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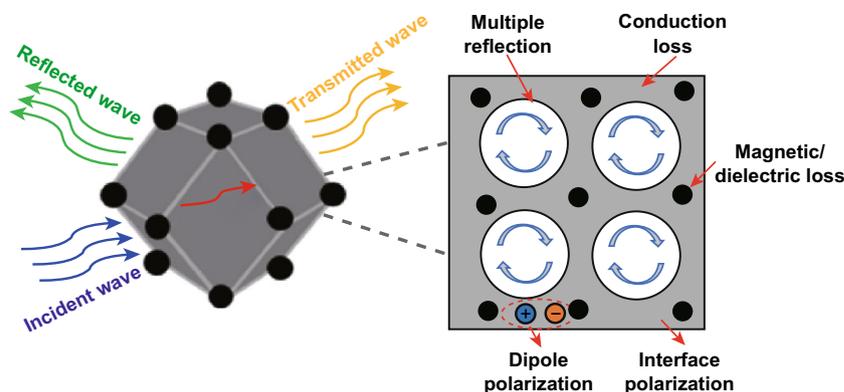
A Review on Metal–Organic Framework-Derived Porous Carbon-Based Novel Microwave Absorption Materials

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HIGHLIGHTS

- The theoretical knowledge in the field of microwave absorption is summarized in detail.
- The recent progress of metal–organic frameworks-derived porous carbon-based nanocomposites as microwave absorption materials is reviewed.

ABSTRACT The development of microwave absorption materials (MAMs) is a considerable important topic because our living space is crowded with electromagnetic wave which threatens human's health. And MAMs are also used in radar stealth for protecting the weapons from being detected. Many nanomaterials were studied as MAMs, but not all of them have the satisfactory performance. Recently, metal–organic frameworks (MOFs) have attracted tremendous attention owing to their tunable chemical structures, diverse properties, large specific surface area and uniform pore distribution. MOF can transform to porous carbon (PC) which is decorated with metal species at appropriate pyrolysis temperature. However, the loss mechanism of pure MOF-derived PC is often relatively simple. In order to further improve the MA performance, the MOFs coupled with other loss materials are a widely studied method. In this review, we summarize the theories of MA, the progress of different MOF-derived PC-based MAMs, tunable chemical structures incorporated with dielectric loss or magnetic loss materials. The different MA performance and mechanisms are discussed in detail. Finally, the shortcomings, challenges and perspectives of MOF-derived PC-based MAMs are also presented. We hope this review could provide a new insight to design and fabricate MOF-derived PC-based MAMs with better fundamental understanding and practical application.



MOF can transform to porous carbon (PC) which is decorated with metal species at appropriate pyrolysis temperature. However, the loss mechanism of pure MOF-derived PC is often relatively simple. In order to further improve the MA performance, the MOFs coupled with other loss materials are a widely studied method. In this review, we summarize the theories of MA, the progress of different MOF-derived PC-based MAMs, tunable chemical structures incorporated with dielectric loss or magnetic loss materials. The different MA performance and mechanisms are discussed in detail. Finally, the shortcomings, challenges and perspectives of MOF-derived PC-based MAMs are also presented. We hope this review could provide a new insight to design and fabricate MOF-derived PC-based MAMs with better fundamental understanding and practical application.

KEYWORDS Metal–organic frameworks; Porous carbon; Microwave absorption material; Reflection loss; Effective absorption bandwidth

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1 Introduction

The rapid development of science and technology made many kinds of electronic devices become irreplaceable role in human's daily life [1–7]. However, the electronic devices make the space rife with electromagnetic waves (EMWs) [8–14]. The EMWs become a new and more hazardous source of pollution as water, air and noise pollution [15–17]. On the one hand, the undesirable EMW may make strong interference to the nearby instruments, causing their malfunctioning and signal interruption. On the other hand, EMW may harm human's health which cause some disease such as cancer and endocrine disorder [18, 19]. Besides, plants can be inactive, variation and even die with the strong EMW radiation [20, 21]. It is urgently need for human to solve the problem of EMW pollution, while in the field of military, many advanced weapons such as warcraft are the key target of the enemy. EMW stealth technology of military equipment is a crucial solution to evade detection and attack. Coating of MAMs on military equipment is an effective anti-detection method [22]. Therefore, the exploration of high-performance MAMs is of great significance in both civil and military fields.

Recently, MAMs have received much attention because they have the ability to attenuate EMW. They can convert EMW into thermal energy or other forms of energy to dissipate [23, 24]. The ideal MAMs are often multiple loss mechanisms and they required to have lightweight, thin thickness, wide absorption bandwidth and strong absorption characteristics [25, 26].

Metal–organic frameworks (MOFs) are a kind of crystalline porous material with periodic network structure, which is composed of inorganic metal center (metal ion or metal cluster) and organic ligand connected by self-assembly [27–30]. Due to large amounts of organic ligands that could be used, MOFs have various of compositions and structures. As we know, more than 20,000 MOFs have been reported so far [31, 32]. MOFs have attracted lots of research interest due to their performance diversity, which are potential to extensive uses in many fields, such as electrochemical energy storage [33, 34], catalysis [35, 36], purification [37, 38] and sensing [39, 40]. Moreover, with MOFs as the precursor, carbon and metal-based compounds can be generated in situ by high-temperature pyrolysis in an inert atmosphere [28, 41, 42]. Fortunately,

the morphologies of MOFs are still well preserved after pyrolysis [43, 44]. The MOF-derived PC nanocomposites also possess the desirable properties from MOFs [45] such as their tunable chemical structures, large specific surface area, uniform pore distribution, diverse morphology and chemical stability, which enable MOF-derived PC to be an ideal candidate for MA.

MOF-derived PC-based nanocomposites have been widely studied in the field of MAMs, but the attenuation mechanism may be relatively simple. In order to improve the attenuation performance, they usually coupled with other lossy materials. Based on the above views, how to design and prepare MOF-derived PC-based MAMs is now a hot research topic [46, 47]. In this review, we summarize the recent progress of several MOF-derived PC-based nanocomposites as MAMs such as Co, Ni, Fe, Zn, Cu, Ti, Zr and rare-earth (RE) MOF-derived PC-based nanocomposites. Besides the pure MOF, multi-metal MOF or tunable chemical composition incorporated with other loss material had also been fabricated as MAMs. Furthermore, MAMs with different morphology had been reviewed. Finally, we put forward some personal insights into the current status and perspectives in the future research direction.

2 Theories of Microwave Absorption

When the incident EMW contacts with the surface of the MAMs, as shown in Fig. 1, three situations may happen. Part of the incident EMW reflects on the surface of the MAMs (reflected EMW), part of it goes to the interior of the MAMs and absorbed by the MAMs (adsorbed EMW), and the rest of the EMW goes through the MAMs (transmitted EMW) [48]. When designing MAMs, researchers expect the incident EMW to be dissipated as much as possible inside the MAMs to reduce reflected EMW and transmitted EMW. Therefore, a good MAM usually needs to meet two conditions: good impedance matching and strong EMW attenuation ability [49, 50]. The good impedance matching requires incident EMW goes into the MAMs as much as possible and reduces the reflection on the material surface [51]. The ideal impedance matching requires that the complex permittivity ($\epsilon_r = \epsilon' - j\epsilon''$) is equal to the complex permeability ($\mu_r = \mu' - j\mu''$). In the formula, ϵ' and μ' represent the ability to store electrical and magnetic energy, while ϵ'' and μ'' refer to the loss of electrical and magnetic energy [52, 53].

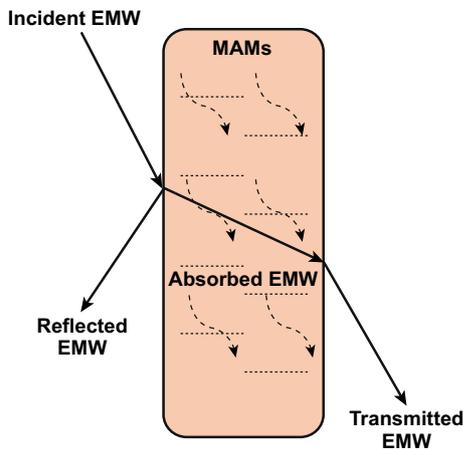


Fig. 1 Schematic diagram of interaction between MAMs and micro-waves

It is not easy to meet this requirement. We can artificially adjust electromagnetic parameters (ϵ and μ) to improve the impedance matching. The EMW attenuation ability means the loss capacity of the EMW which enters into the interior of MAMs [24, 54, 55].

Reflection loss (RL) is often used to evaluate the EMW absorption ability. For example, the RL value of -10 dB is comparable to 90% of MA, and the RL value of -20 dB is comparable to 99% of MA. The RL is calculated by the transmission line theory, which is shown as follows [56–60]:

$$Z = \left| Z_{in}/Z_0 \right| = \left| \sqrt{u_r/\epsilon_r} \tanh \left[j(2\pi fd/c \sqrt{\epsilon_r u_r}) \right] \right| \quad (1)$$

$$R_L = 20 \log \left| (Z_{in} - Z_0)/(Z_{in} + Z_0) \right| \quad (2)$$

Z_0 is the impedance of free space (377Ω), Z_{in} is the input impedance of MAMs, f is the frequency of EMW, d is the thickness of the MAMs, and c is velocity of light, respectively. When $Z = 1$, the wave impedance of the MAMs is exactly the same as that of the free space. The incident EMW can enter the MAMs completely without reflected wave. Therefore, $Z = 1$ is an ideal situation. When the value of Z is equal or close to 1, it is beneficial for improving MA ability [61, 62]. The delta-function is another method to evaluate the EM impedance matching degree. The equation is shown as follows [63–65]:

$$|\Delta| = \left| \sinh^2(Kfd) - M \right| \quad (3)$$

$$K = \frac{4\pi \sqrt{\mu'_r \epsilon'_r} \times \sin \left(\frac{\delta_e + \delta_m}{2} \right)}{c \times \cos \delta_e \times \cos \delta_m} \quad (4)$$

$$M = \frac{4\mu'_r \epsilon'_r \cos \delta_e \times \cos \delta_m \sqrt{\mu'_r \epsilon'_r} \times \sin \left(\frac{\delta_e + \delta_m}{2} \right) 4\mu'_r \epsilon'_r \cos \delta_e \times \cos \delta_m \sqrt{\mu'_r \epsilon'_r} \times \sin \left(\frac{\delta_e + \delta_m}{2} \right)}{(\mu'_r \cos \delta_e - \epsilon'_r \cos \delta_m)^2 + \left[\tan \left(\frac{\delta_e - \delta_m}{2} \right) \right]^2 (\mu'_r \cos \delta_e + \epsilon'_r \cos \delta_m)^2} \quad (5)$$

The small delta value and close to zero indicate good impedance matching. If $|\Delta|$ tends to far away from zero, it gives poor microwave absorption.

The MAMs can be roughly divided into three types according to their loss mechanisms including dielectric loss materials, magnetic loss materials and multiple loss materials, as shown in Table 1. Dielectric loss materials are represented by carbon materials [66, 67], non-magnetic metal powder [68, 69], polymers [70, 71], non-magnetic metal oxides [72, 73], non-oxygen ceramics [74, 75] and so on. They possess features such as high strength, resistance to high temperature, excellent electrical conductivity and low density, but effective absorption bandwidth (EAB) and the MA performance may be not sufficient [76, 77]. Magnetic loss materials are represented by magnetic metal powder and compounds [78, 79], ferrite [80–87], carbonyl iron [88–90] and so on. However, their high density and poor stability limit their practical application [91]. The dielectric and magnetic loss factors, defined as $\tan \delta_E = \epsilon''/\epsilon'$ and $\tan \delta_M = \mu''/\mu'$, are suggested to evaluate on dielectric and magnetic losses [92–94].

