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Breaking Boundaries: Advancing Trisulfur Radical-Mediated Catalysis for High-Performance Lithium–Sulfur Batteries

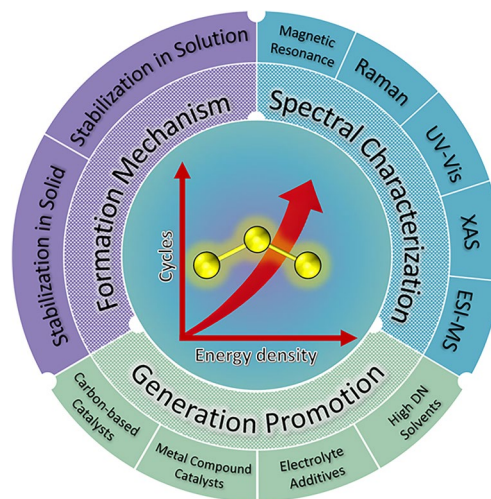
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HIGHLIGHTS

- The review emphasizes the formation of trisulfur radicals in solid-state lapis lazuli analogs and the role of high donor number solvents and/or their co-solvents in stabilizing trisulfur radicals.
- The detection techniques are also discussed for monitoring the generation of trisulfur radicals, which are critical for understanding their behavior and optimizing the design of lithium–sulfur batteries.
- The strategies involving both homogeneous and heterogeneous catalysts are summarized to increase the generation of trisulfur radicals and enhance catalytic reactions in lithium–sulfur batteries for practical applications. The strategies involving both homogeneous and heterogeneous catalysts are summarized to increase the generation of trisulfur radicals and enhance catalytic reactions in lithium–sulfur batteries for practical applications.

ABSTRACT Lithium–sulfur batteries (LSBs) have attracted significant attention due to their high theoretical energy density and low-cost raw materials. However, LSBs still face various challenges in practical applications, particularly the shuttle effect, electrode passivation, and slow kinetics. In recent years, trisulfur radicals (TRs), important intermediates in LSBs, have emerged as a promising and beyond-traditional solution to these problems, which serves as a mediated catalyst to improve the electrochemical performance of LSBs. As a system that is inconsistent with the catalytic conversion process discussed in the traditional LSBs, this review focuses on the generation, detection, promotion, and catalytic roles of TRs, especially emphasizing the formation of TRs in solid-state lapis lazuli analogs and discussing the pros and cons of high donor number solvents and/or their co-solvents in stabilizing TRs. Strategies involving homogeneous/heterogeneous catalysts are discussed for increment of TRs and enhancing catalytic reactions in LSBs. Ultimately, given TRs' significant potential as a key factor in enhancing the performance of LSBs, future perspectives and outlooks are provided to guide the further development of TRs in LSBs. This review provides valuable insights into the design of electrolytes and catalysts for increment of TRs, paving the new practical direction and way for advanced LSBs.

KEYWORDS Lithium–sulfur batteries; Trisulfur radicals; Mediated catalyst; High donor number; Homogeneous/heterogeneous catalyst



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1 Introduction

Typical lithium–sulfur batteries (LSBs), consisted with sulfur cathode and metallic lithium (Li) anode, are one of the most promising energy storage devices [1–3]. As a classical cathodic active material, elemental sulfur (*i.e.*, cyclo-S₈) is natural abundant, cost-effective, and non-toxic. Meanwhile, it possesses a high theoretical specific capacity of 1675 mAh g⁻¹ [4]. Metallic Li is the lightest metal (6.941 g mol⁻¹), which possesses the highest specific capacity (3861 mAh g⁻¹) among the metal electrodes. Moreover, Li has a conspicuous negative electrode potential (−3.04 V vs. standard hydrogen electrode) [5–7]. Coupling cyclo-S₈ and Li, the theoretical energy density of LSBs can reach 2600 Wh kg⁻¹ or 2800 Wh L⁻¹, which is much higher than the state-of-the-art Li-ion batteries [8, 9].

The operation of LSBs relies on the sulfur reduction reaction (SRR) occurring on the surface of sulfur. This process involves the dissolution of intermediate lithium polysulfides (LiPSs, chemical formula: Li₂S_x, x equals to 3–8) and the deposition of Li₂S₂/Li₂S, collectively referred to as the dissolution–deposition process. During this process, the diffusion of soluble LiPSs (such as Li₂S₈, Li₂S₆, Li₂S₄) leads to the "shuttle effect", while the deposition of insulating Li₂S₂/Li₂S can easily passivate the electrode. These issues result in performance degradation and reduced lifespan of LSBs, posing significant challenges to their development and application.

To overcome the problems mentioned above, many efforts have been utilized, such as cathode design [10–12], separator functionalization [13, 14], electrolyte solution modification (additives and redox mediators) [15–17], lithium engineering technologies (surface stabilization and alloying) [18, 19], and electrocatalysts [20] that have been widely used in sulfur cathode. Rational design and use of electrocatalysts can accelerate the SRR process, reducing LiPSs accumulation and thereby mitigating the shuttle effect [21, 22]. Recently, multifunctional catalytic systems, such as triple-phase interface catalysis [23], selective catalysis [24], and hierarchical adsorption catalysis [25], have been proposed. These systems enable electrocatalysts to exhibit excellent adsorption and catalytic performance for LiPSs and simultaneously regulate the deposition of Li₂S₂/Li₂S. Therefore, developing novel and multifunctional electrocatalysts represents a key approach to address the challenges faced by LSBs.

The choice of suitable electrolytes for LSBs is also caught in a dilemma, because the electrolytes not only serve as ion conductor for mass transport but also participate in the conversion reactions of LiPSs. The traditional electrolyte for LSBs is ether-based solution, *i.e.*, 1 M lithium bis(trifluoromethanesulfonyl)imide (LiTFSI) and 0.2 M LiNO₃ in equivolume 1,3-dioxolane (DOL) and 1,2-dimethoxyethane (DME) [26, 27], noted as DOL/DME in this review. Soluble LiPSs are moderately dissolved to constitute a part of the catholyte. It is notable that minimizing amount of the electrolyte solution is essential for pursuing high energy density of LSBs [28, 29]. The ideal goal of practical LSBs in energy density is to achieve 500 Wh kg⁻¹ or 700 Wh L⁻¹, *viz.* a low electrolyte solution/sulfur (E/S) ratio of *ca.* 1 μL mg⁻¹ is needed [30]. However, the E/S ratio of DOL/DME is limited to *ca.* 4.7 μL mg⁻¹, which is far exceed the standard [31].

To reduce the amount of electrolyte or enhance the solubility of polysulfides, one effective approach is to use highly soluble electrolytes (HSEs) [32, 33]. For example, dimethyl sulfoxide (DMSO) that is the first investigated HSE for Li–S flow batteries has higher solubility for both long- and short-order polysulfides than DOL/DME [34]. The saturation concentration of Li₂S₈ exceeds 14 M [S] in pure DMSO, where it is higher than the required threshold of 10.4 M [S]. HSEs typically exhibit a higher donor number (DN). This value is defined as the negative enthalpy change associated with the formation of a 1:1 adduct between a Lewis base and the standard Lewis acid antimony pentachloride (SbCl₅), in dilute solution in the non-coordinating solvent 1,2-dichloroethane with a zero DN [35, 36].

In essence, DN serves as a measure of a solvent's capacity to solvate cations and Lewis acids. For example, the DN of highly soluble DMSO is 29.8 kcal mol⁻¹, whereas the DN of lowly soluble DME is 7.2 kcal mol⁻¹ [35]. Thus, DN can be regarded as a descriptor for dissolving ability of LiPSs [37]. Long-order LiPSs, such as Li₂S₆ in a high-DN solvent, exhibits a blue coloration, which is attributed to the formation of trisulfur radicals or thiozonide anion (S₃⁻) [38]. This anion is analogous to the ozonide anion (O₃⁻), to which they are valence-isoelectronic [39]. Generally, the widely accepted formation mechanism of S₃⁻ radicals in high-DN solvents involves its derivation from the homolytic cleavage of the middle S–S bond in S₆²⁻ [40, 41].

The characterized absorption band (*ca.* 617 nm) of S₃⁻ radicals in high-DN solvents, such as DMSO,

N,N-dimethylformamide (DMF), dimethylacetamide (DMA), or their mixture with low-DN solvent, can be easily distinguished in ultraviolet–visible (UV–Vis) spectrum [42, 43]. In other cases, $S_3^{\cdot-}$ radicals can also be detected in DOL/DME mixtures with high-DN anions [43], or in low-DN tetraethylene glycol dimethyl ether (TEGDME or G4) solvents with more oxygen atoms available for coordination (4 per molecule) [44].

$S_3^{\cdot-}$ radicals serve as a key intermediate to enhance active sulfur interconversion reactions. First of all, the short-order Li_2S_2 or Li_2S can react with the remanent cyclo- S_8 or S_8^{2-} to form highly reactive $S_3^{\cdot-}$ radicals during discharge, driving the full sulfur utilization of LSBs. What's more, sulfur chemistry mediated by $S_3^{\cdot-}$ radicals enables three-dimensional deposition of Li_2S , which can weaken the surface passivation process, accelerating the electrode reaction kinetics [35, 42, 45]. In addition, $S_3^{\cdot-}$ radicals additionally reduces the overpotential associated with Li_2S oxidation during the charging process by providing extra oxidation reaction pathways [45]. However, high-DN solvents easily corrode the metallic Li anode, resulting in poor cycle performance and shortening cycle life of LSBs, which partly limits their further application [32].

There is another form of trisulfur radicals, namely lithium trisulfur radicals (LiS_3^{\cdot}), existing with similar effect of $S_3^{\cdot-}$ radicals, which is derived from the homolytic cleavage of Li_2S_6 in the traditional and low-DN ether-based electrolyte [46]. LiS_3^{\cdot} radicals can be regarded as an endogenous multifunctional electrocatalyst in traditional ether-based electrolyte system. Under the mediation/catalysis of LiS_3^{\cdot} radicals, chemical reactions and electrochemical reactions in LSBs drive each other forward [40]. However, due to thermodynamic reasons, LiS_3^{\cdot} radicals in traditional ether-based electrolytes are easily associated to form Li_2S_6 or undergo disproportionation reactions to form other sulfur species, leading to a small content. This is also the reason why LiS_3^{\cdot} radicals difficultly be detected or distinguished by *ex situ* and steady-state testing techniques. For this reason, radicals trapping or spin trapping agents are used as a compromise method to stabilize $S_3^{\cdot-}/LiS_3^{\cdot}$ radicals by forming radical adducts [47]. Otherwise, the transient generation of LiS_3^{\cdot} radicals during the operation of LSBs with traditional low-DN ether-based electrolytes can be effectively detected, owing to the advanced *in situ* techniques (such as Raman spectroscopy [47], electron spin resonance

(ESR) [48], X-ray absorption spectroscopy (XAS) [45], and UV–Vis [49]).

Due to the low content compared to total and the evanescent properties, LiS_3^{\cdot} radicals are usually considered insufficient to affect the entire sulfur chemical reaction [50]. Nevertheless, a majority of as-reported catalysts, such as metal compounds, do not involve the mediation/catalysis of sulfur radicals, but focus more on their adsorption/catalysis [51, 52]. In recent years, some heterogeneous [53, 54] or homogeneous [55, 56] electrocatalysts were found to have the ability to induce the homolytic reaction of Li_2S_6 to generate more LiS_3^{\cdot} radicals in the traditional ether-based electrolyte. At the same time, born with the rapid development of *in situ* technology and the improvement of the detection limit, LiS_3^{\cdot} radicals were found to change significantly during the catalytic process. It can be speculated that sulfur radicals are likely to be one of the important factors for the catalytic effect of electrocatalysts in LSBs. Therefore, in-depth exploration of the incremental and efficiency improvement strategies and paths of $S_3^{\cdot-}/LiS_3^{\cdot}$ radicals in LSBs are of great significance to the essential improvement of the performance of electrocatalysts in LSBs.

This review systematically summarizes the latest progress on $S_3^{\cdot-}/LiS_3^{\cdot}$ radicals, focusing on their formation mechanisms in solid matter and solutions, spectral characterization techniques. The role of high-DN solvents and their co-solvent strategies, electrolyte additive strategies, and metal compound and carbon-based catalysts are proposed in promoting $S_3^{\cdot-}/LiS_3^{\cdot}$ radicals generation. By comprehensively addressing the formation mechanisms, physicochemical properties, and catalytic behaviors of $S_3^{\cdot-}/LiS_3^{\cdot}$ radicals, this review aims to provide new insights for the design of more efficient electrolytes and electrocatalysts. This effort not only deepens the understanding of the core reaction mechanisms, but also offers theoretical support for developing innovative strategies to improve energy density, rate capability, cycling stability, and overall performance of LSBs.

2 Formation Mechanism of $S_3^{\cdot-}/LiS_3^{\cdot}$ Radicals

2.1 Stabilization Mechanism in Solid Matter

Before the advent of modern chemical pigments, natural mineral Lazurite (Fig. 1a) [57], named from the Persian *lajvard* for blue [58], was always used to prepare an expensive

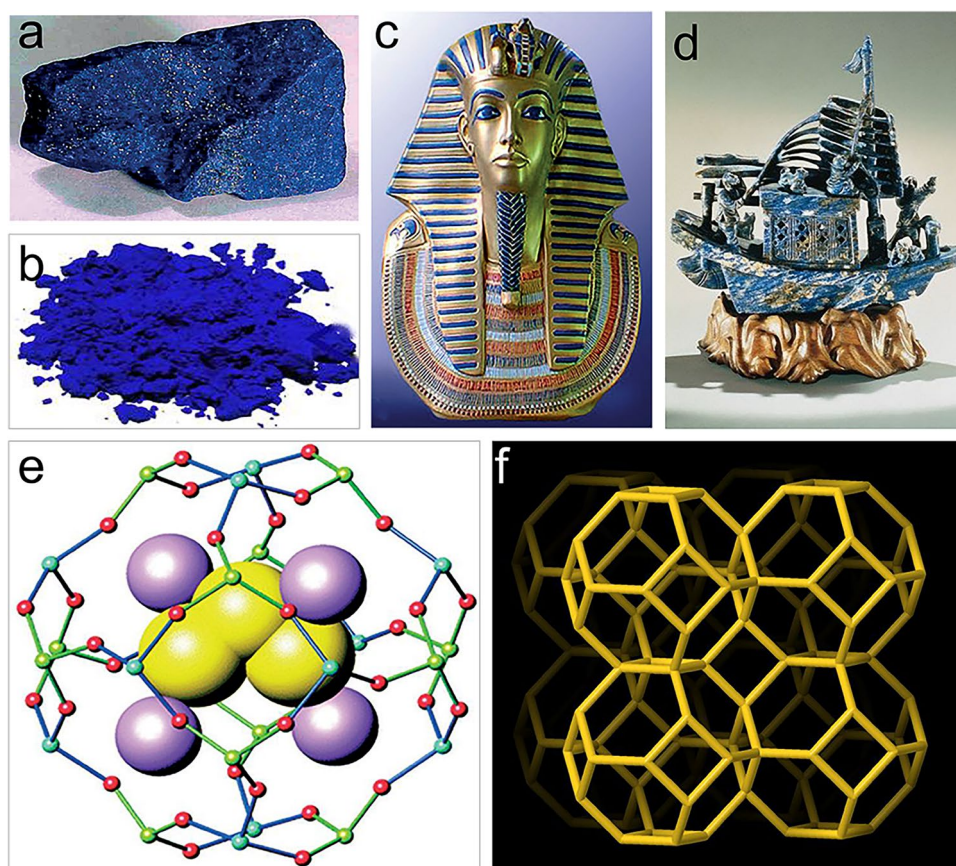


Fig. 1 $S_3^{\bullet-}/LiS_3^{\bullet+}$ radicals stabilized in solid matter. **a** Natural Lazurite mineral [57]. Copyright 2009, Elsevier. **b** The ultramarine pigments made from Lazurite [59]. Copyright 2018, American Chemical Society. **c** Gold mask of Pharaoh Tutankhamun with blue stripes [57]. Copyright 2009, Elsevier. **d** A craft of Chinese junk with a blue surface [61]. Copyright 1999, Royal Society of Chemistry. **e** Model of β -cage with assembled $S_3^{\bullet-}$ radical [57]. Copyright 2009, Elsevier. **f** SOD-type zeolite composed of β -cage [62]

blue pigment (Fig. 1b) [59] and exuded huge attraction to humans. Lazurite was concluded in a well-documented work that it originated from the mines in Badakhshan Province, Afghanistan [60]. It has been mined as a gem or pigment for more than 9000 years and was used as a pigment in painting and crafts since at least the sixth to seventh century. The blue stripe of famous gold mask of Pharaoh Tutankhamun in ancient Egypt (Fig. 1c) [57] and a craft of Chinese junk (Fig. 1d) [61] all use blue pigment coming from Lazurite mineral, showing high artistic value and attractiveness.

To date, we have known that natural blue pigment is a kind of lapis lazuli analogue [40, 63, 64]. Modern technology has confirmed that the blue color of natural pigments is caused by the replacement of some Na^+ and Cl^- on the sodalite (SOD) cage or β -cage of the natural aluminosilicate sodalite ($Na_8[Al_6Si_6O_{24}]Cl_2$) by some $S_3^{\bullet-}$ radicals to become $Na_7[Al_6Si_6O_{24}]S_3$ (Fig. 1e) [57]. The $S_3^{\bullet-}$ radicals

is located in the β -cage of lapis lazuli and is coordinated by seven Na^+ in the center of the surrounding six-membered ring. It is expensive to extract lapis lazuli blue pigment from natural lapis lazuli minerals, but modern mature synthetic processes using zeolite as the main raw material can easily obtain lapis lazuli analogs. Zeolites are composed of $[SiO_4]$ and $[AlO_4]^-$ basic units, connected by Si–O–Al topology, and have hollow cages such as α -, β -, or γ -cage [65]. However, zeolites that can be used to synthesize lapis lazuli analogs are mainly sodalite-like type (Fig. 1f) [62], such as zeolite A (LTA), zeolite X/Y (FAU), zeolite EMT, and zeolite LTN, which are formed by connecting β -cages in different ways. Kowalak et al. mixed zeolite A with alkali metals and synthesized lapis lazuli analogs embedded with $S_3^{\bullet-}$ radicals at high temperature. Their results confirmed that calcination temperature and the type of mixed alkali metal cations (such as Li^+ , Na^+ ,

K⁺) affect the product type and yield [66]. Rejmak *et al.* conducted density functional theory (DFT) calculations on the structural properties of lapis lazuli analogs earlier, providing a reference for the analysis of related experimental spectral data [59].

Unlike zeolites found in nature, silicoaluminophosphate (SAPO) zeolites are synthesized artificially on a laboratory or industrial scale by replacing some of the Si and Al atoms in zeolite molecular sieves with P atoms. SAPO exhibits unique acidity, pore structures, and tunable properties. By ion exchange in the pores or on the surface of SAPO, Zn⁺ replaces the original cations (such as Na⁺, H⁺), resulting in Zn@SAPO, which alters the catalytic properties of SAPO and imparts unique acidity and active centers to the material. In terms of applications, Chen and co-workers demonstrated that the reaction of sublimed sulfur with Zn@SAPO-CHA produces a lapis lazuli analogue containing S₃^{•-} radicals, *i.e.*, (S₃, Zn)@SAPO-CHA [64]. Due to electron transfer between the S₃^{•-} radicals and H₂O molecules, this material can act as a sensor detecting trace amounts of H₂O in air and organic solvents. This property where S₃^{•-} radicals can be accessible and interact with guest molecules offers valuable insights for using lapis lazuli analogues as S₃^{•-} radical donor electrocatalysts in LSBs.

Recent advancements in solid-state LSBs have shown great promise in improving battery performance by addressing issues such as dendrite growth and polysulfide dissolution. However, solid-state LSBs still face challenges such as low ionic conductivity and sluggish solid–solid sulfur redox reaction (SSRR) [67–69]. S₃^{•-} radicals, with their high electrochemical reactivity and stability in lapis lazuli analogs, present an innovative solution to these challenges, with the analogy of assembling functional guest into zeolite cages as solid-state electrolytes [70]. Notably, lapis lazuli analogs are stable in high temperatures and exhibit excellent corrosion resistance [66], which can adapt to the preparation process conditions of solid-state electrolytes. In addition, these radical-carrier materials also possess electrically insulated properties and good ion conductivity, making them meet some basic requirements for solid-state electrolytes [71]. The ionic conductivity of lapis lazuli analogs may be improved by recombination into solid-state electrolytes, and the introduction of S₃^{•-} radicals is expected to promote SSRR kinetics and electrode–solid electrolyte interface activity [72], leading to better cycle stability and higher rate performance of solid-state LSBs.

2.2 Stabilization Mechanism in Solutions

Besides solid Lazurites of lapis lazuli analogs, S₃^{•-} radicals have been shown to be stable in aqueous solution under a pressure of 0.5 GPa and are expected to exist naturally at depth in the Earth's crust where subduction or high-pressure metamorphism occurs [73]. This radical ion is probably important in movement of copper and gold in hydrothermal fluids [74] and further influences sulfide chemistry under these environments.

