

# Synthesis of 3D Hexagram-Like Cobalt–Manganese Sulfides Nanosheets Grown on Nickel Foam: A Bifunctional Electrocatalyst for Overall Water Splitting

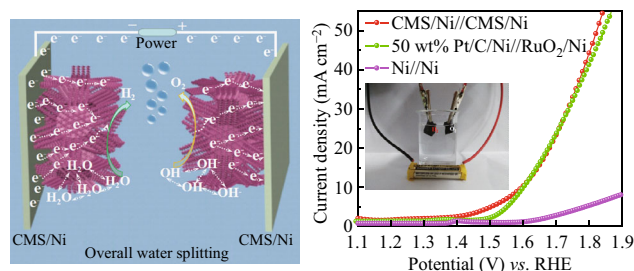
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## Highlights

- Cobalt–manganese sulfides grown on Ni foam (CMS/Ni) with three-dimensionally hexagram-like nanosheets structure were prepared via a solvothermal method.
- As-prepared CMS/Ni shows highly catalytic activity for OER and HER in basic medium and can catalyze water splitting by a 1.50 V dry battery.

**Abstract** The exploration of low-cost and efficient bifunctional electrocatalysts for oxygen evolution reaction and hydrogen evolution reaction through tuning the chemical composition is strongly required for sustainable resources. Herein, we developed a bimetallic cobalt–manganese sulfide supported on Ni foam (CMS/Ni) via a solvothermal method. It has discovered that after combining with the pure  $\text{Co}_9\text{S}_8$  and  $\text{MnS}$ , the morphologies of CMS/Ni have modulated. The obtained three-dimensionally hexagram-like CMS/Ni nanosheets have a significant increase in electrochemical active surface area and charge transport ability. More than that, the synergetic effect of Co and Mn has also presented in this composite. Benefiting from these, the CMS/Ni electrode shows great performance toward hydrogen evolution reaction and oxygen evolution reaction in basic medium, comparing favorably to that of



the pure  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $\text{MnS}/\text{Ni}$ . More importantly, this versatile CMS/Ni can catalyze the water splitting in a two-electrode system at a potential of 1.47 V, and this electrolyzer can be efficiently driven by a 1.50 V commercial dry battery.

**Keywords** Bifunctional electrocatalysts · Oxygen evolution reaction · Hydrogen evolution reaction · Cobalt–manganese sulfides · Water splitting

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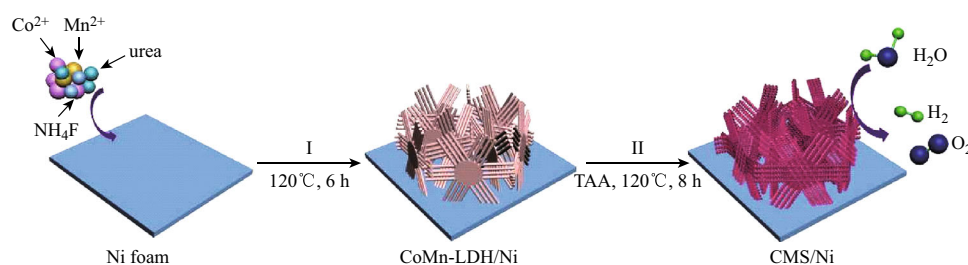
## 1 Introduction

Electrocatalytic water splitting has been regarded as the most promising and feasible technology to produce clean hydrogen fuel from aqueous solutions [1–3]. Hence, efficient electrocatalysts for both the oxygen evolution reaction (OER) at anodes and hydrogen evolution reaction (HER) at cathodes are urgently needed to reduce the energy consumption for overall water splitting [4–6]. Precious metal oxide (e.g.,  $\text{RuO}_2$ ,  $\text{IrO}_2$ ) and noble metal (e.g., Pt, Ir, Rh) electrocatalysts are so far known as the most efficient electrocatalysts toward OER and HER, respectively, but the high cost and scarcity have limited their widespread application [6, 7]. In this regard, tremendous efforts have been devoted to explore alternatively earth-abundant and cost-effective transition metal materials for OER or HER over the past several decades [8–22]. Unfortunately, these electrocatalysts are still not suitable for the real commercial applications. In addition, to simplify the overall water splitting system and cut the cost, developing highly efficient bifunctional electrocatalysts for both OER and HER in the same electrolyte, especially for alkaline electrolyte, has become one of the hottest issues recently [23]. Despite great advances have taken in this field [24–39], it is still in great demand to explore high-performance and non-noble bifunctional electrocatalysts for overall water splitting.

The catalytic activity could improve following the methods of chemical composition tuning and nanostructure modification [40–44]. On the one hand, to tailor the chemical composition of electrocatalysts, an effective way is doping with foreign atoms into the crystal lattice of materials. Following this way, the multicomponent or composite electrocatalysts would obtain [40–49]. The formation of different valence and electronic states of metal ions in these composites facilitates the adsorption and desorption of intermediates in the electrocatalysis process, and the synergistic effect between the metal ions is benefit for their catalytic activity. For example, Wu et al. [42] discovered the different valence states of Ni and synergistic effect between the metal ions in  $\text{Ni}_3\text{ZnC}_{0.7}$ , playing an important role in its catalytic activity for HER and OER. Yang et al. [50] synthe-