The dielectric loss ability mainly stems from electrical conductivity loss and polarization relaxation loss [95, 96]. The electrical conductivity loss is that when the EMWs enters into the MAMs, the charge carriers would form a current under the action of the electric field, and then, the electric energy converts to the thermal energy or other form of energy and dissipated out [97, 98], thus increasing EMW attenuation. However, if the conductivity is too high, the incident EMW will be reflected by a large amount, resulting in impedance mismatch and poor EMW attenuation. The polarization relaxation loss is split into ionic polarization, electronic polarization, dipoles relaxation polarization and interfacial polarization (spatial polarization) [55, 99]. Ion polarization is caused by the relative displacement of cations and anions. Electron polarization is caused by position change of the constituent atoms relative to the nucleus, and thus, the dipole moment is generated. Ion polarization and electron polarization usually occur in the frequency range of

Table 1 Classification table of common MAMs

Types of MAMs	Typical materials	Loss mechanisms
Dielectric loss material	Carbon materials, non-magnetic metal powder, polymer, non-magnetic metal oxides, non-oxygen ceramics, etc.	Electrical conductivity loss, polarization relaxation loss (dipoles relaxation polarization and interfacial polarization)
Magnetic loss material	Magnetic metals and compounds, ferrite, carbonyl iron, etc.	Hysteresis loss, eddy current loss and residual loss
Multiple loss material	Combination of the above	Multiple loss

ultraviolet, visible and infrared light, which is much higher than the microwave frequency range (2–18 GHz), so they are excluded [100, 101]. Dipoles relaxation polarization refers to the polarization caused by the rotation of the dipole moment in the direction of the electric field, and it can greatly influence the dielectric loss [41, 102]. The relaxation loss can be analyzed by Debye equation [103–105]:

$$\varepsilon' = \varepsilon_{\infty} + (\varepsilon_S - \varepsilon_{\infty}) \frac{1}{1 + \omega^2 \tau^2} \quad (6)$$

$$\varepsilon'' = (\varepsilon_S - \varepsilon_{\infty}) \frac{\omega \tau}{1 + \omega^2 \tau^2} \quad (7)$$

We can deduce an equation from Eqs. (3) and (4) as follows:

$$\left(\varepsilon' - \frac{\varepsilon_S + \varepsilon_{\infty}}{2} \right)^2 + (\varepsilon'')^2 = \left(\frac{\varepsilon_S - \varepsilon_{\infty}}{2} \right)^2 \quad (8)$$

where ε_S is the static dielectric constant, ε_{∞} is the dielectric constant of infinite frequency, and τ is the time of relaxation. The circle of this equation is called Cole–Cole semicircle [106, 107]. Each Cole–Cole semicircle represents a polarization relaxation process [108, 109]. The points on the semicircle correspond to the values of the real and imaginary parts of the dielectric constant at a certain frequency calculated by the Debye equation. Interfacial polarization usually appears at the interface of heterogeneous medium, which is caused by the accumulation of electrons or ions at the interface under the action of the external electric field [110]. Generally speaking, dielectric materials can be wideband absorption. However, the disadvantage is that the low-frequency absorption effect is poor and it is difficult to achieve the thin coating wideband absorption.

Magnetic loss refers to the phenomenon that the work is done by the outside world to a magnetic material and then the work is converted into heat during the process of magnetization or demagnetization [111]. It includes hysteresis loss, eddy current loss and residual loss [112, 113]. The hysteresis loss is due to the hysteresis loop relationship

between the magnetic perceptual strength and the magnetic field strength. Normally, the hysteresis loss often occurred in the weak field can be excluded [114]. When a conductor moves in an inhomogeneous magnetic field or is in a time-varying magnetic field, the energy loss caused by the induced current in the conductor is called eddy current loss. It can be defined as [115–117]:

$$C_0 = \frac{\mu''}{(\mu')^2 f} = \frac{2}{3} \pi \mu^0 \delta d^2 \quad (9)$$

where δ is the electrical conductivity of material, and d is the thickness of the MAMs. From the equation, C_0 is a constant at a certain thickness of the MAMs with the change of frequency. This is one of the ways to determine whether EMW loss only results from the eddy current loss [118]. The residual loss refers to other losses except hysteresis loss and eddy current loss [77, 101].

The multiple loss material is not just a single loss mechanism, and it combined the advantages of various losses.

We all know that the synergistic effects between the dielectric loss and magnetic loss contribute to the excellent EMW absorption ability, which result in the good impedance matching and strong EM wave attenuation of the MAMs. However, the conflict between the two sides is still exist. In order to get the EMW into the material as much as possible, it will inevitably reduce the attenuation ability of the MAMs to the EMW. Therefore, it is necessary to coordinate impedance matching and EMW attenuation in practical application. The attenuation constant α can be defined as [119, 120]:

$$\alpha = \frac{\sqrt{2}\pi f}{c} \sqrt{(\mu''\varepsilon'' - \mu'\varepsilon') + \sqrt{(\mu''\varepsilon'' - \mu'\varepsilon')^2 + (\mu''\varepsilon'' + \mu'\varepsilon')^2}} \quad (10)$$

Dissipation of reflected EMW through the interference of MAMs is another key factor to attenuate the EMW, which can be described by the quarter-wavelength matching mechanism ($\lambda/4$) as follows [46, 109]:

$$d_m = \frac{n\lambda}{4} = \frac{nc}{4f_m \sqrt{|\mu_r||\epsilon_r|}} \quad (n = 1, 3, 5) \quad (11)$$

In the formula, λ is the wavelength of EMW, d_m and f_m are the thickness and corresponding frequency of maximum RL values, and $|\mu_r|$ and $|\epsilon_r|$ are the modulus of complex permeability and permittivity at f_m , respectively.

3 MOF-Derived PC-Based Nanocomposites as MAMs

As we know, MOFs are composed of inorganic metal center (metal ion or metal cluster) and organic ligand [121, 122]. Through direct pyrolysis of MOFs, they can be converted into metal-doped carbon, and the structure does not change significantly. We can simple and fast synthesis of MAMs by pyrolysis. However, the pure MOF-derived PC-based nanocomposites have not been functionalized, the absorption loss mechanism is simple and MA performance may be not exciting. By incorporated with materials with different absorbing loss mechanisms, impedance matching can be improved and the MA performance can be enhanced.

3.1 Magnetic Single-Metal MOF-Derived PC-Based Nanocomposites as MAMs

Common magnetic metals are Fe, Co and Ni. They often act as inorganic metal centers to synthesize MOF. When they are directly pyrolyzed, the MA performance may not be very good because the simple loss mechanism may lead to impedance mismatch. They often coupled with dielectric loss material to improve impedance matching and rational design on the microstructure to introduce multiple loss mechanisms.

3.1.1 Co-MOF-Derived PC-Based Nanocomposites as MAMs

The most widely studied MOF-derived PC-based nanocomposites as MAMs are the Co-MOF. The typical Co-MOF is ZIF (zeolitic imidazolate framework)-67, which is prepared through the self-assembly of Co^{2+} and 2-methylimidazole. In Kuang's work, they pyrolyze Co-based MOFs (Co-MOF,

ZIF-67) to synthesize porous Co/C composite under inert atmosphere with different pyrolysis temperature [123]. The morphology before and after sintering has not changed much, just the surface was wrinkled, as shown in Fig. 2. The sample pyrolysis at 500 °C shows better MA performance. The maximum RL of Co/C-500 reached -35.3 dB at 5.8 GHz with a thickness of 4 mm, and the EAB was 5.80 GHz (8.40–14.20 GHz) corresponding to a thickness of 2.5 mm. The magnetic loss of Co, the large dielectric loss value of carbon and the porous structure result in the MA performance, but the RL and EAB are not very satisfactory.

Since the tunable of inorganic metal center and organic ligand, MOFs have various of compositions and structures. Co-MOF-derived PC-based nanocomposites as MAMs have been reported by many groups. Kong synthesized Co/C pyrolysis from the cubic $[\text{Co}(\text{INA})_2]$ MOF by isonicotinic acid as organic ligand [124]. Wang's group constructed Co/C composites via pyrolysis a new Co-based MOF named $[\text{Co}_2\text{O}(\text{cptpy})_2(\text{DMF})]$ (CPT-1-Co, $\text{Hcptpy} = 4'-(4\text{-carboxyphenyl})-4,2':6',4''\text{-terpyridine}$, $\text{DMF} = \text{N,N-dimethylformamide}$), by reacting a multi-dentate ligand, $4'-(4\text{-carboxyphenyl})-4,2':6',4''\text{-terpyridine}$ (Hcptpy), with $\text{Co}(\text{OAc})_2$ salt [125]. More works of Co/C synthesized from pure Co-MOF have also been reported [126, 127]. However, the MA performance of pure Co-MOF-derived PC-based nanocomposites is not quite satisfying. The reason may be the low relatively complex permittivity, and high relatively complex permeability leads to poor impedance matching performance, thus limiting their application in MAMs. So, Co-MOF coupled with other loss material, especially dielectric loss material, is a better solution to improve the MA performance.

When they coupled with MOF-derived PC, they would show unexpected performance. The MOF-derived PC/dielectric loss material nanocomposites have many advantages, such as low cost, easy preparation and low density. Moreover, the additional loss mechanisms are created, and electrical loss, polarization loss, the interfacial and multiple scattering may lead to the good MA performance.

Carbon materials such as graphene, carbon nanotube (CNT) and carbon nanofiber (CNF) have aroused wide attention as MAMs due to their excellent physical and chemical properties, including their lightweight, high specific surface area, mechanical strength, thermal stability, corrosion resistance, electric conductivity and dielectric properties. Graphene, composed of sp^2 -bonded carbon atoms, has a lot of advantages such as electrical, thermal

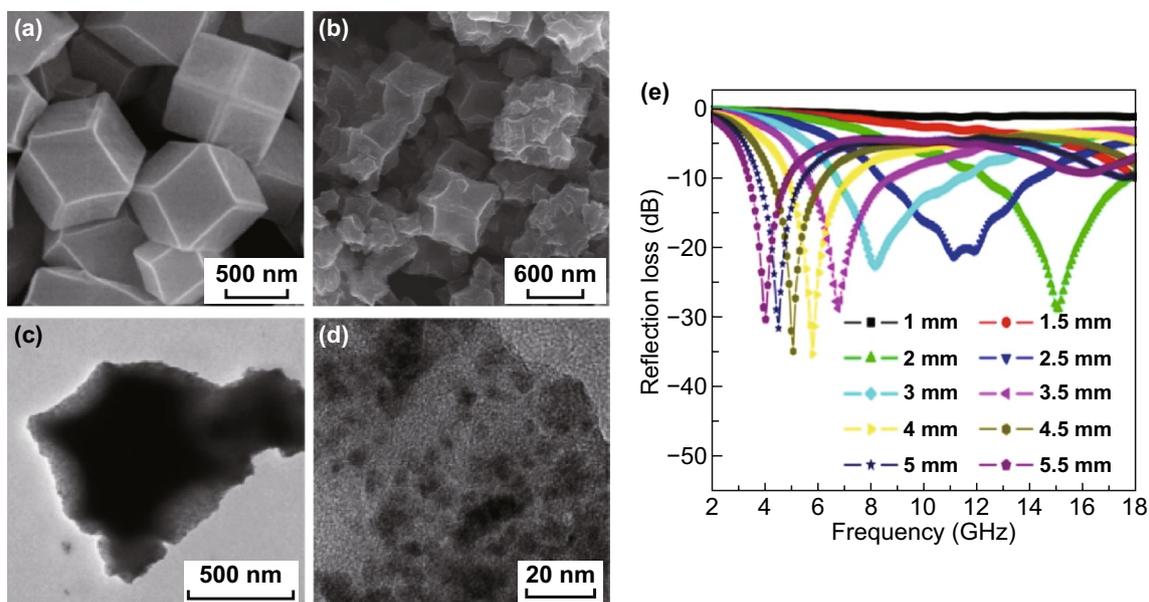


Fig. 2 **a, b** SEM images of Co/C before and after calcination at 500 °C under Ar for 5 h. **c, d** Low and high magnification TEM images of Co/C-500. **e** Calculated results of the reflection loss vs frequency for Co/C-500 with different thicknesses. Reprinted with permission from Ref. [123]

and mechanical properties. Graphene has functional groups and some defects on its surface. The impedance matching of graphene can be improved, and the dipole polarization relaxation can also be generated to improve the MA performance. Graphene can also form a multilayer structure, which can increase the number of reflections and propagation distance of EMW. Dong had synthesized MOF/RGO hybrids by two steps including in situ growth of Co-based MOF on GO nanosheets and a controlled calcination process [128], which is shown in Fig. 3. The maximum *RL* of the sample reached -52 dB at 9.6 GHz with a thickness of 4.1 mm. The *EAB* of this MOF/RGO hybrid can reach 7.72 GHz only under a thickness of 3.2 mm, which surpasses most reported MOF and RGO-based MAMs. It is worthy to point out that the enhanced effect of MOF/RGO interface, improved match between dielectric loss and magnetic loss should be considered as the factor of the high MA performance.