The discovery and study of S₃^{•-} radicals in liquid solutions can be traced back to the observation of blue and red phenomena in polysulfide solutions. The earliest studies showed that sulfur in certain solvents (such as DMSO and DMF) could form solutions with deep blue colors, which aroused great interest among chemists. Initial studies found that sulfur dissolved in these solvents exhibited a deep blue color at low concentrations, which shifted to deep red when the polysulfide solution approaches saturation. This phenomenon was further proved by the unique chemical characteristics of sulfur in these solvents. The color transformation was closely related to the polarity and nucleophilicity of the solvent and was attributed to the solvent's role as an electron pair donor. In addition, sulfur can also produce blue solutions when heated with H₂O and a small amount of alkaline salt. At first, it was believed that the blue and red substances in the solvent should be caused by neutral particles, but later it was confirmed that the sulfur molecules formed free radicals with unpaired electrons through electrochemical reduction or other reactions. Therefore, these radicals could form stable complexes with solvent molecules, thus making the solution show different colors.

It has been revealed that S₈²⁻ will disproportionate into S₆²⁻ and 1/4 S₈ in electrolytes in LSBs [75–77]. Meanwhile, S₃^{•-} radicals can be also directly formed *via* the dissociation reaction of S₆²⁻ → 2S₃^{•-}, which is an entropy-driven process [41, 78]. It is worth noting that the dissociation pathways of LiPSs differ significantly between low-DN ether-based solvents (such as DME) and high-DN solvents (such as DMSO), which has been systematically studied by Zhang and co-workers [46, 79]. In low-DN solvents, Li₂S₆ primarily exists as a neutral molecule, with limited dissociation into Li⁺ and Li S₆⁻ or homolysis into LiS₃^{•-} radicals due to high dissociation constants (Fig. 2a) [46]. This results in a low concentration of radicals and minimal involvement in reaction pathways. In contrast, high-DN solvents, *e.g.*, DMSO,

enhance the dissociation of Li_2S_6 , leading to a significant formation of S_6^{2-} and $\text{S}_3^{\bullet-}$ radicals (Fig. 2b) [79], stabilized by strong polarity.

These findings highlight the critical influence of solvent properties on the dissociation behavior of polysulfides and their impact on LSBs performance. Han *et al.* also proposed generation mechanism of $\text{S}_3^{\bullet-}$ radicals in DMSO (Fig. 2c) [80]. The S_6^{2-} undergoes a series of dissociation and isomerization processes in high-DN solvents like DMSO. Initially, S_6^{2-} forms clusters with DMSO, with intermediate species (such as $\text{Li}_2\text{S}_6(\text{DMSO})_4$, $\text{Li}_2\text{S}_6(\text{DMSO})_8$), following by releasing S_6^{2-} and $\text{Li}_2\text{S}_6(\text{DMSO})_4$ with a free energy $-18.06 \text{ kcal mol}^{-1}$. Subsequently, S_6^{2-} undergoes a singlet-state isomerization and transitions to a triplet state *via* a spin-state crossing, leading to the formation of $\text{S}_3^{\bullet-}$ radicals as form of $\text{LiS}_3^{\bullet-}(\text{DMSO})_2$ with a kinetically feasible free energy barrier $9.62 \text{ kcal mol}^{-1}$. The $\text{S}_3^{\bullet-}$ radical becomes somewhat shielded inside the caged $\text{Li}^+(\text{DMSO})_2$ complex, which probably cannot easily access other solvated Li^+ and result in longer lifetime and solubility. The configuration of $\text{LiS}_3^{\bullet-}(\text{DMSO})_2$ is a natural analogy to superoxide radicals ($\text{O}_2^{\bullet-}$) in the non-aqueous Li-air battery [84].

No matter Li_2S_6 and $\text{LiS}_3^{\bullet-}$, Li^+ predominantly adopts a tetrahedral coordination mode in DMSO (Fig. 2d, e), with lithium atoms directly bonded to nearby oxygen or sulfur atoms, showcasing the remarkable stability of this four-coordinated structure. This coordination behavior is widely observed in Li^+ solvation and LiPSs systems across other solvents or solvent mixtures, such as DMA, DME, DOL, and DOL/DME blends (Fig. 2f-h) [46, 81, 82], playing a critical role in ensuring the structural stability and offering novel and specific design of functional solutions in LSBs. In fact, in the $\text{S}_3^{\bullet-}$ radicals' solution, the solvent molecules and $\text{S}_3^{\bullet-}$ radicals are competing for the coordination with Li^+ . The entire solvated structure should be considered as a whole.

The highly reactive $\text{S}_3^{\bullet-}$ together with its parent S_6^{2-} generated in LSBs is likely involved in the shuttling phenomenon, contributing to parasitic reactions with the metallic Li anode during cycling. Hence, when using the pore restriction strategy of the sulfur cathode to confine $\text{S}_3^{\bullet-}$ radicals, the size of the solvated cluster needs to be considered. After all, the size of $\text{S}_3^{\bullet-}$ radical (maximum 6.4 \AA) is only half that of the solvated cluster (maximum 13.1 \AA), as shown in Fig. 2i [83]. According to the calculation of Zhang *et al.* [79], $\text{S}_3^{\bullet-}$ radicals exhibits A1 irreducible representation in C_{2v} symmetry and gives rise to

specific Raman-active symmetric stretching vibration of the S–S bond, and electronic transitions of $n \rightarrow \pi^*$ resulting in the absorption at visible range of UV–Vis spectra (Fig. 2j). The Raman peak at 545 cm^{-1} and UV–Vis absorption at 617 nm is proven to be related to the existence of $\text{S}_3^{\bullet-}$ radicals via calculation. The calculation together provides theoretical basis and complementary methods for experimental detecting and characterizing $\text{S}_3^{\bullet-}$ radicals, enabling a detailed understanding of its structure and reactivity in polysulfide systems.

To explore the stability of $\text{S}_3^{\bullet-}$ radicals in different solvents, Lu *et al.* used the hard and soft acids and bases (HSAB) theory to explain origin [37]. The S_4^{2-} polysulfide (harder base) is better stabilized by weakly solvated Li^+ (hard acid) prevalent in low-DN solvents, whereas $\text{S}_3^{\bullet-}$, S_8^{2-} , and S_6^{2-} (softer bases) are better stabilized by strongly solvated Li^+ (soft acid) prevalent in high-DN solvents (Fig. 3a) [37]. Among them, $\text{S}_3^{\bullet-}$, dissociated from S_6^{2-} , is identified as the softest base, which enables $\text{S}_3^{\bullet-}$ are more abundant in high-DN solvents. By contrast, owing to the stable properties of S_4^{2-} in low-DN ether-based solvent, $\text{LiS}_3^{\bullet-}$ radicals, dissociated from Li_2S_6 should be less.

The HSAB theory cannot explain all the issues occurring in LSBs. During the discharge process of LSBs, the concentration of $\text{LiS}_3^{\bullet-}$ radicals reaches its maximum just before the onset of the plateau around 2.0 V . The dissociation of Li_2S_6 into $\text{LiS}_3^{\bullet-}$ alone cannot fully explain this phenomenon. Park *et al.* proposed that the formation of $\text{LiS}_3^{\bullet-}$ is also suggested to occur through electrochemical processes via $\text{Li} + \text{Li}_2\text{S}_6 = \text{Li}_2\text{S}_3 + \text{LiS}_3^{\bullet-}$ [86]. This reaction pathway complements the dissociation model and underscores the importance of electrochemical conditions in driving the formation of reactive intermediates like $\text{LiS}_3^{\bullet-}$ radicals, which contributes to the overall complexity of the polysulfide transformation process in LSBs.

In addition to $\text{S}_3^{\bullet-}$ radicals, other polysulfide radicals $\text{S}_n^{\bullet-}$ may also exist in the electrolyte [41, 85]. Their related generation paths are given in Fig. 3b. During the discharging process, the active sulfur (*viz.* cyclo- S_8) on cathode gradually gains two electrons to form dianion (S_8^{2-}) and dissolve in electrolyte [41]. Then, S_8^{2-} will form various $\text{S}_n^{\bullet-}$ ($n = 2-6$) radicals that are known as potentially key intermediates in the electrochemical reduction of cyclo- S_8 in LSBs [87]. Moreover, $\text{S}_n^{\bullet-}$ radicals obtain one more electron to form closed-shell dianions (such as S_6^{2-} and S_3^{2-}) corresponding to them. $\text{S}_n^{\bullet-}$ may also form longer chain closed-shell dianions,

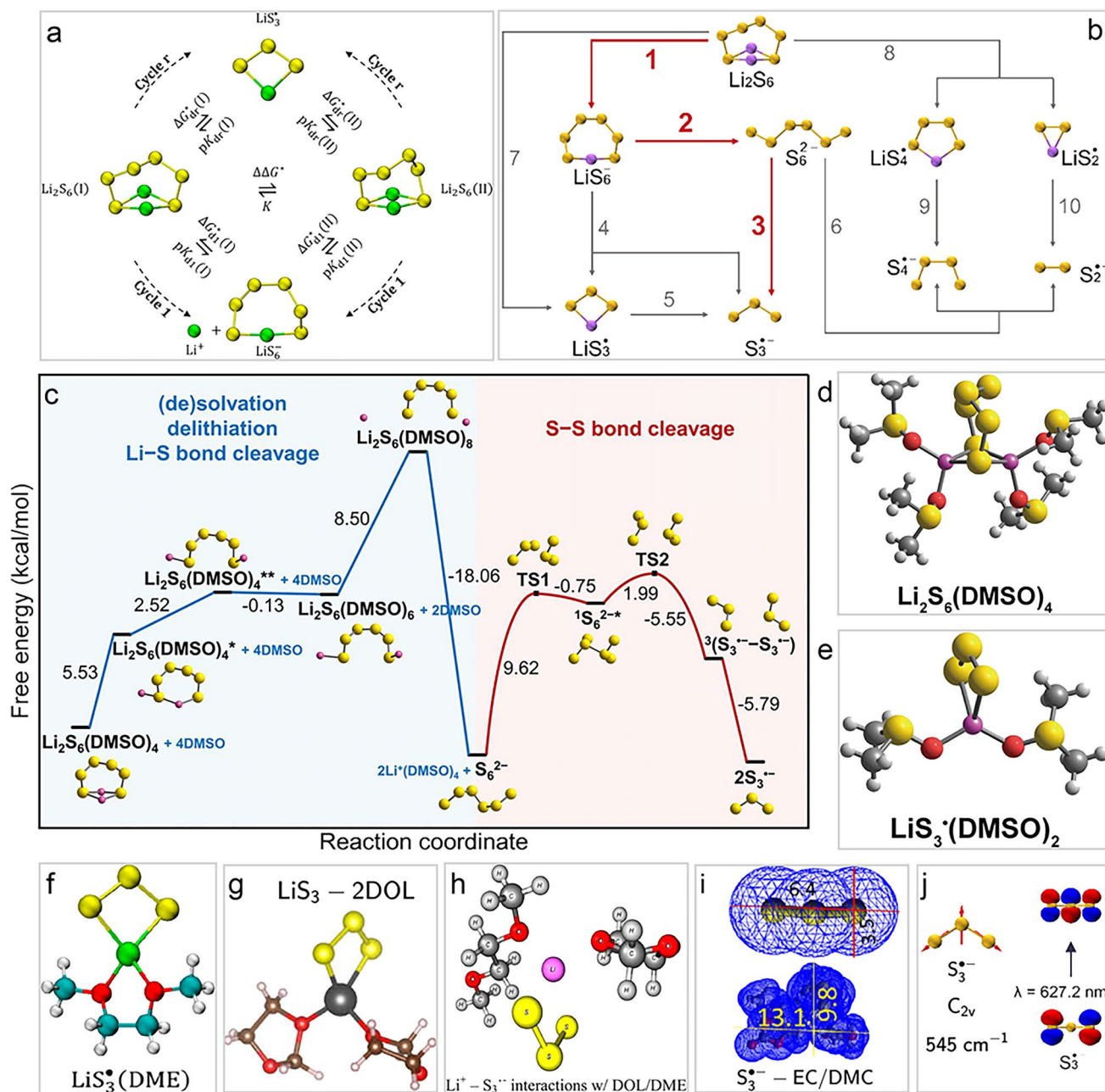


Fig. 2 S_3^-/LiS_3^* radicals in solutions. **a** Chemical equilibria of polysulfides and LiS_3^* radicals in DME [46]. Copyright 2021, American Chemical Society. **b** Dissociation routes of Li_2S_6 in DMSO [79]. Copyright 2023, Royal Society of Chemistry. **c** Generation mechanism of S_3^- radicals in DMSO [80]. Copyright 2023, Royal Society of Chemistry. **d** Optimized solvated structures of $Li_2S_6(DMSO)_4$ cluster and **e** $LiS_3^*(DMSO)_2$ cluster [80]. Copyright 2023, Royal Society of Chemistry. **f** Optimized gas-phase structures of $LiLiS_3^*(DME)$ cluster [46], Copyright 2021, American Chemical Society. **g** $LiS_3^*(DOL)_2$ cluster [81], Copyright 2020, Springer Nature. **h** $LiS_3^*(DOL-DME)$ cluster [82]. Copyright 2015, American Chemical Society. **i** Three-dimensional (3D) molecular models and Van der Waals surfaces of desolvated and dissolved S_3^- radicals for the electrolyte system [83]. Copyright 2020, Springer Nature. **j** Normal mode analyses and electron transition model of S_3^- [79]. Copyright 2023, Royal Society of Chemistry

e.g., S_{10}^{2-} and S_{12}^{2-} via symmetric coupling of S_5^- and S_6^- , respectively, while association of S_3^- and S_4^- yields S_7^{2-} [88, 89]. Similarly, Prendergast and co-workers think that LiS_4^*

and LiS_5^* , generated from the homolytic and heterolytic reaction of Li_2S_8 via $Li_2S_8 \rightarrow 2 LiS_4^*$ and $Li_2S_8 \rightarrow LiS_3^* + LiS_5^*$, respectively, also exist, by fitting the experimental XAS

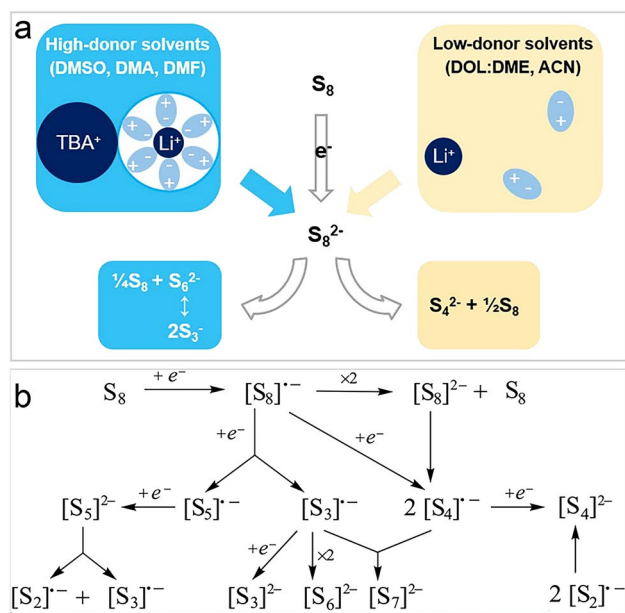


Fig. 3 The formation mechanism of $S_3^{\bullet-}$ radicals. **a** Summary of how polysulfide stability/speciation is affected by its interaction with cations and solvent molecules [37]. Copyright 2018, IOP Publishing. **b** Possible radical anion intermediates in the stepwise electrochemical reduction of S_8 [85]. Copyright 2023, Royal Society of Chemistry

spectrum with calculated ones [90]. Even more, Kawase *et al.* think other radicals, such as $LiS_2^{\bullet-}$, $LiS_6^{\bullet-}$, $LiS_7^{\bullet-}$ and $LiS_8^{\bullet-}$, exist through fitting experimental UV–Vis spectra with calculated ones [91]. However, there is no direct evidence of the existence of $S_n^{\bullet-}$ radicals, except for $S_3^{\bullet-}$ radicals, since the overlapping of spectra signals [92].

3 Spectral Characterization Technique Detecting $S_3^{\bullet-}/LiS_3^{\bullet-}$ Radicals

3.1 Magnetic Resonance Spectroscopy

$S_3^{\bullet-}/LiS_3^{\bullet-}$ radicals in LSBs have been fully characterized by a variety of spectroscopic techniques. Therein, ESR spectroscopy is a critical tool for identifying these paramagnetic radicals in solution and solid catalyst defects (such as oxygen vacancy and sulfur vacancy in metal oxides) with the spin transitions of unpaired electrons. This unpaired electron interacts with external static magnetic field and are motivated from lower to higher energy levels by low-energy microwave (on the order of meV) in ESR test. Besides, ESR tests are usually operated in dark and low-temperature condition. Thus, this applied microwave

radiation has a small effect on the polysulfide ions, enabling ESR technique to have a strong anti-interference ability in detect $S_3^{\bullet-}$ radicals.

Figure 4a shows the *ex situ* ESR of several prepared polysulfides solution with DOL/DME, revealing that $S_3^{\bullet-}$ radicals are present in most situation, due to the rapid, non-stepwise disproportionation of S_8^{2-} , S_6^{2-} , and S_4^{2-} . In view of the high precision of the detection of $S_3^{\bullet-}$ radicals by ESR spectrum, the weak changes of $S_3^{\bullet-}$ radicals during the operation of LSBs can be well captured by *in situ* ESR spectrum (Fig. 4b) [48]. Persistent during electrochemical cycling, the content of the $S_3^{\bullet-}$ radicals shows a significant change, indicating they actively participate in sulfur conversion reaction pathways, making their precise detection by ESR essential for understanding and optimizing polysulfide systems in LSBs.

In addition to directly detecting $S_3^{\bullet-}$ radicals, another magnetic resonance technology, nuclear magnetic resonance (NMR), based on the principle of interaction between nuclear spin and external magnetic field, is also used to indirectly detect the presence of $S_3^{\bullet-}$ radicals. As shown in Fig. 4c, the chemical shift of 7Li changes arises from the dynamic coordination environment of Li^+ with different sulfur-based species, such as $S_3^{\bullet-}$ radicals, in polysulfide systems [82]. These chemical shifts that correspond to different lithium nuclei (approximately -58 ± 10 ppm) do not overlap with the signal range of LiPSs (such as $Li_2S_x, \pm 2$ ppm), but clearly distinguishing $S_3^{\bullet-}$ radical signals from polysulfide signals. Besides, in *in situ* environments, $S_3^{\bullet-}$ radical signals are particularly prominent due to the significant increase in radical concentration during electrochemical reactions, further enhanced by local electric fields and charged surfaces. The unique sensitivity and non-interference of *in situ* NMR provide a supplementary tool for studying $S_3^{\bullet-}$ radicals and their critical role in polysulfide reaction mechanisms of LSBs.

