

Supporting Information for

## Mediating the Local Oxygen-Bridge Interactions of Oxysalt/Perovskite Interface for Defect Passivation of Perovskite Photovoltaics

Ze Qing Lin<sup>1, #</sup>, Hui Jun Lian<sup>1, #</sup>, Bing Ge<sup>1</sup>, Ziren Zhou<sup>1</sup>, Haiyang Yuan<sup>1, \*</sup>, Yu Hou,  
Shuang Yang<sup>1, \*</sup>, Hua Gui Yang<sup>1</sup>

<sup>1</sup>Key Laboratory for Ultrafine Materials of Ministry of Education, Shanghai Engineering Research Center of Hierarchical Nanomaterials, School of Materials Science and Engineering, East China University of Science and Technology, Shanghai 200237, P. R. China

#Ze Qing Lin and Hui Jun Lian contributed equally to this work.

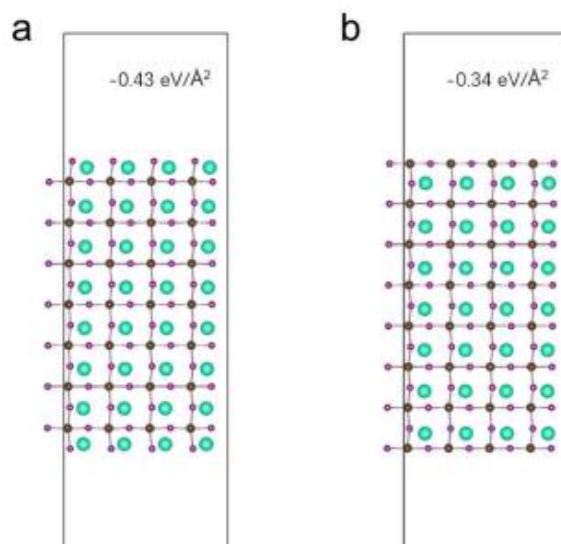
\*Corresponding authors. E-mail: [hyyuan@ecust.edu.cn](mailto:hyyuan@ecust.edu.cn) (Haiyang Yuan), [syang@ecust.edu.cn](mailto:syang@ecust.edu.cn) (Shuang Yang)

### Supplementary Tables and Figures

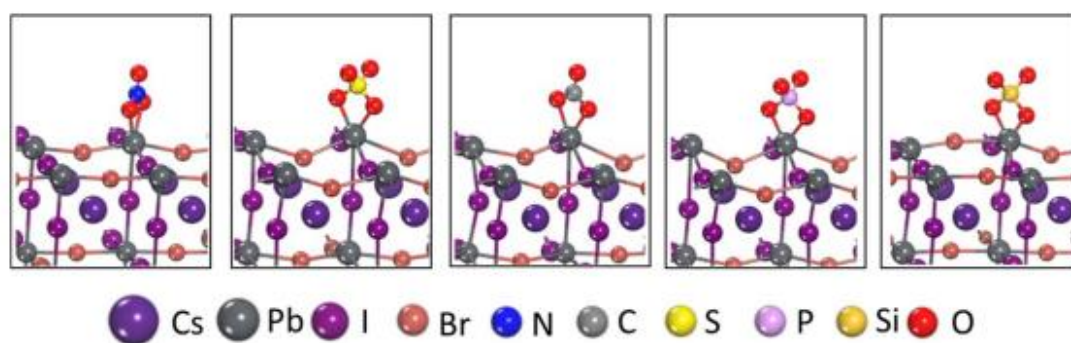
The (001) surface of CsPbI<sub>2</sub>Br is one of the thermodynamically most stable surfaces [S1], thus which is selected in our calculations. Considering the complexity and different terminal properties of CsPbI<sub>2</sub>Br surface, we investigated the possibilities of different terminated CsPbI<sub>2</sub>Br (001), i.e., PbIBr termination and CsI termination, by building the seven layers of CsPbI<sub>2</sub>Br (001) supercell with 20 Å vacuum region (Fig. S1). The surface energies ( $\gamma$ ) of all slab were calculated as the formula [S2]:

$$\gamma = (E_{\text{surf}} - N \cdot E_{\text{bulk}} + \sum n_i \cdot \mu_i) / 2A$$

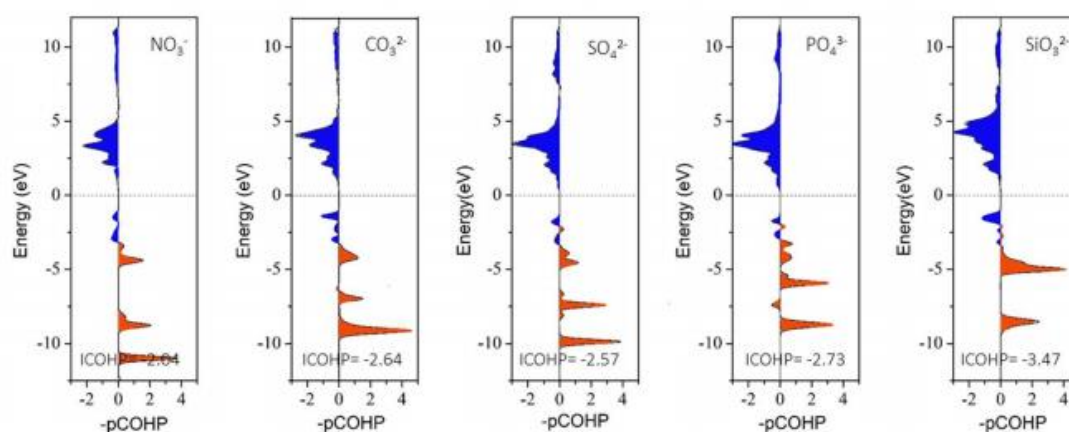
where  $E_{\text{surf}}$ ,  $E_{\text{bulk}}$  are the energies of PbIBr or CsI-terminated CsPbI<sub>2</sub>Br (001) surface, bulk CsPbI<sub>2</sub>Br, respectively.  $N$  is the number of sufficient bulk CsPbI<sub>2</sub>Br units,  $\mu_i$  is the chemical potential, and  $A$  is the area of the CsPbI<sub>2</sub>Br (001). The more negative  $\gamma$  means the better stability of the surface. The calculated results show that  $\gamma$  of the PbIBr-terminated CsPbI<sub>2</sub>Br (001) is more negative than that of CsI-terminated one (-0.43 eV *versus* -0.34 eV). This means that PbIBr-terminal CsPbI<sub>2</sub>Br (001) is more stable and easier to be exposed during preparation, which could be the main exposed termination of CsPbI<sub>2</sub>Br (001). Hence, in our calculations, the PbIBr-terminal CsPbI<sub>2</sub>Br (001) was selected as our model.