Chen's group synthesized CoC-rGO obtained by calcination of ZIF-67-GO hybrids [129]. The introduction of high conductivity of rGO may lead to strong eddy current loss and reduce the permeability. The impedance matching is determined by the additive amount of rGO. The RL_{\max} value reached up to -44.77 dB at thickness of 2.1 mm, and

the *EAB* reached 5.2 GHz at thickness of 1.8 mm, which showed evident advantages compared to CoC or rGO alone with single loss mechanism. Zhang's group synthesized MOF-derived carbonaceous $\text{Co}_3\text{O}_4/\text{Co}/\text{RGO}$ composite at 600 °C in Ar [130]. The sample displays $RL_{\max} - 52.8$ dB at 13.12 GHz with thickness which is only 2.0 mm. The *EAB* is up to 10.72 GHz in the thicknesses range of 2.0–4.0 mm. Since the unique porous structure, the dielectric and magnetic tangent losses of the sample are in the middle levels, it is beneficial for the impedance match and thus results in the higher MA performances. CNTs can be divided into single-walled carbon nanotubes (SCNTs) and multi-walled carbon nanotubes (MCNTs). They have a very large aspect ratio, so a conductive network can be formed. The dielectric constant is large, and its permeability is small, so its impedance matching is poor. Therefore, it is often combined with other magnetic loss materials to improve the MA performance. Dong et al. had prepared a 3D Co/C-MCNTs hybrid network using MCNTs as wires and Co-based MOFs as junctions [131]. The multiple components synergistic effect leads to the good MA performance. The purpose of introducing MCNTs may promote the formation of a conductive network, increasing interfacial polarization. The RL_{\max} of the sample Co/C-MCNTs is -33.4 dB at the frequency

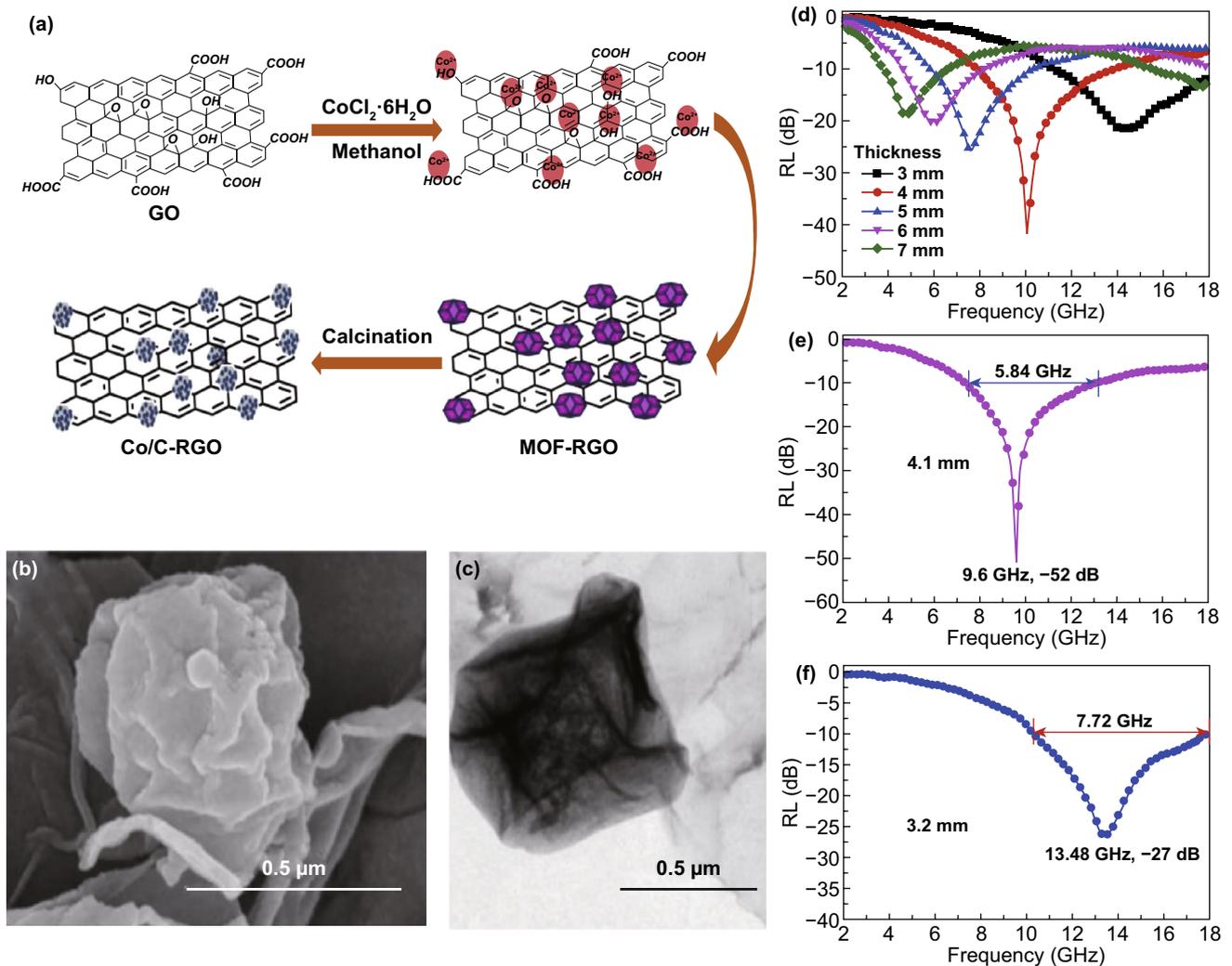


Fig. 3 a Process for the preparation of MOF/RGO. b, c SEM and TEM images of heat-treated MOF/GO hybrid at 500 °C for 6 h. d–f RL curves of MOF/GO-500 and under 3.2 and 4.1 mm with a filler loading ratio of 6 wt% in wax. Reproduced with permission from Ref. [128]

of 3.6 GHz with the thickness of 6 mm. And the EAB is 4.08 GHz at 1.8 mm. Yu had also fabricated Co–C/MWCNTs composites, and the RL_{max} is -48.9 dB at 2.99 mm [132]. Tan synthesized ultra-small Co/CNTs nanohybrid via the pyrolysis of ZIF-67 and (catalytic chemical vapor deposition) CCVD method. It achieves the RL_{max} of -49.16 dB and the in EAB of 4.2 GHz (12.4–16.6 GHz) [133]. Chen synthesized MWCNTs@carbonaceous CoO composites with good MA properties [134]. When the annealing temperature is 500 and 600 °C, the carbonaceous Co_3O_4 can be obtained. When the annealing temperature is 700 °C, Co_3O_4 was all reduced to CoO. The value of RL_{max} is up to -50.2 dB with 1.84 mm thickness. CNF has high dielectric constant, so

the impedance match may be not very good, and we usually improve the MA performance by combining with magnetic loss materials. Zhang et al. reported necklace-like CNFs@MOF-based carbonaceous Co/CoO composite, which was synthesized by wet and pyrolysis method [135]. The optimum RL value is -53.1 dB at 6.56 GHz with the thickness 3.54 mm, and EAB is up to 13.52 GHz with the thickness range of 2.0–5.0 mm. The unique structure will form many defects, which can generate much interfacial polarization, which lead to more dielectric loss. The small-sized nanoparticle can improve dipole polarization. And impedance matching is also optimized to improve the MA performance.

Polymer has the advantages of low density, anti-corrosion and adjustable conductivity, and the electric conductivity and dielectric constant are high. So, it has shown promising prospect in the field of MAMs. In Wang's report, a chain-like PPy (Polypyrrole) aerogel decorated with MOF-based nanoporous Co/C (Co/C@PPy) has been successfully prepared by a self-assembled polymerization method [136]. The composite Co/C@PPy can reach the optimal RL value of -44.76 dB at 17.32 GHz with the thickness of 2.0 mm. And the EAB of 6.56 GHz (11.04–17.60 GHz) is achieved with the thickness of 2.5 mm. The performance is attributed to a proper impedance matching and a high dielectric loss highly enhanced by the PPy aerogel. Besides, the unique chain-like PPy aerogel and the porous feature of Co/C itself can induce more multiple reflection and scattering of EMW.

Non-magnetic metal oxide is a common dielectric loss material and coupled with magnetic loss material to regulate impedance matching which is an effective strategy to solve the absorption problems. Zinc oxide (ZnO), as an important semiconductor with a wide band gap, has been extensively investigated as MAMs, due to its excellent dielectric properties and lightweight [137]. Hu's group constructed a novel 3D hetero-structured Co/NPC@ZnO/rGO by the direct pyrolysis of ZIF-67@ZnO NPs wrapped on rGO nanosheets [138]. ZnO can be utilized to regulate the complex permittivity over the measured frequency range and upgrade the impedance matching property of the sample. The RL_{\max} can reach up to -45.4 dB at only 2 mm, and the EAB achieved 5.4 GHz (from 11.9 to 17.3 GHz). Vanadium sesquioxide (V_2O_3), with relatively high electrical conductivity at room temperature and superior dielectric loss, is usually used in the field of MAMs. Yan's group designed and synthesized Co/C@ V_2O_3 hollow spheres with an RL of -40.1 dB and the EAB of 4.64 GHz at a small thickness of only 1.5 mm [139]. The sample exhibits both excellent impedance matching and light weight due to the rational combination of hollow V_2O_3 spheres and porous Co/C. One-dimensional chain-like MnO@Co/C composite derived from MnO_x @ZIF-67 has also been reported [140]. Similar work was also reported by Co/N/C@ MnO_2 sample [141]. Hierarchical MnO_2 sheets are used to decrease the excessive complex permittivity of Co/N/C and improving impedance match, and polydopamine (PDA) is carbon source. The sample with a filler loading of 15 wt% shows the RL_{\max} of -58.9 dB and EAB of 5.5 GHz. The excellent MA performance of Co/N/C@ MnO_2 composites is attributed to synergistic effects of excellent impedance

match, and dramatical EM attenuation ability arises from multiple helpful constituents, abundant interfaces and extraordinary hollow structure.

Non-oxygen ceramics presents high strength, good thermal stability and chemical resistivity. But the MA performance is not very good. In order to improve the MA performance, we usually combine non-oxygen ceramics with magnetic loss materials. Dong constructed kebab-like nanocomposites composed of SiC stringing polyhedral Co-MOF [142]. The excellent MA performance attributed to the reduced dielectric constant, enlarged aspect ratio and enhanced interface polarization. Under the thickness of 3 mm, the RL_{\max} attained -47 dB at the frequency of 9.32 GHz, and the EAB achieved 5.92 GHz in a frequency range 12.08–18 GHz with the sample thickness of 2.0 mm.