3.2 Raman Spectroscopy

Raman spectroscopy is another tool based on photon scattering technology capable of detecting $S_3^{\bullet-}$ radicals with its characteristic vibrational peak at 531 cm^{-1} and other LiPSs, enabling the study of sulfur radical dynamics and their role in the polysulfide reaction mechanisms during the operation of LSBs. Liu *et al.* identify key intermediate species and their transformations by monitoring the

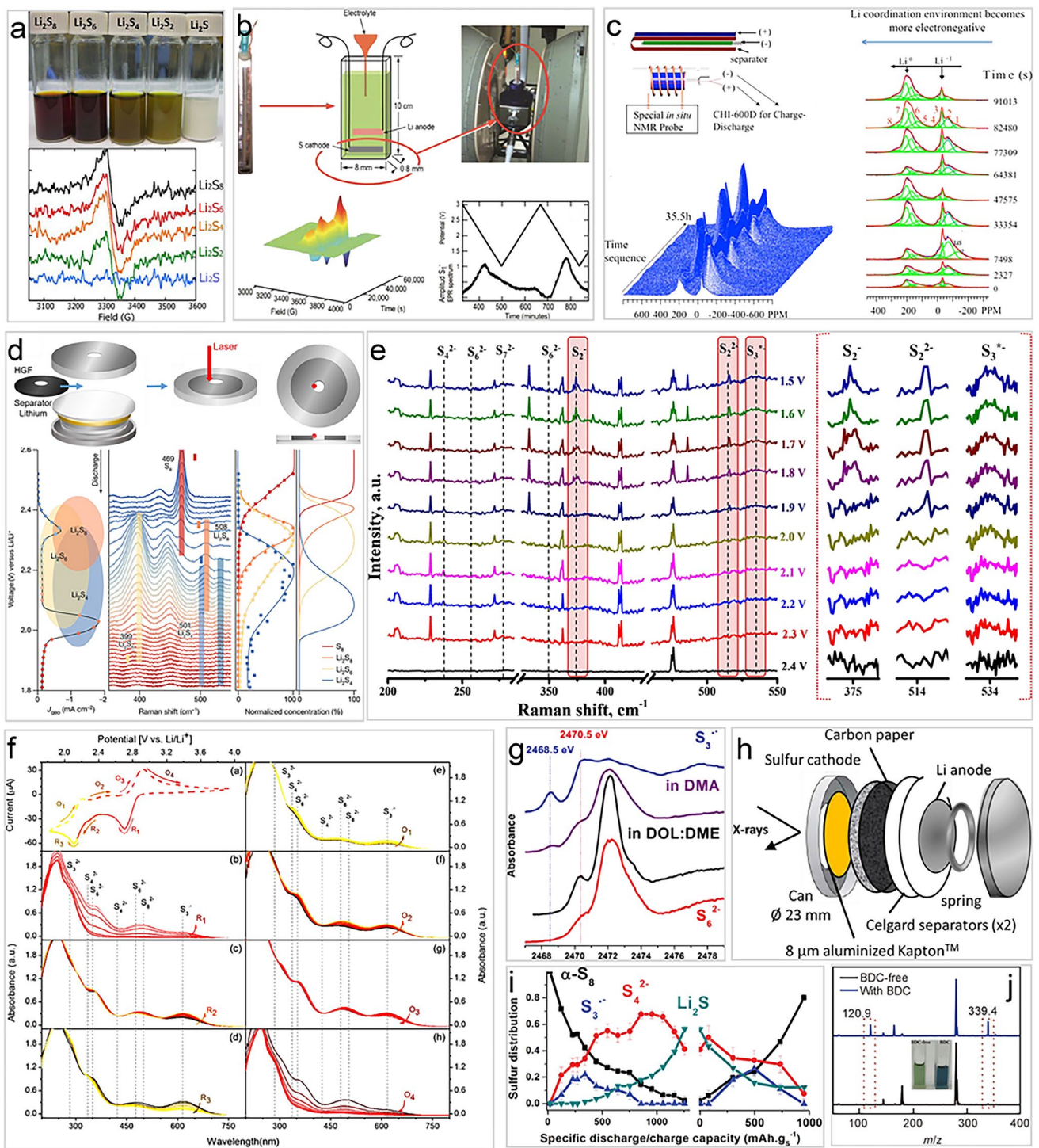


Fig. 4 Spectrum technologies to detect S_3^-/LiS_3^* radicals. **a** The upper: Li_2S_x in DOL/DME solvents, and the bottom: corresponding EPR spectra [48]. Copyright 2015, IOP Publishing. **b** EPR spectra for *in situ* testing and capturing of radical resonance signals [48]. Copyright 2015, IOP Publishing. **c** *In situ* NMR probe and the acquired spectra [82]. Copyright 2015, American Chemical Society. **d** *In situ* Raman results with the carbon-based catalytic electrode and DOL/DME electrolyte [50]. Copyright 2024, Springer Nature. **e** *In situ* Raman measurements for LiPS redox species evolution on carbon-based cathode and TEGDME electrolyte [93]. Copyright 2022, IOP Publishing. **f** *In operando* UV – Vis spectra of each reaction step with carbon paper cathode and DMSO-based S_8 catholyte [35]. Copyright 2016, American Chemical Society. **g** K-edge XANES spectra from discharging LSBs using the liquid electrolytes compared with the reference materials [45]. Copyright 2015, John Wiley and Sons. **h** Schematic of the operando cell for *in situ* XAS [94]. Copyright 2013, American Chemical Society. **i** Linear combination fit analysis of the XANES spectra upon cycling [45]. Copyright 2015, John Wiley and Sons. **j** ESI-MS results of BDC-free and BDC electrolyte solutions [49]. Copyright 2024, Springer Nature

characteristic vibrational peaks of polysulfides during discharge (Fig. 4d) [50]. For example, the initially detected S_8 peak (469 cm^{-1}) gradually disappears, while peaks corresponding to Li_2S_8 (508 cm^{-1}), Li_2S_6 (399 cm^{-1}), and Li_2S_4 (501 cm^{-1}) emerge and reach their respective maxima, reflecting the electrochemical conversion and equilibrium reactions between polysulfides. Additionally, a minor Raman peak (531 cm^{-1}) attributed to the LiS_3^* was detected, which is associated with a homolytic reaction of the electrochemically inactive Li_2S_6 ($Li_2S_6 \leftrightarrow 2LiS_3^*$).

Although the concentration of LiS_3^* radicals in the low-DN electrolyte, such as traditional ether-based electrolyte, is relatively low (less than 3% of Li_2S_6), LiS_3^* was thought to have limited impact on the equilibrium of the overall reaction network. However, because of the high sensitivity of Raman spectroscopy that enables the detection of these critical and transient species, this provides valuable insights and evidence into the mechanism revelation of radicals involvement in LSBs. Similarly, Thangavel *et al.* used TEGDME, a low-DN solvent, as the electrolyte solvent coupled with a carbon cloth electrode, under which the *in situ* Raman spectroscopy signal was detected only corresponding to weak S_3^{2-} radicals (Fig. 4e) [93]. Herein, Li_2S_6 is considered electrochemically inactive when using carbon electrodes, as described in this and the previous reference. However, when the electrode was replaced with metallic Ni, the S_3^{2-} radicals' signal was significantly enhanced. This enhancement is likely due to the strong polar–polar interactions between the Ni electrode and Li_2S_6 , *i.e.*, the Ni electrode adsorbs and stabilizes Li_2S_6 . This phenomenon highlights the critical role of metal electrodes in promoting the formation of trisulfur radicals. Inspired by this, emerging single-atom catalysts (SACs), particularly those based on transition metals, have shown potential in enhancing the electrochemical reactions in LSBs [95]. SACs provide a high density of active sites with precise control over atomic coordination [96], which can effectively stabilize polysulfides, improve reaction kinetics, and facilitate the formation of trisulfur radicals. The unique properties of SACs could thus offer significant advantages in optimizing LSB performance. Similar phenomena will be discussed later in Sect. 4.4. The increment of Li_2S_6 leads to the dissociation of more LiS_3^* and altering the electrochemical reaction pathway. These results make Raman technology vital in detecting S_3^{2-}/LiS_3^* radicals of LSB system, but it need to consider the effect of light source

(commonly used 532, 632.8 nm *etc.*) on polysulfide, which has been proved by Zhang and co-workers [79].

3.3 UV–Vis Spectroscopy

UV–Vis spectroscopy is another effective method based on electron exciting for monitoring blue color S_3^{2-} radicals solution, differing from Raman spectroscopy through vibrational signatures. UV–Vis spectroscopy using ultraviolet and visible light sources ($< 700\text{ nm}$) identifies the electronic transitions signature of S_3^{2-} radicals, such as $n \rightarrow \pi^*$, through its characteristic absorption peak at 617 nm, enabling dynamic, time-resolved tracking of S_3^{2-} radicals concentrations during LSBs operation. Its sensitivity to electronic environments allows selective detection of S_3^{2-} radicals even in complex polysulfide systems, providing valuable information on reaction kinetics.

Zou *et al.* used UV–Vis to track the evolution of polysulfide species with cyclic voltammetry (CV) operation, revealing key reduction and oxidation mechanisms (Fig. 4f) [35]. During reduction, cyclo- S_8 undergoes stepwise transformations to S_8^{2-} , S_6^{2-} , S_4^{2-} , S_3^{2-} , and S_3^{2-} in DMSO. Each reaction shows distinct kinetics, with S_8^{2-} reaching a steady state quickly, while S_6^{2-} disproportionation ($S_8^{2-} \rightarrow S_6^{2-} + 1/4 S_8 \downarrow$) and S_3^{2-} radicals generation ($S_6^{2-} \rightarrow 2S_3^{2-}$ or $3 S_4^{2-} \rightarrow 2S_3^{2-} + 2 S_3^{2-}$) take longer. During oxidation, polysulfides are reformed and ultimately converted back to cyclo- S_8 . This observation indicates that S_3^{2-} radicals are the most stable and dominant reaction intermediates in high-DN solvents. The study highlights the central role of S_3^{2-} radicals in redox processes ($S_3^{2-} + e^- \rightarrow S_3^{2-}$) and demonstrates the value of UV–Vis spectroscopy in correlating electrochemical potential with the dynamic behavior of polysulfides. However, it may need to consider the photo-sensitive characteristic of the dissociation reaction of S_6^{2-} into S_3^{2-} radicals when making a quantitative analysis [79].

3.4 XAS Spectroscopy

XAS is a highly powerful analytical technique that provides information about the valence state, local structure, and coordination environment of elements in a sample. The high-energy X-ray (up to 10 keV) with 0.01–10 nm wavelength is enough to excite the inner electrons (such as K-shell or L-shell electrons) out of the atom, producing an X-ray

absorption spectrum. Through XAS testing, researchers can gain deep insights into the electronic structure, chemical state and local geometry of the atom of trisulfur radicals and other polysulfides [90], including their coordination with catalysts, metal ions or solvent molecules. This coordination significantly influences their electrochemical behavior, reaction kinetics, and stability during cycling of in LSBs. By analyzing both the X-ray absorption near edge structure (XANES) and extended X-ray absorption fine structure (EXAFS) regions, the local atomic environment, oxidation states, and bond distances can be correlated with the reactivity and stability of sulfur species [45, 97]. Therefore, the ability of *in situ* XAS to track the evolution and relationship of various sulfur species is crucial for understanding structure–property relationship, offering valuable insights into how these features affect battery performance and help to design better sulfur cathodes and electrolytes.

Wujcik *et al.* simulated the XANES spectra of the LiS_3^{\bullet} radicals dissolved in TEGDME, based on *ab initio* molecular dynamics (AIMD) sampling performed at 298 K [45]. This simulation shows that the terminal sulfur $1s-3p$ (π^*) transition of LiS_3^{\bullet} radical give rise to the characteristic peak at 2468.5 eV, which can be easily distinguished from other polysulfide dianions. Since the physical state (crystalline, amorphous, or solute) does not affect the spectral features in the XANES region, the XAS spectrum of an ultramarine pigment of lapis lazuli analog containing S_3^{2-} radicals can be used as a reference spectrum for detecting S_3^{\bullet} radicals in solution (Fig. 4g). Experimental results show that there is a relatively weak and narrow peak (2468.5 eV) below the pre-edge peak (2470.5 eV) of polysulfide dianions in the reference spectrum, which is in accordance with theoretical calculation, thereby proving that the characteristic peak at 2468.5 eV belongs to the S_3^{\bullet} radical for the blue solution of polysulfides.

According to XAS experimental spectra, Cuisinier *et al.* further demonstrated that S_6^{2-} readily dissociated into S_3^{\bullet} radicals in high-DN solvents, such as DMA and DMSO, resulting in high radical concentrations [45]. By using *in situ* XANES technology and the special electrode (Fig. 4h) [94], the content of S_3^{\bullet} varies with the discharge and charge process up to 25% of the total sulfur (Fig. 4i) [45]. Thus, its role as an internal redox mediator is confirmed, enabling 24% more sulfur utilization of DMA-based electrolyte than low-DN TEGDME electrolyte. Conversely, S_3^{\bullet} radicals are

not stabilized in a measurable (*i.e.*, $\ll 5\%$) concentration in traditional ether-based electrolytes at the experimental timescale (*ca.* a few minutes). In addition, S_3^{\bullet} radicals are found to react with DOL at elevated temperatures, while DME remains inert. So, DOL as a co-solvent with high-DN solvent should take into account the effect of temperature. These findings highlight the critical role of S_3^{\bullet} radicals dynamics, solvent interactions, and the importance of anode protection for utilizing high-DN solvents in LSBs. There is no doubt that the higher energy of XAS than UV–Vis makes it difficult to quantitatively analyze S_3^{\bullet} radicals and polysulfides.

3.5 Mass Spectrometry

S_3^{\bullet} radicals are challenging to be detected in ether-based solvents due to their low stability and rapid conversion to other polysulfide species. However, biphenyl-4,4'-dicarboxylic acid (BDC) serves as a stabilizer, enhancing the formation and stability of S_3^{\bullet} radicals coordination compound through strong Lewis acid–base interactions. Dou *et al.* confirms that the presence of S_3^{\bullet} radicals becomes prominent in the presence of BDC using UV–Vis spectroscopy in the form of a strong characteristic absorption peak at 612 nm (upper panel of Fig. 4j) [49]. This stabilization of coordination compound also enables electrospray ionization-mass spectrometry (ESI–MS) to detect specific mass-to-charge ratios (m/z) corresponding to $[\text{BDC-S}_3^{\bullet}]$ complexes at $m/z = 339.4 \text{ kg C}^{-1}$ (bottom panel of Fig. 4j), providing direct evidence of the radical and its interactions. Together, these findings highlight the critical role of BDC in facilitating the detection and study of S_3^{\bullet} radicals in systems where it would otherwise remain elusive.

4 Promoting Strategies for the Generation of $\text{S}_3^{\bullet}/\text{LiS}_3^{\bullet}$ Radicals

The unique electronic structure of sulfur radicals, including unpaired electrons, electron delocalization, and SOMO-LUMO energy level characteristics, gives them high electron transfer efficiency over sulfur in redox reactions. This makes sulfur radicals highly reactive in

Table 1 Comparison of parameters and performances of LSBs with different strategies for generation promotion of trisulfur radicals

Strategy	Catalyst/Material type	Initial capacity (mAh g ⁻¹)	E/S (μL mg ⁻¹)	Sulfur Loading (mg cm ⁻¹)	Rate(C)	Cycles	Dosage	References
High-DN solvent	DMSO	1250	~ 10.0	~ 6.0	–	–	50 v%	[42]
High-DN solvent	DMI ^a	1595	~ 5.0	10.0	0.1	~ 50	50 v%	[100]
Co-solvent	TMU ^b /DOL	1524	3.0	2.5	0.1	180	50 v%	[43]
High-DN solvent additive	NMP ^c	1250	7.1	4.2	0.3	340	1 v%	[101]
High-DN anion additive	Br ⁻	1535	–	~ 3.0	0.2	80	1 M	[102]
High-DN anion additive	FL ^d	1099	5.0	5.5	0.5	520	50 mM	[103]
High-DN anion additive	DPDTe ^e	1227	5.0	5.0	1.0	300	50 mM	[104]
Metal compound catalyst	WO _{3-x}	1221	20.0	2.5	4.0	300	4.5 wt% ^f	[53]
Metal compound catalyst	VS _{2-x}	1471	15.0	5.6	0.2	400	10 wt% ^f	[105]
Metal compound catalyst	Co ₉ S ₈ /MoS ₂ -rGO	1129	13.8	3.2	0.5	300	30 wt% ^f	[106]
MXene	Ti ₃ C ₂ T _x	~ 1000	8.0	10.5	–	80	36 wt% ^f	[107]
Carbon-based catalyst	UN/O-CNS	1211	–	4.2	1.0	1500	32 wt% ^f	[54]

^aDMI refers to 1,3-dimethyl-2-imidazolidinone. ^bTMU refers to tetramethylurea. ^cNMP refers to N-methyl-2-pyrrolidone. ^dFL refers to fluorenone. ^eDPDTe refers to diphenyl ditelluride. ^fCalculated based on the overall mass of the cathode composite

organic synthesis, promoting C–S bond formation, cyclization reactions, and sulfurization reactions, and they are widely used in the green synthesis of sulfur-containing compounds and heterocyclic molecules [98, 99]. Similarly, these characteristics make sulfur radicals key intermediates in LSBs, contributing to enhance sulfur utilization and reaction kinetics.

The S₃⁻ radical is relatively stable and primarily exists in high-DN solvents such as DMSO, while the LiS₃[•] radical is less stable and exists in lower concentrations in traditional low-DN solvents such as DOL/DME. Regarding reactivity, the LiS₃[•] radical should have higher reactivity than S₃⁻ due to the higher vertical electron affinity (1.9 eV) of LiS₃[•] against the more negative value (~ -3.0 eV) of S₃⁻ [79]. However, note that both of these radicals are highly reactive species in their respective electrolyte systems. The enhanced stability of S₃⁻ in high-DN solvents contributes to improved sulfur utilization and reduced discharge/charge overpotential in LSBs compared with those using traditional low-DN solvents. On the other hand, the high reactivity of LiS₃[•] radicals makes low-DN LSBs promising candidates, as these radicals can act as a mediating catalyst, due to better compatibility with the lithium anode than high-DN solvent systems. The following sections reviews strategies that promote the stability of reactive radicals, while minimizing the adverse effects of high-DN solvents, as summarized in Table 1.

4.1 High-DN Solvents and Their Co-Solvents

High-DN solvents (such as DMA, DMSO, DMF) refer to the use of only a high-DN solvent (such as DMSO) as an electrolyte solvent, which show significant potential in LSBs due to their higher solubility for polysulfides than the traditional DOL/DME electrolytes, enabling high-energy-density LSBs under lean E/S ratio conditions. In addition, high-DN electrolytes facilitate polysulfide disproportionation and dissociation reactions, producing the intermediate S₃⁻ radicals (Fig. 5a) [32]. These intermediates act as redox mediators, accelerating electrochemical reactions and improving sulfur utilization. Additionally, owing to the mediated role of S₃⁻ radicals, high-DN electrolytes promote the deposition of Li₂S in a 3D particle-like morphology, avoiding the passivation issues caused by insulating films, thereby maintaining electrode conductivity and extending cycle life.

Unfortunately, high-DN solvents have poor lithium metal compatibility and high viscosity, which lead to frequent side reactions and low ionic conductivity respectively, thereby limiting battery cyclability and rate ability. For example, those solvents containing strongly polar C=O or C–N bonds, such as DMA and DMF, readily react with the strongly reducing Li metal, leading to the formation of decomposition products [42]. Among them, 1,3-dimethyl-2-imidazolidinone (DMI), as a high-DN solvent, rarely exhibits excellent performance in LSBs (Fig. 5b) [100], thanks to its unique

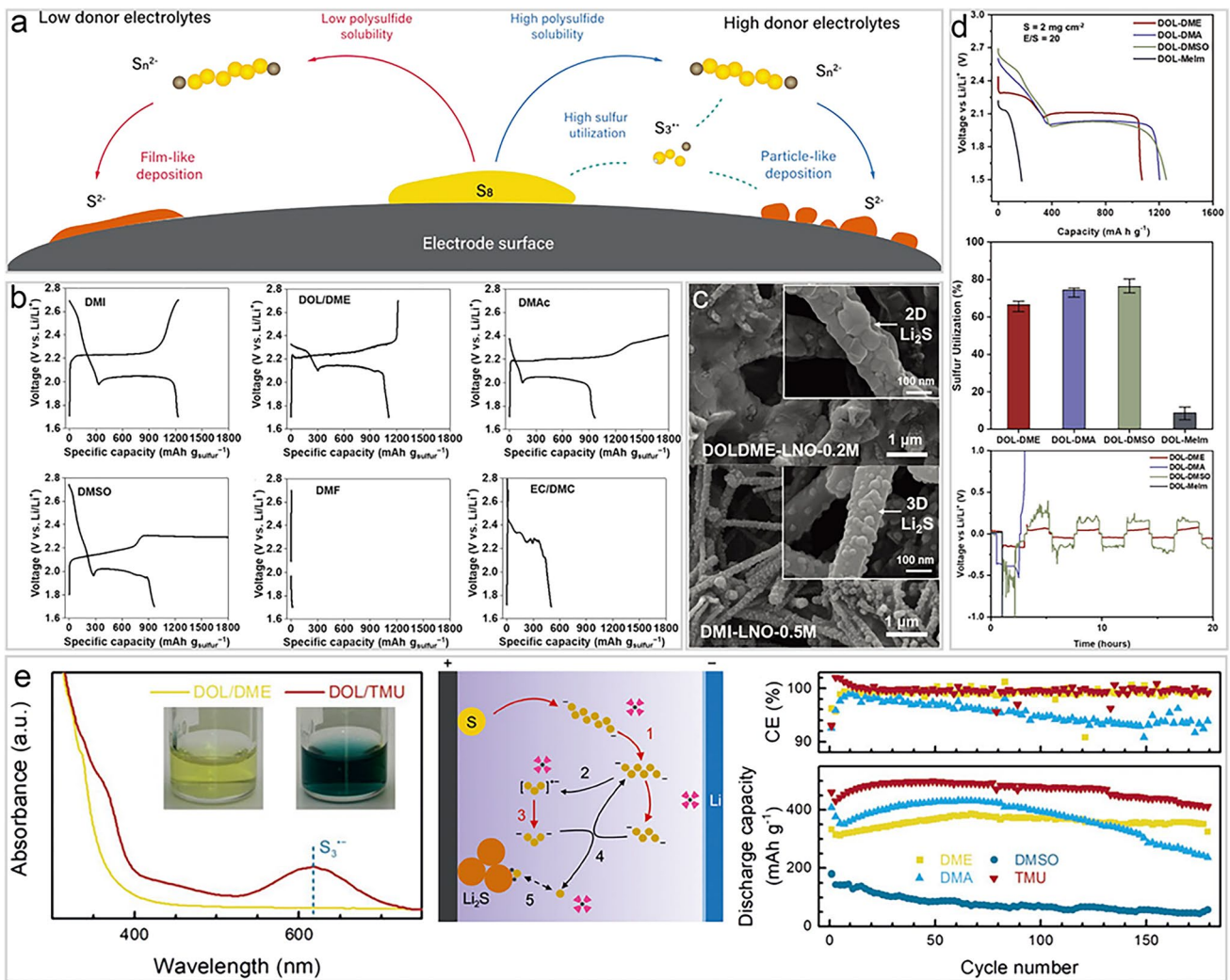


Fig. 5 High-DN solvents and co-solvents to promote the generation of $S_3^{•-}/\text{Li}_2S_3$ radicals. **a** Schematic illustration of the lithiation process of low and high donor electrolytes in LSBs [32]. Copyright 2020, John Wiley and Sons. **b** Charge–discharge profiles of LSBs with different electrolytes [100]. Copyright 2020, John Wiley and Sons. **c** SEM images of Li_2S deposits on carbon electrodes [100]. Copyright 2020, John Wiley and Sons. **d** Discharge profiles and sulfur utilizations of sulfur cathode, and galvanostatic cycling of lithium symmetric cells in four blended electrolytes [42]. Copyright 2019, John Wiley and Sons. **e** The left: UV–Vis spectra of Li_2S_6 solution in different solvents, the middle: properties, mechanism and performance of TMU-based electrolyte, the right: cycling performance of Li/polysulfide in different solvent of coin cells [43]. Copyright 2018, John Wiley and Sons

molecular structure and compatibility with the metallic Li anode. By replacing the oxygen atoms in ethylene carbonate (EC) with nitrogen, DMI stabilizes the carbonyl group, reducing its reactivity with polysulfides. It also promotes the formation and stabilization of $S_3^{•-}$ radicals, activating additional reaction pathways that enhance sulfur utilization. Furthermore, DMI encourages the deposition of Li_2S in a 3D particle form, preventing electrode passivation (Fig. 5c) [100]. However, it still needs to be used in conjunction with LiNO_3 additives to further stabilize the metallic Li anode

interface, reduce the shuttle effect, and improve cycling performance.