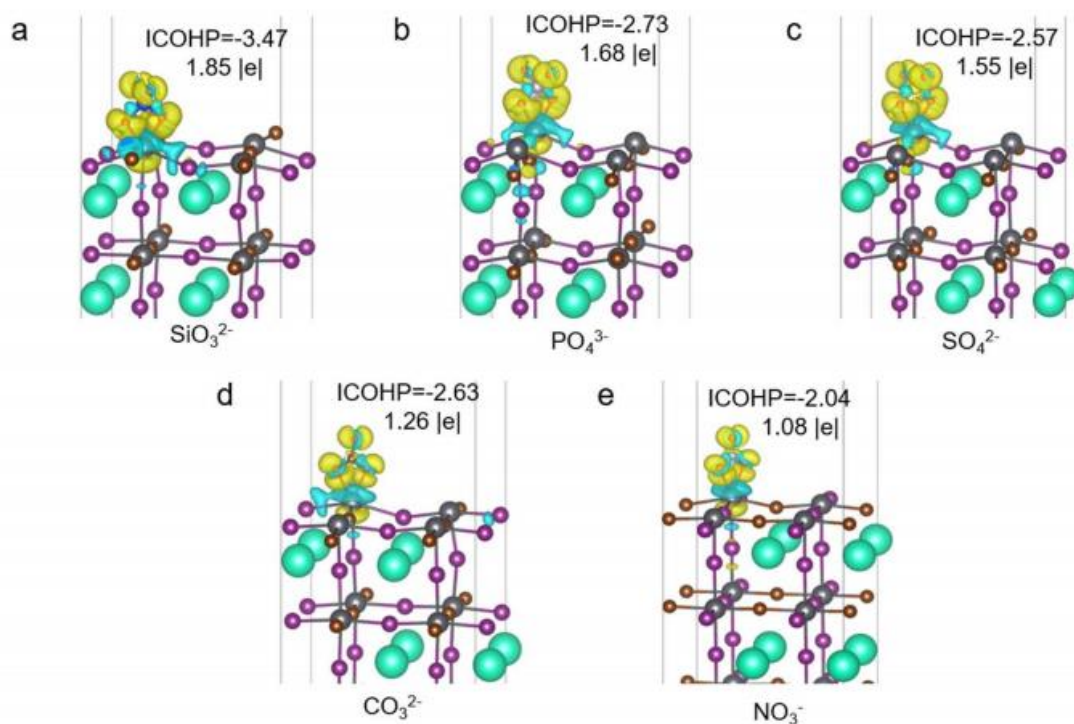
**Fig. S1** Optimized CsPbI<sub>2</sub>Br (001) supercell models with (a) CsI-terminated and (b) PbI<sub>2</sub>-terminated



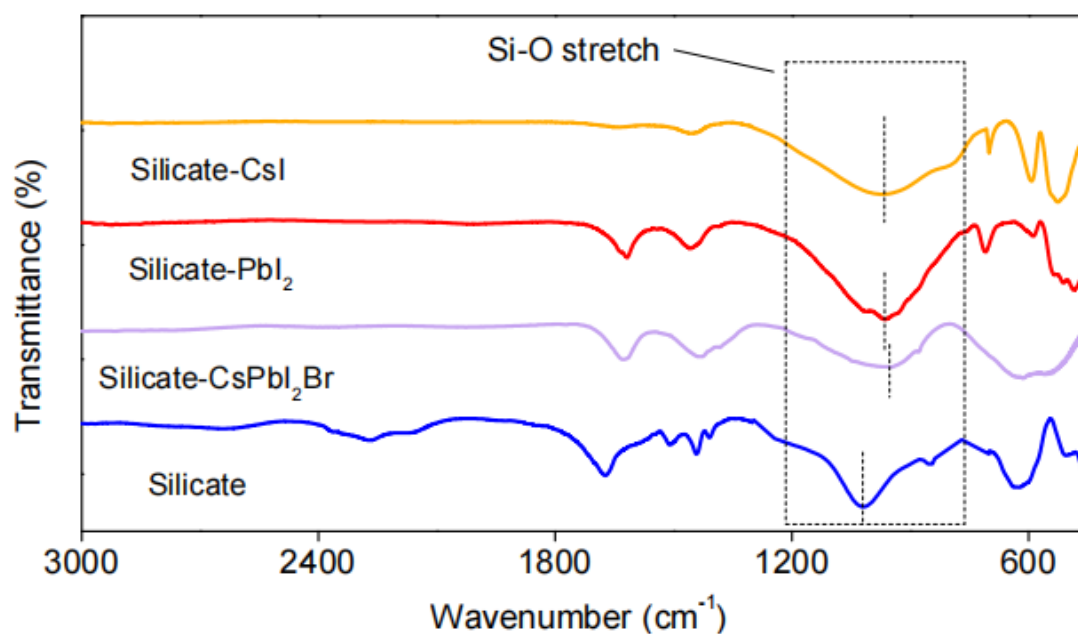
**Fig. S2** Atomic structure of optimized CsPbI<sub>2</sub>Br (001) surface with oxyacid anions with different central atoms



