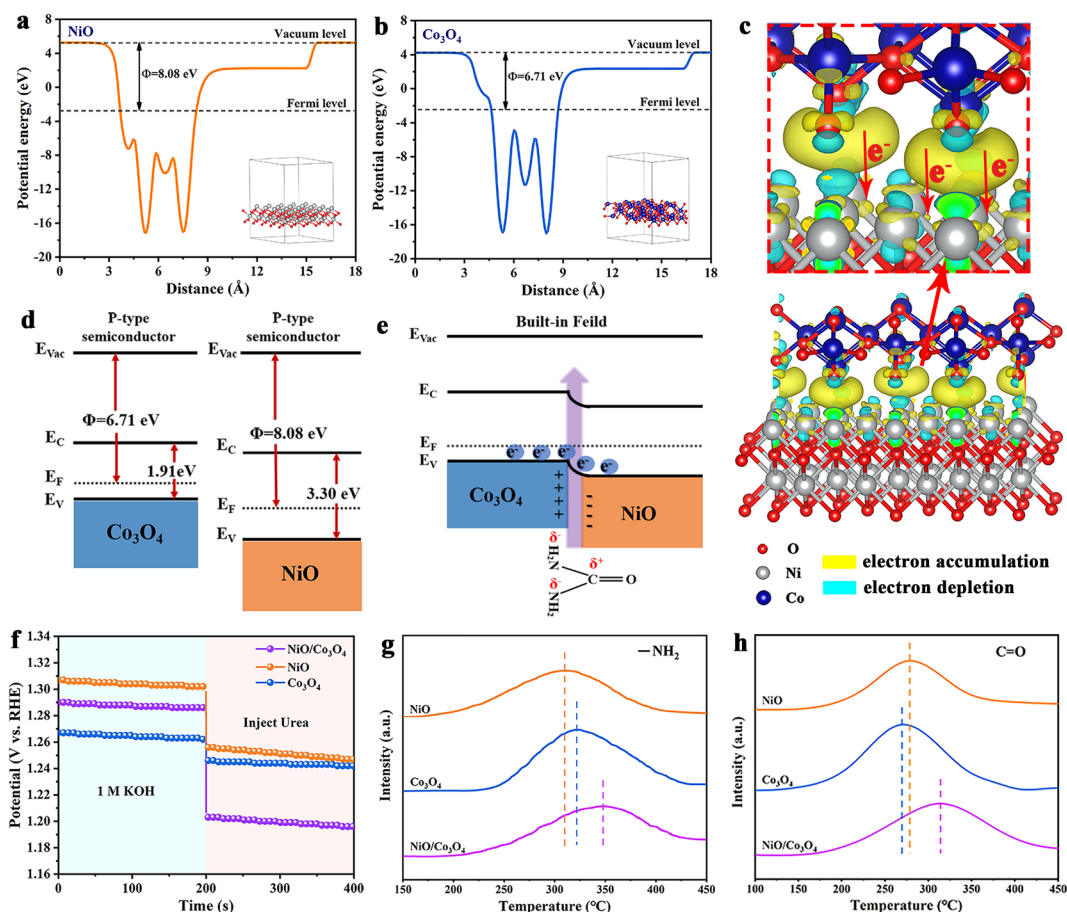


To verify the effect of heterojunctions on the adsorption behavior of urea, NiO,  $\text{Co}_3\text{O}_4$  and  $\text{NiO}/\text{Co}_3\text{O}_4$  were subjected to open circuit potential (OCP) and TPD tests [43, 44]. NiO,  $\text{Co}_3\text{O}_4$  and  $\text{NiO}/\text{Co}_3\text{O}_4$  were placed in a solution of 1 M KOH to test their OCP; after 200 s of stabilization, urea (0.33 M) was injected into the solution and stirred evenly (Fig. 4f). The adsorption of organic molecules on the surface of the electrode would lead to the exchange of  $\text{OH}^-$  in the Helmholtz layer of the electrode, resulting in a decrease in the OCP. The larger OCP reduction value of  $\text{NiO}/\text{Co}_3\text{O}_4$  indicates that it exhibits stronger adsorption capacity than NiO and  $\text{Co}_3\text{O}_4$  toward urea molecules. Subsequently, the adsorption behavior of NiO,  $\text{Co}_3\text{O}_4$  and  $\text{NiO}/\text{Co}_3\text{O}_4$  on urea groups was further studied through TPD testing. In the butylamine/He atmosphere, the desorption temperature for  $\text{Co}_3\text{O}_4$  is higher

than that for NiO (Fig. 4g), demonstrating the stronger adsorption of  $\text{Co}_3\text{O}_4$  for the  $-\text{NH}_2$  group. The desorption temperature of NiO for CO exceeds that of  $\text{Co}_3\text{O}_4$ , indicating that NiO has a stronger adsorption capacity for C=O groups (Fig. 4h). Surprisingly, the desorption temperature of  $\text{NiO}/\text{Co}_3\text{O}_4$  for  $-\text{NH}_2$  and C=O is much higher than that of NiO and  $\text{Co}_3\text{O}_4$ ; relatively speaking, the synergistic effect of  $\text{NiO}/\text{Co}_3\text{O}_4$  heterojunction interface enhances the adsorption capacity of the catalyst for  $-\text{NH}_2$  and C=O.