sized a  $\text{Co(II)}_{1-x}\text{Co(0)}_{x/3}\text{Mn(III)}_{2x/3}\text{S}$  nanoparticles combining with B/N-codoped mesoporous nanocarbon. They investigated the formation of different valence and electronic states of Co and Mn ions in facilitating the catalytic activity. On the other hand, optimizing the nanostructure of electrocatalysts can increase the quantity of effectively active sites. Comparing to nanoparticles materials, the electrocatalysts directly supported on conductive substrates, such as Ni foam, Ni mesh, Cu mesh, carbon cloth and carbon paper, with binder free can get to this point easily [41]. Recently, Ni foam has exhibited considerable potential in optimizing the nanostructure of materials [51–54]. It has a unique three-dimensional (3D) porous structure and high conductivity. For instance, our group [32] prepared urchin-like sphere arrays  $\text{Co}_3\text{O}_4$  supported on 3D Ni foam showing great performance for HER and OER, which is benefit from its urchin-like nanostructure with rich mesopores and low charge-transfer resistance. Hu et al. [41] developed a Co–Mn carbonate hydroxide ( $\text{CoMnCH}$ ) nanosheet arrays on Ni foam exhibiting superior activity for HER and OER in basic medium.

Based on the above mentioned, a cobalt–manganese sulfides composite with a unique nanostructure was synthesized to exhibit an efficient catalytic performance for OER and HER. However, to the best of our knowledge, the cobalt–manganese sulfide with efficient catalytic performance has never been reported. Herein, the 3D hexagram-like CMS/Ni was prepared via a simple two-step hydrothermal method (Scheme 1). Firstly, the  $\text{CO}_3^{2-}$  and  $\text{OH}^-$  ions were released by the hydrolysis of urea and gradually co-precipitated with  $\text{Co}^{2+}$  and  $\text{Mn}^{2+}$  ions to form the 3D hexagram-like precursor  $\text{CoMn-LDH/Ni}$ . Subsequently, the 3D hexagram-like CMS/Ni nanosheets were synthesized after a sulfurization process employing the thioacetamide (TAA) as the sulfur sources. The electrochemical catalytic activity of this composite was evaluated by linear sweep voltammetry, electrochemical impedance spectroscopy and chronoamperometry. The electrocatalytic performance of CMS/Ni for HER and OER is optimized with respect to those of the pure  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $\text{MnS/Ni}$ . The CMS/Ni was designed as an efficient electrolyzer for overall water splitting.



**Scheme 1** Scheme of the synthesis procedure for CMS/Ni

## 2 Experimental Sections

### 2.1 Materials

MnCl<sub>2</sub>·4H<sub>2</sub>O, Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O, urea and NH<sub>4</sub>F were purchased from Aladdin. Ni foam was purchased from Kunshan Electronic Limited Corporation. All chemicals were directly used without any purification.

### 2.2 Synthesis of CoMn-LDH/Ni

Typically, 198 mg MnCl<sub>2</sub>·4H<sub>2</sub>O, 582 mg Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O, 180 mg urea and 37 mg NH<sub>4</sub>F were dissolved into a beaker containing 40 mL distilled water and 10 mL absolute ethanol to form a homogeneous solution under stirring for 10 min. A piece of Ni foam (3 × 3 cm<sup>2</sup>) which was cleaned by sonication sequentially in 3 mol L<sup>-1</sup> HCl solution and absolute ethanol for 15 min each was immersed into the above solution, and then, the mixture was transferred into an autoclave (80 mL). The autoclave was sealed and heated at 120 °C for 6 h. After cooling down to room temperature, the precursor CoMn-LDH/Ni (XRD pattern is exhibited in Fig. S1) was taken out and washed by deionized water for four times and dried at 60 °C.

### 2.3 Synthesis of CMS/Ni

To obtain the CMS/Ni, 400 mg thioacetamide was dissolved in 50 mL deionized water. Then, the clean solution was transferred into an autoclave containing a piece of prepared CoMn-LDH/Ni. After heating at 120 °C for 12 h, the product was taken out and severally washed by absolute ethanol and deionized water for four times. Finally, the CMS/Ni was obtained after dehydrating in an oven at 60 °C overnight. The mass loading of CMS is 4.1 mg cm<sup>-2</sup>. For comparison, the Co<sub>9</sub>S<sub>8</sub>/Ni and MnS/Ni were also synthesized through the same processes without adding the Mn<sup>2+</sup> or Co<sup>2+</sup> ions.

### 2.4 Materials Characterization

The X-ray diffraction (XRD) patterns were tested by a MSAL-XD2 X-ray diffractometer with Cu K $\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ). The scanning electron microscopy (SEM) was performed by a Philips SEM-XL30S microscope operated at 15 kV. High-resolution transmission electron microscope (HRTEM, JEOL JEM-2100F) coupled with an energy-dispersive X-ray spectroscopy (EDS) analyzer was carried out with an accelerating voltage of 200 kV. The nitrogen sorption isotherms were carried out by a Micromeritics TriStar 3000 Analyzer at 77 K. The X-ray

photoelectron spectroscopy (XPS) was analyzed by an ESCALab250.