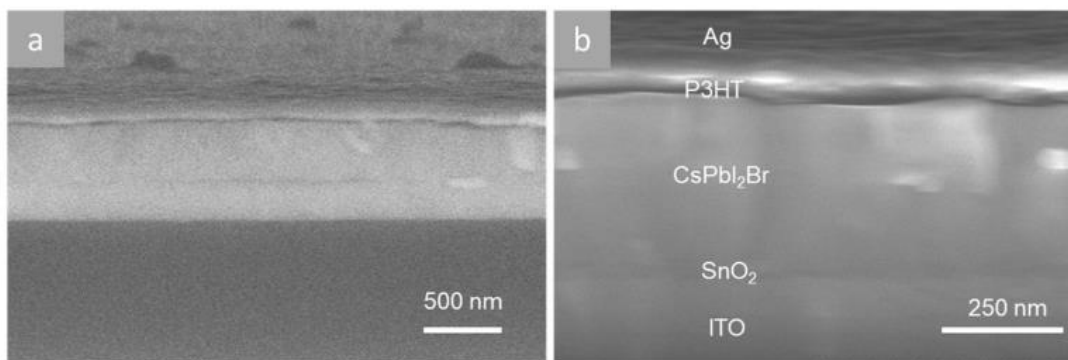
**Fig. S3** Projected crystal orbital Hamilton population (pCOHP) between Pb and O atom in different oxyacid anions. The values of ICOHP calculated by the energy integral below Fermi level indicated the interaction between silicate and Pb site is the strongest



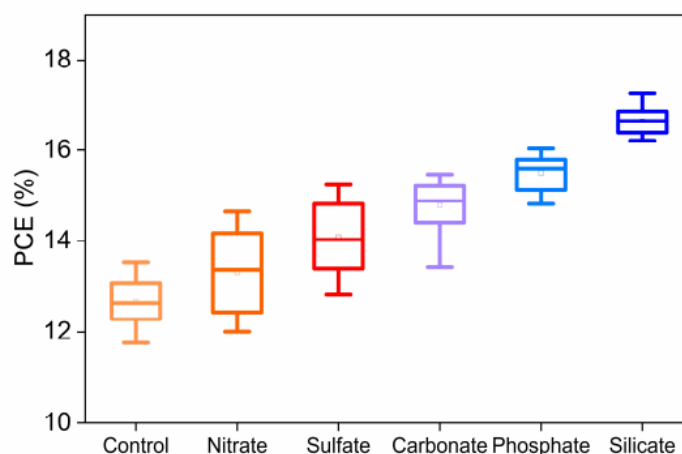
**Fig. S4** Charge density difference of oxysalt/CsPbI<sub>2</sub>Br interface: (a) SiO<sub>3</sub><sup>2-</sup>, (b) PO<sub>4</sub><sup>3-</sup>, (c) SO<sub>4</sub><sup>2-</sup>, (d) CO<sub>3</sub><sup>2-</sup> and (e) NO<sub>3</sub><sup>-</sup>



**Fig. S5** FTIR spectra of sodium silicate powders without and with perovskite components. The asymmetric stretch ( $\nu_{as}$ ) peaks of silicate are highlighted by dotted lines

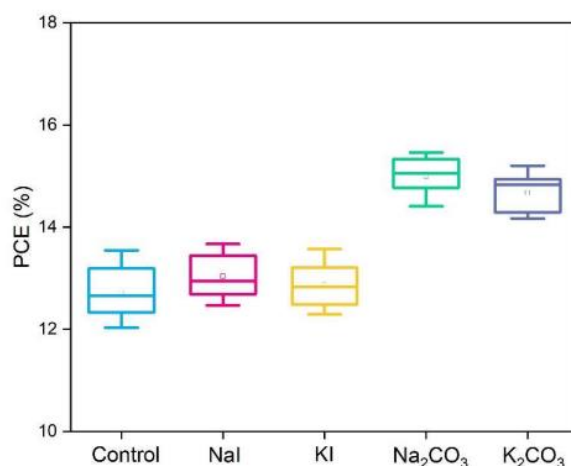


**Fig. S6** Cross-sectional SEM of a typical CsPbI<sub>2</sub>Br perovskite solar cell. The device structure adopted in this work is indium tin oxide (ITO) glass/tin (IV) oxide (SnO<sub>2</sub>)/CsPbI<sub>2</sub>Br/poly(3-hexylthiophene-2,5-diyl) (P3HT)/Ag