Rational design on the microstructure of MOFs-derived PC-based nanocomposites is an effective strategy to prepare high-performance MAMs, through designing some special structures such as foam structure, core-shell structure and hollow structure, which improve the multiple reflections and interfacial polarization so as to enhance MA performance. Zheng's group successfully synthesized NRGO (nitrogen-doped)/MWCNT composite foams by hydrothermal and high-temperature calcination strategy [143]. The 3D networks were well constructed by overlapped flaky RGO in the composite foams, and the calcination temperature showed notable effects on the micromorphology. The RL_{\max} is -69.6 dB at 12.5 GHz, and EAB achieved 4.3 GHz (13.2–17.5 GHz) at a low thickness of 1.5 mm, as shown in Fig. 4. The excellent MA performance of the foams was derived from a well-constructed 3D network structure, nitrogen doping, polarization relaxation and conduction loss. Wu and his workmates designed a 3D hybrid carbon sponge composite with a hierarchical micro/nanostructure and hollow skeleton [144]. The conductive network, diverse interface, porous and tubular structures, as well as the synergistic effect between metallic Co nanocrystals and carbon species, resulting in the RL_{\max} is -51.2 dB with a ultrathin thickness of 1.6 mm and EAB is 5.4 GHz. MOFs-derived hollow Co/C microspheres were produced by Li's group [145]. They use cetyltrimethylammonium bromide (CTAB) as self-sacrificing template to make hollow microstructures well preserved in the resultant carbon matrix. The hollow microstructures improve the dielectric loss, magnetic loss and enhancing attenuation ability. So rational design on the microstructure of

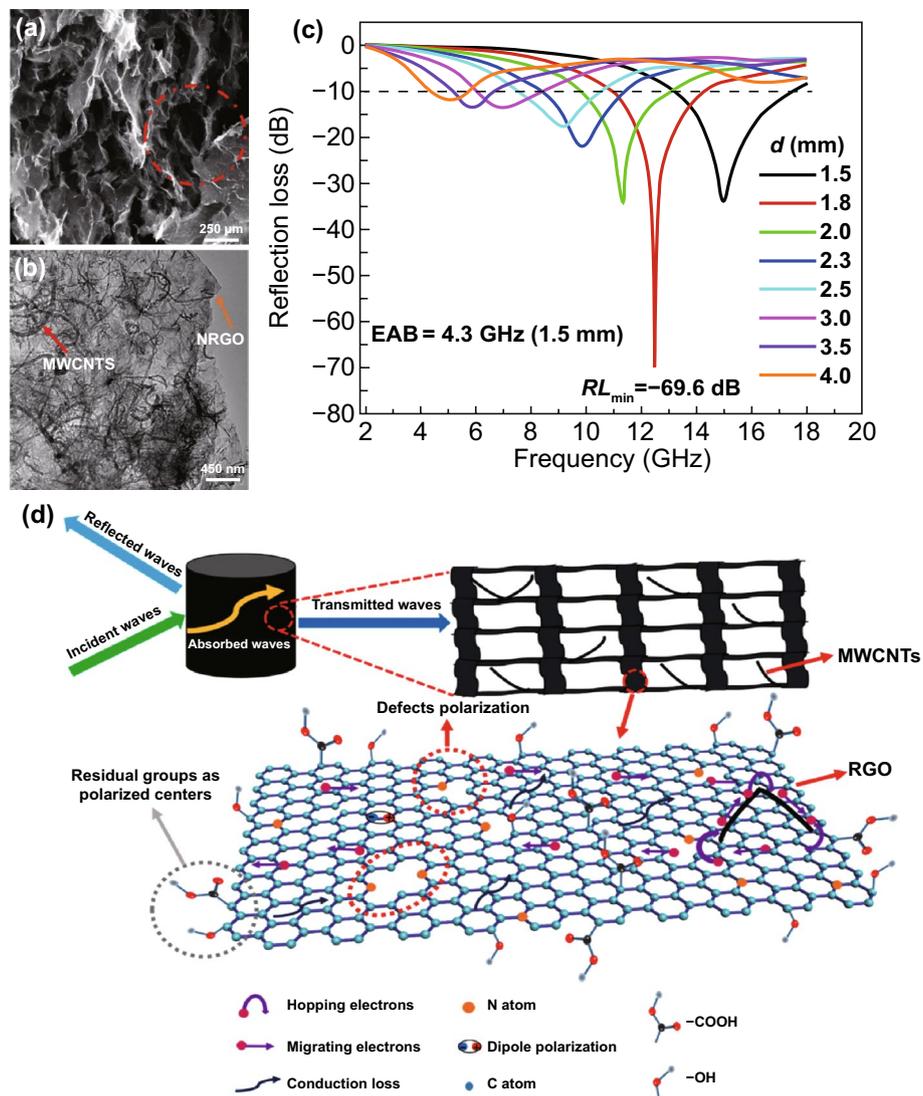


Fig. 4 a, b SEM and TEM images of the samples. c RL-f curves of the sample. d Schematic diagram of the EM absorbing mechanisms of NRGO/MWCNT composite foams. Reprinted with permission from Ref. [143]

MOFs-derived PC nanocomposites is an effective strategy to develop high-performance MAMs.

3.1.2 Ni-MOF-Derived PC-Based Nanocomposites as MAMs

Magnetic metal Ni-based MOF is also often used in MAMs. Zou and his workmates synthesized Ni@C composites by thermal decomposition of pure Ni-MOF [146]. The sample which calcinated at temperature 800 °C showed a RL_{max} of -55.7 dB, and the EAB is 6.0 GHz at a thickness of 1.85 mm. The excellent MA performance is related to the

hollow structure and the synergistic effect between carbon and nickel nanoparticles. Ji had fabricated Ni nanoparticles-embedded nanoporous carbon (NPC/Ni), and the sample prepared at 700 °C exhibits nice MA performance with the RL_{max} value of -39.4 dB and EAB of 4.2 GHz [147]. In Yang’s report, they synthesized Ni-based MOF hollow spheres with various surface morphologies via a simple hydrothermal method [148]. The surface morphologies are controlled by the hydrothermal reaction time. The surface morphologies are smooth, hair-like and rod-like corresponding to the reaction time which is 6, 8, and 10 h, respectively.

The RL_{\max} of the 10 h sample reached -58 dB at 6 GHz with a thickness of 1.5 mm, and the EAB was 6.2 GHz (5–11.2 GHz) with a thickness of 4.6 mm, as shown in Fig. 5. The difference in the surface morphologies results in variation of the magnetic anisotropy, which leads to multi-resonance behavior of the permeability. Liu's group synthesized two kinds of Ni@C derived from the Ni-based MOFs with two kinds of organic ligands (dimethylimidazole as a ligand named as Ni-ZIF and trimesic acid as a ligand named as Ni-BTC) [149]. The RL_{\max} of the Ni@C-ZIF microspheres is -86.8 dB with the thickness of 2.7 mm, and the EAB was 7.4 GHz (4–11.4 GHz) with the thickness ranging from 1.5 to 4.0 mm. The impedance matching, multiple reflection, interfacial polarization among Ni and C and the N-doping were beneficial to the excellent MA performance.

The dielectric loss material is also usually introduced to Ni-MOF to improve impedance matching. Chen's group fabricated multi-component composite SiC/Ni/NiO/C by

annealing SiC NPs and the Ni-MOF in argon [150]. The maximum RL is -50.52 dB at 13 GHz for a film thickness of 4.0 mm, and the EAB is 2.96 GHz (14.76–17.72 GHz) with thickness of 2.5 mm. The high permittivity of the SiC/Ni/NiO/C nanocomposites is expected to enhance absorption of EMW, as shown in Fig. 6. The excellent MA performance also stems from the multi-interface structure which provides interfacial polarization and plasmon resonance.

Structure design is also very important to improve the MA performance of Ni-MOF-derived PC. Du synthesized hierarchical yolk-shell nanostructure (NiO/Ni/GN@Air@NiO/Ni/GN) derived from Ni-MOF by solvothermal reactions [151]. This special structure can effectively enhance the MA performance. And it can also tune the dielectric properties of the NiO/Ni/GN@Air@NiO/Ni/GN composites to achieve good impedance matching. The RL_{\max} of -34.5 dB is obtained at 17.2 GHz with the thin thickness of 1.7 mm. And the EAB can be obtained in the frequency

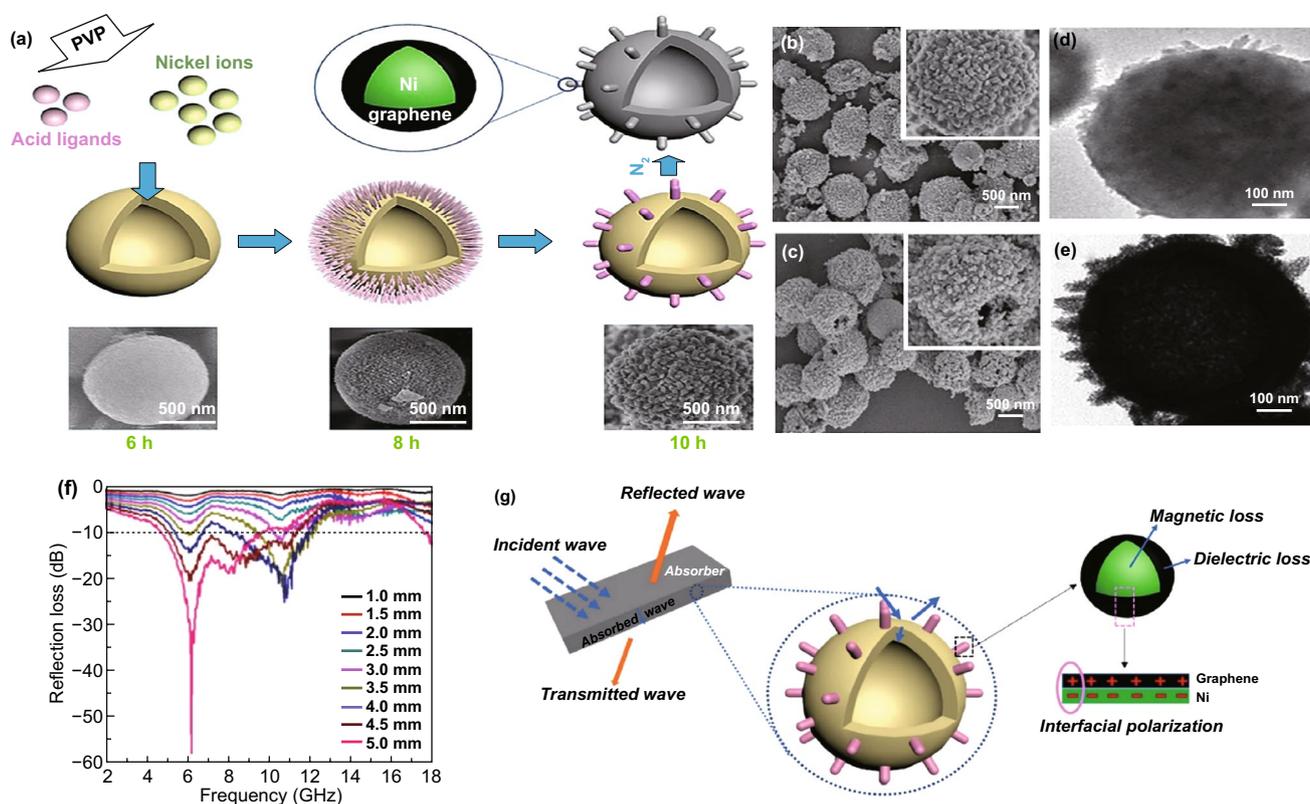


Fig. 5 a Illustration for the formation of Ni-MOF hollow spheres with controllable surface architecture. b, c SEM images and d, e TEM images of Ni-MOFs samples with 10 h before and after annealing at 600 °C. f Electromagnetic wave reflection losses of the Ni-MOF sample with different reaction times. g Absorbing mechanism of as-prepared samples. Reprinted with permission from Ref. [148]