### 2.5 Electrochemical Measurements

All the electrochemical measurements were carried out in a conventional three-electrode system. The CMS/Ni and other obtained samples were used as the working electrodes. Pt foil and Hg/HgO electrodes were separately used as the counter and reference electrodes. The 1.0 mol L<sup>-1</sup> KOH was used as the electrolyte. Linear sweep voltammetry (LSV) was analyzed with a scan rate of 2 mV s<sup>-1</sup>. All potentials in this work were calibrated by the Nernst equation:

$$E_{\text{RHE}} = E_{\text{Hg/HgO}} + (0.098 + 0.059 \text{ pH}) \text{ V} \quad (1)$$

$$\eta = E_{\text{RHE}} - 1.23 \text{ V} \quad (2)$$

where  $\eta$  is the overpotential. In addition, the Tafel slope was modeled by Tafel equation:

$$\eta = b \log j + a \quad (3)$$

where  $\eta$  is the overpotential,  $b$  is the Tafel slope,  $j$  is the current density and  $a$  is a constant. Chronoamperometry measurements were tested at a static potential, and the electrochemical impedance spectroscopy (EIS) was also performed in the frequency range from 10 kHz to 10 mHz with an amplitude of 5 mV. The electrochemically active surface areas (ECSAs) were evaluated by the electrochemical double-layer capacitance ( $C_{\text{dl}}$ ) via collecting cyclic voltammograms (CVs). The different CV cycles (5, 10, 15, 20, 30, and 50 mV s<sup>-1</sup>) were tested in the non-Faradaic potential region from 0.124 to 0.224 V vs RHE to determine the  $C_{\text{dl}}$ .

## 3 Results and Discussion

### 3.1 Structure and Morphology of Materials

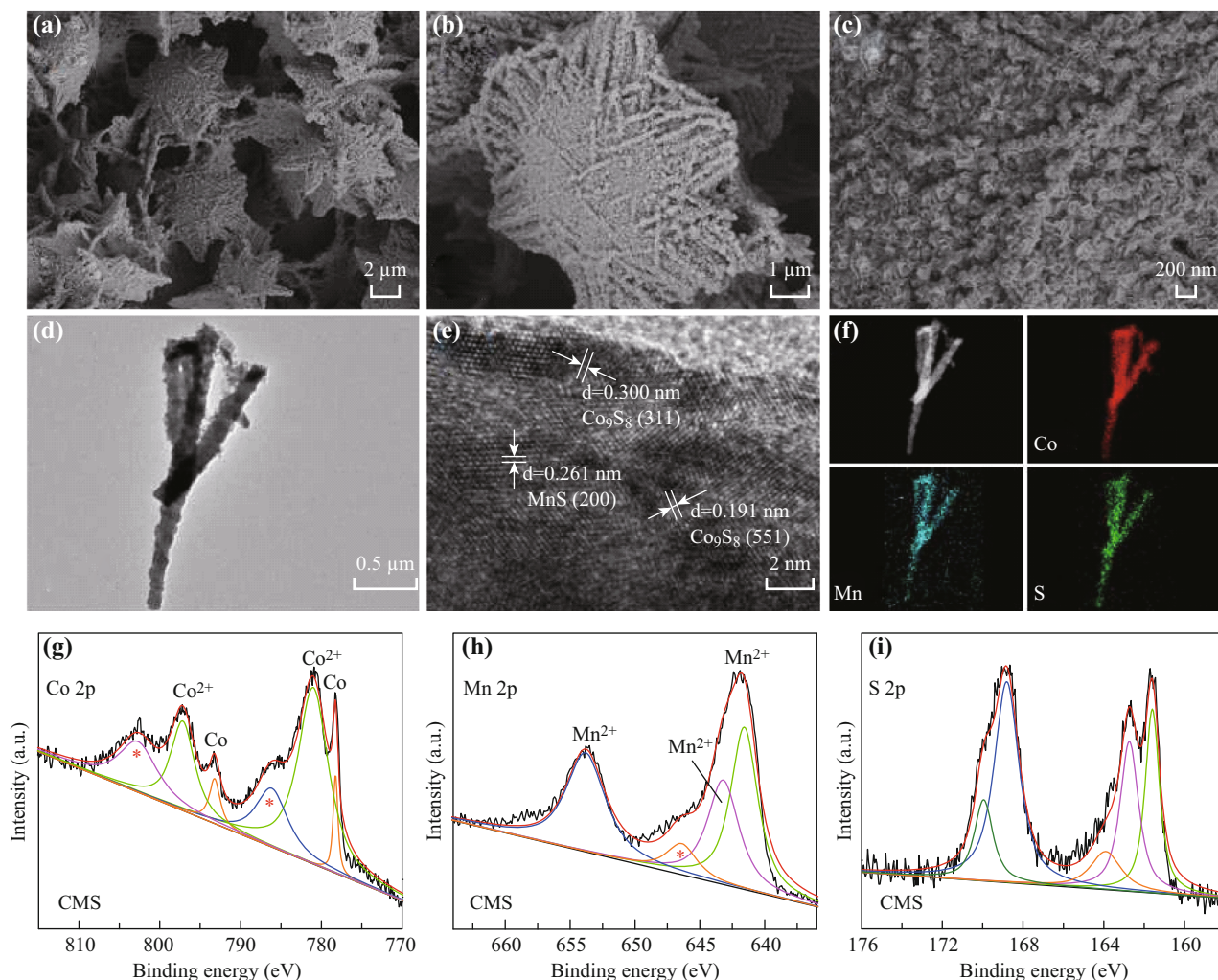
The XRD patterns of the as-prepared samples were performed using their powders scraped down from the Ni foam. As shown in Fig. S2a, the XRD pattern of Co<sub>9</sub>S<sub>8</sub> exhibits the cubic crystalline phase with diffraction peaks at 29.8°, 31.1°, 47.6°, and 51.9°, which are severally corresponding to the (311), (222), (511), and (440) planes of Co<sub>9</sub>S<sub>8</sub> (No. 02-1459). Meanwhile, the typical diffraction peaks of MnS located at 34.3° and 49.3° are matched well with the (200) and (220) planes of cubic MnS (No. 65-2919). Interestingly, the XRD pattern of CMS reveals the crystal structures of the Co<sub>9</sub>S<sub>8</sub> and MnS are still maintained in CMS after modulating with them as a composite. Moreover, the EDS analysis reveals the atomic

ratio of Co, Mn and S is  $\sim 0.9:0.1:1.0$  in the composite (Fig. S2b), while the Cu elemental is coming from copper mesh. Figure S2c, d shows the atomic ratios of  $\text{Co}_9\text{S}_8$  and MnS are  $\sim 9.0:8.0$  and  $1.0:1.0$ , respectively.