It should be pointed out here that in addition to being observed through SEM electron microscopy, Li_2S 3D or 2D deposition can also be judged through the easier-to-achieve current curve fitting of the deposition process, according to Armstrong, Fleischmann and Thirsk (AFT) and Bewick, Fleischmann and Thirsk (BFT) models [108], or Avrami equation [43, 109, 110]. Avrami equation is described as: $Y(t) = 1 - \exp(-\kappa t^n)$, where $Y(t)$ is the transformed volume

fraction, κ is rate constant, n is Avrami exponent, t is time. When $n=2$, it corresponds to 2D instantaneous nucleation, that is, it becomes the BFT model, and when $n=3$, it corresponds to 3D instantaneous nucleation, that is, it becomes the AFT model. The Avrami equation can be used to further explore the regulatory effect of trisulfur radicals on Li_2S deposition.

To overcome the shortcomings of high-DN solvents, researchers have proposed a co-solvent strategy rationally. By the way, a high-DN solvent is mixed with a traditional ether-based solvent (such as DOL) in equivolume to combine the advantages of both solvents, thereby lowering its corrosiveness toward metallic Li while retaining its advantages in enhancing polysulfide solubility and promoting reactions. This solvent system significantly improves the cycling stability of the battery and enhances sulfur utilization, even under low E/S ratios. Furthermore, the co-solvent strategy optimizes the solvent combination to promote the formation of particulate Li_2S deposits, preventing electrode passivation and further improving battery performance (Fig. 5d) [42].

In particular, the high-DN solvent tetramethylurea (TMU) that has good compatibility with the metallic Li anode significantly enhances the performance of LSBs when used in combination with the traditional ether-based solvent DOL (Fig. 5e) [43]. In the DOL/TMU electrolyte environment with 1.0 M LiTFSI and 0.30 M LiNO_3 as the salts, $\text{S}_3^{\bullet-}$ radicals still exhibit visible blue color because of the high stability of high-DN solvent against $\text{S}_3^{\bullet-}$ radicals, indicating $\text{S}_3^{\bullet-}$ radicals exist in practical LSB system without being affected by lithium salts. TMU improves polysulfide solubility, promotes the generation of $\text{S}_3^{\bullet-}$ radicals, and activates multiple reaction pathways, thereby increasing the battery's efficiency and energy density. Additionally, $\text{S}_3^{\bullet-}$ radicals of TMU facilitate the 3D deposition of Li_2S , preventing electrode passivation and enhancing specific capacity and cycling stability. Under the high E/S ratios, this co-solvent system significantly boosts battery performance, providing higher energy density and longer cycle life. However, extended cycling is limited by electrolyte depletion under harsh conditions, necessitating consideration of anode electrolyte consumption when reducing the E/S ratio in LSBs.

4.2 High-DN Solvent Additives

High-DN solvent additive refers to that the volume of high-DN solvent in the electrolyte is relatively small (generally

about 1% of the traditional ether-based solvent electrolyte). A small amount of high-DN solvent can effectively stabilize the $\text{S}_3^{\bullet-}$ radicals and avoid the corrosion of metallic Li anode at high concentration. In addition, because the proportion of high-DN solvents is very low, the side reaction caused by high concentration is avoided, and the metallic Li anode compatibility is significantly improved. Liang and co-workers proposed using high-DN solvent such as N-methyl-2-pyrrolidone (NMP) as electrolyte additives (<1 vol%) [101]. This strategy minimizes the direct reactions between the solvent and the metallic Li anode, preventing corrosion of the metallic Li, while retaining the solvent's advantage in promoting and stabilizing $\text{S}_3^{\bullet-}$ radicals (Fig. 6a). At low concentrations, NMP preferentially coordinate with Li^+ ions, forming a stable solvated layer that inhibits corrosion of the metallic Li. Simultaneously, it enhances the 3D nucleation of Li_2S , decreasing oxide reaction energy barrier and significantly improving sulfur conversion efficiency and the reversibility of the battery. This strategy not only reduces side reactions but also improves the cycling stability and capacity retention of LSBs, significantly extending the battery's lifespan and providing an efficient and stable solution for the optimization of LSB electrolytes with a significant content of $\text{S}_3^{\bullet-}$ radicals. However, considering that high-DN solvents may affect the stability of the metallic Li anode, especially under lean electrolyte conditions, it is necessary to explore other methods to stabilize $\text{S}_3^{\bullet-}$ radicals.

4.3 High-DN Anion Supporting Electrolytes or Electrolyte Additives

In addition to high-DN solvents, high-DN anion supporting electrolytes or electrolyte additives, that is, adding a small amount of salts (usually more than 0.5 M for supporting electrolytes and less than 0.1 M for electrolyte additives) with high-DN anions (such as Br^-) to traditional ether-based solvent electrolytes, can also effectively stabilize $\text{S}_3^{\bullet-}$ radicals through their strong coordination ability. This strategy provides a more stable and efficient solution without the need for additional protection of the metallic Li anode, offering a new direction for the optimization of LSBs. Compared with low-DN anions (such as the commonly used lithium salt TFSI⁻, $\text{DN} = 5.4 \text{ kcal mol}^{-1}$), high-DN anions, such as Br^- ($\text{DN} = 33.7 \text{ kcal mol}^{-1}$) and NO_3^- ($\text{DN} = 22.2 \text{ kcal mol}^{-1}$), can enhance the solubility of

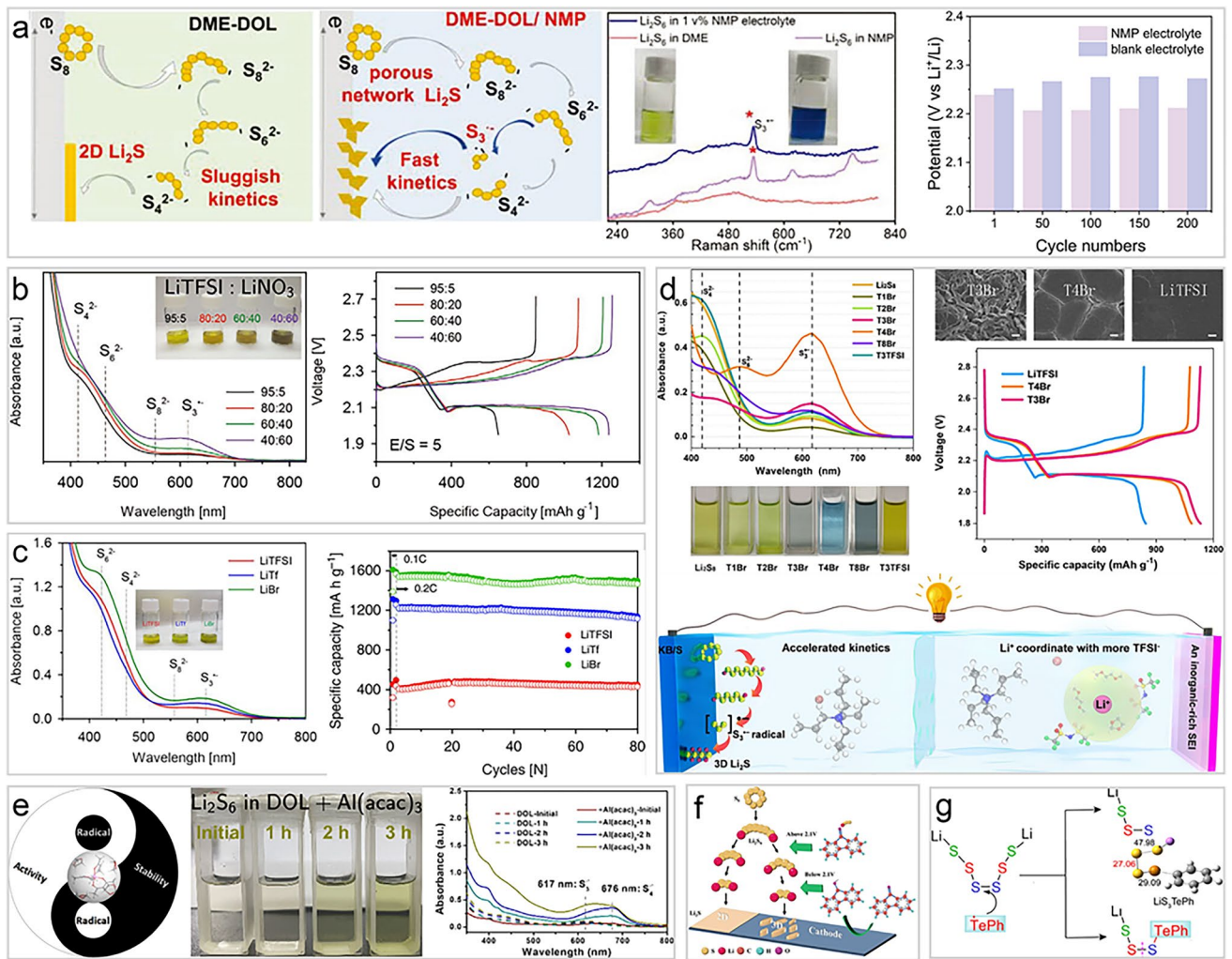


Fig. 6 High-DN solvent and anion additives to promote generation of S_3^-/LiS_3^* radicals. **a** The first and second: schematic of Li_2S nucleation behaviors, the third: Raman of Li_2S_6 solutions, the fourth: Li_2S oxidation overpotential of cells [101]. Copyright 2022, John Wiley and Sons. **b** The left: UV–Vis spectra of the LiPSs in the co-salt electrolytes, the right: discharge/charge profiles of LSB cells [111]. Copyright 2020, John Wiley and Sons. **c** UV–Vis spectra of Li_2S_8 solutions and capacities comparison of cells [102]. Copyright 2019, Springer Nature. **d** The upper left: UV–Vis spectra of the Li_2S_8 solution with the different salts, the upper right: SEM images of the cycled Li anodes, and corresponding discharge/charge profiles, the bottom: schematic of T3Br’s influence mechanism [56]. Copyright 2023, John Wiley and Sons. **e** The left: schematic of balancing stability and activity of S_3^- radicals by metal complexes, the middle: time dependent photographs, and the right: UV–Vis spectra of Li_2S_6 with $Al(acac)_3$ [55]. **f** Schematic of organic additives regulating LSB reactions [103]. Copyright 2023, Elsevier. **g** Simulation results of LiPSs reacting with organic additives [104]. Copyright 2022, John Wiley and Sons

LiPSs due to their strong electron-donating ability, suppress the passivation of the metallic Li anode, and promote the 3D particulate deposition of Li_2S .

Chu *et al.* find that the concentration of the S_3^- radicals is elevated with the higher content $LiNO_3$ as one of the mixed lithium salts (Fig. 6b) [111], demonstrating the role of 0.6 M NO_3^- with higher DN on stabilizing S_3^- radicals. The results also indicate that the S_3^- radicals still exhibit visible blue color in the traditional ether-based solvents containing

high-DN anions. Based on its high electron-donating ability, the full cell could achieve a high sulfur utilization, and the capacity could reach above 1200 mA h g^{-1} . Similarly, this research team also observed that the concentration of S_3^- radicals is increased with increasing the high-DN Br^- content in the traditional ether-based solvent electrolyte (Fig. 6c) [102]. These high-DN Br^- stabilize the S_3^- radicals, improving sulfur conversion efficiency and reaction kinetics, which enhances the reversibility and cycling stability of the

battery. Furthermore, these high-DN salt anions promote the 3D deposition of Li_2S , reduce electrode passivation, decrease polarization, and further enhance cycling performance and capacity retention.

Building on the important role of high-DN Br^- anions in stabilizing the generation of $\text{S}_3^{\bullet-}$ radicals, Meng *et al.* further promoted the generation of $\text{S}_3^{\bullet-}$ radicals used 0.1 M quaternary ammonium salts (QASs) with tetra-alkyl ammonium cations (defined as $\text{T}[x+1]^+$, x denoting the number of $-(\text{CH}_2)-$ units on single chain, and $x=0, 1, 2, 3,$ or 7) as an electrolyte additive by manipulating the cations [56]. Figure 6d shows that QASs with symmetric carbon chains of specific lengths are more effective at triggering the generation of $\text{S}_3^{\bullet-}$ radicals, particularly the T3^+ and T4^+ structures. Among them, the T4Br additive provides the best stabilization of $\text{S}_3^{\bullet-}$, causing the polysulfide solution to exhibit a distinct blue color. Notably, T4^+ paired with the low-DN TFSI^- anion to form T4TFSI cannot trigger the generation of $\text{S}_3^{\bullet-}$ radicals, indicating that the cation and high-DN anion have a synergistic coupling effect in promoting the generation of T4^+ . Furthermore, since the cation QAS promotes the formation of the solid electrolyte interface (SEI) on the metallic Li anode by altering the solvated structure of Li^+ , T3Br exhibits the best protection of the metallic Li anode, resulting in superior electrochemical performance in LSBs. This further underscores the importance of the synergistic effect between cations and high-DN anions in enhancing the performance of LSBs based on $\text{S}_3^{\bullet-}$ radicals' catalysis/mediation and metallic Li anode protection. This strategy provides new insights for further optimizing electrolyte formulations for LSBs, particularly by combining large cations and high-DN anions to improve battery performance.

Although stabilizing $\text{S}_3^{\bullet-}$ radicals is crucial for improving the performance of LSBs, it needs to be done within a certain limit, *i.e.*, excessive stabilization can cause the radicals to lose their electrochemical activity, thus reducing the utilization of active materials in the battery. To balance the electrochemical activity and stability of $\text{S}_3^{\bullet-}$ radicals in LSBs, Zhao and co-workers proposed adding 0.04 M $\text{Al}(\text{acac})_3$ to a low DOL solvent, forming an $\text{Al}(\text{acac})_3$ complex through the ion-dipole interaction between DOL and Al^{3+} (Fig. 6e) [55]. In this solvated metal complexes, the oxygen atoms in DOL donate electrons to Al^{3+} , weakening the attraction of Al^{3+} to sulfur radicals (*i.e.*, $\text{S}_3^{\bullet-}$ and $\text{S}_4^{\bullet-}$). This allows the radicals to maintain their reductive activity while avoiding over-stabilization that would lead to a loss of activity.

This mechanism ensures the efficient conversion of sulfur radicals in LSBs and other multielectron transfer systems, while maintaining long-term stability and optimizing the performance and utilization of active materials. It is worth mentioning that this research proposed the $\text{S}_4^{\bullet-}$ stability mechanism for the first time, and its UV-Vis absorption peak at 676 nm and Raman shift at 517 cm^{-1} sets it apart from $\text{S}_3^{\bullet-}$ at 617 nm and 535 cm^{-1} , respectively. The work has important reference significance for the development of other free radicals (*viz.* $\text{S}_2^{\bullet-}$, $\text{S}_4^{\bullet-}$, $\text{S}_5^{\bullet-}$, $\text{S}_6^{\bullet-}$, $\text{S}_7^{\bullet-}$, and $\text{S}_8^{\bullet-}$) for their application in LSBs.

In addition to inorganic electrolyte additives, Zhao's team also used 0.1 M fluorenone (FL) as an organic electrolyte additive to stabilize $\text{S}_3^{\bullet-}$ radicals [103]. As shown in Fig. 6f, carbonyl groups of FL can capture and stabilize $\text{S}_3^{\bullet-}$ radicals through its electron-accepting ability, which promotes the three-dimensional deposition of Li_2S , reducing the internal resistance of the battery and improving capacity and cycle stability. In addition, FL can be reduced to $\text{FL}^{\bullet-}$ radical within the operating voltage of LSBs. As an electron transfer medium, it can accelerate the reduction process of $\text{Li}_2\text{S}_4 \rightarrow \text{Li}_2\text{S}$ and the oxidation activation energy of Li_2S , thus improving the kinetic reaction and battery performance.

Similarly, Zhang *et al.* used 0.05 M diphenyl ditelluride (DPDTe) as an organic electrolyte additive to promote the generation of $\text{S}_3^{\bullet-}$ radicals in LSBs [104]. During the discharge process, DPDTe is electrochemically reduced to form the PhTe^{\bullet} free radical, which can undergo rapid radical exchange with Li_2S_6 , generating more electrochemically reactive LiS_3^{\bullet} and LiS_3TePh (Fig. 6g). The former has been extensively discussed for its catalytic mediation role in LSBs, while the latter is further reduced to form LiSTePh and Li_2S_2 . The Li_2S_2 is continuously attacked by PhTe^{\bullet} radicals, generating LiSTePh as an intermediate and ultimately Li_2S as the final product. The formed LiSTePh can be easily lithiated to form Li_2S and regenerate PhTe^{\bullet} radicals, completing a Te-radical-mediated catalytic cycle. This dual-free-radical synergistic effect based on organic telluride electrolyte additives enables LSBs to exhibit impressive cycling stability and rate performance.