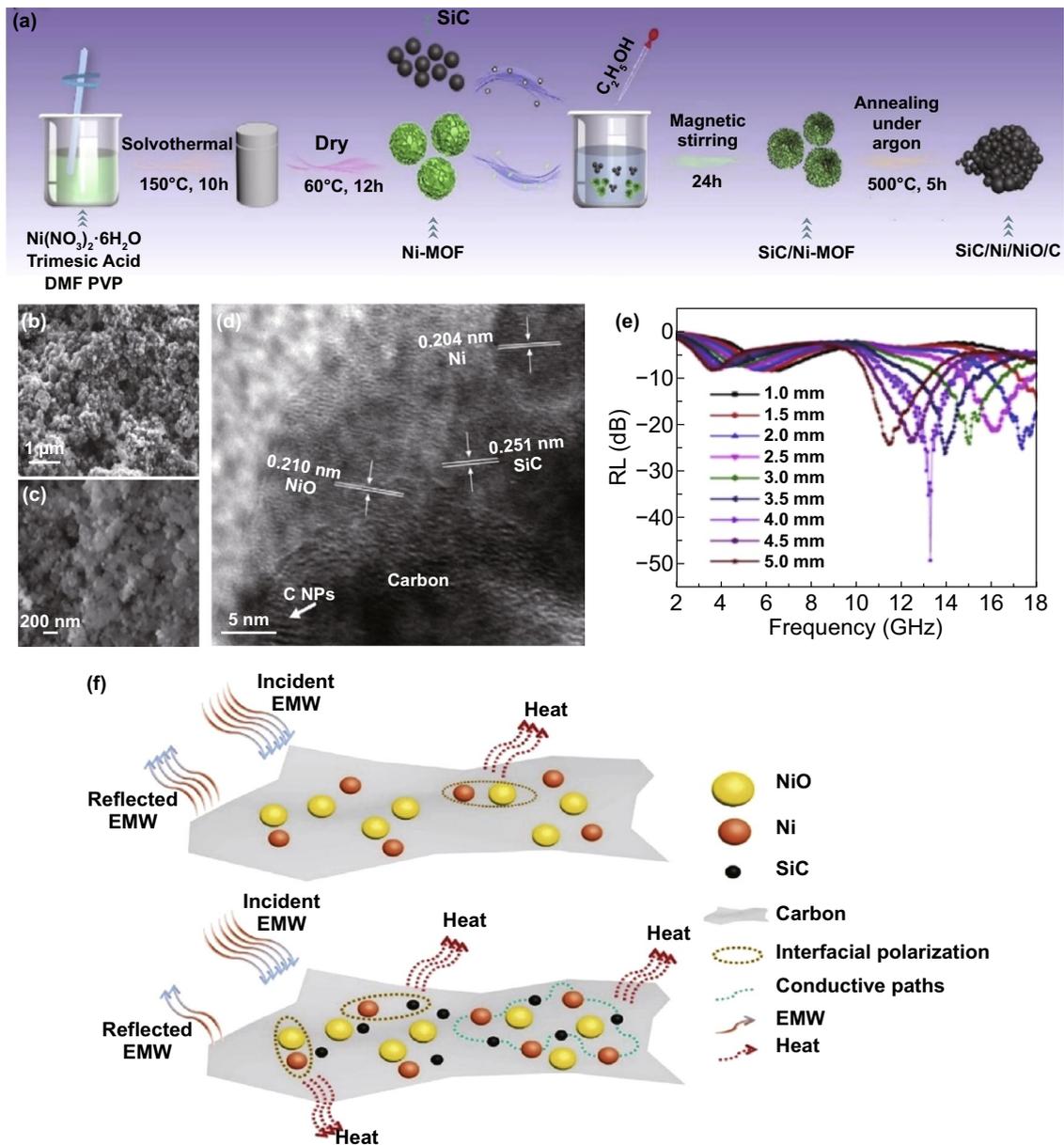


Fig. 6 **a** Preparation of the SiC/Ni/NiO/C nanocomposites. **b**, **c** SEM images and **d** HRTEM image of the SiC/Ni/NiO/C nanocomposites. **e** RL curves of SiC/Ni/NiO/C with a filler loading of 20 wt% in paraffin. **f** Schematic of the electromagnetic wave (EMW) attenuation mechanisms of Ni/NiO/C and SiC/Ni/NiO/C nanocomposites. Reprinted with permission from Ref. [150]

range 7.8–18 GHz with absorber thicknesses of 1.7–5.0 mm. The 3D porous flower-like Ni/C composites were prepared by Zou's group through the pyrolysis of Zn-doped Ni-MOF under N_2 atmosphere [152]. These 3D flower-like structures have massive porous and large spacing flakes, which increases the EMW scatter. The RL_{max} is -52.4 dB with a thickness of 1.6 mm, and EAB is 5 GHz.

3.1.3 Fe-MOF-Derived PC-Based Nanocomposites as MAMs

Because of the good chemical stability, high saturation magnetization and simple preparation, metal iron and ferrite are often used to enhance magnetic loss in MA. So dielectric loss material is also introduced to improve impedance

matching. Xu et al. reported Fe/C nanocubes, which are prepared through an in situ derivation from Prussian blue MOF by controlled high-temperature pyrolysis [153]. The maximum RL of the sample obtained at 650 °C reached -22.6 dB at 4 GHz with a thickness of 5 mm, and the EAB was 7.2 GHz (10.8–18.0 GHz) corresponding to a thickness of 2 mm, as shown in Fig. 7. The good MA performance of the Fe/C nanocubes results in the synergetic effect of dielectric loss and magnetic loss.

Hu's group successfully fabricated magnetic $\text{Fe}_3\text{C}/\text{C}$ (denoted as FC-650) and $\text{Fe}_3\text{C}/\text{Fe}/\text{C}$ (denoted as FC-700) carbon-matrix composites via carbonization of Material Institute Lavoisier (MIL)-101(Fe) [154]. Both $\text{Fe}_3\text{C}/\text{C}$ and $\text{Fe}_3\text{C}/\text{Fe}/\text{C}$ owned flower-like structures formed by 2D flakes. $\text{Fe}_3\text{C}/\text{C}$ possessed RL_{max} of -39.43 dB at 14.00 GHz, at the thickness of 2.00 mm. And the EAB is 14.32 GHz (from 3.68 to 18.00 GHz). The impedance matching and the well-designed structures lead to the excellent MA performance.

The morphology usually had significant effect on the MA performance. Kong pyrolyzed two MOFs with different topologies (MOFs: MIL-101-Fe and MIL-88B-Fe) under

same pyrolysis condition, identical chemical composition and microstructure [155]. The RL_{max} is -59.2 dB with a thickness of 4.32 mm, and the EAB is 6.5 GHz with a thickness of 2 mm which are achieved by Fe/C-700@101 (700: pyrolysis temperature; 101: MIL-101 precursor) and Fe/ $\text{Fe}_3\text{C}/\text{C}$ -800@101, respectively. This article reveals the significant impact of morphology on MA performance.

In order to improve the impedance matching, dielectric loss material is usually introduced to the Fe-MOF. Hu's group reported the synthesis of novel MOF (Fe)/PANI (polyaniline) core-shell composite via hydrothermal and in situ chemical polymerization methods [156]. The RL_{max} of the composite can reach -41.4 dB at 11.6 GHz, and the EAB is up to 5.5 GHz with only 2 mm. The loss mechanism is due to the enhanced interfacial polarization, dipole polarization and charge transfer and attenuation constant. The introduction of PANI enlarges the dielectric constant. Meanwhile, the higher dielectric constant of MOF (Fe)/PANI may be attributed to the improved interfacial polarization and appearance of localized defects as bipolaron/polaron, and the multiple reflection and scattering in the

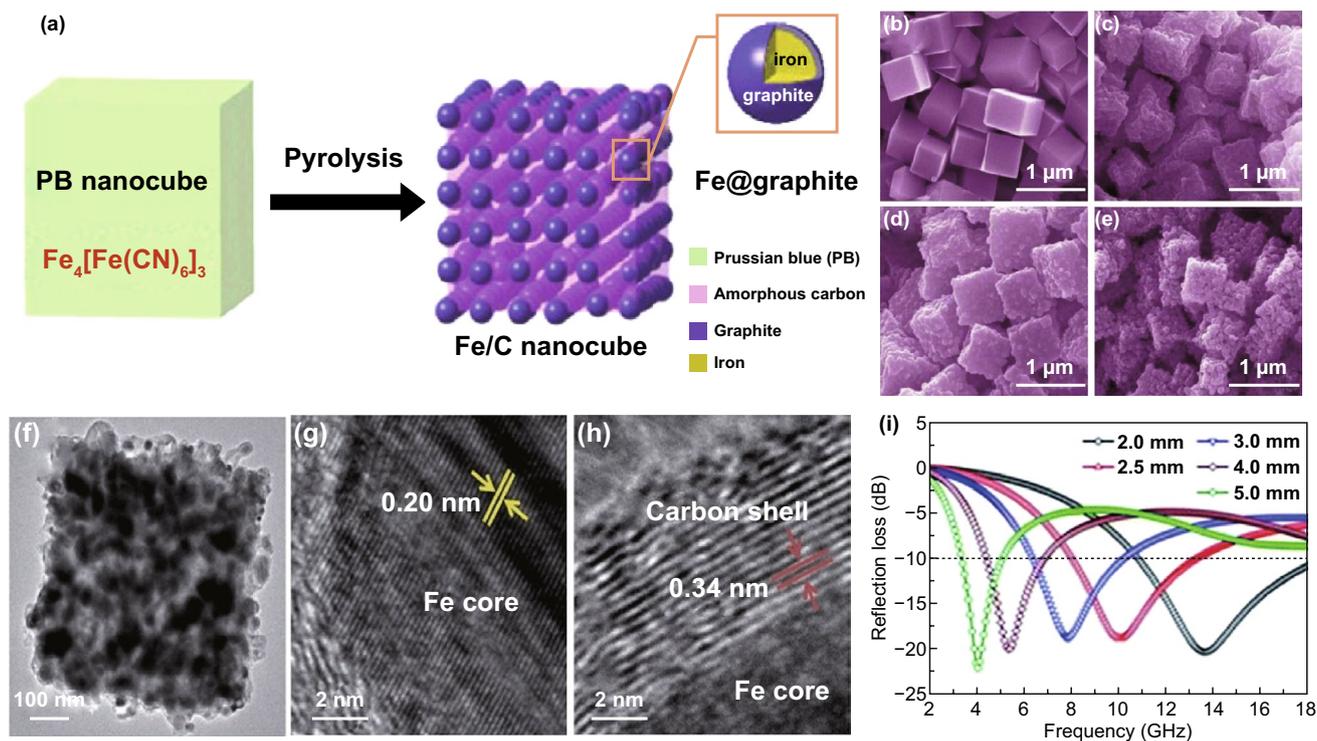


Fig. 7 a Schematic illustration of converting PB nanocubes into Fe/C nanocubes by a pyrolysis technique. b–e SEM images of the as-prepared PB nanocubes and Fe/C nanocubes obtained at different pyrolysis temperatures: 600, 650 and 700 °C. f TEM image of the 650 °C sample. g HRTEM images of the Fe core and h the graphitic carbon shell. i Reflection losses of the 650 °C sample with variable absorber thicknesses. Reprinted with permission from Ref. [153]

pores of the samples, dielectric loss, magnetic loss, good impedance matching also beneficial for EMW absorption. Lu's group reported Fe_3O_4 @ carbon (Fe_3O_4 @NPC) composites by a simple one-pot synthesis method and subsequent in situ formation under thermal decomposition conditions [157]. The Fe_3O_4 @NPC composites exhibited MA performance with a maximum RL of -65.5 dB at 9.8 GHz with a thickness of 3 mm and the EAB of 4.5 GHz, as

shown in Fig. 8. The $\tan \delta_m$ value was higher than $\tan \delta_e$ value, which indicated that magnetic loss contributed more than dielectric loss to the EMW attenuation. Thus, the improvement of the absorption performance was mainly originated from the magnetic loss. The synergistic effects of the dielectric loss and the magnetic loss are effective in enhancing the MA performance.