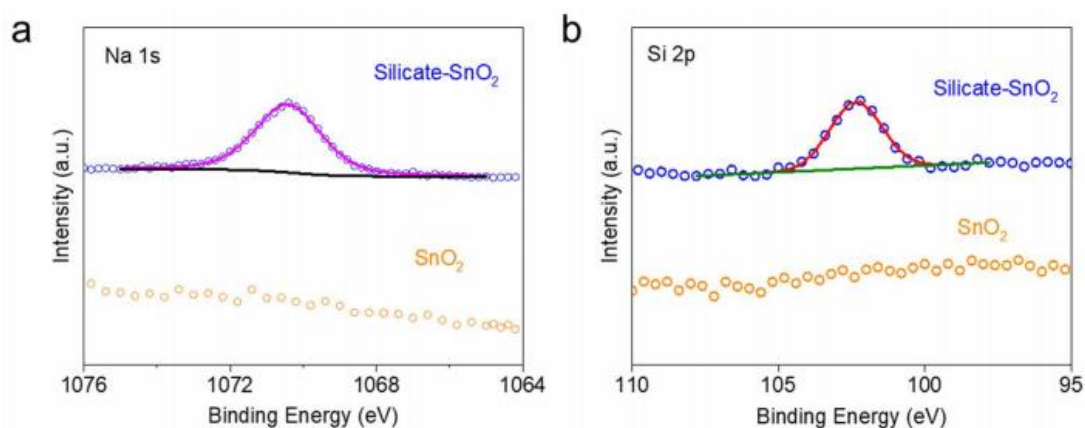


**Fig. S7** PCE distributions of CsPbI<sub>2</sub>Br solar cells without and with oxysalt passivation

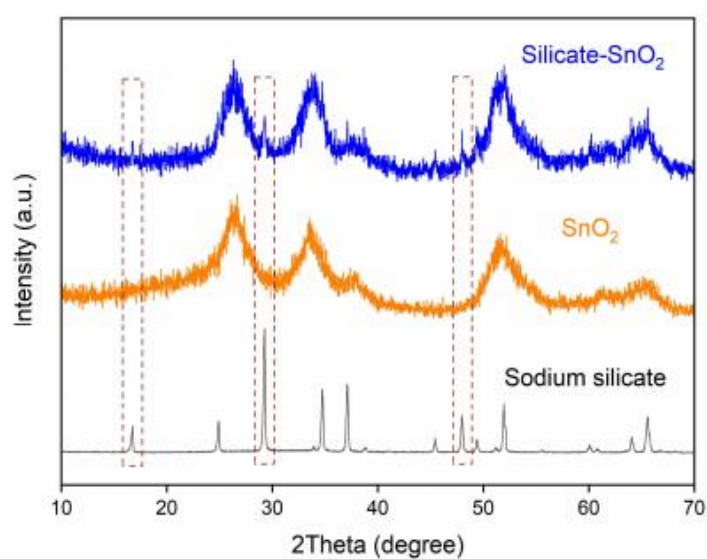
For each kind of device, the solid transverse lines in the boxes are the average PCEs, analysed from 20 devices, and the error bars show the highest and lowest PCE values.



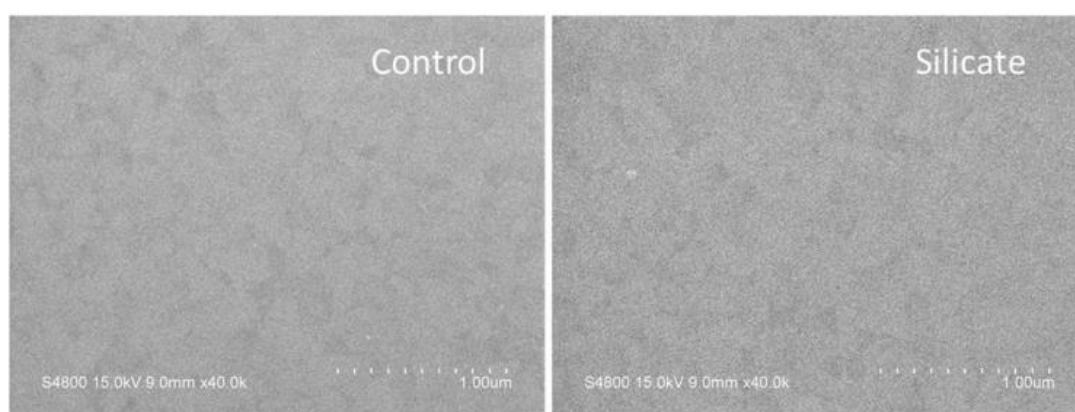
**Fig. S8** PCE distributions of CsPbI<sub>2</sub>Br solar cells without and with different passivation materials



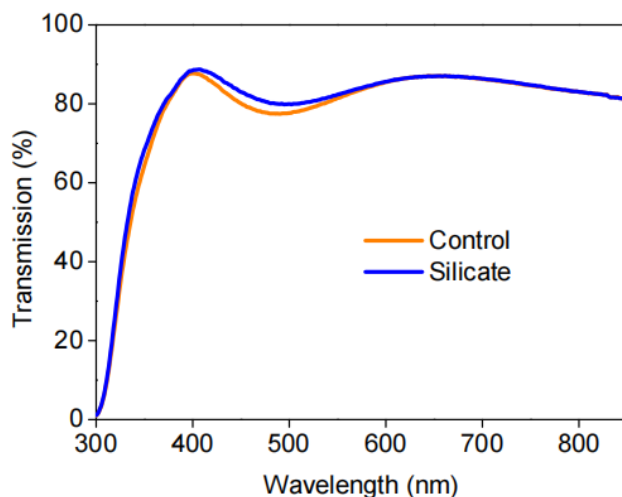
**Fig. S9** (a) Na 1s and (b) Si 2p XPS spectra of SnO<sub>2</sub> and silicate-SnO<sub>2</sub> films