4.4 Metal Compound Catalysts

Metal oxide catalysis of sulfur chemical conversion reactions plays an important role in LSBs, particularly oxygen-deficient metal oxides, which exhibits excellent catalytic

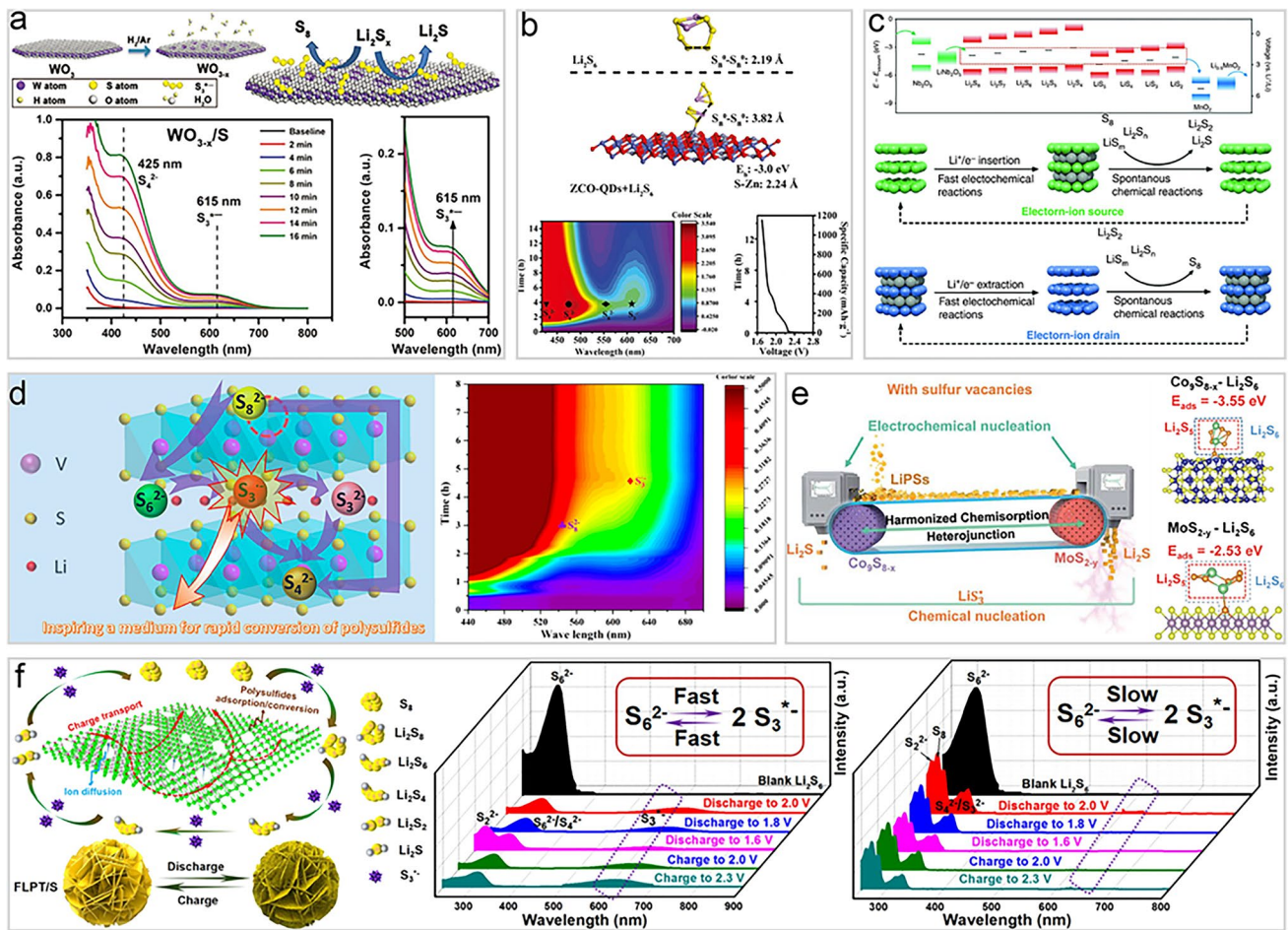


Fig. 7 Metal compound catalysts to promote the generation of S_3^-/LiS_3^+ radicals. **a** The top: schematic of WO_{3-x} and the conversion of Li_2S_x on its surface, the bottom: time dependent UV–Vis spectra of sulfur cathodes with WO_{3-x} [53]. Copyright 2018, John Wiley and Sons. **b** The top: optimized geometric configurations of Li_2S_6 adsorbed on ZCO-QDs, the bottom: contour maps of *in situ* UV–Vis spectra, and the corresponding discharge profiles of ZCO-QDs based cathode [113]. Copyright 2021, John Wiley and Sons. **c** The top: electron transfer with electron-ion reservoirs, the bottom: schematic of reactions coupling to enhance kinetics during operation [81]. Copyright 2020, Springer Nature. **d** The left: schematic of the adsorption-catalytic behavior of polysulfides on VS_{2-x} , the right: *in situ* UV–Vis spectra of the VS_{2-x} based cathodes [105]. Copyright 2022, American Chemical Society. **e** The left: catalytic mechanism of sulfur vacancy heterojunctions, the right: calculated adsorption energies of Li_2S_6 on MoS_{2-y} and Co_9S_{8-x} crystals [106]. Copyright 2022, Royal Society of Chemistry. **f** The left: schematic of the interaction of LiPSs on MXenes, the right: UV–Vis spectra of cathodes at different states [107]. Copyright 2019, American Chemical Society

performance in accelerating polysulfide conversion reactions [112]. Lin et al. prepared oxygen-deficient tungsten oxide (WO_{3-x}) and used it as a sulfur host for LSBs. Oxygen defects reduce the oxidation state of metal oxides and promote the formation and stabilization of highly active S_3^- radicals on their surfaces, which was verified by UV–Vis (Fig. 7a) [53]. Moreover, oxygen deficiency creates more active sites, enhancing the metal oxide surface’s ability to adsorb polysulfides, thereby suppressing the shuttle effect. As a result, WO_{3-x} improved the kinetics of polysulfide

conversion and significantly enhanced the cycle stability and high-rate performance of LSBs.

Oxygen-deficient metal oxides, in addition to their traditional roles in polysulfide adsorption and catalysis, may also stabilize S_3^- radicals. However, the underlying mechanism still requires further exploration. Liu *et al.* found that the $ZnCo_2O_4$ composite oxide can significantly promote the generation of LiS_3^+ radicals (Fig. 7b) [113]. DFT calculations show that the abundant metal active sites on the surface of $ZnCo_2O_4$ adsorb Li_2S_6 through metal–S bonds, reducing its stability and promoting its cleavage into LiS_3^+ radicals. These

radicals can further consume solid sulfur through reactions such as $2\text{S}_3^{\cdot-} + 1/4\text{S}_8 \rightarrow \text{S}_8^{2-}$, generating other reducible polysulfides, thereby improving sulfur utilization and the cycle stability of the battery. Notably, although oxygen vacancies were also detected on the surface of ZnCo_2O_4 , they were not considered to stabilize the $\text{S}_3^{\cdot-}$ radicals. Instead, they enhance the adsorption of Li_2S_4 , thereby promoting the efficiency of its reduction reaction.

In addition to adsorption and catalytic effects, metal oxides can also catalyze sulfur chemical transformations through their electron–ion source and drain functions (Fig. 7c) [81]. Based on molecular orbital theory, Lu *et al.* suggested that the band gap center (BGC, derived from the midpoint of HOMO and LUMO energy levels) of the tetra-coordinated lithium polysulfides (Li_2S_n -4DOL) is higher than that of the bi-coordinated lithium sulfur radicals (LiS_n^{\cdot} -2DOL), indicating that the reduction of LiS_n^{\cdot} -2DOL occurs prior to Li_2S_n -4DOL. Once LiS_n^{\cdot} -2DOL is consumed, Li_2S_n -4DOL spontaneously converts to regenerate LiS_n^{\cdot} -2DOL. Integrating this understanding with band theory, they employed a mixed metal oxide catalyst ($\text{Nb}_2\text{O}_5/\text{MnO}_2$) featuring both high and low BGC values (corresponding to the valence band and conduction band centers) as a cathode catalyst for LSBs. During the discharge process, lithiated LiNb_2O_5 acts as an electron and ion source, providing electrons and Li^+ ions to LiS_n^{\cdot} -2DOL, thereby accelerating the electrochemical reduction of $\text{Li}_2\text{S}_n/\text{LiS}_n^{\cdot}$ to Li_2S . Conversely, during the charging process, delithiated MnO_2 acts as an electron and ion drain, extracting electrons and Li^+ ions from $\text{Li}_2\text{S}_n/\text{LiS}_n^{\cdot}$, thus promoting the oxidation of $\text{Li}_2\text{S}_n/\text{LiS}_n^{\cdot}$ to S_8 . Notably, lithium sulfur radicals (including LiS_3^{\cdot} radicals), which arise from the homolytic or heterolytic cleavage of Li_2S_x , serve as critical intermediates in this catalytic process. Although their concentration may be limited due to chemical equilibrium, the electron–ion source and drain catalyst significantly enhances their role in facilitating these reactions.

Metal sulfide catalysts are also widely applied in LSBs to promote the electrochemical conversion and reaction kinetics of sulfur species [114]. Wang *et al.* applied sulfur-defect-enriched VS_2 nanosheets (VS_{2-x}) as catalysts in LSBs and found that both VS_{2-x} and its lithiation intermediate $\text{Li}_y\text{VS}_{2-x}$ significantly enhance the content of $\text{S}_3^{\cdot-}$ radicals by promoting the dissociation of S_6^{2-} (Fig. 7d) [105]. These free radicals facilitate the ring-opening reaction of cyclo- S_8 , accelerating the conversion of sulfur, reducing the shuttle

effect of polysulfides, and improving the sulfur utilization, cycling stability, and reaction kinetics of the battery.

Notably, the introduction of sulfur vacancies also ensures the long-term stability of the catalyst, allowing it to continuously exert its catalytic effect during the charge and discharge processes. Wei and co-workers also introduced sulfur vacancy catalysts ($\text{Co}_9\text{S}_8/\text{MoS}_2$ heterojunction) in LSBs to promote the content of $\text{S}_3^{\cdot-}$ radicals (Fig. 7e) [106]. In contrast, they proposed that the LiS_3^{\cdot} radicals are not directly dissociated from Li_2S_6 adsorbed on the sulfide, but rather, when Li_2S_6 interacts with sulfur vacancies, one of the sulfur atoms in the Li_2S_6 molecule is asymmetrically adsorbed to the sulfur vacancy, leading to the formation of the Li_2S_5 intermediate, according to theoretical calculations. Li_2S_5 is a relatively unstable and highly reactive species, which can further convert into LiS_3^{\cdot} radicals and other polysulfide species. This study provides new insights into the use of vacancy defects in metal compounds to stabilize $\text{S}_3^{\cdot-}$ radicals in LSBs.

Special metallic compound MXene, as a class of emerging two-dimensional transition metal carbides or nitrides [115], have gradually become important materials in LSBs due to their unique nanostructures, excellent conductivity, and surface chemical properties [116, 117]. MXene materials such as $\text{Ti}_3\text{C}_2\text{T}_x$ have abundant polar surface sites that can effectively adsorb and catalyze the transformation of polysulfides, thereby improving the performance of LSBs [118–120]. Xiao *et al.* prepared a flower-like porous MXene material $\text{Ti}_3\text{C}_2\text{T}_x$ (FLPT) and applied it to the cathode of LSBs (Fig. 7f) [107]. The surface of FLPT is rich in polar groups, such as hydroxyl and oxygen, which, through Lewis acid–base interactions, strongly adsorb polysulfides (especially S_6^{2-}). According to the enhanced characteristic adsorption at 617 nm through in situ UV–Vis spectroscopy, this strong adsorption can effectively promote the generation and stabilization of $\text{S}_3^{\cdot-}$ radicals, thereby accelerating the sulfur species transformation in LSBs. Furthermore, the high conductivity and nanosheet structure of FLPT further promote the rapid charge transfer and efficient ion transport, allowing the $\text{S}_3^{\cdot-}$ radicals to quickly participate in electrochemical reactions on the electrode surface. The three-phase interface effect (FLPT support, sulfur species, and electrolyte) also enhances the enrichment and stabilization of $\text{S}_3^{\cdot-}$ radicals, further facilitating the oxidation–reduction reactions of sulfur. Therefore, FLPT materials not only enhance the stability of $\text{S}_3^{\cdot-}$ radicals

through strong chemical adsorption and rapid charge transfer mechanisms but also optimize the electrochemical performance of LSBs, improving the battery's capacity and cycle stability. This mechanism of stabilizing $S_3^{\bullet-}$ radicals provides new insights into the development and application of MXene materials in LSBs, highlighting their potential in advancing the performance and practical deployment of next-generation energy storage systems.

4.5 Carbon-Based Catalysts

Carbon-based catalysts, with their excellent conductivity, abundant active sites, and tunable surface chemistry, exhibit significant potential in LSBs [121, 122]. Generally, the higher specific surface area and more porous structure of carbon-based catalysts provide more surface and active sites for adsorption and better diffusion of LiPSs [121], including trisulfur radicals. Specially, carbon-based catalysts with heteroatom doping are crucial for anchoring trisulfur radicals and enhancing their stability. Note that the type and quantity of doped heteroatoms have different effects on the degree of graphitization [121], which has a positive correlation with electrical conductivity. For example, nitrogen doping can promote carbon graphitization, while oxygen doping generally does not favor graphitization. Therefore, the balance between electron conductivity and trisulfur radical stability should be considered for heteroatom doping of carbon-based catalysts.

Zhang and co-workers developed a carbon-based catalyst (UN/O-CNS) through heteroatom doping (such as N/O co-doping) [54], which not only effectively anchors LiS_3^{\bullet} radicals (Fig. 8a), but also reduces the energy barriers of SRR and sulfur oxidation reactions (SOR) (Fig. 8b), thereby significantly enhancing the electrochemical conversion process. UV-Vis testing of the supernatant after UN/O-CNS adsorbed Li_2S_6 solution revealed that the peak intensities of polysulfide ions and LiS_3^{\bullet} radicals were significantly reduced compared to the control group (Fig. 8c), indicating that the N/O dual active sites firmly anchor LiS_3^{\bullet} radicals, preventing their aggregation or side reactions due to high reactivity. Post-cycling UN/O-CNS cathode materials displayed additional peaks at $G=2.035$ and $G=2.053$ (Fig. 8d) [54], further confirming the presence of anchored LiS_3^{\bullet} radicals. Compared to models with pure N or O doping, the N/O co-doped model showed significant charge redistribution

(Fig. 8e), with a substantial electron concentration at the triangular bond positions, implying stronger charge transfer and higher binding energy (Fig. 8f). Consequently, under the influence of its high specific surface area, porous structure, and synergistic catalytic active sites, UN/O-CNS achieves efficient capture of LiS_3^{\bullet} radicals, stabilizes the radicals through triangular bonding, effectively suppresses the polysulfide shuttle effect, and enhances the conversion efficiency of sulfur species.

Chen *et al.* discovered that N-doped porous carbon (NPC) can effectively stabilize $S_3^{\bullet-}$ radicals [123]. N-doped carbon-based materials provide active sites such as pyrrolic-N and pyridinic-N. DFT calculations show that $S_3^{\bullet-}$ radicals preferentially interact with these N atoms, exhibiting low adsorption energies of -2.09 and -2.00 eV, respectively (Fig. 8g). ESR tests reveal that in the discharged NPC-S composites, in addition to carbon radicals ($g=2.0023$), distinct signals of $S_3^{\bullet-}$ radicals ($g_y=2.0355$, $g_z=2.0526$) are present (Fig. 8h), further confirming the ability of NPC to capture $S_3^{\bullet-}$ radicals. This adsorption and stabilization mechanism of $S_3^{\bullet-}$ radicals by NPC significantly improves the cycling stability and Coulombic efficiency of LSBs, effectively suppressing the shuttle effect.

To elucidate the mechanism of heteroatom in carbon-based electrocatalyst, Feng *et al.* developed a heteroatom-doped carbon-based electrocatalytic model, using DFT calculations to analyze the impact of heteroatom doping on active sites on the carbon material surface [124]. The study showed that doping heteroatoms can effectively modulate the adsorption and conversion of lithium sulfur radicals LiS_y^{\bullet} ($y=1-3$) and short-chain Li_2S_x ($x=1-4$), significantly improving reaction efficiency (Fig. 8i). Furthermore, by using the adsorption energy of LiS_y^{\bullet} radicals on the catalyst as a key descriptor, the study predicts the reaction pathway, rate-determining step, and overpotential (Fig. 8j). This research provides valuable theoretical insights into the mechanism of heteroatom-doped carbon-based electrocatalysts in promoting the generation of sulfur radicals including trisulfur radicals, contributing to the further optimization of LSBs performance. In recent years, machine learning has begun to be applied in the design of LSB materials. Feeding large amounts of computational data (such as reaction path, transition state, electronic structure changes of the active site) generated by DFT into machine learning models enables rapid prediction and screening of materials with

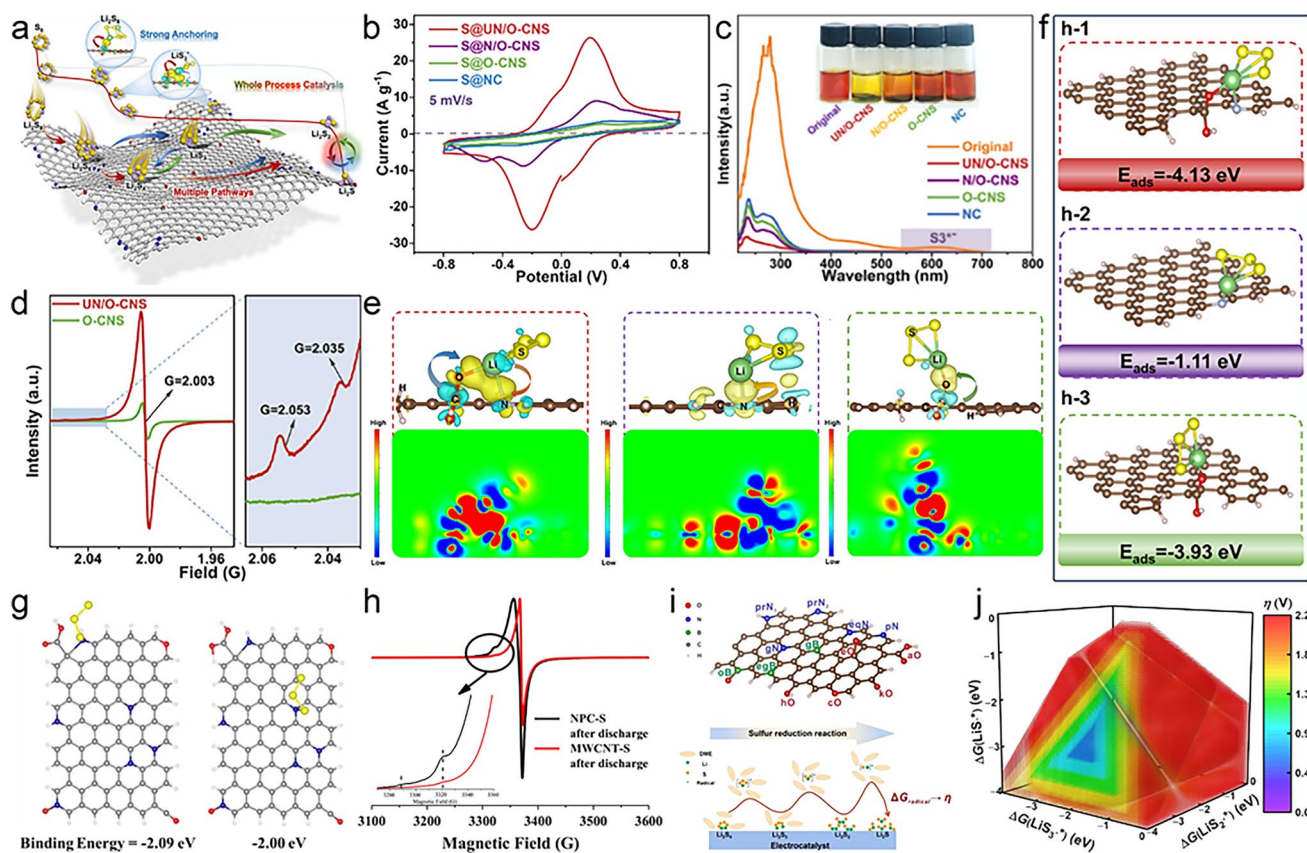


Fig. 8 Carbon-based catalysts to promote the generation of $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals. **a** Schematic of strategies with carbon-based electrocatalyst [54]. Copyright 2024, Elsevier. **b** CV curves of symmetric cells with Li_2S_6 electrolyte [54]. Copyright 2024, Elsevier. **c** UV-Vis spectra of materials after adsorption of Li_2S_6 . Copyright 2024, Elsevier. **d** EPR spectra of carbon-based catalysts after discharge. Copyright 2024, Elsevier. **e** Electron density differences and **f** binding configurations of LiS_3^{\bullet} on the different heteroatom-doped graphene [54]. Copyright 2024, Elsevier. **g** Optimized configurations of $S_3^{\bullet-}$ radicals absorbed on the N-doped carbon [123]. Copyright 2015, Elsevier. **h** ESR spectra of carbon/sulfur composite in the discharged state [123]. Copyright 2015, Elsevier. **i** Schematic of the reaction mechanism on the second discharge platform of LSBs with heteroatom-doped graphene [124]. Copyright 2022, John Wiley and Sons. **j** The overpotential with the adsorption Gibbs free energies of radicals [124]. Copyright 2022, John Wiley and Sons

excellent electrocatalytic properties [125]. This multidisciplinary approach can significantly improve the efficiency of developing materials that promote trisulfur radical generation and reduce the reliance on expensive experimental and computational resources.

Different from the adsorption mechanism of $S_3^{\bullet-}$ radicals, Kumar *et al.* proposed a grafting mechanism. Specifically, activated carbon cloth (ACC) with abundant carbon radicals serve as an effective sulfur host [126]. The dangling bond carbon radicals on ACC can couple with $S_3^{\bullet-}$ radicals, modulating the chemical conversion pathways and reaction kinetics of sulfur cathode. This radical grafting mechanism ultimately enhances the rate performance and cycling stability of sodium-sulfur batteries.