### 3.5 AEMWE Performance

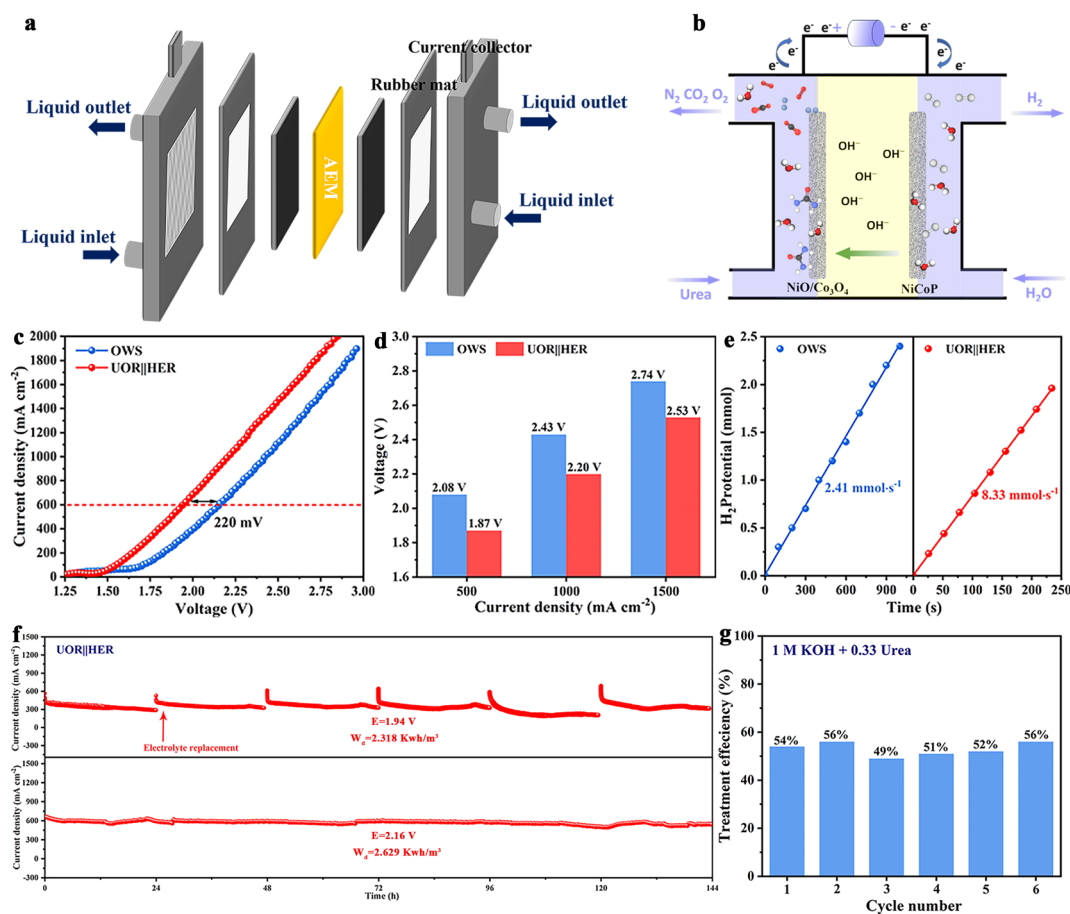
Considering the excellent electrocatalytic performance of  $\text{NiO}/\text{Co}_3\text{O}_4$  in UOR and NiCoP in HER, assembling into AEMWE using  $\text{NiO}/\text{Co}_3\text{O}_4$  as anode and NiCoP as cathode



**Fig. 4** Electrostatic potential of **a** NiO, **b**  $\text{Co}_3\text{O}_4$ , **c** Charge density difference in heterostructure of NiO and  $\text{Co}_3\text{O}_4$ . Schematic diagrams of the band structure of NiO and  $\text{Co}_3\text{O}_4$  **d** before and **e** after contact. **f** OCP test. TPD adsorption spectra of NiO,  $\text{Co}_3\text{O}_4$  and  $\text{NiO}/\text{Co}_3\text{O}_4$  in **g** butylamine/He and **h** CO atmospheres