Figure S3 presents the SEM images of  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $\text{MnS}/\text{Ni}$ . The nanosheet structures of  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $\text{MnS}/\text{Ni}$  are still retained a rough surface with respect to their precursors. However, the morphologies are modified after combining with  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $\text{MnS}/\text{Ni}$ . It can be observed in Fig. 1a, b that the 3D hexagram-like CMS/Ni nanosheets were obtained after sulfonating the  $\text{CoMn-LDH}/\text{Ni}$  (Fig. S4), and each “hexagram” is entirely covered by uniform nanosheets (Fig. 1c). The TEM image shown in Fig. 1d reveals the 3D hexagram-like CMS/Ni nanosheets which are made up of solid nanorods. Figure 1e shows the HRTEM image, where the distinct lattice fringes of  $d = 0.300$  nm and  $d = 0.191$  nm are corresponding to (311) and (511) crystal plane of  $\text{Co}_9\text{S}_8$ , respectively, and

the  $d = 0.261$  nm is attributed to (200) crystal plane of MnS. This result indicates that the CMS is a composite material, which is comprised by the interrelated  $\text{Co}_9\text{S}_8$  and MnS. Additionally, it can be seen that the elements of Co, Mn and S are uniformly dispersed in CMS/Ni, as shown in Fig. 1f.

The Co 2p XPS spectra of CMS/Ni in Fig. 1g reveal two distinct peaks at 781.0 and 797.3 eV corresponding to the Co  $2p_{3/2}$  and Co  $2p_{1/2}$ , respectively, with two related satellite peaks at 786.3 and 802.9 eV. These are the characteristic peaks of  $\text{Co}^{2+}$  [50], and the other two peaks at 778.2 and 793.2 eV are assigned to the metallic Co [50]. Nevertheless, there is not metallic Co presented on the Co 2p XPS spectra of  $\text{Co}_9\text{S}_8/\text{Ni}$  counterpart (Fig. S5a). Moreover, there are no diffraction peaks of metallic Co presenting in the XRD pattern of CMS/Ni, which may be attributed to its low content in CMS/Ni. As Mn 2p XPS spectra of CMS/Ni shown in Fig. 1h, the peak located at



**Fig. 1** The characterization results of CMS/Ni: **a–c** SEM images in different magnifications, **d** TEM image, **e** HRTEM image, **f** elemental mapping images and **g–i** XPS spectra