5 Comprehensive Comparison of $S_3^{\bullet-}/LiS_3^{\bullet}$ Radicals Detection and Increment

5.1 Advancing Detection Techniques

In the detection of $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals in LSBs, the selection of spectroscopic techniques needs to consider multiple key factors, including precision, sensitivity, *in situ* operability, *in operando* operability, and photostability, as illustrated in radar map of Fig. 9. Photostability refers to the stability of the process in which polysulfides dissociate to form $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals under light sources, where higher stability indicates less interference from the light source on the radical signal. *In situ* operability involves real-time observation of the

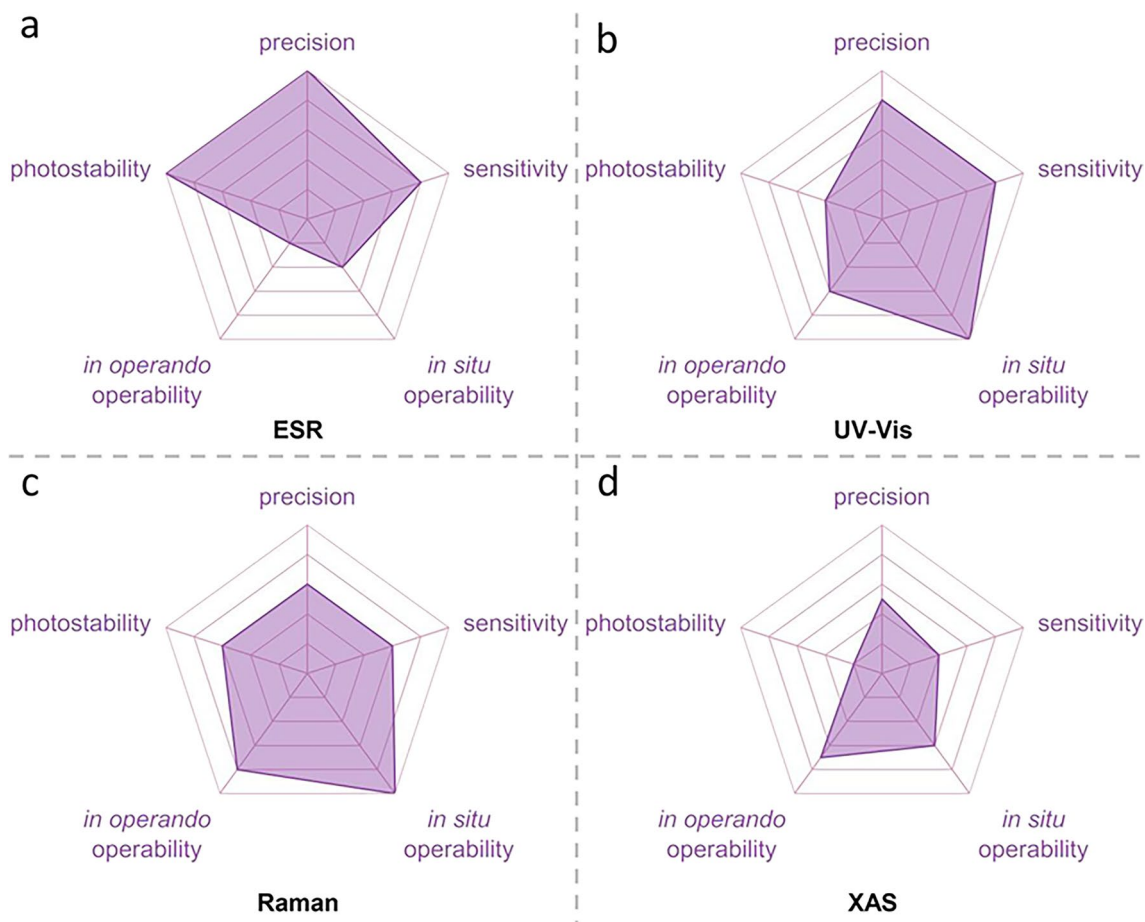


Fig. 9 Radar map of optical testing technique with aspect of precision, photostability, *in operando* and *in situ* operability for **a** ESR, **b** UV-Vis, **c** Raman, and **d** XAS

generation and behavior of $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals under operating conditions, while *in operando* operability further emphasizes the dynamic correlation between $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals behavior and the electrochemical performance of the battery.

ESR stands out due to its high precision and sensitivity, allowing direct detection of low-concentration $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals (Fig. 9a). It also exhibits good photostability, making it well-suited for detailed studies of the generation and transformation mechanisms of radicals. However, its demanding experimental conditions, complex equipment, and high costs limit its widespread application. In contrast, UV-Vis and Raman spectroscopy offer significant advantages in *in situ* and real-time *in operando* monitoring (Fig. 9b, c), with convenient operation that makes them suitable for studying the dynamic generation and transformation of $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals. However, these techniques have relatively lower sensitivity and light

stability, with light sources potentially interfering with the radicals' signal, making them more suitable as complementary methods. XAS is well-suited for exploring the electronic structure and chemical environment of $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals, with its high photon energy providing atomic-level resolution. However, its precision and sensitivity are moderate, and the high photon energy can lead to the decomposition of radicals or polysulfides, resulting in poor photostability and high equipment costs (Fig. 9d).

With the development of advanced spectroscopic techniques, many emerging methods have shown potential for $S_3^{\bullet-}/LiS_3^{\bullet}$ radical research. For instance, time-resolved spectroscopy can dynamically capture the temporal processes of radical generation and transformation, providing key insights into radical reaction kinetics. Furthermore, two-photon spectroscopy and ultrafast laser spectroscopy offer high-resolution and short timescale

observation, enabling the capture of short-lived radical states and providing new tools for elucidating their transient behaviors. Synchrotron X-ray spectroscopy further enhances the resolution and sensitivity of XAS, and when combined with in situ electrochemical cells, it allows for an in-depth correlation between S_3^-/LiS_3^* radicals behaviors and electrochemical processes. Integrating multiple spectroscopic techniques, such as the combination of Raman with ESR or XAS with UV–Vis, can offer a more comprehensive analysis of the generation mechanisms and stabilization processes of S_3^-/LiS_3^* radicals.

Each spectroscopic technique has unique characteristics in S_3^-/LiS_3^* radical detection, and their selection requires a trade-off between precision, sensitivity, photostability, and operability depending on experimental needs. Moreover, with the introduction of advanced spectroscopic techniques and the realization of multitechnology synergies, future research is expected to systematically unravel the generation and stabilization mechanisms of S_3^-/LiS_3^* radicals and their relationships with LSBs performance, providing crucial insights for LSB design and optimization.

5.2 Constructing Catalytic System with High Content of S_3^-/LiS_3^* Radicals

Different electrolyte strategies exhibit unique advantages and limitations in stabilizing S_3^-/LiS_3^* radicals and improving the performance of LSBs (Fig. 10). High-DN solvents, with their strong coordination capabilities, significantly promote the dissociation of polysulfides and stabilize S_3^-/LiS_3^* radicals (Fig. 10a), thereby enhancing sulfur utilization. However, their strong reactivity with metallic Li leads to poor compatibility with metallic Li anodes and reduced cycling stability. These issues can be mitigated through rational solvent selection, such as combining high-DN solvents with low-DN solvents to reduce corrosive interactions. Additionally, functional additives (such as $LiNO_3$) can strengthen the SEI and enhance metallic Li protection.

Co-solvents, a mixture of high-DN solvents and traditional ether-based solvents, demonstrate a balanced performance, with good ionic conductivity, improved lithium compatibility, and enhanced cycling stability (Fig. 10b). However, the dilution of high-DN solvents reduces their ability to stabilize S_3^-/LiS_3^* radicals. To optimize this

strategy, adjusting the solvent ratio and viscosity can enhance polysulfide solubility and diffusion efficiency while further suppressing the shuttle effect. Furthermore, selecting low-viscosity ether solvents can improve ion transport, thereby boosting overall battery performance.

High-DN solvent additives, involving the addition of a small amount of high-DN solvents, effectively stabilize S_3^-/LiS_3^* radicals while avoiding severe lithium corrosion associated with higher concentrations (Fig. 10c). This approach offers an excellent balance between stability and efficiency, making it suitable for applications requiring well-rounded performance. Additionally, combining this strategy with other functional additives, such as high-DN anions or lithium salts, can further enhance the comprehensive performance of the electrolyte.

High-DN anions supporting electrolyte or electrolyte additives, achieved by introducing salts with high-DN anions (such as Br^-) in traditional ether-based electrolytes, provide excellent metallic Li protection, significantly improving cycling stability and sulfur utilization (Fig. 10d). However, their direct stabilization effect on S_3^-/LiS_3^* radicals is relatively limited. To address this, optimizing the solvent-salt ratio can improve ionic conductivity and polysulfide conversion efficiency. Furthermore, leveraging machine learning (ML) and computational simulations can accelerate the development of more effective additive combinations tailored for specific applications.

Therefore, integrating electrolyte engineering design into the discussion of electrolyte strategies offers a more systematic framework for optimizing LSBs performance. By combining solvent selection, additive design, viscosity control, and advanced computational techniques, it is possible to balance trisulfur radical stabilization, sulfur conversion efficiency, and cycling stability while advancing the technological breakthrough and practical implementation of LSBs.

Carbon-based catalysts, metal compound catalysts, MXenes, and sodium or β -cage zeolites serve as heterogeneous electrocatalysts with distinct advantages and limitations in stabilizing S_3^-/LiS_3^* radicals and improving LSBs, as shown in Fig. 11. Carbon-based catalysts are notable for their excellent electronic conductivity and cycling stability but exhibit limited ability to stabilize S_3^-/LiS_3^* radicals and adsorb LiPSs which hinders their effectiveness in suppressing the shuttle effect (Fig. 11a). To address this, surface modification with heteroatoms (such as nitrogen, sulfur, or metal atoms) can significantly enhance their chemical

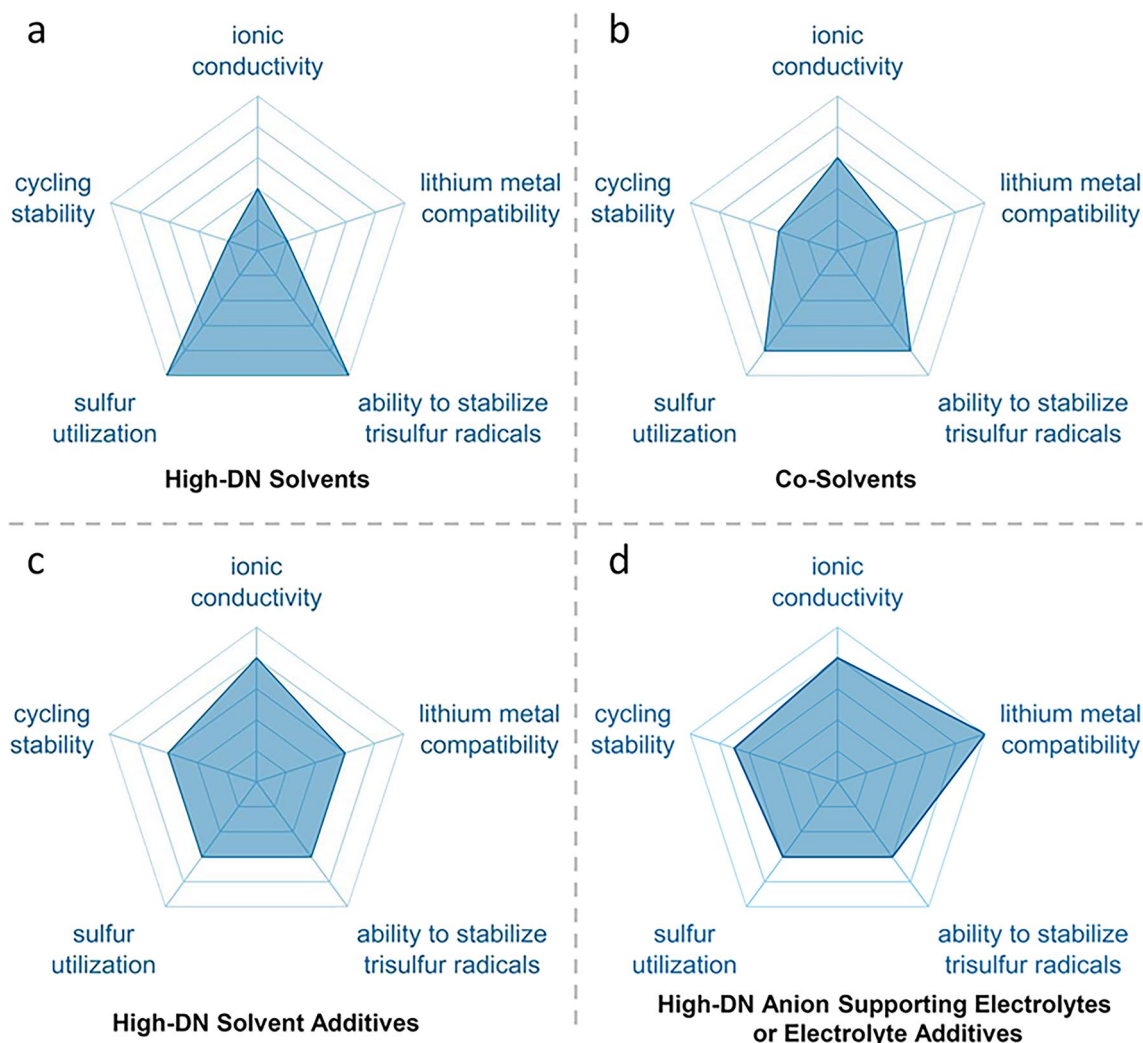


Fig. 10 Radar map of homogeneous electrocatalyst strategy with aspect of ionic conductivity, cycling stability, sulfur utilization, ability to stabilize S_3^-/LiS_3^+ radicals and lithium metal compatibility for **a** high-DN solvents, **b** co-solvents, **c** high-DN solvent additives, and **d** high-DN anion supporting electrolytes or electrolyte additives

adsorption capability and catalytic activity for polysulfides, thereby improving overall performance. Metal compound catalysts, on the other hand, demonstrate outstanding adsorption ability for LiPSs (Fig. 11b), effectively suppressing the shuttle effect and enhancing cycling stability and sulfur utilization. However, their relatively low electronic and ionic conductivity poses a limitation. This can be mitigated by coupling metal compounds with highly conductive carbon materials or introducing defect structures (such as oxygen or sulfur vacancies) to improve their conductivity and catalytic activity.

MXenes, with their high electronic and moderate ionic conductivity, achieve a good balance in suppressing the shuttle effect and enhancing reaction kinetics, though their

ability to stabilize S_3^-/LiS_3^+ radicals is slightly inferior (Fig. 11c). Surface modification or intercalation engineering (such as introducing functional groups or metal cations) could further enhance their ability to capture and stabilize S_3^-/LiS_3^+ radicals. Sodium or β -cage zeolites excel in ionic conductivity, strong LiPSs adsorption, and S_3^-/LiS_3^+ radicals' stabilization (Fig. 11d), making them a promising candidate for suppressing the shuttle effect and promoting sulfur conversion. However, their limited electronic conductivity restricts their application under high-rate charge/discharge conditions. This can be addressed by compositing zeolites with conductive materials or tuning their pore structures (such as introducing metal ions to enhance electron transfer).

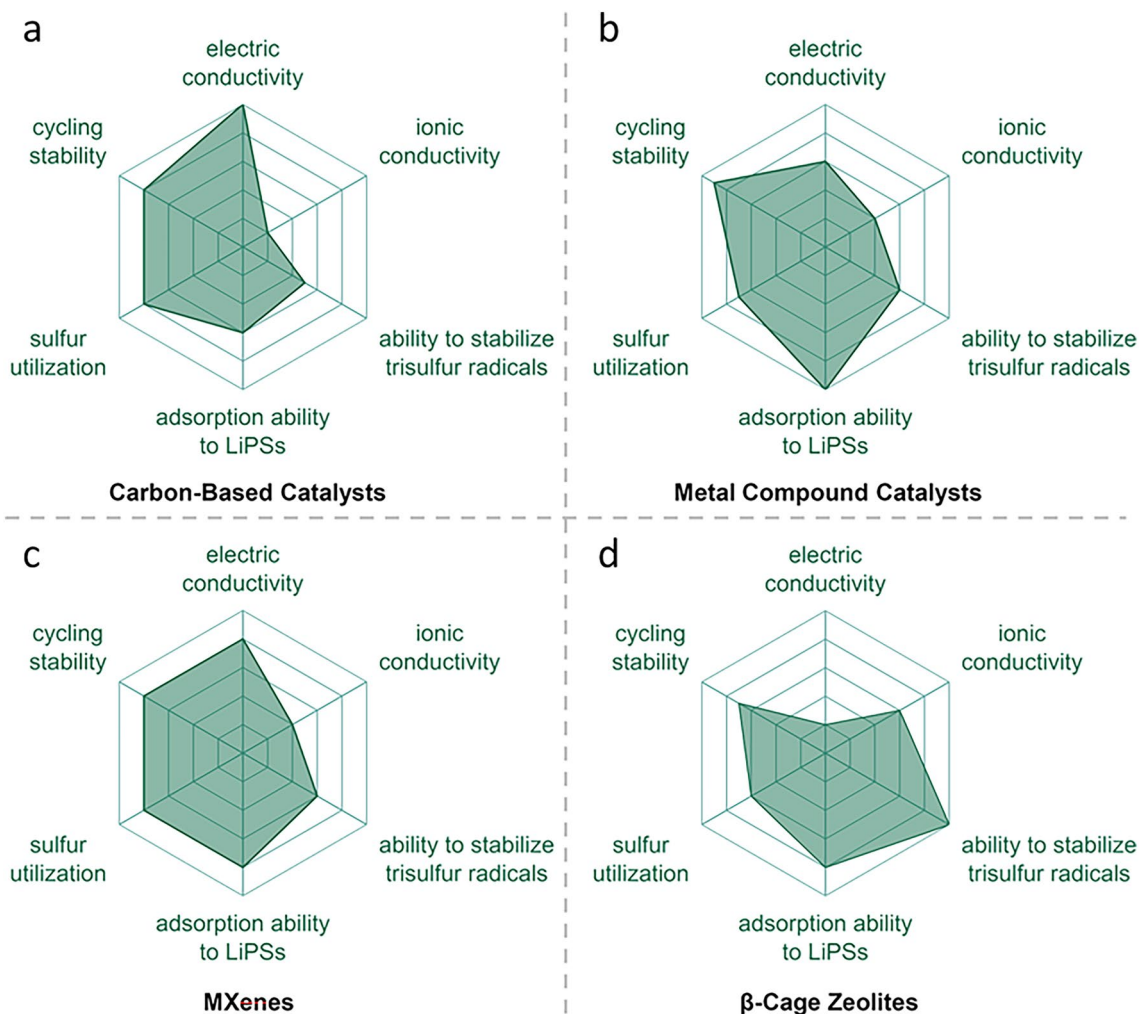


Fig. 11 Radar map of heterogeneous electrocatalyst strategy with aspect of ionic conductivity, cycling stability, sulfur utilization, ability to stabilize $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals and metallic Li compatibility for **a** carbon-based catalysts, **b** metal compound catalysts, **c** MXenes, and **d** β -cage zeolites

Thus, optimizing different catalysts through strategies such as surface modification, introducing heterogeneous structures, and functional design is essential to balance their performance in stabilizing $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals, adsorbing polysulfides, and enhancing overall electrochemical performance. These modifications and combination strategies provide crucial directions for the development of more efficient and stable catalysts for LSBs.

6 Summaries and Perspectives

$S_3^{\bullet-}/LiS_3^{\bullet}$ radicals serve as essential intermediates in LSBs, facilitating sulfur conversion reactions, regulating Li_2S deposition, and mitigating key challenges such as shuttle

effects and electrode passivation. Below, we summarize several key issues discussed in this review that are central to the development of sulfur radicals in LSBs.

- i. Theoretical calculations play an increasingly important role in understanding the formation and transformation mechanisms of $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals. First-principles calculations provide molecular-level insights into the reactions of $S_3^{\bullet-}/LiS_3^{\bullet}$ radicals within the electrodes and electrolytes, guiding the design of electrolyte and catalyst materials to enhance battery performance. By integrating AIMD and classical dynamics simulations based on reactive force field (ReaxFF), the influence of electrolytes and catalysts on the stability of $S_3^{\bullet-}$

LiS_3^* radicals and polysulfide shuttle behavior can be systematically studied, offering valuable theoretical support for electrolyte optimization. Additionally, theoretical calculations provide essential guidance for spectroscopic analysis, revealing the impact of electrolyte systems on the spectroscopic features of $\text{S}_3^-/\text{LiS}_3^*$ radicals, thereby facilitating the precise design of electrolytes and catalysts to improve the stability and energy efficiency of LSBs.