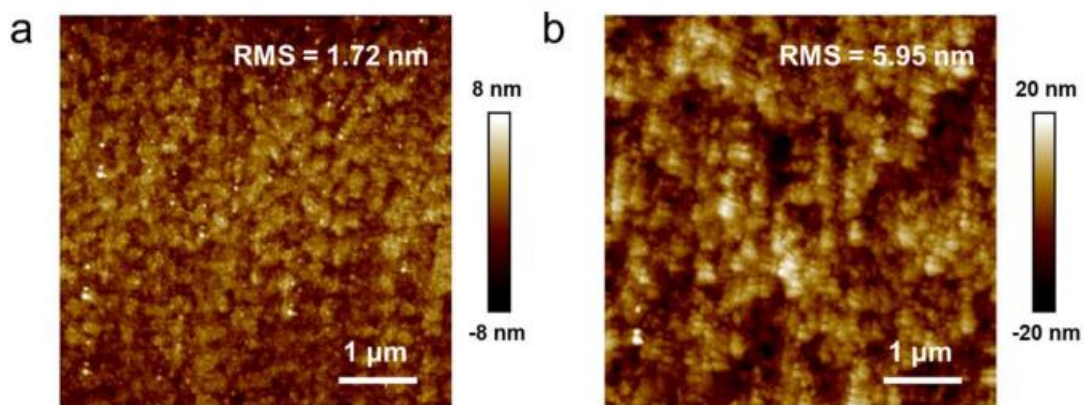
**Fig. S10** XRD patterns of sodium silicate, SnO<sub>2</sub> and silicate-SnO<sub>2</sub> powders. SnO<sub>2</sub> and silicate-SnO<sub>2</sub> powders were scratched from the as-casted films. The concentration of sodium silicate in precursor solution was 0.1 M



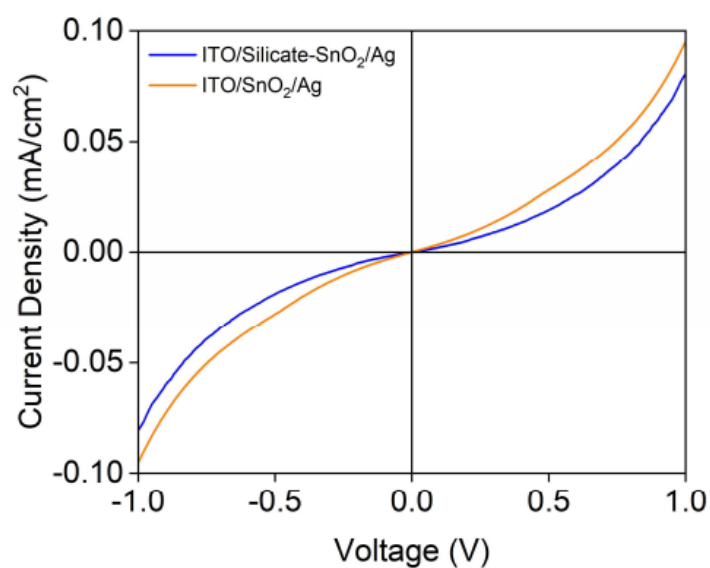
**Fig. S11** SEM images of SnO<sub>2</sub> films with and without silicate. Both films exhibit uniform morphology with full coverage of the ITO glass substrate



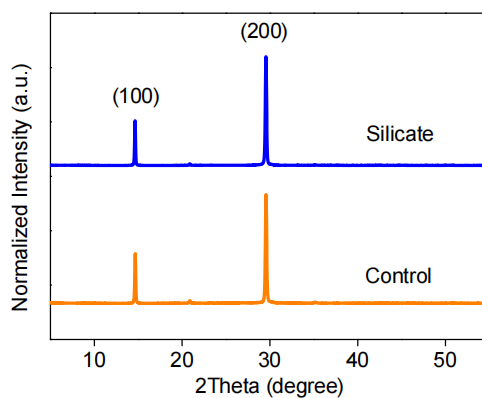
**Fig. S12** Transmission spectra of SnO<sub>2</sub> films without and with silicate. The optical transmittance of both samples is similar in the UV and visible region.



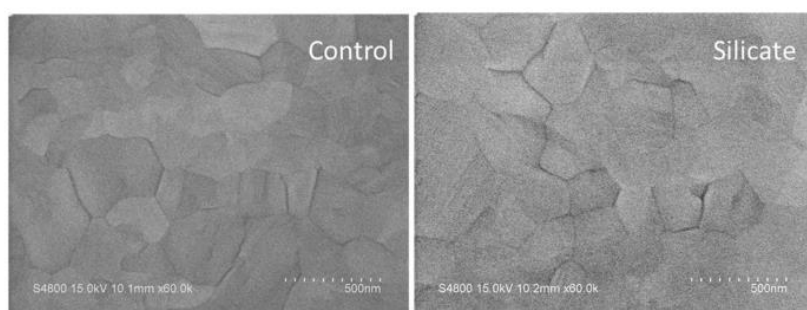
**Fig. S13** AFM images of (a) SnO<sub>2</sub> and (b) silicate-SnO<sub>2</sub> films with an area of 5×5 μm<sup>2</sup>. RMS represents root-mean-square roughness of the film