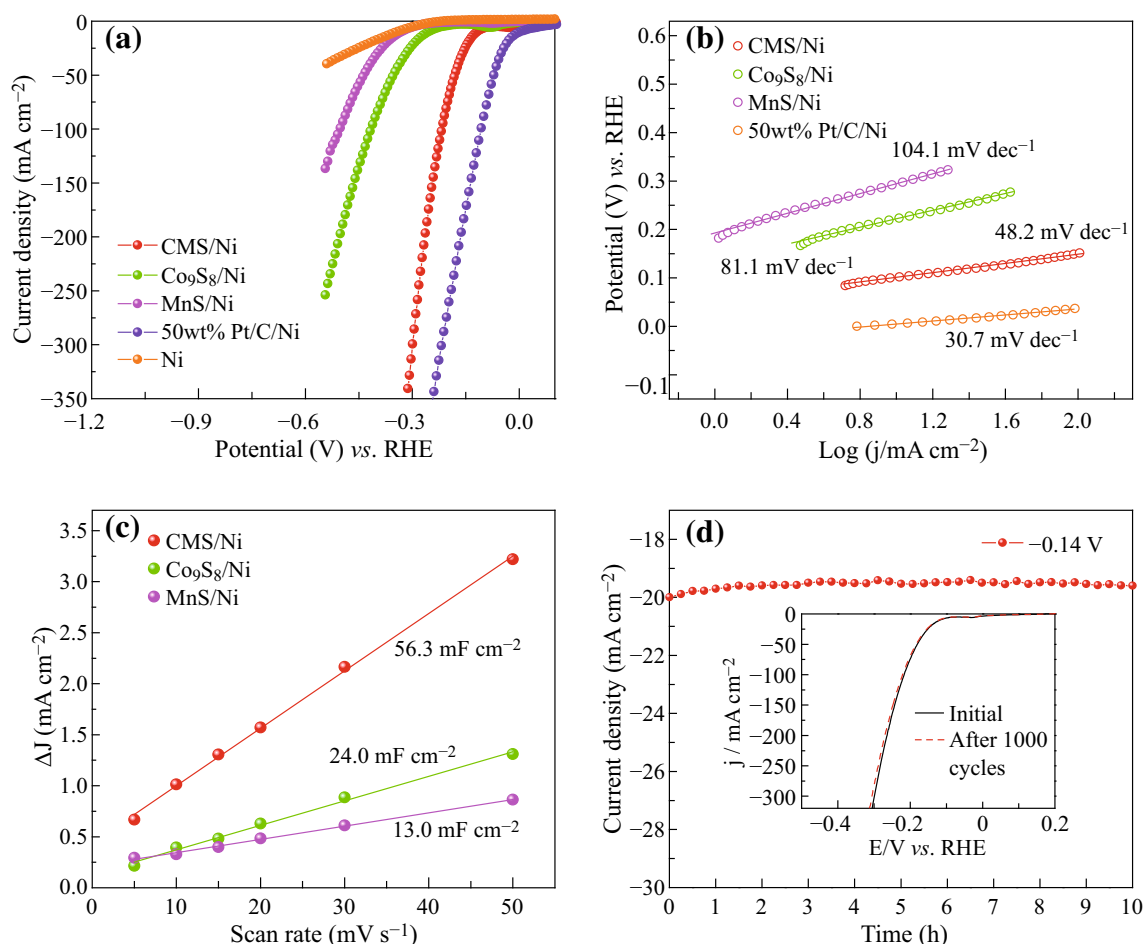


643.3 eV confirms the oxidized  $\text{Mn}^{3+}$  species in CMS/Ni [50, 55], while there is no  $\text{Mn}^{3+}$  detected from XPS spectra of MnS/Ni counterpart (Fig. S5b). Simultaneously,  $\text{Mn}^{2+}$  reveals with two characteristic peaks at binding energies of 641.6 eV ( $\text{Mn } 2p_{3/2}$ ) and 653.8 eV ( $\text{Mn } 2p_{1/2}$ ). The above results are similar to Yang's work [50]. The occurrence of metallic Co and  $\text{Mn}^{3+}$  in CMS/Ni can be explained by the higher reduction potential of  $\text{Co}^{2+}/\text{Co}$  ( $\text{Co}^{2+} + 2e^- \rightarrow \text{Co}$ ,  $-0.277 \text{ V vs. NHE}$ ) comparing to  $\text{Mn}^{2+}/\text{Mn}$  ( $\text{Mn}^{2+} + 2e^- \rightarrow \text{Mn}$ ,  $-1.18 \text{ V vs. NHE}$ ). This means that the  $\text{Co}^{2+}$  as oxidizer is easier to be reduced from  $\text{Co}^{2+}$  to metallic Co than the same reaction of  $\text{Mn}^{2+}$ . The reduction in  $\text{Co}^{2+}$  to metallic Co would result in the oxidation of  $\text{Mn}^{2+}$  to  $\text{Mn}^{3+}$  at the same time ( $\text{Mn}^{2+} - e^- \rightarrow \text{Mn}^{3+}$ ,  $1.5 \text{ V vs. NHE}$ ) [56]. The different valance states of metal cations in CMS/Ni are benefit for improving the catalytic performance [42, 50]. Figure 1i shows the peaks of S  $2p_{3/2}$  and S  $2p_{1/2}$  are located at 161.6 and 162.7 eV, respectively, which are derived from metal-sulfur bonds [50]. Furthermore, a peak at 168.8 eV with its

satellite peak at 170.0 eV is attributed to the superficial oxidation of CMS/Ni in air [55].

### 3.2 Hydrogen Evolution Activity

The electrocatalytic activity of CMS/Ni toward HER was characterized by LSV measurement with a scan rate of  $2 \text{ mV s}^{-1}$ . For comparison, the  $\text{Co}_9\text{S}_8/\text{Ni}$ , MnS/Ni, bare Ni foam and commercial 50 wt% Pt/C coated on Ni foam (50 wt% Pt/C/Ni [30], loading  $4.1 \text{ mg cm}^{-2}$ ) were also evaluated. As the polarization curves illustrated in Fig. 2a, the 50 wt% Pt/C/Ni possesses the most excellent HER performance with a near-zero onset potential. The CMS/Ni reveals a much smaller onset potential at  $-88 \text{ mV}$  and larger HER current than those of  $\text{Co}_9\text{S}_8/\text{Ni}$ , MnS/Ni and bare Ni foam. This result highlights the catalytic activity of CMS/Ni has significantly enhanced after combining  $\text{Co}_9\text{S}_8/\text{Ni}$  with MnS/Ni as a composite, which may be ascribed to the synergistic effect of Co and Mn. The HER performance of CMS/Ni is also comparable with the CoMn-LDH/Ni, as



**Fig. 2** **a** Polarization curves, **b** Tafel plots, **c** plots of the current density for CMS/Ni,  $\text{Co}_9\text{S}_8/\text{Ni}$ , MnS/Ni at potential of 0.174 V against the scan rates and **d** chronoamperometric curve (the inset is polarization curves before and after 1000 CV cycles)