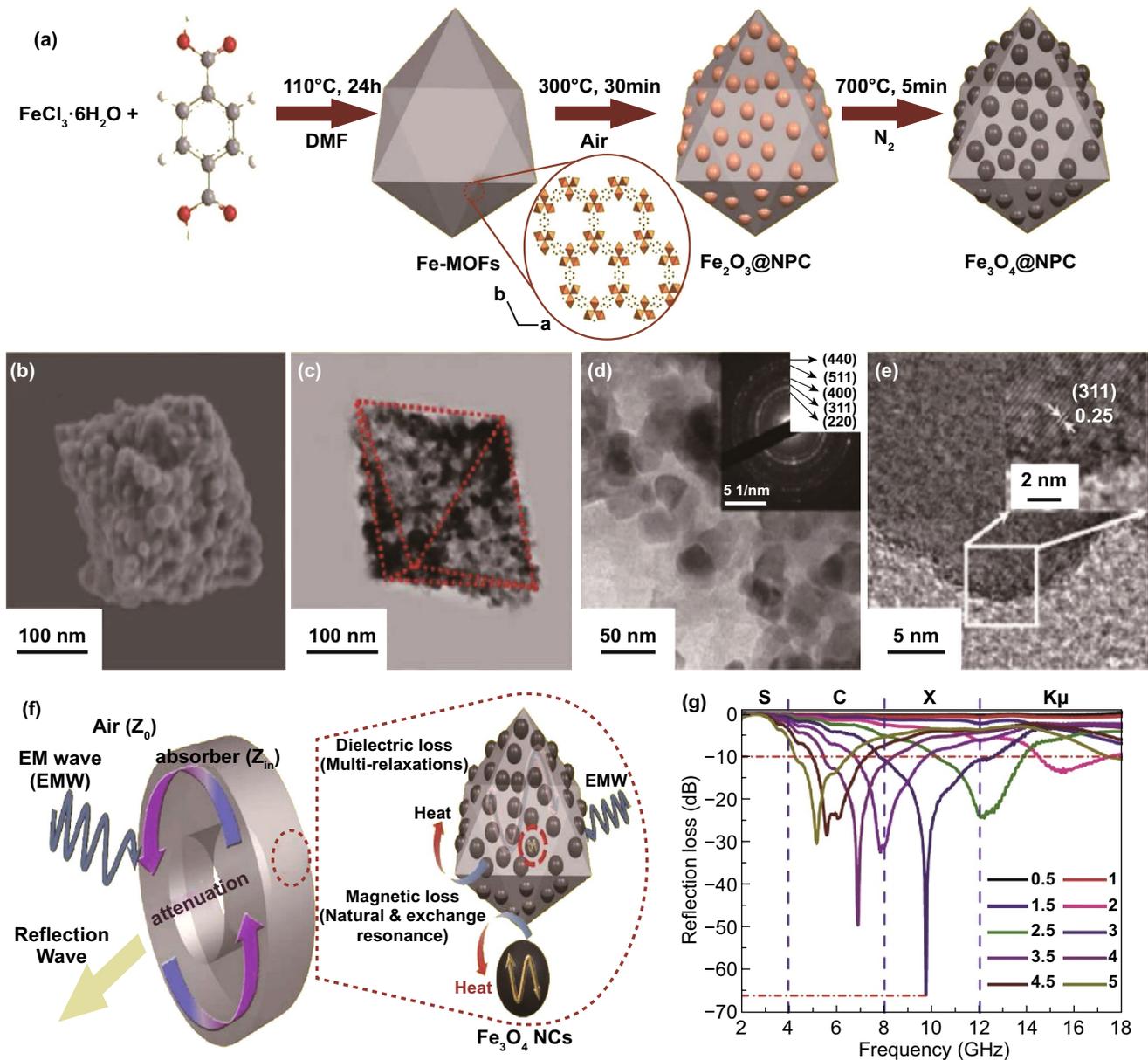


Fig. 8 a Schematic illustration of the Fe_3O_4 @NPC composites formation process. b–e SEM, low magnification TEM images, high magnification TEM images (inset: SAED patterns) and HRTEM images of Fe_3O_4 @NPC composites. f Schematic illustration of the electromagnetic wave absorption mechanism. g Electromagnetic wave reflection loss with various thicknesses for Fe_3O_4 @NPC composites. Reprinted with permission from Ref. [157]

3.2 Non-magnetic Single-Metal MOF-Derived PC-Based Nanocomposites as MAMs

Non-magnetic single-metal MOF-derived PC-based nanocomposites usually act as dielectric performance but have negative characteristics in attenuation and impedance matching. Therefore, selecting a high dielectric candidate to combine with non-magnetic single-metal MOF or magnetic loss material is critical.

3.2.1 Zn-MOF-Derived PC-Based Nanocomposites as MAMs

Ji's group fabricated ZnO/nanoporous carbon (NPC)/reduced graphene oxide (RGO) materials through a simple

and valid hydrothermal method derived from Zn-MOF [158]. The RL_{\max} is -50.5 dB with a thickness of 2.4 mm, and the EAB is 7.4 GHz with a thickness of 2.6 mm, which is shown in Fig. 9. The dielectric constant of ZnO/NPC/RGO samples could be modulated by regulating the combination ratio. Too high or too low permittivity can hardly satisfy an ideal absorber. They had also prepared novel ZnO/carbon porous nanofibers derived from Zn-MOF and polyacrylonitrile (PAN) nanofibers [159]. Xie reported polypyrrole (PPy)/Zn-MOF nanocomposites show tunable electrical conductivity as well as a tunable MA performance [160]. The EAB reaches 7.24 GHz with the thickness of 2.6 mm, and the RL_{\max} is -49 dB with the thickness of 2.9 mm. The MA performance is attributed to the electrical conduction loss and interfacial polarization relaxation.

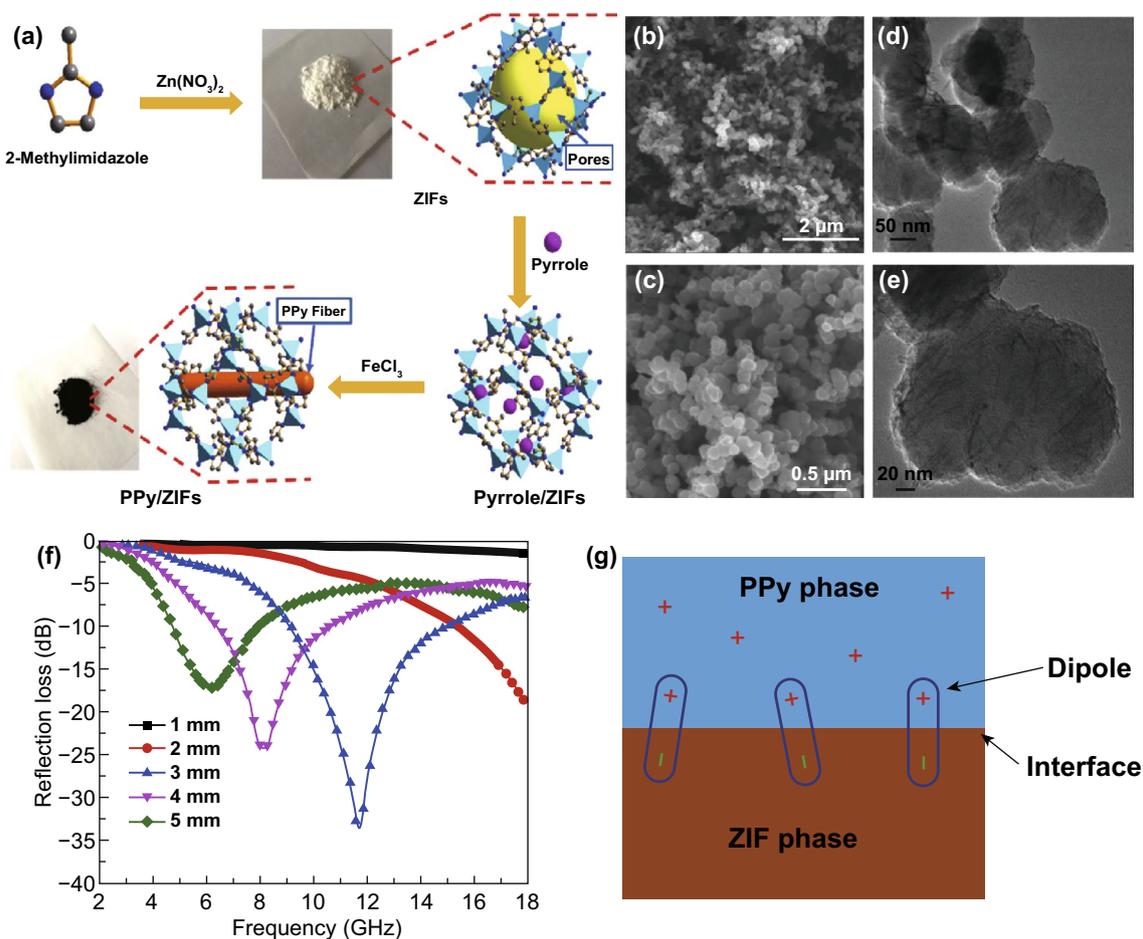


Fig. 9 a Preparation route of PPy/ZIFs nanocomposites. b, c SEM images, d, e TEM images and f MA performance of PPy/ZIFs. g Interfacial polarization of interfacial polarization. Reprinted with permission from Ref. [160]

3.2.2 Ti-MOF-Derived PC-Based Nanocomposites as MAMs

Ji's group had also synthesized a novel nanoporous carbon material (TiO₂/C) by annealing titanium-based MOFs (MIL-125 (Ti); MIL stands for Material from Institute Lavoisier) [161]. The RL_{\max} is -49.6 dB, and the EAB is 4.6 GHz (13.4–18 GHz) with the thickness of 1.6 mm. The outstanding MA performance may be due to the high $\tan \delta_e$, α and polarization loss.

3.2.3 Cu-MOF-Derived PC-Based Nanocomposites as MAMs

Zeng's group synthesized Ni/NiO/Cu@C composites by using Cu MOFs as the precursor [162]. The RL_{\max} value is -38.1 dB at a layer thickness of 3.2 mm. The introduction of Ni offers magnetic loss, and interfacial polarization is changed by increasing the interface area and electrical conductivity.

3.2.4 Zr-MOF-Derived PC-Based Nanocomposites as MAMs

Liu's group developed cobalt-decorated porous ZrO₂/C hybrid octahedrons by pyrolysis of Co(NO₃)₂ impregnated NH₂-UIO-66(Zr-MOF) [44]. The sample results in RL_{\max} of -57.2 dB at 15.8 GHz, corresponding to a matching thickness of 3.3 mm. The EAB reaches 11.9 GHz (6.1–18 GHz). The excellent MA performance of Co/ZrO₂/C can be ascribed to the strong interface polarization and the suitable impedance matching, and the synergistic effect among the components. Wang had also synthesized ZrO₂/C octahedra from UIO-66 [163]. The RL_{\max} value of -58.7 dB (16.8 GHz, 1.5 mm) has been achieved. And the EAB could cover 91.3% (3.4–18.0 GHz) of the measured frequency within the thickness range of 1.0–5.0 mm.

3.2.5 Rare-Earth MOF-Derived PC-Based Nanocomposites as MAMs

Li's group had reported the synthesis of a series of rare-earth MOFs based on MH (maleic hydrazide) ligands [13]. RE-MOFs have many advantages such as hierarchical porous structures, low density and large pore volume.

These properties will meet the requirements of MAMs. They successfully synthesized four novel RE-MOFs [Y₂(MH)₆]_n·DMF (1), [Er₂(MH)₆]_n (2), [Yb₂(MH)₆]_n (3) and [La(MH)₃]_n (4) by the traditional hydrothermal method. Different MA performances can be attributed to different structures and different central ions. The maximum RL values of MOF 1, MOF 2, MOF 3 and MOF 4 are -22.78 dB at 5 mm, -19.99 dB at 4.5 mm, -28.14 dB at 2 mm and -13.07 GHz at 2 mm, respectively. And the effective absorption bandwidth is 2.24 GHz (6.8–9.04 GHz), 2.12 GHz (from 6.8 to 8.72 GHz), 0.96 GHz (15.76–16.72 GHz) and 0.32 GHz (16.72–17.04 GHz) for MOF 1, MOF 2, MOF 3 and MOF 4. The property may be resulted in the synergetic effects of permittivity and permeability.