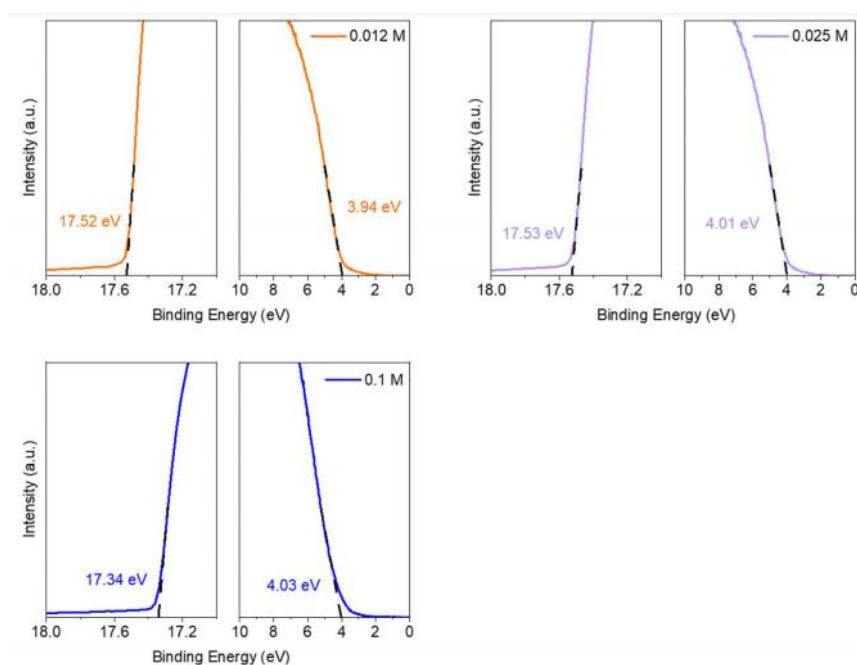
**Fig. S14** Dark *J-V* characteristics of ITO/SnO<sub>2</sub>/Ag and ITO/silicate-SnO<sub>2</sub>/Ag. Both films were spin-coated for 6 cycles to increase the film thickness.



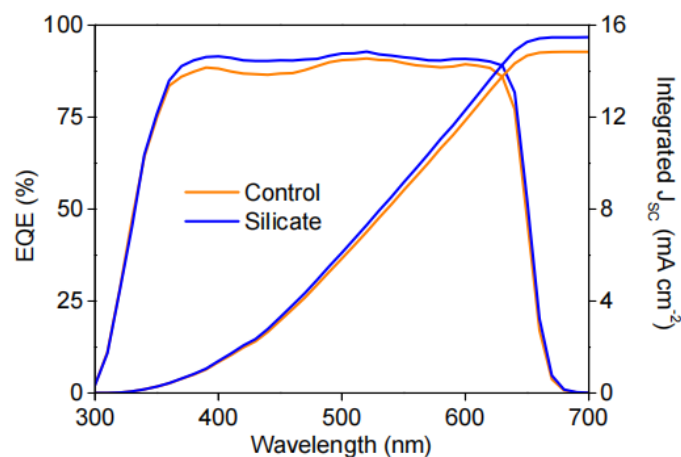
**Fig. S15** XRD patterns of CsPbI<sub>2</sub>Br films deposited on SnO<sub>2</sub> films without and with silicate. The structure of CsPbI<sub>2</sub>Br films deposited on both SnO<sub>2</sub> films can be indexed as cubic perovskite structure with preferred [100] orientation.



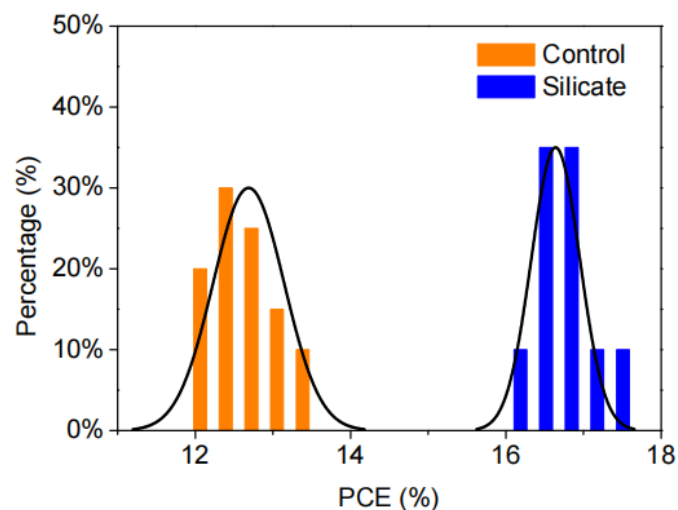
**Fig. S16** SEM images of perovskites films deposited on SnO<sub>2</sub> films without and with silicate. Both CsPbI<sub>2</sub>Br films have compact morphology with similar grain size



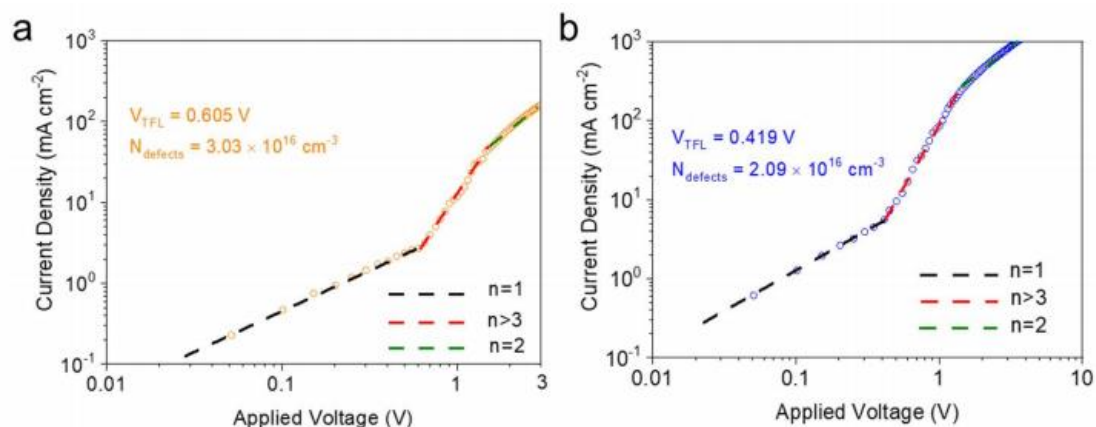
**Fig. S17** UPS spectra of the SnO<sub>2</sub> film with a varied concentration of silicate salts. The dotted lines indicate the secondary electron cutoff position and the valance band onset of the films obtained by linear extrapolating the binding energy edge to the baseline.