shown in Fig. S6a, b. Noticeably, the CMS/Ni drives a high current density of  $100 \text{ mA cm}^{-2}$  at an overpotential of 217 mV, which is lower than 419 mV for  $\text{Co}_9\text{S}_8/\text{Ni}$  and 506 mV for  $\text{MnS}/\text{Ni}$ . Such an overpotential of CMS/Ni is superior to most of the recently reported bifunctional electrocatalysts listed in Table S1. The efficient catalytic activity of CMS/Ni is also supported by the Tafel slopes in Fig. 2b. The Tafel slope of CMS/Ni is  $48.2 \text{ mV dec}^{-1}$  which is obviously lower than  $81.1 \text{ mV dec}^{-1}$  for  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $104.1 \text{ mV dec}^{-1}$  for  $\text{MnS}/\text{Ni}$ , implying great HER kinetics and catalytic activity. In addition, the EIS measurements of CMS/Ni,  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $\text{MnS}/\text{Ni}$  were measured at a static potential of  $-0.33 \text{ V}$  to further elucidate the charge transport of the as-prepared materials. As exhibited in Fig. S7, the CMS/Ni shows lower resistance of  $0.98 \Omega$  than the pure  $\text{Co}_9\text{S}_8/\text{Ni}$  ( $1.80 \Omega$ ) and  $\text{MnS}/\text{Ni}$  ( $2.14 \Omega$ ), suggesting an improvement of conductivity. The lower charge-transfer resistance of CMS/Ni may be attributed to its uniquely 3D hexagram-like nanosheets structure contacting with the electrolyte efficiently, and the metallic Co generated in CMS (see Fig. 1g) after integrating with two  $\text{Co}_9\text{S}_8$  and  $\text{MnS}$  to be the composite [50].

On the other hand, the superior activity of CMS/Ni for HER, in comparison with that of the pure  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $\text{MnS}/\text{Ni}$ , results from the significant increase in electrochemical active surface areas (ECSAs) to expose more accessible catalytic active sites. The ECSAs of CMS/Ni,  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $\text{MnS}/\text{Ni}$  were measured by the capacitance measurements through cyclic voltammograms in a non-Faradaic at different scan rates (Fig. S8). The ECSA of an electrocatalyst is proportional to its  $C_{dl}$  value. It can be seen that the  $C_{dl}$  values of  $24.0 \text{ mF cm}^{-2}$  for  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $13.0 \text{ mF cm}^{-2}$  for  $\text{MnS}/\text{Ni}$  are tremendously increase to  $56.3 \text{ mF cm}^{-2}$  for CMS/Ni, implying the CMS/Ni has more effective active sites, as shown in Fig. 2c.

The HER stability of CMS/Ni was further evaluated at a constant potential of  $-0.14 \text{ V}$ . As shown in Fig. 2d, the CMS/Ni reveals a great stability with a negligible decay of the current density after 10 h continuous measurements. Simultaneously, the LSV polarization curve tested at  $100 \text{ mV s}^{-1}$  after 1000 cycles is similar to the first cycle, as the inset in Fig. 2d. The superior durability of CMS/Ni is benefit from its high structure stability because the hexagram-like structure for CMS/Ni just exhibits a little aggregation after stability measurement in Fig. S9a. The excellent catalytic performance and great stability highlight the great potential of CMS/Ni for practical application.

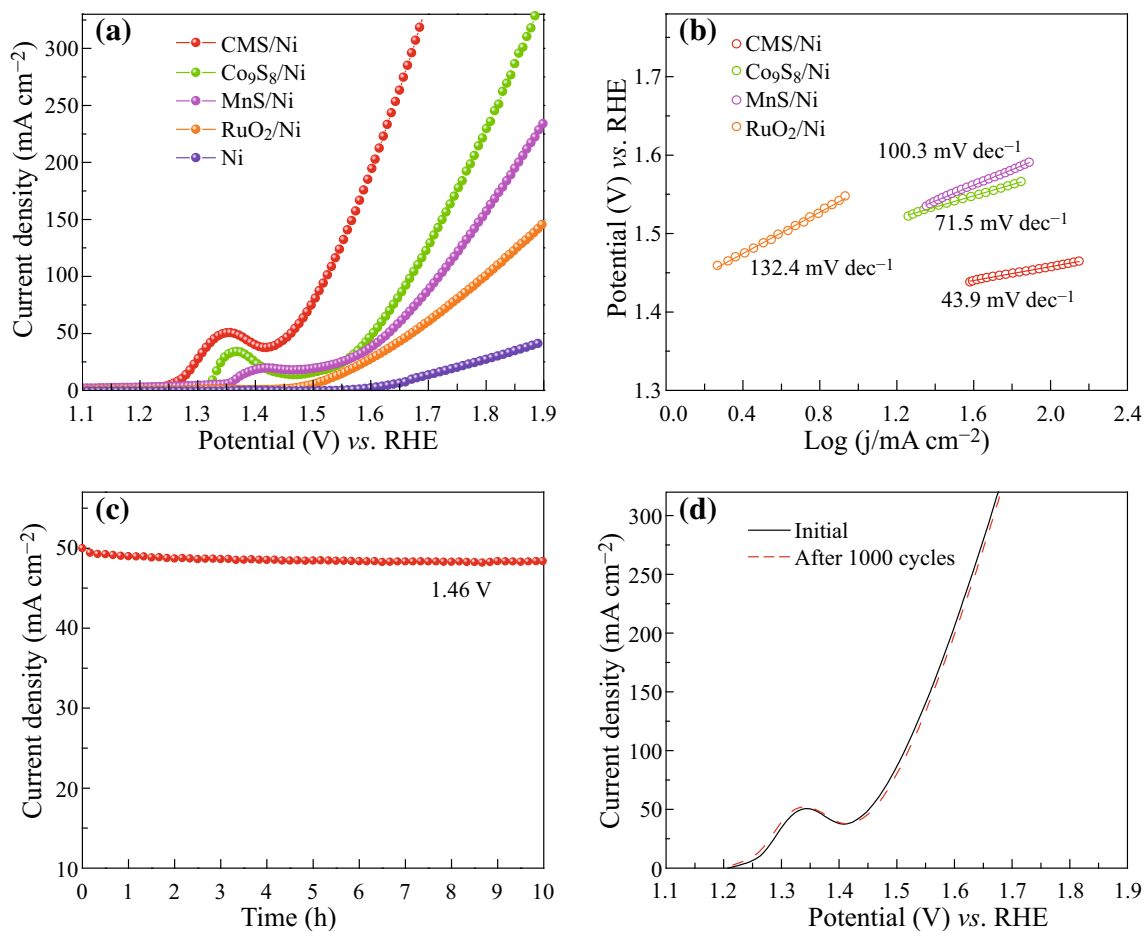
### 3.3 Oxygen Evolution Activity

The electrocatalytic activity of CMS/Ni for OER was also characterized by LSV measurement with a scan rate of  $2 \text{ mV s}^{-1}$ . The  $\text{Co}_9\text{S}_8/\text{Ni}$ ,  $\text{MnS}/\text{Ni}$ , bare Ni foam and