3.3 Multi-metal MOF-Derived PC-Based Nanocomposites as MAMs

The MA performance of multi-metal MOF-derived PC-based nanocomposite is often better than single-metal MOF because the multi-metal MOF combines the advantages of two or more materials, endows the mixture with new chemical and physical properties and effectively regulates the electromagnetic parameters of the MAMs.

3.3.1 Multi-magnetic Metal MOF-Derived PC-Based Nanocomposites as MAMs

NiCo nanoparticles/nanoporous carbon (NiCo/NPC) composites with multilayered structure were synthesized through in situ pyrolysis of the bimetallic NiCo-MOF by Lu's group [164]. The synergistic interactions of magnetic loss and dielectric loss among NiCo NPs, graphitized carbon layer and NPC were beneficial to the optimization of the impedance matching and the enhancement of EMW attenuation. The multilayered nanoporous carbon matrix leads to the multiple reflection and scatterings, interface and dipole polarization as well as the natural resonance and exchange resonance. The RL_{\max} value is -51 dB at 17.9 GHz with EAB of 4.5 GHz (13.5–18 GHz) and a thickness of 1.5 mm at 600 °C. Dong had also fabricated porous and hollow CoNi@C microspheres derived from CoNi-MOFs [165]. The RL_{\max} can reach -44.8 dB at 10.7 GHz, and the EAB



can reach up to 13.3 GHz (4.7–18.0 GHz) with the thickness of 1.6–4.0 mm, as shown in Fig. 10. The simultaneous enhancement of attenuation ability and impedance matching together contribute to the improved MA performance. The attenuation ability comes from interfacial polarization, eddy current loss, multiple reflection and scattering. The impedance matching stems from magnetic CoNi alloy and dielectric graphitized carbon. Liu had also reported CoNi/C nanocomposites derived from bimetallic CoNi-MOF [166]. The RL_{\max} of -74.7 dB could be achieved with a thickness of 1.8 mm at 15.6 GHz. The EAB ranged from 2.9 GHz to 18 GHz. The porous Co–Ni/C nanocomposites combined advantages of excellent impedance matching and strong interfacial loss between metallic NPs and porous carbon composites. Similarly, FeCo alloy/carbon composites [167]

and Fe_3Ni/C composites [168] had also shown the excellent MA performance. Hollow sphere trimetallic FeCoNi@C MAMs via high-temperature carbonization were obtained using FeCoNi-based MOF-74 (FeCoNiMOF) as the precursor [169].

Other multi-magnetic metal MOF-derived PC-based nanocomposites as MAMs had also been reported. In order to further improve impedance matching, the dielectric loss is often introduced to the multi-magnetic metal MOF such as FeCo@C@CNGs(carbon nanocages) [170], NiCo alloy/carbon nanorod@CNT [171], Fe–Co/NC/rGO [172], FeNi@CNT/CNRs(carbon nanorods) [173] and CoFe@C@MnO₂ [174]. All of these samples show good impedance matching and outstanding EMW attenuation capability.

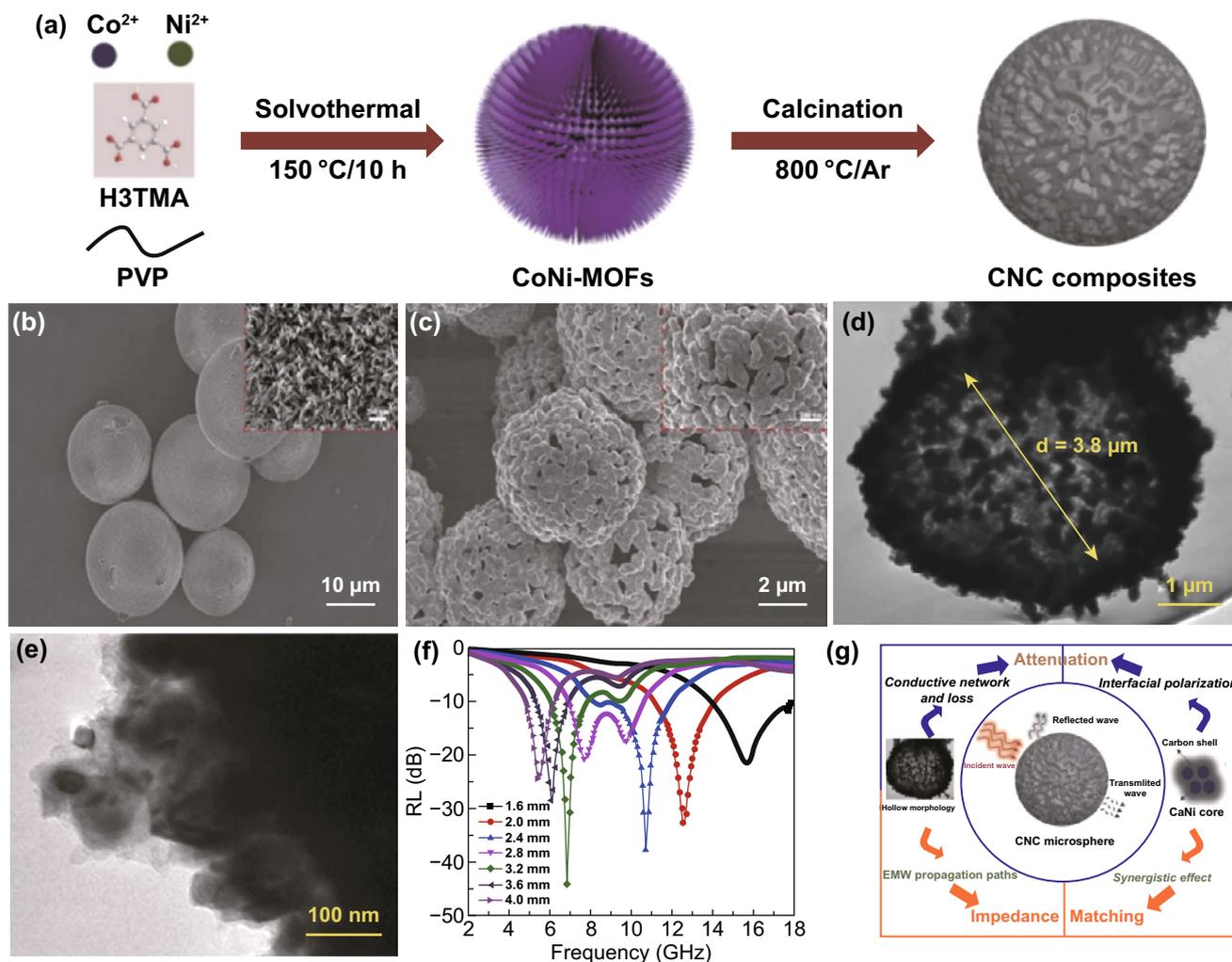


Fig. 10 a Illustration for the synthetic process of hollow CNC microspheres. b, c SEM images, d, e TEM images and f RL curves of CoNi@C samples. g Schematic illustration of microwave absorption mechanisms for CNC microspheres. Reprinted with permission from Ref. [165]

3.3.2 Magnetic and Non-magnetic Metal MOF-Derived PC-Based Nanocomposites as MAMs

Non-magnetic metal MOFs play the role of dielectric loss. Zn is most widely used in this occasion. Since the unique evaporation character of Zn metal under high pyrolysis temperature, the porous low-dielectric amorphous carbon/Zn shell derived from Zn-MOF was formed to decrease the permittivity for a better impedance match. Zheng's group fabricated nitrogen-doped CoO/Co/C nanocomposites by high-temperature pyrolysis of Co/Zn-ZIFs [175]. Zn was evaporated during the high-temperature pyrolysis process at 700 °C. The RL_{\max} reached -66.7 dB at 7.2 GHz with a thickness of 3.3 mm, and the EAB is 5.1 GHz (12.6–17.7 GHz) with thickness of 1.8 mm, as shown in Fig. 11. The excellent MA performance ascribed to the enhanced polarization relaxation, and synergistic effects of dielectric loss, conduction loss and magnetic loss.

Jiang had fabricated CoZn-MOF and then calcined it at different high temperatures to gain the metal Co embedded in porous and N-doped graphitized carbon matrix (Co@pNGC) [176]. Zn species was also evaporated at high temperature. The RL_{\max} is -50.7 dB at 11.3 GHz, and the EAB reaches 5.5 GHz (12.3–17.8 GHz), corresponding to a thickness of 2.0 mm. The strong dielectric loss is derived from interfacial polarization, migration, hopping of electrons and the magnetic loss from the Co nanoparticles.

4 Comparison of MA Performance of Different MOF-Derived PC-Based Nanocomposites

As is mentioned above, many MOF-derived PC-based nanocomposites exhibited the appreciable MA performance. In Table 2, we sum up the performance of the MAMs mentioned above. As is described, there are many kinds of MOF-derived PC-based materials used in the MAMs. Most of the MOF-derived PC-based MAMs have better MA performance than the comparison MAMs. The MA performance of pure Ni, Co, Fe-MOF-derived PC-based is not satisfactory. When they coupled with

dielectric loss material, the MA performance will be significantly improved. Non-magnetic metal MOF-derived PC-based MAMs such as Zn, Ti, Cu, Zr and RE are usually coupled with magnetic loss material to get impedance matching, while the multiple metal MOF-derived PC-based MAMs have shown excellent MA performance. They usually have multiple loss mechanism, so the synergistic effect between each part will be beneficial to impedance matching and electromagnetic wave attenuation. The structure of MOF also has a significant effect on the MA performance. Through design of MOF with different structures such as foam structure, core-shell structure, hollow structure, etc., the multiple reflections and interfacial polarization can be achieved. Therefore, MOF-derived PC-based nanocomposite is a promising material in the field of high-performance MAMs in the future.

5 Conclusion

The recent progress of MOF-derived PC-based nanocomposites as MAMs has been systematically summarized by this review. In view of these studies, we find that MOF-derived PC-based MAMs from in situ pyrolysis of MOFs will be a promising method for the development of lightweight and highly effective MAMs. After pyrolysis, the PC-based MAMs from the MOFs exhibit porosity, low density, good electrical conductivity and dielectric loss. And the inorganic metal center can result in magnetic loss (magnetic metal) or dielectric loss (non-magnetic metal). To further improve the MA performance, the MOF-derived PC-based nanocomposites often coupled with other loss material. The well-designed nanocomposites with multiple advantages will show good impedance matching and strong EM attenuation capability because of the multiple loss mechanism and synergistic effect of the multi-components. Therefore, the MOF-derived PC-based nanocomposites coupled with multiple loss material are an attractive development direction of MAMs in the future.