(NiO/Co<sub>3</sub>O<sub>4</sub>||NiCoP), its SEUWT coupled hydrogen production performance was studied (Fig. 5a, b) [45]. LSV curves of NiO/Co<sub>3</sub>O<sub>4</sub>||NiCoP during two-electrode electrolysis were compared in different solutions. At a current density of 600 mA cm<sup>-2</sup>, UOR||HER possesses a voltage of 220 mV lower than that of overall water splitting (OWS), indicating that UOR||HER improves the electrolysis efficiency (Fig. 5c). Compared with OWS, UOR||HER saves 210, 230 and 210 mV voltage at 500, 1000 and 1500 mA cm<sup>-2</sup>, reducing overall energy consumption (Fig. 5d). EIS result displays that the reaction kinetic of UOR||HER system is faster, which is more conducive to charge transfer (Fig. S32). Notably, the hydrogen generation rate of UOR||HER reaches 8.33 mmol s<sup>-1</sup>, which is nearly 3.5 times higher than OWS (2.44 mmol s<sup>-1</sup>), implying its distinguished H<sub>2</sub> generation ability (Fig. 5e). Figure 5f shows that OWS can

operate stably for 144 h at 2.16 V with negligible activity decay. In addition, the long-term stability test of UOR||HER was conducted using the chronoamperometric method at a voltage of 1.94 V and 6 consecutive long-term experiments were conducted in the electrolyte. (Each experiment lasted for 24 h, and the electrolyte was replaced after each test.) During this process, each standard cubic meter of H<sub>2</sub> generates approximately ~2.318 kW h of electricity, which is lower than OWS (~2.629 kW h) and effectively saves electricity consumption. Due to the continuous consumption of urea during testing, the current density decreased and eventually remained stable, which is due to the severe OER competition caused by the decrease in urea concentration [46]. However, when replacing the electrolyte with fresh urea solution, the required current density will return to its initial state. The improved chemical technology of



**Fig. 5** **a** Electrolyzer cell structure model. **b** Two-electrode schematic diagram. **c** LSV curves of NiO/Co<sub>3</sub>O<sub>4</sub>||NiCoP in OWS and UOR||HER. **d** Compare the voltage at different current densities between OWS and UOR||HER. **e** Comparison of H<sub>2</sub> production rate in OWS and UOR||HER. **f** Long-term stability test for OWS and UOR||HER. **g** Treatment efficiency of urea in UOR||HER system. (Note: All data in this figure were without *iR* compensation)

diacetylacetoxime was used to analyze the concentration of urea-rich wastewater after UOR (Fig. S33). Impressively, the average degradation efficiency of urea in 6 cycles reached 53% (Fig. 5g). Compared with other HER||UOR systems, NiO/Co<sub>3</sub>O<sub>4</sub>||NiCoP electrocatalysts are superior to recently reported works (Table S5). The morphology and structure of NiO/Co<sub>3</sub>O<sub>4</sub> and NiCoP after UOR show no significant changes (Figs. S34 and S35). The systematic research has verified the impressive stability of NiO/Co<sub>3</sub>O<sub>4</sub> and NiCoP in the practical application of SEUWT coupled hydrogen production.

## 4 Conclusions

In summary, the NiO/Co<sub>3</sub>O<sub>4</sub> was synthesized using a hydrothermal–calcination method for SEUWT coupled hydrogen production. DFT calculations indicate that self-driven electron transfer at the NiO/Co<sub>3</sub>O<sub>4</sub> interface can induce charge redistribution, resulting in NiO and Co<sub>3</sub>O<sub>4</sub> becoming electron-rich and electron-deficient regions, respectively. Based on TPD and experimental results, it was verified that NiO (electron-rich) and Co<sub>3</sub>O<sub>4</sub> (electron-deficient) are superior to adsorbed C=O (electron-withdrawing group) and –NH<sub>2</sub> (electron-donating group), respectively, and the interaction between heterogeneous interfaces regulates the adsorption behavior of the catalyst for urea molecules, accelerating the reaction kinetics of UOR. In AEMWE, the UOR||HER system only requires 1.94 V to reach 600 mA cm<sup>−2</sup>, and H<sub>2</sub> production increases by about 3.5 times compared to OWS. Meanwhile, the UOR||HER system and OWS system exhibited long-term stability for 144 h at 1.94 and 2.16 V, respectively, and the average urea treatment efficiency in UOR||HER reached 53%. This work is expected to provide broad application prospects in scalable purifying urea-rich wastewater and energy-saving hydrogen production.

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**Author Contributions** Chunming Yang was involved in conceptualization, funding acquisition, review, supervision,

writing—editing and project administration. Huijuan Pang contributed to methodology, data curation, formal analysis and writing—original draft. Xiang Li was responsible for methodology and data curation. Xueyan Zheng, Tingting Wei, Xu Ma and Qi Wang conducted investigation and validation. Chuantao Wang and Danjun Wang took part in conceptualization, review and supervision. Bin Xu participated in supervision, validation and writing—editing. All authors participated in the discussion of the results and commented on the manuscript.

## Declarations

**Conflict of Interest** The authors declare no interest conflict. They have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper. Bin Xu is an editorial board member for Nano-Micro Letters and was not involved in the editorial review or the decision to publish this article. All authors declare that there are no competing interests.

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