**Fig. S18** EQE spectra and integrated short-circuit current density of the control and silicate passivated devices



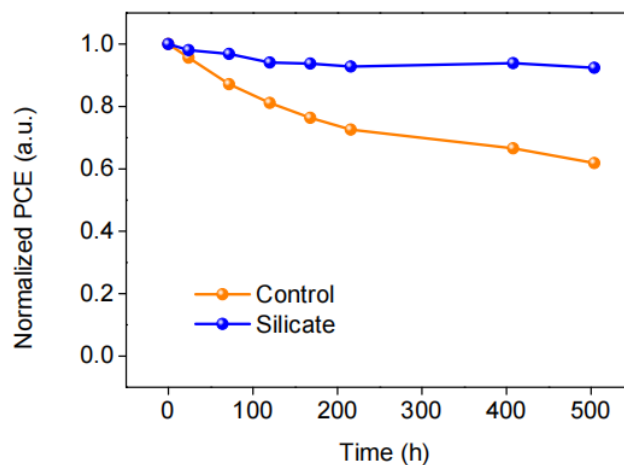
**Fig. S19** Statistics of PCE distribution for control and silicate passivated PSC devices. For each type of device, the average PCE values are obtained from 20 individual cells.



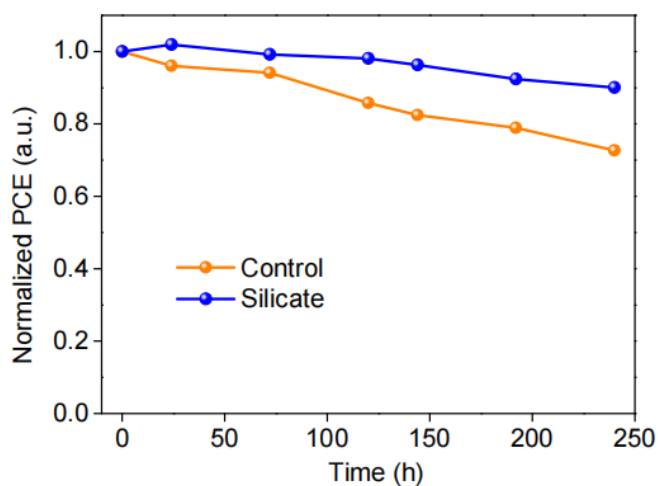
**Fig. S20** SCLC measurements of electron-only perovskite devices **a** without and **b** with silicate passivation. The device structure is ITO/SnO<sub>2</sub>/CsPbI<sub>2</sub>Br/PCBM/Ag. The measurements were carried out under dark conditions at room temperature.



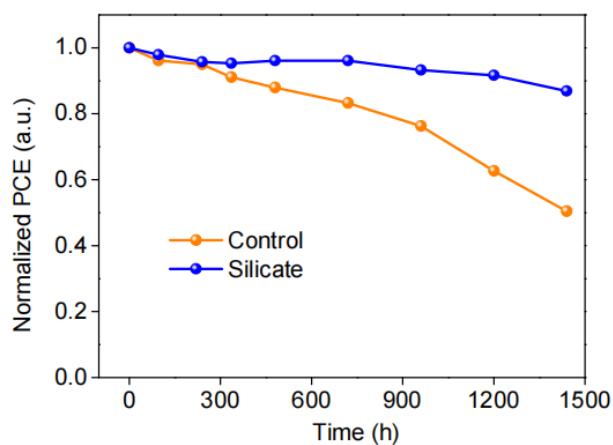
### Nano-Micro Letters



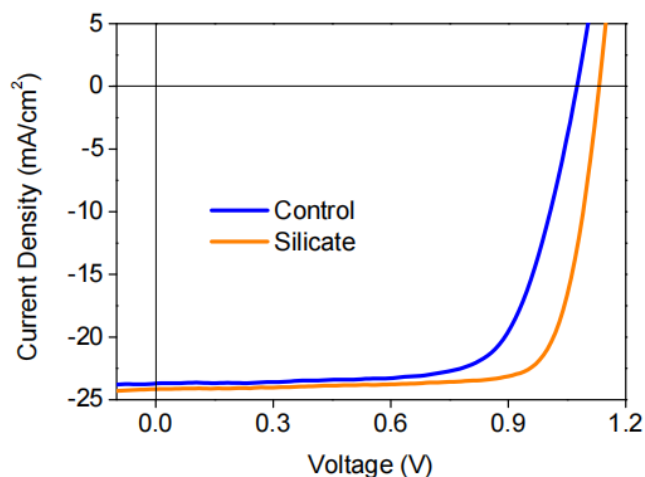
**Fig. S21** Operational stability measurement of control and silicate passivated CsPbI<sub>2</sub>Br devices under continuous one-sun illumination ( $100 \text{ mW} \cdot \text{cm}^{-2}$ ) in N<sub>2</sub>-filled glovebox.