Many achievements have been made in MOF-derived PC-based nanocomposites as MAMs, but most are just at the research stage, far away from the practical use. The EAB and the maximum RL values are not enough to meet the actual needs. We can rationally design the MAMs with the suitable preparation conditions to realize the special microstructure, which can

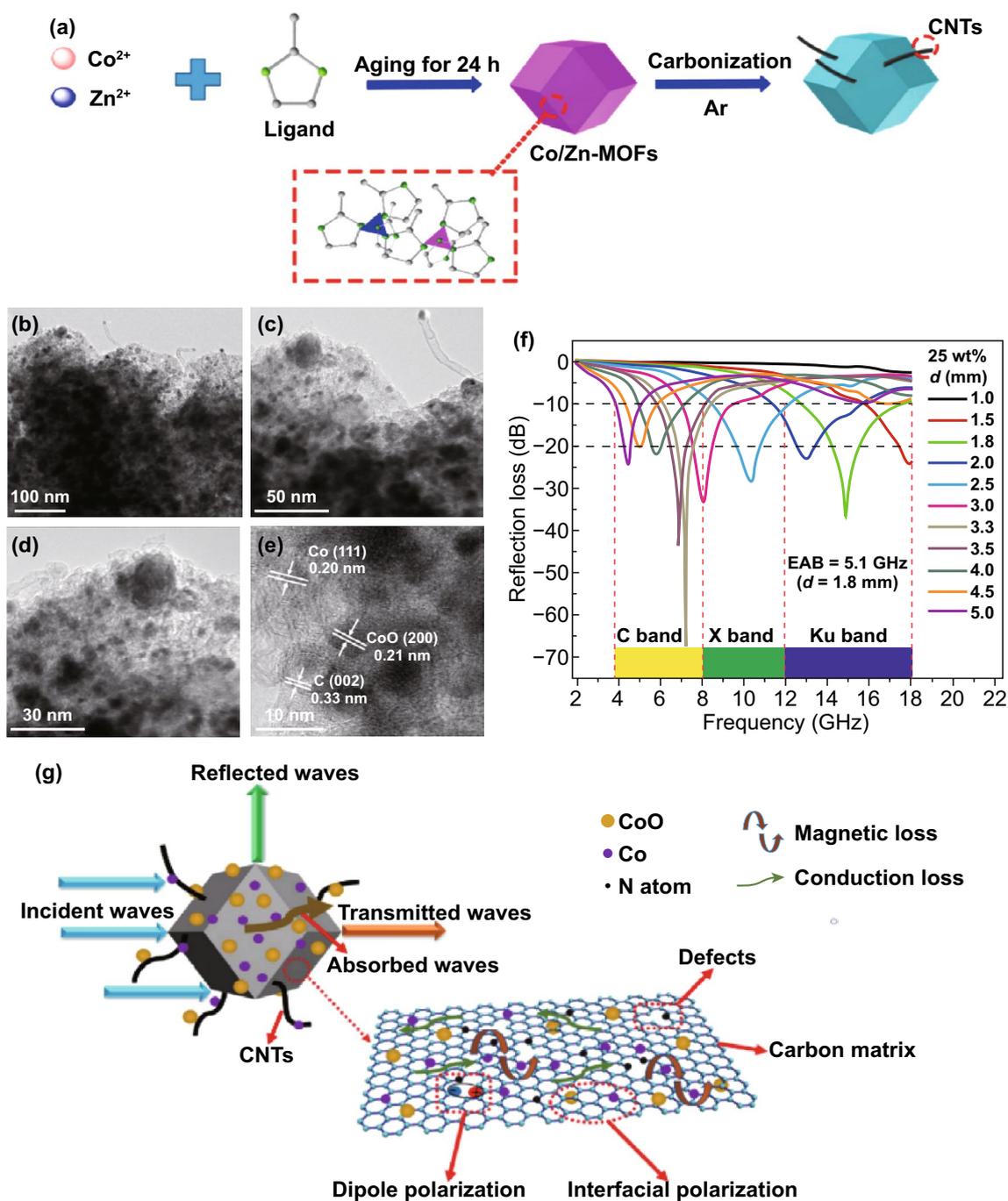


Fig. 11 **a** Schematic illustration of the preparation procedures of CoO/Co/C nanocomposites. **b–e** TEM images with different magnifications of CoO/Co/C nanocomposites. **f** RL curves of CoO/Co/C nanocomposites. **g** Schematic illustration of the possible microwave absorption mechanisms of nitrogen-doped CoO/Co/C nanocomposites. Reprinted with permission from Ref. [175]

improve the scattering of EM, and the multiple loss mechanism is realized by the synergistic effect of multiple components. As more than 20,000 kinds of MOFs have been used in various fields, we only review the common Ni, Co, Fe, Zn, Ti, Cu,

Zr and RE metal as the central elements, and they have shown considerable MA performance. But they are just the tip of the iceberg of the big family of MOFs. We should also pay more attention to other metal elements. Through the modulation of

Table 2 The MA performance of different kinds of MOF-derived PC-based nanocomposites as MAMs

Type	MAMs	RL_{max}			EAB (< -10 dB) (GHz)		Refs.
		Value (dB)	f_m (GHz)	Thickness (mm)	Value (GHz)	Thickness (mm)	
MOF-derived PC-based nanocomposites	Co/C-500	-35.3	5.8	4	5.8 (8.4–14.2)	2.5	[123]
	Co/C-650	-47.6	5.11	2	5.1 (12.1–17.2)	2	[124]
	Co/C-700	-15.7	15.1	1.7	5.4 (12.3–17.7)	1.7	[125]
	Co/C-700	-30.31	11.03	3	4.93 (8.31–13.24)	3	[126]
	Co/C-800	-39.6	9.6	2	3.8 (10.7–14.5)	2	[127]
	MOF/RGO-500	-52	9.6	4.1	7.72 (10.28–18)	3.2	[128]
	CoC-rGO-2	-44.77	12.1	2.1	5.2 (12.8–18)	1.8	[129]
	Co ₃ O ₄ /Co/RGO	-52.8	13.12	2	10.72 (4.88–15.6)	2–4	[130]
	3D Co/C-MCNT	-20.3	13.84	1.8	4.08	1.8	[131]
	Co/C-MCNTs	-48.9	9	2.99	–	–	[132]
	Co/CNT	-49.16	14.16	2.5	4.2 (12.4–16.6)	2.5	[133]
	MWCNTs@carbonaceous CoO	-50.2	14.3	1.84	4.32 (12.32–16.64)	1.84	[134]
	CNFs@carbonaceous Co/CoO	-53.1	6.56	3.54	13.52 (3.68–14.64, 15.44–18)	2–5	[135]
	Co/C@PPy	-44.76	17.32	2.0	6.56 (11.04–17.60)	2.5	[136]
	Co/NPC@ZnO/rGO	-45.4	14.2	2	5.4 (11.9–17.3)	2	[138]
	Co/C@V ₂ O ₃	-40.1	14.1	1.5	4.64 (13.36–18)	1.5	[139]
	MnO@Co/C	-49.06	6.48	3.4	2.24	3.4	[140]
	Co/N/C@MnO ₂	-58.9	5.56	3.7	5.5	–	[141]
	NRGO/MWCNT	-69.6	12.5	1.8	4.3 (13.2–17.5)	1.5	[143]
	Co/CNTs/CS	-51.2	12	2.2	4.1 (10.3–14.4)	2.2	[144]
Co/C-HS	-66.5	17.6	1.53	14.3 (3.7–18.0)	1–5	[145]	
Ni-MOF-derived PC-based nanocomposites	Ni@C-800	-57	13.8	1.85	6 (12–18)	1–5	[146]
	NPC/Ni	-39.4	–	–	4.2	–	[147]
	Ni@C-ZIF	-86.8	7.25	2.7	7.4 (4–11.4)	1.5–4	[149]
	SiC/Ni/NiO/C	-50.52	13	4	2.96 (14.76–17.72)	2.5	[150]
	NiO/Ni/GN@Air@NiO/Ni/GN	-34.5	17.2	1.7	10.2 (7.8–18)	1.7–5.0	[151]
	Porous flower-like Ni/C	-52.4	16.1	1.6	5	1.6	[152]
Fe-MOF-derived PC-based nanocomposites	Fe/C nanocubes	-22.6	4	5	7.2 (10.8–18)	2	[153]
	FC-650	-39.43	14	2	5.36 (11.76–17.12)	2	[154]
	Fe/C-700-101	-59.2	5	4.32	5	1.8	[155]
	MOF (Fe)/PANI	-41.4	11.6	2	5.5 (9.8–15.3)	2	[156]
	Fe ₃ O ₄ @NPC	-65.5	9.8	3	4.5	3	[157]
Zn-MOF-derived PC-based nanocomposites	ZnO/NPC/RGO	-50.5	14	2.4	7.4 (9.6–17)	2.6	[158]
	Fe ₃ O ₄ /CNT	-43	15.2	1.5	8.3 (9.7–18)	1.75	[159]
	PPy/ZIFs	-49	12.1	2.9	7.24 (10.76–18)	2.6	[160]
Ti-MOF-derived PC-based nanocomposites	TiO ₂ /C	-49.6	15.8	1.6	4.6 (13.4–18 GHz)	1.6	[161]
Cu-MOF-derived PC-based nanocomposites	Ni/NiO/Cu@C	-38.1	14.8	3.2	–	–	[162]
Zr-MOF-derived PC-based nanocomposites	ZrO ₂ /C	-57.2	15.8	3.3	11.9 (6.1–18)	3.3	[44]

Table 2 (continued)

Type	MAMs	RL_{\max}			EAB (< -10 dB) (GHz)		Refs.	
		Value (dB)	f_m (GHz)	Thickness (mm)	Value (GHz)	Thickness (mm)		
RE-MOF-derived PC-based nanocomposites	ZrO ₂ /C	-58.7	16.8	1.5	14.6 (3.4–18.0)	1–5	[163]	
	[Yb ₂ (MH) ₆] _n	-28.14	–	2	–	–	[13]	
	[Y ₂ (MH) ₆] _n -DMF	–	–	–	2.24 (6.8–9.04)	5		
Multi-magnetic metal MOF-derived PC-based nanocomposites	NiCo/NPC	-51	17.9	1.5	4.5 (13.5–18)	1.5	[164]	
	CoNi@C	-44.8	6.8	3.2	13.3 (4.7–18.0)	1.6–4	[165]	
	CoNi/C	-74.7	15.6	1.8	15.1 (2.9–18.0)	0.3–5	[166]	
	FeCo alloy/carbon	-57.4	17.7	1.26	4.2 (11.0–15.2)	–	[167]	
	Fe ₃ Ni/C	-46.2	10.44	2.65	5.24 (12.76–18)	2	[168]	
	FeCoNi@C	-69.03	5.52	2.1	8.08 (9.92–18)	2.47	[169]	
	Core-shell FeCo@carbon/PDA	-67.8	15.8	1.75	5.3 (11.0–16.3)	2	[170]	
	NiCo alloy/C nanorod@CNT	-58.8	14.0	2.2	6.5 (11.5–18)	2.2	[171]	
	Fe-Co/NC/rGO	-43.26	11.28	2.5	9.12 (8.88–18)	2.63	[172]	
	FeNi@CNT/CNRs	-47.0	–	2.3	4.5	1.6	[173]	
	CoFe@C@MnO ₂ nanocubes	-64	15.6	1.3	9.2 (8.8–18)	1.6	[174]	
	Magnetic and non-magnetic metal MOF-derived PC-based nanocomposites	CoO/Co/C	-66.7	7.2	3.3	5.1 (12.6–17.7)	1.8	[175]
		Co@pNGC	-50.7	11.3	2.5	4.0 (12.2–16.2)	1.2	[176]
	Comparison MAMs	CNT	-21	5	3.5	0.5	3.5	[177]
		rGO	-6.9	7	2	–	–	[178]
graphene foam		-34	13.1	–	14.3 (3.7–18)	–	[59]	
Carbon nanotube/graphene foams		-39.5	11.6	–	16	–	[179]	
Carbon nanotube grown on the carbon fiber		-42	11.4	2.5	2.7	2.5	[180]	
3D PPy aerogel		-22.5	12	3	5.0 (10.0–15.0)	3	[181]	
PANI nanoparticle		-18.8	17.2	2	3.9 (14.1–18.0)	2	[182]	
ZnO nanoparticles		-37.7	8.96	2.1	3.55 (7.5–11.05)	2.1	[183]	
C ₃ N ₄ nanosheets		-36.1	14.6	19.5	1.7	19.5	[184]	
SiC		-24.8	11	3	4.2 (8.2–12.4)	3	[185]	
Fe powder	-5.2	11	3	–	–	[186]		
Fe ₃ O ₄ @C	-40	15.9	1.5	3.9 (14.1–18)	1.5	[93]		
Flaky carbonyl iron particles	-14	0.6	1	1.6 (0.4–2)	1	[187]		

inorganic metal center and organic ligand, different kinds of MOFs are constructed to achieve the optimal MA performance. The “thin, wide, light, strong” is the goal to develop MAMs, but most of the studies merely focus on the EAB and the maximum RL values, while the thickness and the weight of the MAMs have

been usually ignored. In fact, low density is also an important parameter to evaluate the MAMs. One of the pyrolysis products of MOFs is carbon; therefore, MOFs are promising materials to employ new lightweight MAMs, especially in military applications. In conclusion, MOF-derived PC-based nanocomposites

had already shown its great potential as MAMs. We firmly believe that the MOF-derived PC-based nanocomposites will be widely used in the field of MAMs in the future.

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