commercial  $\text{RuO}_2$  coated on Ni foam ( $\text{RuO}_2/\text{Ni}$  [30], loading  $4.1 \text{ mg cm}^{-2}$ ) were evaluated for comparison. Figure 3a shows the polarization curves of the as-prepared samples, in which the oxidation peaks from 1.33 to 1.41 V are ascribed to the transition from  $\text{M}^{2+}$  to  $\text{M}^{3+}$  [30]. Owing to the intense oxidation peaks, the overpotential for materials to generate the anodic current of  $10 \text{ mA cm}^{-2}$  for OER is not accurate. Therefore, we report the overpotentials at  $100 \text{ mA cm}^{-2}$  here. As observed, a current density of  $100 \text{ mA cm}^{-2}$  is easily reached at the overpotential of 298 mV for CMS/Ni, better than those of  $\text{Co}_9\text{S}_8/\text{Ni}$  (439 mV),  $\text{MnS}/\text{Ni}$  (492 mV),  $\text{RuO}_2/\text{Ni}$  (568 mV) and even most of the other recently reported bifunctional materials exhibited in Table S1. The above results suggest the efficient OER activity of CMS/Ni, which is also superior to the  $\text{CoMn-LDH}/\text{Ni}$  (Fig. S6c, d). A tremendous improvement of catalytic performance for CMS/Ni with respect to its monometallic counterparts can also be supported by its smallest Tafel slope. As shown in Fig. 3b, the smallest Tafel slopes of  $43.9 \text{ mV dec}^{-1}$  for CMS/Ni with respect to  $\text{Co}_9\text{S}_8/\text{Ni}$  ( $71.5 \text{ mV dec}^{-1}$ ),  $\text{MnS}/\text{Ni}$  ( $100.3 \text{ mV dec}^{-1}$ ) and  $\text{RuO}_2/\text{Ni}$  ( $132.4 \text{ mV dec}^{-1}$ ) imply the great OER kinetics activity.

The stability of CMS/Ni for OER was calculated at a static potential of 1.46 V in  $1.0 \text{ mol L}^{-1}$  KOH solution. Figure 3c presents a negligible decrease in current density after 10 h continuing OER measurement, indicating the superior durability. Additionally, this result can be further confirmed by the LSV curves in Fig. 3d, because the polarization curve after 1000 cycles is similar to the initial cycle. In particular, it can be seen that the morphology of CMS/Ni slightly changes after stability testing in Fig. S9b.

All the above results indicate that the CMS/Ni possesses superior HER and OER catalytic activity comparing to the pure  $\text{Co}_9\text{S}_8/\text{Ni}$  and  $\text{MnS}/\text{Ni}$ , which could be involved the following factors: (1) The obtained CMS/Ni has a uniquely 3D hexagram-like nanosheet structure, which not only provides a large electrochemical active surface areas (ECSAs) to expose more accessible catalytic active sites, but also contacts with the electrolyte efficiently and facilitates the transportation of  $\text{O}_2$ ,  $\text{H}_2$  bubbles [28, 32, 57]; (2) the 3D hexagram-like CMS directly supported on Ni foam substrate enhances the structure stability and improves electrons transport ability of CMS/Ni, which are beneficial to improving the catalytic activity; and (3) the XPS results (shown in Fig. 1g, h) indicate that the different valence states of Co and Mn are presented in the CMS/Ni, which can facilitate the adsorption and desorption of intermediates in the electrocatalysis process. The synergistic effect between Co and Mn is helpful for the catalytic activity [42, 49, 50, 58].

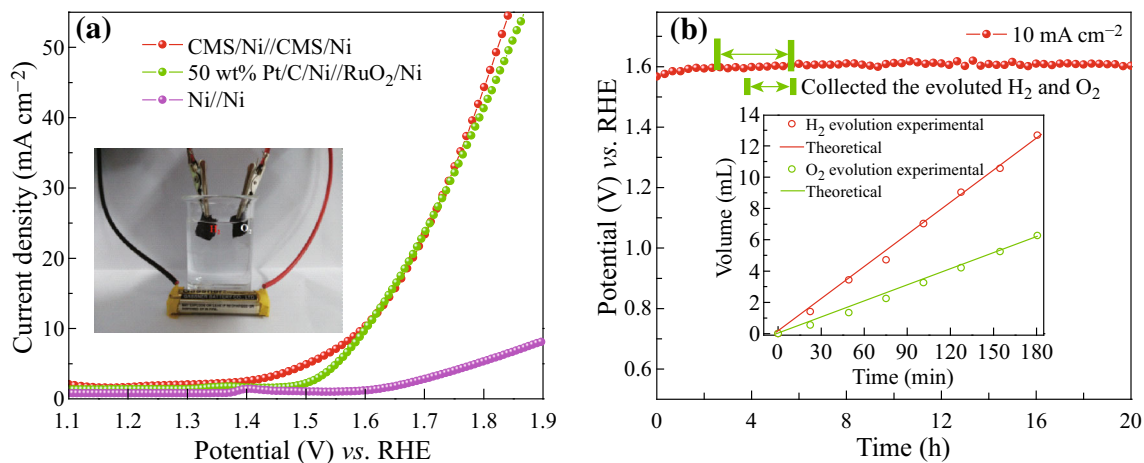


**Fig. 3** **a** Polarization curves, **b** Tafel plots, **c** chronoamperometric curve and **d** polarization curves before and after 1000 CV cycles