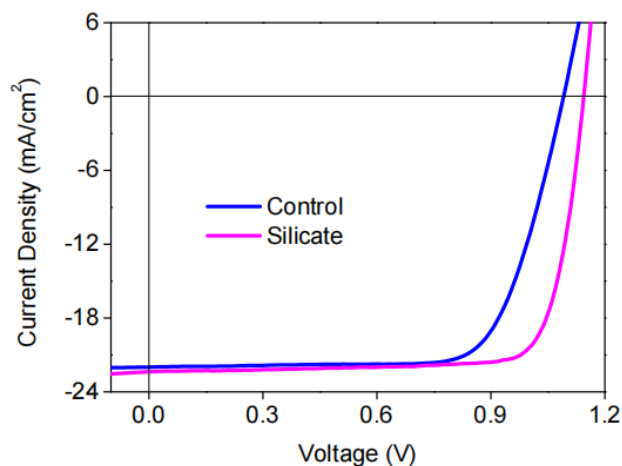
**Fig. S22** Thermal stability measurement of control device and silicate passivated devices heated to 85 °C in N<sub>2</sub>-filled glovebox



**Fig. S23** Long-term stability measurements of unencapsulated PSCs in dry ambient air. The relative humidity for the stability test is  $\sim 15 \pm 3\%$ .



**Fig. S24** J–V characteristics of the FAPbI<sub>3</sub> solar cells with and without silicate passivation measured under AM 1.5G irradiation. The device configuration is ITO/SnO<sub>2</sub>/FAPbI<sub>3</sub>/Spiro-OMeTAD/Ag. Devices were measured under simulated AM 1.5G irradiation at the reverse scan.



**Fig. S25** J–V characteristics of the MAPbI<sub>3</sub> solar cells with and without silicate passivation measured under AM 1.5G irradiation. The device configuration is FTO/NiO<sub>x</sub>/MAPbI<sub>3</sub>/PCBM/BCP/Ag. Devices were measured under simulated AM 1.5G irradiation at the reverse scan.

**Table S1** Carrier lifetimes of CsPbI<sub>2</sub>Br films without and with oxysalt passivation obtained by fitting TRPL spectra

Samples	$A1/(A1+A2)$ (%)	$\tau_1$ (ns)	$A2/(A1+A2)$ (%)	$\tau_2$ (ns)
Control	51%	4.02	49%	15.65
Nitrate	44%	4.22	56%	20.22
Sulfate	42%	4.45	58%	26.71
Carbonate	42%	4.71	58%	30.09
Phosphate	41%	5.57	59%	30.54
Silicate	38%	5.90	62%	31.99

**Table S2** Photovoltaic parameters of typical CsPbI<sub>2</sub>Br solar cells with different oxysalt passivation

Devices	Jsc (mA/cm <sup>2</sup> )	Voc (V)	FF	PCE
Control	15.14	1.16	0.70	12.35%
Nitrate	15.41	1.19	0.74	13.61%
Sulfate	15.51	1.23	0.78	14.80%
Carbonate	15.67	1.25	0.76	14.99%
Phosphate	15.76	1.29	0.76	15.41%
Silicate	15.83	1.33	0.80	16.84%

Devices were measured under simulated AM 1.5G irradiation at the reverse scan

**Table S3** Photovoltaic parameters of typical CsPbI<sub>2</sub>Br solar cells with different concentrations of silicate in SnO<sub>2</sub> layers

Concentration	J <sub>sc</sub> (mA cm <sup>-2</sup> )	V <sub>oc</sub> (V)	FF	PCE (%)
w/o	15.14	1.16	0.70	12.35%
0.012 M	15.36	1.24	0.75	14.33%
0.025 M	15.71	1.30	0.78	15.97%
0.05 M	15.83	1.33	0.80	16.84%
0.1 M	14.22	1.28	0.76	13.83%

Devices were measured under simulated AM 1.5G irradiation at the reverse scan.

**Table S4** Comparison of the photovoltaic parameters of recently reported CsPbI<sub>2</sub>Br solar cells

J <sub>sc</sub> (mA cm <sup>-2</sup> )	V <sub>oc</sub> (V)	FF	PCE (%)	Reference
15.86	1.36	0.80	17.26	This work
15.78	1.34	0.81	17.03	Nat. Commun. 2020, 11, 4237
16.34	1.334	0.801	17.46	Matter 2021, 4, 1
14.25	1.41	0.77	15.53	J. Am. Chem. Soc. 2020, 142, 9725
16.95	1.18	0.80	16.15	Nano Lett. 2019, 19, 5176.
16.82	1.15	0.75	14.69	Joule 2019, 3, 2485.
15.45	1.21	0.79	14.85	Nano Lett. 2019, 19, 6213.
15.86	1.32	0.75	15.50	ACS Energy Lett. 2019, 4, 2491.
14.00	1.28	0.78	14.00	Nat. Commun. 2019, 10, 4686.
15.30	1.30	0.81	16.20	Adv. Mater. 2019, 31, e1901152.

## Supplementary References

- [S1] J. He, J. Liu, Y. Hou, Y. Wang, S. Yang et al., Surface chelation of cesium halide perovskite by dithiocarbamate for efficient and stable solar cells. *Nat. Commun.* **11**, 4237 (2020). <https://doi.org/10.1038/s41467-020-18015-5>
- [S2] S. Tan, T. Huang, I. Yavuz, R. Wang, M. H. Weber et al., Surface reconstruction of halide perovskites during post-treatment. *J. Am. Chem. Soc.* **143**, 6781-6786 (2021). <https://doi.org/10.1021/jacs.1c00757>