- ii. The generation and catalytic behavior of $\text{S}_3^-/\text{LiS}_3^*$ radicals can be effectively monitored using advanced *in situ* characterization techniques, such as ESR, UV–Vis, Raman spectroscopy, and synchrotron XAS. These techniques offer critical insights into the stability, kinetics/dynamics, and electronic structure of $\text{S}_3^-/\text{LiS}_3^*$ radicals. However, challenges persist due to radicals' transient nature, low concentration in traditional ether-based electrolytes, and the photosensitivity of their formation reactions. The introduction of radical trapping agents, such as nitrones and pyridinium cations, has significantly enhanced the stability of $\text{S}_3^-/\text{LiS}_3^*$ radicals and amplified spectral signals, enabling clearer elucidation of their generation and transformation. Integrating complementary techniques, such as ESR with XAS or UV–Vis with Raman spectroscopy, provides a more comprehensive understanding of radicals behavior and reaction pathways. Combining radical traps with UV–Vis and Raman improves signal detection, while synchrotron-based XAS reveals molecular-level interactions. Looking ahead, time-resolved spectroscopy and ultrafast laser techniques can capture radicals' rapid dynamics, and novel trapping agents and, alongside advanced material designs, will optimize radical-mediated catalysis and accelerate the development of high-energy–density, long-cycle-life LSBs.
- iii. β -cage zeolites, a type of lapis lazuli analog containing trisulfur radicals, hold great potential as sulfur hosts for improving the performance of LSBs. These materials can be easily synthesized through reactions with sulfur and possess unique structural characteristics, enabling dual functionalities: mediating the stabilization of $\text{S}_3^-/\text{LiS}_3^*$ radicals and providing catalytic adsorption capabilities. Surface engineering through heteroatom doping or defect introduction can significantly enhance their adsorption and catalytic performance toward LiPSs. Integrating β -cage zeolites with conductive networks like graphene or carbon nanotubes effectively overcomes their limited electronic conductivity. Meanwhile, exploring other zeolite structures, such as SAPO molecular sieves and other artificial zeolites, offers new opportunities for $\text{S}_3^-/\text{LiS}_3^*$ radicals assembly. SAPO zeolites, with their tunable acidic sites and adjustable framework structures, show promise in precisely controlling the pathways and stability of $\text{S}_3^-/\text{LiS}_3^*$ radical formation, potentially enhancing catalytic activity and polysulfide adsorption. By introducing diverse framework structures and functionalized designs, these artificial zeolites can broaden their applications in energy storage, becoming strong candidates for next-generation high-efficiency sulfur hosts.
- iv. Although high-DN solvents are considered effective media for stabilizing $\text{S}_3^-/\text{LiS}_3^*$ radicals, their high reactivity and viscosity can lead to metallic Li anode corrosion and excessive electrolyte consumption, limiting the cycle life and stability of the battery. High-DN anion-supported electrolytes or organic/inorganic additives, which stabilize $\text{S}_3^-/\text{LiS}_3^*$ radicals through the synergistic effects of anions and cations, present a promising solution. However, the chemical and electrochemical stability of these additives must be carefully addressed to prevent long-term performance degradation. Future research should focus on designing anion-cation pairs that balance radical stabilization with metallic Li anode compatibility, leveraging computational chemistry to identify low-reactivity high-DN anions. Additionally, integrating high-DN additives into hybrid or solid-state electrolytes, combined with dynamic protective interfaces or artificial SEI layers, can effectively mitigate lithium corrosion. Advanced *in situ* and *in operando* characterization techniques will also be essential to reveal the interactions between high-DN components, polysulfides, and lithium surfaces in real time.
- v. Metal compounds with vacancies or defects, heteroatom-doped carbon materials, MXenes, and other novel solid-state catalyst as heterogeneous catalysts, not only possess traditional adsorption catalysis functions but also exhibit significant capabilities in promoting the generation of $\text{S}_3^-/\text{LiS}_3^*$ radicals. These materials show immense potential in enhancing sulfur utilization,



suppressing the shuttle effect, and accelerating reaction kinetics in LSBs. However, their exact mechanisms in stabilizing S_3^-/LiS_3^* radicals and increasing their concentration remain unclear, requiring further investigation. Future research could employ advanced characterization techniques to explore their dynamic behavior during S_3^-/LiS_3^* radical generation and transformation. Additionally, rational design strategies, including the introduction of vacancy defects, heteroatom doping, and integration with conductive materials, could further optimize catalyst performance. Evaluating their long-term stability under practical conditions, such as high sulfur loading and low electrolyte content, is essential to address issues like structural degradation and active site passivation.

- vi. Machine learning (ML) technology has shown tremendous potential in material design and battery performance optimization. By constructing appropriate descriptors and performing high-throughput calculations, ML can efficiently identify key factors influencing the stability and formation of S_3^-/LiS_3^* radicals and reveal beneficial cathode materials, catalytic environments, and electrolyte characteristics that promote S_3^-/LiS_3^* radical formation. Notably, advancements in constructing potential energy surfaces have significantly reduced the cost of traditional computational methods, extending the scale of simulations from the microscopic to the macroscopic level, providing additional insights into the role of S_3^-/LiS_3^* radicals in battery performance. Furthermore, deep learning models can delve into the complex relationships between material properties and S_3^-/LiS_3^* radicals stability by mining multidimensional data, offering theoretical guidance for designing efficient electrolytes and catalysts.

In summary, future research should focus on optimizing the design of high-DN solvents and additives, as well as the development of advanced heterogeneous catalysts that can effectively stabilize and promote trisulfur radicals without compromising the stability of the metallic Li anode. Furthermore, integrating computational models and ML with advanced characterization techniques will be essential for elucidating the formation mechanisms and catalytic roles of S_3^-/LiS_3^* radicals. These efforts will pave the way for designing more efficient and durable LSBs and potentially other metal – sulfur batteries, offering

enhanced energy density, rate capability, longer cycle life, and improved overall performance.

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Declarations

Conflict of Interest The authors declare no interest conflict. They have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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References

1. G. Zhou, H. Chen, Y. Cui, Formulating energy density for designing practical lithium–sulfur batteries. *Nat. Energy* **7**(4), 312–319 (2022). <https://doi.org/10.1038/s41560-022-01001-0>
2. Y. Chen, T. Wang, H. Tian, D. Su, Q. Zhang et al., Advances in lithium-sulfur batteries: from academic research to commercial viability. *Adv. Mater.* **33**(29), e2003666 (2021). <https://doi.org/10.1002/adma.202003666>
3. Y. Guo, Q. Niu, F. Pei, Q. Wang, Y. Zhang et al., Interface engineering toward stable lithium–sulfur batteries. *Energy Environ. Sci.* **17**(4), 1330–1367 (2024). <https://doi.org/10.1039/d3ee04183b>
4. T. Wang, J. He, Z. Zhu, X.B. Cheng, J. Zhu et al., Heterostructures regulating lithium polysulfides for advanced

- lithium-sulfur batteries. *Adv. Mater.* **35**(47), 2303520 (2023). <https://doi.org/10.1002/adma.202303520>
5. M. Winter, J.O. Besenhard, M.E. Spahr, P. Novák, Insertion electrode materials for rechargeable lithium batteries. *Adv. Mater.* **10**(10), 725–763 (1998). [https://doi.org/10.1002/\(SICI\)1521-4095\(199807\)10:10%3c725::AID-ADMA725%3e3.0.CO;2-Z](https://doi.org/10.1002/(SICI)1521-4095(199807)10:10%3c725::AID-ADMA725%3e3.0.CO;2-Z)
 6. H. Irfan, A.M. Shanmugharaj, Organosilane based artificial solid electrolyte interface layer for stable metallic lithium anode. *Appl. Surf. Sci.* **586**, 152806 (2022). <https://doi.org/10.1016/j.apsusc.2022.152806>
 7. S. Zhang, S.E. Saji, Z. Yin, H. Zhang, Y. Du et al., Rare-earth incorporated alloy catalysts: synthesis, properties, and applications. *Adv. Mater.* **33**(16), e2005988 (2021). <https://doi.org/10.1002/adma.202005988>
 8. M. Zhao, H.-J. Peng, B.-Q. Li, J.-Q. Huang, Kinetic promoters for sulfur cathodes in lithium-sulfur batteries. *Acc. Chem. Res.* **57**, 545–557 (2024). <https://doi.org/10.1021/acs.accounts.3c00698>
 9. J.W. Choi, D. Aurbach, Promise and reality of post-lithium-ion batteries with high energy densities. *Nat. Rev. Mater.* **1**(4), 16013 (2016). <https://doi.org/10.1038/natrevmats.2016.13>
 10. J. Liu, M. Xue, Y. Zhou, Z.-Y. Wang, B. Zhang et al., Capturing polysulfides with a functional anhydride compound for lithium-sulfur batteries. *ACS Appl. Energy Mater.* **5**(6), 7719–7727 (2022). <https://doi.org/10.1021/acsaem.2c01185>
 11. J. Wu, B. Zhang, S. Liu, Y. Song, S. Ye et al., Grafting polysulfides into a functional N-halo compound for high-performance lithium-sulfur battery. *Sci. China Mater.* **63**(10), 2002–2012 (2020). <https://doi.org/10.1007/s40843-020-1345-3>
 12. Y.-W. Xu, B.-H. Zhang, G.-R. Li, S. Liu, X.-P. Gao, Covalently bonded sulfur anchored with thiol-modified carbon nanotube as a cathode material for lithium-sulfur batteries. *ACS Appl. Energy Mater.* **3**(1), 487–494 (2020). <https://doi.org/10.1021/acsaem.9b01761>
 13. J. He, Y. Chen, A. Manthiram, Vertical Co₉S₈ hollow nanowall arrays grown on a Celgard separator as a multifunctional polysulfide barrier for high-performance Li-S batteries. *Energy Environ. Sci.* **11**(9), 2560–2568 (2018). <https://doi.org/10.1039/C8EE00893K>
 14. Y. Pang, J. Wei, Y.G. Wang, Y.Y. Xia, Synergetic protective effect of the ultralight MWCNTs/NCQDs modified separator for highly stable lithium-sulfur batteries. *Adv. Energy Mater.* **8**(10), 1702288 (2018). <https://doi.org/10.1002/aenm.201702288>
 15. Y.Y. Wang, J.K. Gu, B.H. Zhang, G.R. Li, S. Liu et al., Specific adsorption reinforced interface enabling stable lithium metal electrode. *Adv. Funct. Mater.* **32**(18), 2112005 (2022). <https://doi.org/10.1002/adfm.202112005>
 16. J. Wu, B. Zhang, J. Liu, S. Liu, T. Yan et al., Grafting and depositing lithium polysulfides on cathodes for cycling stability of lithium-sulfur batteries. *ACS Appl. Mater. Interfaces* **13**(34), 40685–40694 (2021). <https://doi.org/10.1021/acsaami.1c11904>
 17. M. Zhao, H.J. Peng, J.Y. Wei, J.Q. Huang, B.Q. Li et al., Dictating high-capacity lithium-sulfur batteries through redox-mediated lithium sulfide growth. *Small Meth.* **4**(6), 1900344 (2020). <https://doi.org/10.1002/smt.201900344>
 18. G. Li, Y. Gao, X. He, Q. Huang, S. Chen et al., Organosulfide-plasticized solid-electrolyte interphase layer enables stable lithium metal anodes for long-cycle lithium-sulfur batteries. *Nat. Commun.* **8**(1), 850 (2017). <https://doi.org/10.1038/s41467-017-00974-x>
 19. X. Liang, Q. Pang, I.R. Kochetkov, M.S. Sempere, H. Huang et al., A facile surface chemistry route to a stabilized lithium metal anode. *Nat. Energy* **2**, 17119 (2017). <https://doi.org/10.1038/nenergy.2017.119>
 20. H. Li, R. Meng, C. Ye, A. Tadich, W. Hua et al., Developing high-power LiSS batteries via transition metal/carbon nanocomposite electrocatalyst engineering. *Nat. Nanotechnol.* **19**(6), 792–799 (2024). <https://doi.org/10.1038/s41565-024-01614-4>
 21. F. Shi, L. Zhai, Q. Liu, J. Yu, S.P. Lau et al., Emerging catalytic materials for practical lithium-sulfur batteries. *J. Energy Chem.* **76**, 127–145 (2023). <https://doi.org/10.1016/j.jechem.2022.08.027>
 22. J. Li, L. Gao, F. Pan, C. Gong, L. Sun et al., Engineering strategies for suppressing the shuttle effect in lithium-sulfur batteries. *Nano-Micro Lett.* **16**, 12 (2024). <https://doi.org/10.1007/s40820-023-01223-1>
 23. H. Yuan, H.J. Peng, B.Q. Li, J. Xie, L. Kong et al., Conductive and catalytic triple-phase interfaces enabling uniform nucleation in high-rate lithium-sulfur batteries. *Adv. Energy Mater.* **9**(1), 1802768 (2019). <https://doi.org/10.1002/aenm.201802768>
 24. W. Hua, H. Li, C. Pei, J. Xia, Y. Sun et al., Selective catalysis remedies polysulfide shuttling in lithium-sulfur batteries. *Adv. Mater.* **33**(38), e2101006 (2021). <https://doi.org/10.1002/adma.202101006>
 25. P. Wang, T. Xu, B. Xi, J. Yuan, N. Song et al., A Zn₈ double-cavity metallacalix[8]arene as molecular sieve to realize self-cleaning intramolecular tandem transformation of Li-S chemistry. *Adv. Mater.* **34**(51), e2207689 (2022). <https://doi.org/10.1002/adma.202207689>
 26. L.F. Nazar, M. Cuisinier, Q. Pang, Lithium-sulfur batteries. *MRS Bull.* **39**(5), 436–442 (2014). <https://doi.org/10.1557/mrs.2014.86>
 27. M. Barghamadi, A.S. Best, A.I. Bhatt, A.F. Hollenkamp, M. Musameh et al., Lithium-sulfur batteries: the solution is in the electrolyte, but is the electrolyte a solution? *Energy Environ. Sci.* **7**(12), 3902–3920 (2014). <https://doi.org/10.1039/C4EE02192D>
 28. A. Bhargava, J. He, A. Gupta, A. Manthiram, Lithium-sulfur batteries: attaining the critical metrics. *Joule* **4**(2), 285–291 (2020). <https://doi.org/10.1016/j.joule.2020.01.001>
 29. Y. Cao, M. Li, J. Lu, J. Liu, K. Amine, Bridging the academic and industrial metrics for next-generation practical batteries. *Nat. Nanotechnol.* **14**, 200–207 (2019). <https://doi.org/10.1038/s41565-019-0371-8>
 30. Y.-T. Liu, S. Liu, G.-R. Li, X.-P. Gao, Strategy of enhancing the volumetric energy density for lithium-sulfur batteries.



- Adv. Mater. **33**(8), e2003955 (2021). <https://doi.org/10.1002/adma.202003955>
31. F.Y. Fan, Y.-M. Chiang, Electrodeposition kinetics in Li-S batteries: effects of low electrolyte/sulfur ratios and deposition surface composition. *J. Electrochem. Soc.* **164**(4), A917–A922 (2017). <https://doi.org/10.1149/2.0051706jes>
32. H. Shin, M. Baek, A. Gupta, K. Char, A. Manthiram et al., Recent progress in high donor electrolytes for lithium–sulfur batteries. *Adv. Energy Mater.* **10**(27), 2001456 (2020). <https://doi.org/10.1002/aenm.202001456>
33. W. Chen, T. Lei, C. Wu, M. Deng, C. Gong et al., Designing safe electrolyte systems for a high-stability lithium–sulfur battery. *Adv. Energy Mater.* **8**(10), 1702348 (2018). <https://doi.org/10.1002/aenm.201702348>
34. H. Pan, X. Wei, W.A. Henderson, Y. Shao, J. Chen et al., On the way toward understanding solution chemistry of lithium polysulfides for high energy Li–S redox flow batteries. *Adv. Energy Mater.* **5**(16), 1500113 (2015). <https://doi.org/10.1002/aenm.201500113>
35. Q. Zou, Y.-C. Lu, Solvent-dictated lithium sulfur redox reactions: an operando UV-vis spectroscopic study. *J. Phys. Chem. Lett.* **7**(8), 1518–1525 (2016). <https://doi.org/10.1021/acs.jpcclett.6b00228>
36. V. Gutmann, Solvent effects on the reactivities of organometallic compounds. *Coord. Chem. Rev.* **18**(2), 225–255 (1976). [https://doi.org/10.1016/S0010-8545\(00\)82045-7](https://doi.org/10.1016/S0010-8545(00)82045-7)
37. Q. He, Y. Gorlin, M.U.M. Patel, H.A. Gasteiger, Y.-C. Lu, Unraveling the correlation between solvent properties and sulfur redox behavior in lithium-sulfur batteries. *J. Electrochem. Soc.* **165**(16), A4027–A4033 (2018). <https://doi.org/10.1149/2.0991816jes>
38. T. Chivers, Ubiquitous trisulphur radical ion $S_3^{\cdot-}$. *Nature* **252**(5478), 32–33 (1974). <https://doi.org/10.1038/252032a0>
39. R. Linguerri, N. Komih, J. Fabian, P. Rosmus, Electronic states of the ultramarine chromophore $S_3^{\cdot-}$. *Z. Für Phys. Chem.* **222**(1), 163–176 (2008). <https://doi.org/10.1524/zpch.2008.222.1.163>
40. T. Chivers, P.J.W. Elder, Ubiquitous trisulfur radical anion: fundamentals and applications in materials science, electrochemistry, analytical chemistry and geochemistry. *Chem. Soc. Rev.* **42**(14), 5996–6005 (2013). <https://doi.org/10.1039/c3cs60119f>
41. R. Steudel, T. Chivers, The role of polysulfide dianions and radical anions in the chemical, physical and biological sciences, including sulfur-based batteries. *Chem. Soc. Rev.* **48**(12), 3279–3319 (2019). <https://doi.org/10.1039/c8cs00826d>
42. A. Gupta, A. Bhargav, A. Manthiram, Highly solvating electrolytes for lithium-sulfur batteries. *Adv. Energy Mater.* **9**(6), 1803096 (2019). <https://doi.org/10.1002/aenm.201803096>
43. G. Zhang, H.-J. Peng, C.-Z. Zhao, X. Chen, L.-D. Zhao et al., The radical pathway based on a lithium-metal-compatible high-dielectric electrolyte for lithium-sulfur batteries. *Angew. Chem. Int. Ed.* **57**(51), 16732–16736 (2018). <https://doi.org/10.1002/anie.201810132>
44. C. Barchasz, F. Molton, C. Duboc, J.-C. Leprêtre, S. Patoux et al., Lithium/sulfur cell discharge mechanism: an original approach for intermediate species identification. *Anal. Chem.* **84**(9), 3973–3980 (2012). <https://doi.org/10.1021/ac2032244>
45. M. Cuisinier, C. Hart, M. Balasubramanian, A. Garsuch, L.F. Nazar, Radical or not radical: revisiting lithium–sulfur electrochemistry in nonaqueous electrolytes. *Adv. Energy Mater.* **5**(16), 1401801 (2015). <https://doi.org/10.1002/aenm.201401801>
46. B. Zhang, J. Wu, J. Gu, S. Li, T. Yan et al., The fundamental understanding of lithium polysulfides in ether-based electrolyte for lithium–sulfur batteries. *ACS Energy Lett.* **6**(2), 537–546 (2021). <https://doi.org/10.1021/acsenenergylett.0c02527>
47. R. Bouchal, C. Pechberty, A. Boulaoued, N. Lindahl, P. Johansson, Tri-sulfur radical trapping in lithium–sulfur batteries. *J. Power Sources Adv.* **28**, 100153 (2024). <https://doi.org/10.1016/j.powera.2024.100153>
48. Q. Wang, J. Zheng, E. Walter, H. Pan, D. Lv et al., Direct observation of sulfur radicals as reaction media in lithium–sulfur batteries. *J. Electrochem. Soc.* **162**, A474–A478 (2015). <https://doi.org/10.1149/2.0851503jes>
49. R. Dou, Q. Wang, X. Ren, L. Lu, In-situ UV-vis spectroscopy of trisulfur radicals in lithium-sulfur batteries. *Chem. Res. Chin. Univ.* **40**(2), 279–286 (2024). <https://doi.org/10.1007/s40242-024-4027-3>
50. R. Liu, Z. Wei, L. Peng, L. Zhang, A. Zohar et al., Establishing reaction networks in the 16-electron sulfur reduction reaction. *Nature* **626**(7997), 98–104 (2024). <https://doi.org/10.1038/s41586-023-06918-4>
51. W. Zhang, H. Pan, N. Han, S. Feng, X. Zhang et al., Balancing adsorption, catalysis, and desorption in cathode catalyst for Li–S batteries. *Adv. Energy Mater.* **13**(43), 2301551 (2023). <https://doi.org/10.1002/aenm.202301551>
52. M. Zhang, W. Chen, L. Xue, Y. Jiao, T. Lei et al., Adsorption-catalysis design in the lithium-sulfur battery. *Adv. Energy Mater.* **10**(2), 1903008 (2020). <https://doi.org/10.1002/aenm.201903008>
53. H. Lin, S. Zhang, T. Zhang, H. Ye, Q. Yao et al., Elucidating the catalytic activity of oxygen deficiency in the polysulfide conversion reactions of lithium–sulfur batteries. *Adv. Energy Mater.* **8**(30), 1801868 (2018). <https://doi.org/10.1002/aenm.201801868>
54. Y. Cui, W. Fang, J. Zhang, J. Li, H. Wu et al., Controllable sulfur redox multi-pathway reactions regulated by metal-free electrocatalysts anchored with LiS_3^* radicals. *Nano Energy* **122**, 109343 (2024). <https://doi.org/10.1016/j.nanoen.2024.109343>
55. X. Song, C. Wang, Z. Shen, K. Guo, J. Wu et al., Solvated metal complexes for balancing stability and activity of sulfur free radicals. *eScience* **4**(4), 100225 (2024). <https://doi.org/10.1016/j.esci.2023.100225>
56. R. Meng, X. He, S.J.H. Ong, C. Cui, S. Song et al., A radical pathway and stabilized Li anode enabled by halide quaternary ammonium electrolyte additives for lithium-sulfur batteries. *Angew. Chem. Int. Ed.* **62**(38), e202309046 (2023). <https://doi.org/10.1002/anie.202309046>