### 3.4 Overall Water Splitting

Based on the excellent activity toward OER and HER, the CMS/Ni was directly assembled as the cathodic and anodic

electrodes in two-electrode cell (CMS/Ni//CMS/Ni) in 1.0 mol L<sup>-1</sup> KOH electrolyte for overall water splitting. As shown in Fig. 4a, the LSV curve of the cell indicates that water is electrolyzed only at the voltage of 1.47 V to



**Fig. 4** **a** Polarization curves of CMS/Ni//CMS/Ni, 50 wt Pt/C/Ni//RuO<sub>2</sub>/Ni and Ni//Ni for overall water splitting (the inset is a device for overall water splitting). **b** Chronopotentiometry curve of CMS/Ni//CMS/Ni (the inset is volume of O<sub>2</sub> and H<sub>2</sub> generated for theoretically calculated and experimentally measured versus time)

generate H<sub>2</sub> and O<sub>2</sub> bubbles. This result is supported by a device, which is driven by a 1.50 V dry battery (the inset in Figs. 4a and S10). For comparison, the 50 wt% Pt/C/Ni, RuO<sub>2</sub>/Ni and bare Ni foam were also used as the two-electrode electrolyzers. A current density of 10 mA cm<sup>-2</sup> for CMS/Ni//CMS/Ni can be achieved at a cell voltage of 1.60 V, equaling to that of 1.60 V for 50 wt% Pt/C/Ni//RuO<sub>2</sub>/Ni. Although this cell voltage (1.60 V) of CMS/Ni//CMS/Ni is higher than 1.52 V for Cu@CoS<sub>x</sub>/CF–Cu@CoS<sub>x</sub>/CF [59], 1.45 V for MoO<sub>x</sub>/Ni<sub>3</sub>S<sub>2</sub>/NF//MoO<sub>x</sub>/Ni<sub>3</sub>S<sub>2</sub>/NF [60] and 1.53 V for Ni<sub>x</sub>Co<sub>3-x</sub>S<sub>4</sub>/Ni<sub>3</sub>S<sub>2</sub>/NF//Ni<sub>x</sub>Co<sub>3-x</sub>S<sub>4</sub>/Ni<sub>3</sub>S<sub>2</sub>/NF [61], it is superior to that of NiCo<sub>2</sub>S<sub>4</sub>NW/NF//NiCo<sub>2</sub>S<sub>4</sub>NW/NF (1.68 V) [25], FeNi<sub>3</sub>N/NF//FeNi<sub>3</sub>N/NF (1.62 V) [27], Ni/NiP//Ni/NiP (1.61 V) [30], Co<sub>3</sub>O<sub>4</sub>@Ni//Co<sub>3</sub>O<sub>4</sub>@Ni (1.64 V) [32], Ni<sub>2.5</sub>Co<sub>0.5</sub>Fe/NF//Ni<sub>2.5</sub>Co<sub>0.5</sub>Fe/NF (1.62 V) [35], Ni(OH)<sub>2</sub>/NF//Ni(OH)<sub>2</sub>/NF (1.68 V) [36], Co<sub>1</sub>Mn<sub>1</sub>CH/NF//Co<sub>1</sub>Mn<sub>1</sub>CH/NF (1.68 V) [41], Co(S<sub>0.71</sub>Se<sub>0.29</sub>)<sub>2</sub>//Co(S<sub>0.22</sub>Se<sub>0.78</sub>)<sub>2</sub> (1.63 V) [43] and the two-electrode electrolyzers listed in Table S1. In addition, the chronopotentiometry curve of CMS/Ni//CMS/Ni was measured at a constant current of 10 mA cm<sup>-2</sup> for 20 h, exhibiting a great stability for CMS/Ni//CMS/Ni in Fig. 4b. The volume–time plots for generating O<sub>2</sub> and H<sub>2</sub> suggest that the Faradaic efficiency of CMS/Ni//CMS/Ni electrolyzer is nearly 100% (see the inset in Fig. 4b and experimental details are shown in SI).

## 4 Conclusions

In summary, a simple anion exchange method was employed to successfully prepare 3D hexagram-like CMS/Ni. The 3D hexagram-like CMS/Ni nanosheets have large electrochemical active surface area to expose more active sites and low charge-transfer resistance. Noticeably, the synergetic effect of Co and Mn is also presented in this composite. Consequently, it exhibits superior catalytic activity in basic medium with low overpotentials of 217 mV for HER and 298 mV for OER to reach a current density of 100 mA cm<sup>-2</sup>. More importantly, the assembled CMS/Ni//CMS/Ni device for overall water splitting can be driven by a 1.50 V dry battery, indicating the great potential for practical applications. Therefore, this work provides a scalable method to synthesize bi- or multi-metallic sulfide composites and extends the preparation of the other novel electrocatalysts for water splitting.

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