57. S. Kowalak, A. Jankowska, *Chapter 22 - Inorganic Sulphur Pigments Based on Nanoporous Materials*, 1st edn. (Elsevier, Amsterdam, 2009), pp.591–620
58. M. Matin, M. Pollard, Historical accounts of cobalt ore processing from the Kashan mine, Iran. *Iran* **53**(1), 171–183 (2015). <https://doi.org/10.1080/05786967.2015.11834755>
59. P. Rejmak, Structural, optical, and magnetic properties of ultramarine pigments: a DFT insight. *J. Phys. Chem. C* **122**(51), 29338–29349 (2018). <https://doi.org/10.1021/acs.jpcc.8b09856>
60. J. Wyart, P. Bariand, J. Filippi, C. Stockton, Lapis-lazuli from sar-E-Sang, badakhshan. Afghanistan. *Gems Gemol.* **17**(4), 184–190 (1981). <https://doi.org/10.5741/gems.17.4.184>
61. D. Reinen, G.-G. Lindner, The nature of the chalcogen colour centres in ultramarine-type solids. *Chem. Soc. Rev.* **28**(2), 75–84 (1999). <https://doi.org/10.1039/a704920j>
62. IZA-SC. Database of zeolite structures. URL: <https://www.iza-structure.org/databases>
63. N. Gobeltz-Hauteceour, A. Demortier, B. Lede, J.P. Lelieur, C. Duhayon, Occupancy of the sodalite cages in the blue ultramarine pigments. *Inorg. Chem.* **41**(11), 2848–2854 (2002). <https://doi.org/10.1021/ic010822c>
64. Q. Gao, Y. Xiu, G.-D. Li, J.-S. Chen, Sensor material based on occluded trisulfur anionic radicals for convenient detection of trace amounts of water molecules. *J. Mater. Chem.* **20**(16), 3307–3312 (2010). <https://doi.org/10.1039/B925233A>
65. Q. Zhang, S. Gao, J. Yu, Metal sites in zeolites: synthesis, characterization, and catalysis. *Chem. Rev.* **123**(9), 6039–6106 (2023). <https://doi.org/10.1021/acs.chemrev.2c00315>
66. S. Kowalak, A. Jankowska, S. Zeidler, A.B. Wieckowski, Sulfur radicals embedded in various cages of ultramarine analogs prepared from zeolites. *J. Solid State Chem.* **180**(3), 1119–1124 (2007). <https://doi.org/10.1016/j.jssc.2007.01.004>
67. H. Song, K. Münch, X. Liu, K. Shen, R. Zhang et al., All-solid-state Li–S batteries with fast solid–solid sulfur reaction. *Nature* **637**, 846–853 (2025). <https://doi.org/10.1038/s41586-024-08298-9>
68. B.B. Gicha, L.T. Tufa, N. Nwaji, X. Hu, J. Lee, Advances in all-solid-state lithium-sulfur batteries for commercialization. *Nano-Micro Lett.* **16**(1), 172 (2024). <https://doi.org/10.1007/s40820-024-01385-6>
69. B. Qi, X. Hong, Y. Jiang, J. Shi, M. Zhang et al., A review on engineering design for enhancing interfacial contact in solid-state lithium–sulfur batteries. *Nano-Micro Lett.* **16**(1), 71 (2024). <https://doi.org/10.1007/s40820-023-01306-z>
70. X. Chi, M. Li, X. Chen, J. Xu, X. Yin et al., Enabling high-performance all-solid-state batteries via guest wrench in zeolite strategy. *J. Am. Chem. Soc.* **145**(44), 24116–24125 (2023). <https://doi.org/10.1021/jacs.3c07858>
71. X. Zhu, L. Wang, Z. Bai, J. Lu, T. Wu, Sulfide-based all-solid-state lithium-sulfur batteries: challenges and perspectives. *Nano-Micro Lett.* **15**(1), 75 (2023). <https://doi.org/10.1007/s40820-023-01053-1>
72. M. Liu, L.-J. Hu, Z.-K. Guan, T.-L. Chen, X.-Y. Zhang et al., Tailoring cathode-electrolyte interface for high-power and stable lithium-sulfur batteries. *Nano-Micro Lett.* **17**(1), 85 (2024). <https://doi.org/10.1007/s40820-024-01573-4>
73. G.S. Pokrovski, L.S. Dubrovinsky, The S_3^- ion is stable in geological fluids at elevated temperatures and pressures. *Science* **331**(6020), 1052–1054 (2011). <https://doi.org/10.1126/science.1199911>
74. G.S. Pokrovski, M.A. Kokh, D. Guillaume, A.Y. Borisova, P. Gisquet et al., Sulfur radical species form gold deposits on Earth. *Proc. Natl. Acad. Sci. U.S.A.* **112**(44), 13484–13489 (2015). <https://doi.org/10.1073/pnas.1506378112>
75. F. Gaillard, E. Levillain, J.P. Lelieur, Polysulfides in dimethylformamide: only the radical anions S_3^- and S_4^- are reducible. *J. Electroanal. Chem.* **432**(1–2), 129–138 (1997). [https://doi.org/10.1016/S0022-0728\(97\)00192-7](https://doi.org/10.1016/S0022-0728(97)00192-7)
76. R. Steudel, Y. Steudel, Polysulfide chemistry in sodium–sulfur batteries and related systems—a computational study by G3X (MP2) and PCM calculations. *Chem. Eur. J.* **19**, 3162–3176 (2013). <https://doi.org/10.1002/chem.201203397>
77. T. Fujinaga, T. Kuwamoto, S. Okazaki, M. Hojo, Electrochemical reduction of elemental sulfur in acetonitrile. *Bull. Chem. Soc. Jpn* **53**(10), 2851–2855 (1980). <https://doi.org/10.1246/bcsj.53.2851>
78. S. Dev, E. Ramli, T.B. Rauchfuss, S.R. Wilson, Synthesis and structure of $[M(N\text{-methylimidazole})_6]S_8$ ($M = \text{manganese, iron, nickel, magnesium}$). Polysulfide salts prepared by the reaction $N\text{-methylimidazole} + \text{metal powder} + \text{sulfur}$. *Inorg. Chem.* **30**(11), 2514–2519 (1991). <https://doi.org/10.1021/ic00011a011>
79. B. Zhang, Z. Wang, H. Ji, H. Zhang, L. Li et al., Unveiling light effect on formation of trisulfur radicals in lithium-sulfur batteries. *Chem. Commun.* **59**(28), 4237–4240 (2023). <https://doi.org/10.1039/d3cc00120b>
80. X. Han, X. Xu, Mechanistic insights into trisulfur radical generation in lithium–sulfur batteries. *J. Mater. Chem. A* **11**(35), 18922–18932 (2023). <https://doi.org/10.1039/d3ta03366j>
81. F. Liu, G. Sun, H.B. Wu, G. Chen, D. Xu et al., Dual redox mediators accelerate the electrochemical kinetics of lithium-sulfur batteries. *Nat. Commun.* **11**(1), 5215 (2020). <https://doi.org/10.1038/s41467-020-19070-8>
82. J. Xiao, J.Z. Hu, H. Chen, M. Vijayakumar, J. Zheng et al., Following the transient reactions in lithium-sulfur batteries using an in situ nuclear magnetic resonance technique. *Nano Lett.* **15**(5), 3309–3316 (2015). <https://doi.org/10.1021/acs.nanolett.5b00521>
83. S. Babar, C. Lekakou, Molecular modeling of electrolyte and polysulfide ions for lithium-sulfur batteries. *Ionics* **27**(2), 635–642 (2021). <https://doi.org/10.1007/s11581-020-03860-7>
84. I. Gunasekara, S. Mukerjee, E.J. Plichta, M.A. Hendrickson, K.M. Abraham, A study of the influence of lithium salt anions on oxygen reduction reactions in Li-air batteries. *J. Electrochem. Soc.* **162**(6), A1055–A1066 (2015). <https://doi.org/10.1149/2.0841506jes>
85. T. Chivers, R.T. Oakley, Structures and spectroscopic properties of polysulfide radical anions: a theoretical perspective. *Molecules* **28**(15), 5654 (2023). <https://doi.org/10.3390/molecules28155654>



86. K. Park, J.H. Cho, J.-H. Jang, B.-C. Yu, A.T. De La Hoz et al., Trapping lithium polysulfides of a Li-S battery by forming lithium bonds in a polymer matrix. *Energy Environ. Sci.* **8**(8), 2389–2395 (2015). <https://doi.org/10.1039/C5EE01809A>
87. H. Li, S. Chiba, Synthesis of α -tertiary amines by polysulfide anions photocatalysis via single-electron transfer and hydrogen atom transfer in relays. *Chem Catal.* **2**(5), 1128–1142 (2022). <https://doi.org/10.1016/j.checat.2022.03.006>
88. M.K. Mondal, L. Zhang, Z. Feng, S. Tang, R. Feng et al., Tricoordinate nontrigonal pnictogen-centered radical anions: isolation, characterization, and reactivity. *Angew. Chem. Int. Ed.* **58**, 15829–15833 (2019). <https://doi.org/10.1002/ange.201910139>
89. P. Liebing, M. Kühling, C. Swanson, M. Feneberg, L. Hilfert et al., Catenated and spirocyclic polychalcogenides from potassium carbonate and elemental chalcogens. *Chem. Commun.* **55**(99), 14965–14967 (2019). <https://doi.org/10.1039/c9cc08347b>
90. K.H. Wujcik, T.A. Pascal, C.D. Pemmaraju, D. Devaux, W.C. Stolte et al., Characterization of polysulfide radicals present in an ether-based electrolyte of a lithium-sulfur battery during initial discharge using in situ X-ray absorption spectroscopy experiments and first-principles calculations. *Adv. Energy Mater.* **5**(16), 1500285 (2015). <https://doi.org/10.1002/aenm.201500285>
91. A. Kawase, S. Shirai, Y. Yamoto, R. Arakawa, T. Takata, Electrochemical reactions of lithium-sulfur batteries: an analytical study using the organic conversion technique. *Phys. Chem. Chem. Phys.* **16**(20), 9344–9350 (2014). <https://doi.org/10.1039/C4CP00958D>
92. G. Zhang, Z.W. Zhang, H.J. Peng, J.Q. Huang, Q. Zhang, A toolbox for lithium-sulfur battery research: methods and protocols. *Small Meth.* **1**(7), 1700134 (2017). <https://doi.org/10.1002/smt.201700134>
93. K. Mahankali, L.M.R. Arava, Nanoscale visualization of reversible redox pathways in lithium-sulfur battery using in situ AFM-SECM. *J. Electrochem. Soc.* **169**(6), 060501 (2022). <https://doi.org/10.1149/1945-7111/ac70ff>
94. M. Cuisinier, P.-E. Cabelguen, S. Evers, G. He, M. Kolbeck et al., Sulfur speciation in Li-S batteries determined by operando X-ray absorption spectroscopy. *J. Phys. Chem. Lett.* **4**(19), 3227–3232 (2013). <https://doi.org/10.1021/jz401763d>
95. H. Tian, A. Song, H. Tian, J. Liu, G. Shao et al., Single-atom catalysts for high-energy rechargeable batteries. *Chem. Sci.* **12**(22), 7656–7676 (2021). <https://doi.org/10.1039/d1sc00716e>
96. H. Tian, H. Tian, S. Wang, S. Chen, F. Zhang et al., High-power lithium-selenium batteries enabled by atomic cobalt electrocatalyst in hollow carbon cathode. *Nat. Commun.* **11**(1), 5025 (2020). <https://doi.org/10.1038/s41467-020-18820-y>
97. B. Sheng, D. Cao, C. Liu, S. Chen, L. Song, Support effects in electrocatalysis and their synchrotron radiation-based characterizations. *J. Phys. Chem. Lett.* **12**(47), 11543–11554 (2021). <https://doi.org/10.1021/acs.jpcclett.1c02805>
98. P. Song, W. Rao, T. Chivers, S.-Y. Wang, Applications of trisulfide radical anion $S_3^{\cdot-}$ in organic synthesis. *Org. Chem. Front.* **10**(13), 3378–3400 (2023). <https://doi.org/10.1039/d3qo00496a>
99. Z.-Y. Gu, J.-J. Cao, S.-Y. Wang, S.-J. Ji, The involvement of the trisulfur radical anion in electron-catalyzed sulfur insertion reactions: facile synthesis of benzothiazine derivatives under transition metal-free conditions. *Chem. Sci.* **7**(7), 4067–4072 (2016). <https://doi.org/10.1039/C6SC00240D>
100. M. Baek, H. Shin, K. Char, J.W. Choi, New high donor electrolyte for lithium-sulfur batteries. *Adv. Mater.* **32**(52), e2005022 (2020). <https://doi.org/10.1002/adma.202005022>
101. N. Zhong, C. Lei, R. Meng, J. Li, X. He et al., Electrolyte solvation chemistry for the solution of high-donor-number solvent for stable Li-S batteries. *Small* **18**(16), e2200046 (2022). <https://doi.org/10.1002/sml.202200046>
102. H. Chu, H. Noh, Y.-J. Kim, S. Yuk, J.-H. Lee et al., Achieving three-dimensional lithium sulfide growth in lithium-sulfur batteries using high-donor-number anions. *Nat. Commun.* **10**(1), 188 (2019). <https://doi.org/10.1038/s41467-018-07975-4>
103. Z. Shen, Q. Gao, X. Zhu, Z. Guo, K. Guo et al., In-situ free radical supplement strategy for improving the redox kinetics of Li-S batteries. *Energy Storage Mater.* **57**, 299–307 (2023). <https://doi.org/10.1016/j.ensm.2023.02.023>
104. W. Zhang, F. Ma, Q. Wu, Z. Zeng, W. Zhong et al., Dual-functional organotelluride additive for highly efficient sulfur redox kinetics and lithium regulation in lithium-sulfur batteries. *Energy Environ. Mater.* **6**(3), e12369 (2023). <https://doi.org/10.1002/eem2.12369>
105. P. Zeng, Z. Zhou, B. Li, H. Yu, X. Zhou et al., Insight into the catalytic role of defect-enriched vanadium sulfide for regulating the adsorption-catalytic conversion behavior of polysulfides in Li-S batteries. *ACS Appl. Mater. Interfaces* **14**(31), 35833–35843 (2022). <https://doi.org/10.1021/acsami.2c09791>
106. R. Xu, H. Tang, Y. Zhou, F. Wang, H. Wang et al., Enhanced catalysis of radical-to-polysulfide interconversion via increased sulfur vacancies in lithium-sulfur batteries. *Chem. Sci.* **13**(21), 6224–6232 (2022). <https://doi.org/10.1039/d2sc01353c>
107. Z. Xiao, Z. Yang, Z. Li, P. Li, R. Wang, Synchronous gains of areal and volumetric capacities in lithium-sulfur batteries promised by flower-like porous $Ti_3C_2T_x$ matrix. *ACS Nano* **13**(3), 3404–3412 (2019). <https://doi.org/10.1021/acs.nano.8b09296>
108. M. Jafarian, M.G. Mahjani, F. Gobal, I. Danaee, Effect of potential on the early stage of nucleation and growth during aluminum electrocrystallization from molten salt ($AlCl_3$ - $NaCl$ - KCl). *J. Electroanal. Chem.* **588**(2), 190–196 (2006). <https://doi.org/10.1016/j.jelechem.2005.12.028>
109. L.C.H. Gerber, P.D. Frischmann, F.Y. Fan, S.E. Doris, X. Qu et al., Three-dimensional growth of Li_2S in lithium-sulfur batteries promoted by a redox mediator. *Nano Lett.* **16**(1), 549–554 (2016). <https://doi.org/10.1021/acs.nanolett.5b04189>
110. F.Y. Fan, W. Craig Carter, Y.-M. Chiang, Mechanism and kinetics of Li_2S precipitation in lithium-sulfur batteries. *Adv.*

- Mater. **27**(35), 5203–5209 (2015). <https://doi.org/10.1002/adma.201501559>
111. H. Chu, J. Jung, H. Noh, S. Yuk, J. Lee et al., Unraveling the dual functionality of high-donor-number anion in lean-electrolyte lithium-sulfur batteries. *Adv. Energy Mater.* **10**(21), 2000493 (2020). <https://doi.org/10.1002/aenm.202000493>
112. B. Lin, Y. Zhang, W. Li, J. Huang, Y. Yang et al., Recent advances in rare earth compounds for lithium–sulfur batteries. *eScience* **4**(3), 100180 (2024). <https://doi.org/10.1016/j.esci.2023.100180>
113. Y. Liu, Z. Ma, G. Yang, Z. Wu, Y. Li et al., Multifunctional ZnCo₂O₄ quantum dots encapsulated in carbon carrier for anchoring/catalyzing polysulfides and self-repairing lithium metal anode in lithium–sulfur batteries. *Adv. Funct. Mater.* **32**, 2109462 (2021). <https://doi.org/10.1002/adfm.202109462>
114. C.Y. Zhang, C. Zhang, J.L. Pan, G.W. Sun, Z. Shi et al., Surface strain-enhanced MoS₂ as a high-performance cathode catalyst for lithium–sulfur batteries. *eScience* **2**(4), 405–415 (2022). <https://doi.org/10.1016/j.esci.2022.07.001>
115. H. Wang, Z. Cui, S.-A. He, J. Zhu, W. Luo et al., Construction of ultrathin layered MXene-TiN heterostructure enabling favorable catalytic ability for high-areal-capacity lithium-sulfur batteries. *Nano-Micro Lett.* **14**(1), 189 (2022). <https://doi.org/10.1007/s40820-022-00935-0>
116. S. Tian, Q. Zeng, G. Liu, J. Huang, X. Sun et al., Multi-dimensional composite frame as bifunctional catalytic medium for ultra-fast charging lithium-sulfur battery. *Nano-Micro Lett.* **14**(1), 196 (2022). <https://doi.org/10.1007/s40820-022-00941-2>
117. T. Zhang, L. Zhang, Y. Hou, MXenes: synthesis strategies and lithium-sulfur battery applications. *eScience* **2**(2), 164–182 (2022). <https://doi.org/10.1016/j.esci.2022.02.010>
118. Q. Gu, Y. Cao, J. Chen, Y. Qi, Z. Zhai et al., Fluorine-modulated MXene-derived catalysts for multiphase sulfur conversion in lithium–sulfur battery. *Nano-Micro Lett.* **16**, 266 (2024). <https://doi.org/10.1007/s40820-024-01482-6>
119. S. Deng, W. Sun, J. Tang, M. Jafarpour, F. Nüesch et al., Multifunctional SnO₂ QDs/MXene heterostructures as laminar interlayers for improved polysulfide conversion and lithium plating behavior. *Nano-Micro Lett.* **16**(1), 229 (2024). <https://doi.org/10.1007/s40820-024-01446-w>
120. H. Pan, Z. Cheng, Z. Zhou, S. Xie, W. Zhang et al., Boosting lean electrolyte lithium-sulfur battery performance with transition metals: a comprehensive review. *Nano-Micro Lett.* **15**(1), 165 (2023). <https://doi.org/10.1007/s40820-023-01137-y>
121. Y. Gong, J. Li, K. Yang, S. Li, M. Xu et al., Towards practical application of Li-S battery with high sulfur loading and lean electrolyte: will carbon-based hosts win this race? *Nano-Micro Lett.* **15**(1), 150 (2023). <https://doi.org/10.1007/s40820-023-01120-7>
122. W. Sun, Z. Song, Z. Feng, Y. Huang, Z.J. Xu et al., Carbon-nitride-based materials for advanced lithium-sulfur batteries. *Nano-Micro Lett.* **14**(1), 222 (2022). <https://doi.org/10.1007/s40820-022-00954-x>
123. J. Chen, D. Wu, E. Walter, M. Engelhard, P. Bhattacharya et al., Molecular-confinement of polysulfides within mesoscale electrodes for the practical application of lithium sulfur batteries. *Nano Energy* **13**, 267–274 (2015). <https://doi.org/10.1016/j.nanoen.2015.01.031>
124. S. Feng, Z.-H. Fu, X. Chen, B.-Q. Li, H.-J. Peng et al., An electrocatalytic model of the sulfur reduction reaction in lithium-sulfur batteries. *Angew. Chem. Int. Ed.* **61**(52), e202211448 (2022). <https://doi.org/10.1002/anie.202211448>
125. Z. Han, A. Chen, Z. Li, M. Zhang, Z. Wang et al., Machine learning-based design of electrocatalytic materials towards high-energy lithium/sulfur batteries development. *Nat. Commun.* **15**(1), 8433 (2024). <https://doi.org/10.1038/s41467-024-52550-9>
126. A. Kumar, A. Ghosh, M. Forsyth, D.R. MacFarlane, S. Mitra, Free-radical catalysis and enhancement of the redox kinetics for room-temperature sodium–sulfur batteries. *ACS Energy Lett.* **5**(6), 2112–2121 (2020). <https://doi.org/10.1021/acsenenergylett.0c00913>

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